

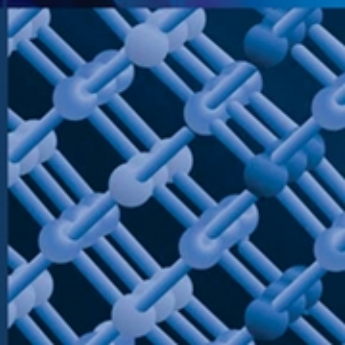
PAUL HARRISON

Quantum Wells, Wires and Dots

2nd Edition

Theoretical and Computational Physics
of Semiconductor Nanostructures

 WILEY



QUANTUM WELLS, WIRES AND DOTS

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QUANTUM WELLS, WIRES AND DOTS

**Theoretical and Computational
Physics of Semiconductor
Nanostructures**

Second Edition

Paul Harrison
The University of Leeds, UK

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To Claire, Hannah and Joe

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PREFACE

I have been working on theoretical and computational studies of the electronic, optical and magnetic properties of semiconductor heterostructures for well over a decade. During this time I have had to follow through various theoretical derivations from either books or research papers and frankly I've struggled time and time again. There never seems to be enough detail and what is commonly a couple of lines in a research paper can literally turn into many pages of precise mathematics.

With the advent of computers and their wide application to science, the emphasis of theoretical work has changed. Years ago, theoreticians aimed to produce short neat relationships, which summarised physical effects. The concentration dependency of the metal-insulator transition of Mott is an excellent example of a complex process summarised in a compact equation, which could then be evaluated by hand. The modern approach to the same problem would be to take a microscopic model of a semiconductor, calculate the bandstructure, and then vary the impurity concentration, thus deducing an empirical relationship from the numerics. The material system would then be changed and the process repeated. It would therefore appear that the old way was preferable, but in these times of materials modelling and device design, *quantitative results* are what we're paid for.

What this book is not

This book is unusual in that what it doesn't contain is as important as what it does contain, and furthermore, without the omissions this book would not exist. This book is not a book about condensed matter, nor is it a book about the physical properties of semiconductor heterostructures. It is not a book reporting the wealth of experimental measurements made on low-dimensional semiconductor systems, nor is it meant to be a general light reading book that you might cuddle up with in bed!

This book isn't even a review of all the methods that are available for calculating various properties. This book is merely an in-depth look at how quantities *can* be calculated. It is not meant to be the definitive guide; I'm sure there are better methods available than some of those presented here.

So all that remains is to say...

What this book is

This book is aimed at providing *all* of the essential information, both theoretical and computational, in order that the reader can, starting from essentially nothing, understand how the electronic, optical and transport properties of semiconductor heterostructures are calculated. However, perhaps more importantly, starting from this low common denominator, this text is designed to lead the reader through a series of simple example theoretical and computational implementations, and slowly build from solid foundations, to a level where the reader can begin to initiate theoretical investigations or explanations of their own.

I believe that there are two aspects to theoretical work, with the first being to analyse and interpret experimental data, while the second is to advance new ideas. My hope is that this book will certainly facilitate the former and I believe that I will at least provide the knowledge and skills base from which quantified predictions can be developed from the beginnings of an idea.

I hope that this book will appeal to readers from outside the low dimensional semiconductor community. Some of the examples developed will certainly be relevant to the semiconductor community at large, while the microscopic calculations presented could be of interest to other areas of condensed matter, such as carbon nanostructures, high-temperature superconductors, etc.

I have attempted here to write a book almost in the style of a mathematics course text. In such books they often describe briefly why differential equations or integration are important and then move on to show the standard techniques for solution, followed by examples and perhaps the application to real problems. Sometimes the books just state 'This is a binomial expansion and here's what you do'. In some ways this book follows both those routes. I expect that many readers will look at this present book having already a great deal of appreciation for their own particular problem. I would think that they have already quantified it in terms of, e.g. 'I must calculate the exciton binding energy'. Then they'll find that this book shows them exactly how to carry this

out, and indeed provides the computer codes for them to achieve their aim quickly. I wouldn't expect a reader to pick this work up knowing nothing about solid state physics—it is not aimed at that particular person—and as I've stated already, there are many excellent books available which describe in detail the electronic, optical, transport, and other properties of semiconductors and semiconductor heterostructures. It is not my aim here to compete with these in any way; indeed I hope this present work will complement the earlier works.

Therefore this book was written to fill a need, namely collecting and documenting together derivations. It is a chance to set the mathematics in stone. By incorporating all of the steps in a derivation there is no possibility of hiding away and 'glossing over' any point that is not fully understood. In many ways this means leaving oneself bare, for any mistakes or errors *will* be spotted, but this will give the opportunity for them to be corrected and hence allow the text to converge (perhaps through later editions) to a true record, which will be of use to future generations of students and researchers alike.

PAUL HARRISON

The University of Leeds

June 1999

The opportunity of correcting a dozen or so errors in the equations and one or two of the figures was taken for this, the first reprinting of this work. A list of these changes can be found on the book's web site.

PAUL HARRISON

The University of Leeds

April 2001

Perhaps the biggest change for this second edition has been brought about by the decision to drop the accompanying CDROM. This was done for a couple of reasons, the first one being that many of the people I met who had bought the book told me they only trusted their own codes and saw writing them as part of their own personal development. The second reason was that dropping the CDROM allowed the publisher to drop the price. With the introduction of a paperback version too, it is hoped that the book will be more accessible to students. To compensate for this the contents of the CDROM from the first edition will be published exactly as was on the book's website. Some of these codes have had updates already published so you'll need to consult with the relevant section, some of the codes will have been affected by the errata since discovered and some of the codes have basically been rewritten and improved, but you won't have access to them!

There is quite a bit of new material. In particular I've added sections on effects of magnetic fields on quantum wells, excited impurity levels, screening of the optical

phonon interaction, acoustic and optical deformation potential scattering and spin-orbit coupling in the pseudopotential calculation. Perhaps more importantly, realising my own limitations I've commissioned contributions from some of my colleagues and this has resulted in new chapters on strained quantum wells and **k.p** theory. To keep the style of the book consistent I've edited these contributed works into my style.

PAUL HARRISON

The University of Leeds

March 2005

ACKNOWLEDGEMENTS

I am indebted to so many people with whom I've worked over the years that these acknowledgements have the potential for turning into a substantial work in themselves. Almost everybody with whom I've shared an office or a research project will look through this book and be able to say 'Ah yes, I helped him with that...'. I am truly grateful to everybody and it is such discussions that have in many ways motivated me to write this book.

In particular I would like to thank people such as John Davies, Winston Hagston and John Killingbeck who motivated me in my undergraduate days and spurred me on to my PhD work. At Newcastle I was lucky to be part of an excellent group with some truly great computational physicists. Perhaps the most important of these were Ian Morrison who in less than an hour really explained the bulk pseudopotential derivation to me and Jerry Hagon who put up with my endless computing questions. I must also include my tutors, Milan Jaros and David Herbert and, of course, my colleagues Andy Beavis and Richard Turton.

Without doubt the largest contribution to my knowledge base came during my formative years as a postdoctoral research assistant with Winston Hagston. Besides sparking my interest in quantum mechanics as an undergraduate and then luring me back to Hull, Winston showed the way in two of the major derivations that I document here. In particular, the first of the quantum confined impurity calculations I attempted,

CHAPTER 1

SEMICONDUCTORS AND HETEROSTRUCTURES

1.1 THE MECHANICS OF WAVES

De Broglie (see reference [4]) stated that a particle of momentum p has an associated wave of wavelength λ given by the following

$$\lambda = \frac{h}{p} \quad (1.1)$$

Thus, an electron in a vacuum at a position \mathbf{r} and away from the influence of any electromagnetic potentials, could be described by a *state function* which is of the form of a wave, i.e.

$$\psi = e^{i(\mathbf{k} \cdot \mathbf{r} - \omega t)} \quad (1.2)$$

where t is the time, ω the angular frequency and the modulus of the wave vector is given by:

$$k = |\mathbf{k}| = \frac{2\pi}{\lambda} \quad (1.3)$$

The quantum mechanical momentum has been deduced to be a linear operator [12] acting upon the *wave function* ψ , with the momentum \mathbf{p} arising as an eigenvalue, i.e.

$$-i\hbar\nabla\psi = \mathbf{p}\psi \quad (1.4)$$

where

$$\nabla = \frac{\partial}{\partial x}\hat{\mathbf{i}} + \frac{\partial}{\partial y}\hat{\mathbf{j}} + \frac{\partial}{\partial z}\hat{\mathbf{k}} \quad (1.5)$$

which when operating on the electron vacuum wave function in equation (1.2) would give the following:

$$-i\hbar\nabla e^{i(\mathbf{k}\cdot\mathbf{r}-\omega t)} = \mathbf{p}e^{i(\mathbf{k}\cdot\mathbf{r}-\omega t)} \quad (1.6)$$

and therefore

$$-i\hbar\left(\frac{\partial}{\partial x}\hat{\mathbf{i}} + \frac{\partial}{\partial y}\hat{\mathbf{j}} + \frac{\partial}{\partial z}\hat{\mathbf{k}}\right)e^{i(k_x x + k_y y + k_z z - \omega t)} = \mathbf{p}e^{i(\mathbf{k}\cdot\mathbf{r}-\omega t)} \quad (1.7)$$

$$\therefore -i\hbar\left(ik_x\hat{\mathbf{i}} + ik_y\hat{\mathbf{j}} + ik_z\hat{\mathbf{k}}\right)e^{i(k_x x + k_y y + k_z z - \omega t)} = \mathbf{p}e^{i(\mathbf{k}\cdot\mathbf{r}-\omega t)} \quad (1.8)$$

Thus the eigenvalue:

$$\mathbf{p} = \hbar\left(k_x\hat{\mathbf{i}} + k_y\hat{\mathbf{j}} + k_z\hat{\mathbf{k}}\right) = \hbar\mathbf{k} \quad (1.9)$$

which not surprisingly can be simply manipulated ($p = \hbar k = (h/2\pi)(2\pi/\lambda)$) to reproduce de Broglie's relationship in equation (1.1).

Following on from this, classical mechanics gives the kinetic energy of a particle of mass m as

$$T = \frac{1}{2}mv^2 = \frac{(mv)^2}{2m} = \frac{p^2}{2m} \quad (1.10)$$

Therefore it may be expected that the quantum mechanical analogy can also be represented by an eigenvalue equation with an operator:

$$\frac{1}{2m}(-i\hbar\nabla)^2\psi = T\psi \quad (1.11)$$

i.e.

$$-\frac{\hbar^2}{2m}\nabla^2\psi = T\psi \quad (1.12)$$

where T is the kinetic energy eigenvalue, and given the form of ∇ in equation (1.5) then:

$$\nabla^2 = \frac{\partial^2}{\partial x^2} + \frac{\partial^2}{\partial y^2} + \frac{\partial^2}{\partial z^2} \quad (1.13)$$

When acting upon the electron vacuum wave function, i.e.

$$-\frac{\hbar^2}{2m}\nabla^2 e^{i(\mathbf{k}\cdot\mathbf{r}-\omega t)} = T e^{i(\mathbf{k}\cdot\mathbf{r}-\omega t)} \quad (1.14)$$

then

$$-\frac{\hbar^2}{2m} (i^2 k_x^2 + i^2 k_y^2 + i^2 k_z^2) e^{i(\mathbf{k} \cdot \mathbf{r} - \omega t)} = T e^{i(\mathbf{k} \cdot \mathbf{r} - \omega t)} \quad (1.15)$$

Thus the kinetic energy eigenvalue is given by:

$$T = \frac{\hbar^2 k^2}{2m} \quad (1.16)$$

For an electron in a vacuum away from the influence of electromagnetic fields, then the total energy E is just the kinetic energy T . Thus the dispersion or energy versus momentum (which is proportional to the wave vector k) curves are parabolic, just as for classical free particles, as illustrated in Fig. 1.1.

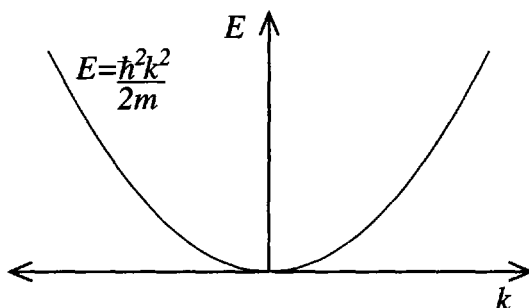


Figure 1.1 The energy versus wave vector (proportional to momentum) curve for an electron in a vacuum

In summary, the equation describing the total energy of a particle in this wave description is called the time-independent Schrödinger equation and for this case with only a kinetic energy contribution can be summarised as follows:

$$-\frac{\hbar^2}{2m} \nabla^2 \psi = E \psi \quad (1.17)$$

A corresponding equation also exists which includes the time-dependency explicitly; this is obtained by operating on the wave function by the linear operator $i\hbar \partial / \partial t$, i.e.

$$i\hbar \frac{\partial}{\partial t} e^{i(\mathbf{k} \cdot \mathbf{r} - \omega t)} = i\hbar(-i\omega) e^{i(\mathbf{k} \cdot \mathbf{r} - \omega t)} \quad (1.18)$$

i.e.

$$i\hbar \frac{\partial}{\partial t} \psi = \hbar \omega \psi \quad (1.19)$$

Clearly this eigenvalue $\hbar \omega$ is also the total energy but in a form usually associated with waves, e.g. a photon. These two operations on the wave function represent the two complimentary descriptions associated with *wave-particle duality*. Thus the second, i.e., time-dependent, Schrödinger equation is given by the following:

$$i\hbar \frac{\partial}{\partial t} \psi = E \psi \quad (1.20)$$

1.2 CRYSTAL STRUCTURE

The vast majority of the mainstream semiconductors have a face-centred cubic Bravais lattice, as illustrated in Fig 1.2. The lattice points are defined in terms of linear combinations of a set of *primitive lattice vectors*, one choice for which is:

$$\mathbf{a}_1 = \frac{A_0}{2}(\hat{\mathbf{j}} + \hat{\mathbf{k}}), \quad \mathbf{a}_2 = \frac{A_0}{2}(\hat{\mathbf{k}} + \hat{\mathbf{i}}), \quad \mathbf{a}_3 = \frac{A_0}{2}(\hat{\mathbf{i}} + \hat{\mathbf{j}}) \quad (1.21)$$

The *lattice vectors* then follow as the set of vectors:

$$\mathbf{R} = \alpha_1 \mathbf{a}_1 + \alpha_2 \mathbf{a}_2 + \alpha_3 \mathbf{a}_3 \quad (1.22)$$

where α_1 , α_2 , and α_3 are integers.

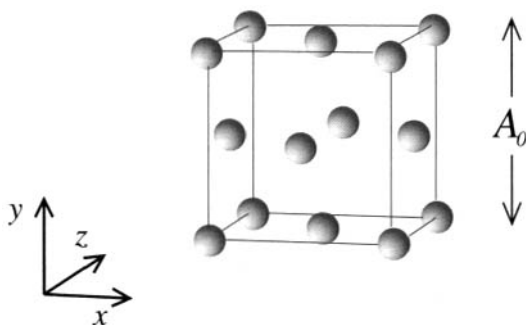


Figure 1.2 The face-centred cubic Bravais lattice

The complete crystal structure is obtained by placing the atomic basis at each Bravais lattice point. For materials such as Si, Ge, GaAs, AlAs, InP, etc., this consists of two atoms, one at $(\frac{1}{8}, \frac{1}{8}, \frac{1}{8})$ and the second at $(-\frac{1}{8}, -\frac{1}{8}, -\frac{1}{8})$, in units of A_0 .

For the group IV materials, such as Si and Ge, as the atoms within the basis are the same then the crystal structure is equivalent to diamond (see Fig. 1.3 (left)). For III–V and II–VI compound semiconductors such as GaAs, AlAs, InP, HgTe and CdTe, the cation sits on the $(-\frac{1}{8}, -\frac{1}{8}, -\frac{1}{8})$ site and the anion on $(+\frac{1}{8}, +\frac{1}{8}, +\frac{1}{8})$; this type of crystal is called the *zinc blende* structure, after ZnS, see Fig 1.3 (right). The only exception to this rule is GaN, and its important $\text{In}_x\text{Ga}_{1-x}\text{N}$ alloys, which have risen to prominence in recent years due to their use in green and blue light emitting diodes and lasers (see for example [13]) these materials have the *wurtzite* structure (see reference [2] p. 47).

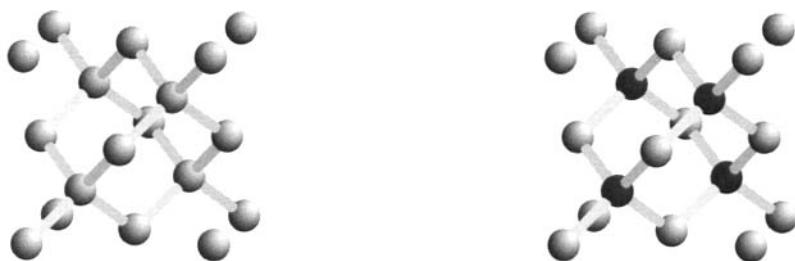


Figure 1.3 The diamond (left) and zinc blende (right) crystal structures

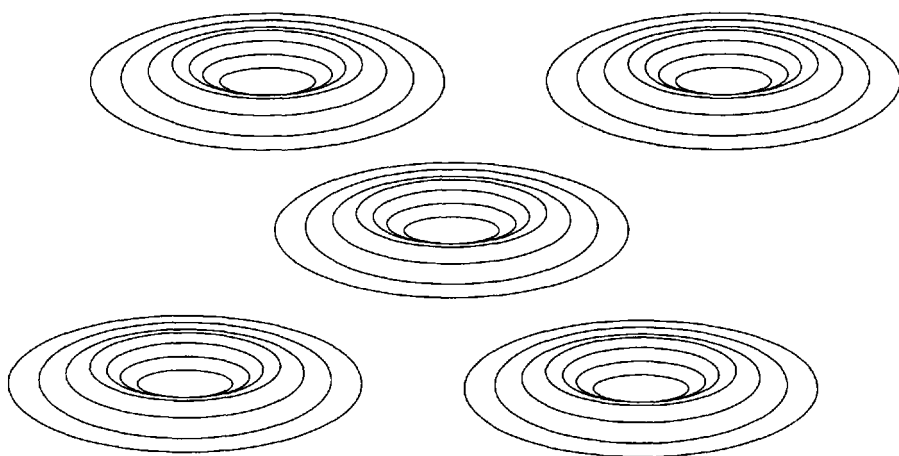


Figure 1.4 Schematic illustration of the ionic core component of the crystal potential across the $\{001\}$ planes—a three-dimensional array of spherically symmetric potentials

From an electrostatics viewpoint, the crystal potential consists of a three-dimensional lattice of spherically symmetric ionic core potentials screened by the inner shell electrons (see Fig. 1.4), which are further surrounded by the covalent bond charge distributions which hold everything together.

1.3 THE EFFECTIVE MASS APPROXIMATION

Therefore the crystal potential is complex; however using the principle of simplicity* imagine that it can be approximated by a constant! Then the Schrödinger equation derived for an electron in a vacuum would be applicable. Clearly though, a crystal isn't a vacuum so allow the introduction of an empirical fitting parameter called the effective mass, m^* . Thus the time-independent Schrödinger equation becomes:

$$-\frac{\hbar^2}{2m^*} \nabla^2 \psi = E\psi \quad (1.23)$$

and the energy solutions follow as:

$$E = \frac{\hbar^2 k^2}{2m^*} \quad (1.24)$$

This is known as the *effective mass approximation* and has been found to be very suitable for relatively low electron momenta as occur with low electric fields. Indeed, it is the most widely used parameterisation in semiconductor physics (see any good solid state physics book, e.g. [1, 2, 7]). Experimental measurements of the effective mass have revealed it to be anisotropic—as might be expected since the crystal potential along say the [001] axis is different than along the [111] axis. Adachi [14] collates reported values for GaAs and its alloys; the effective mass in other materials can be found in Landolt and Bornstein [15].

In GaAs, the reported effective mass is around $0.067 m_0$, where m_0 is the rest mass of an electron. Fig. 1.5 plots the dispersion curve for this effective mass, in comparison with that of an electron in a vacuum.

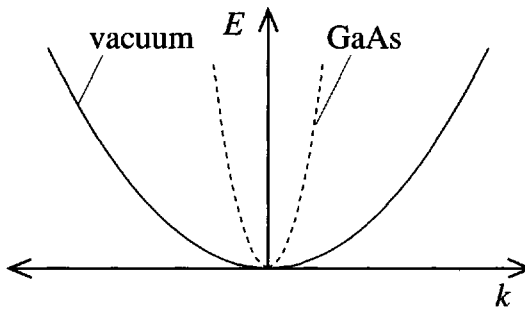


Figure 1.5 The energy versus wave vector (proportional to momentum) curves for an electron in GaAs compared to that in a vacuum

*Choose the simplest thing first; if it works use it, and if it doesn't, then try the next simplest!

1.4 BAND THEORY

It has also been found from experiment that there are two distinct energy *bands* within semiconductors. The lower band is almost full of electrons and can conduct by the movement of the empty states. This band originates from the valence electron states which constitute the covalent bonds holding the atoms together in the crystal. In many ways, electric charge in a solid resembles a fluid, and the analogy for this band, labelled the *valence band* is that the empty states behave like bubbles within the fluid—hence their name *holes*.

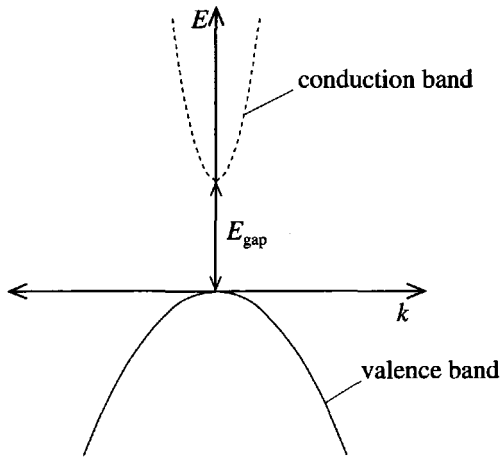


Figure 1.6 The energy versus wave vector curves for an electron in the conduction band and a hole in the valence band of GaAs

In particular, the holes rise to the uppermost point of the valence band and just as it is possible to consider the release of carbon dioxide through the motion of beer in a glass, it is actually easier to study the motion of the bubble (the absence of beer), or in this case the motion of the hole.

In a semiconductor, the upper band is almost devoid of electrons. It represents excited electron states which are occupied by electrons promoted from localised covalent bonds into extended states in the body of the crystal. Such electrons are readily accelerated by an applied electric field and contribute to current flow. This band is therefore known as the *conduction band*.

Fig. 1.6 illustrates these two bands. Notice how the valence band is inverted—this is a reflection of the fact that the ‘bubbles’ rise to the top, i.e. their lowest energy states are at the top of the band. The energy difference between the two bands is known as the *bandgap*, labelled as E_{gap} on the figure. The particular curvatures used in both bands are indicative of those measured experimentally for GaAs, namely effective masses of around $0.067 m_0$ for an electron in the conduction band, and $0.6 m_0$ for a (heavy-)hole in the valence band. The convention is to put the zero of the energy at

the top of the valence band. Note the extra qualifier ‘heavy’. In fact, there is more than one valence band, and they are distinguished by their different effective masses. Chapter 11 will discuss band structure in more detail; this will be in the context of a microscopic model of the crystal potential which goes beyond the simple ideas introduced here.

1.5 HETEROJUNCTIONS

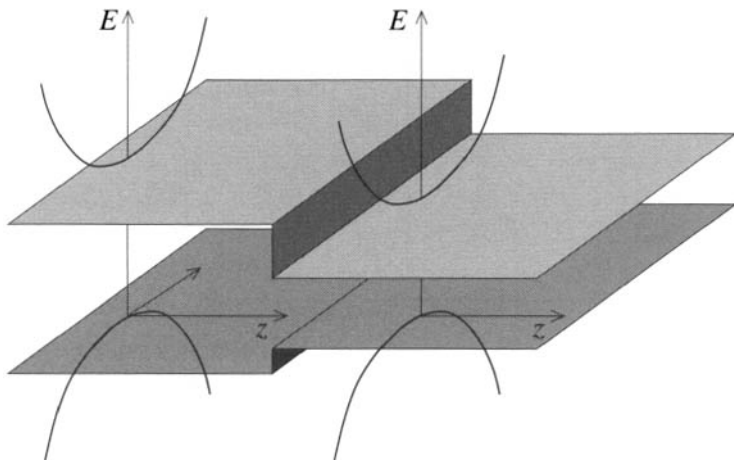


Figure 1.7 Two dissimilar semiconductors with different bandgaps joined to form a heterojunction; the curves represent the unrestricted motion parallel to the interface

The effective mass approximation is for a bulk crystal, which means the crystal is so large with respect to the scale of an electron wave function that it is effectively infinite. Within the effective mass approximation, the Schrödinger equation has been found to be as follows:

$$-\frac{\hbar^2}{2m^*} \frac{\partial^2}{\partial z^2} \psi(z) = E\psi(z) \quad (1.25)$$

When two such materials are placed adjacent to each other to form a *heterojunction*, then this equation is valid within each, remembering of course that the effective mass could be a function of position. However the bandgaps of the materials can also be different (see Fig. 1.7).

The discontinuity in either the conduction or the valence band can be represented by a constant potential term. Thus the Schrödinger equation for any one of the bands, taking the effective mass to be the same in each material, would be generalised to

$$-\frac{\hbar^2}{2m^*} \frac{\partial^2}{\partial z^2} \psi(z) + V(z)\psi(z) = E\psi(z) \quad (1.26)$$

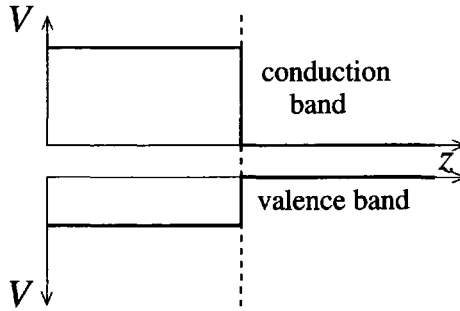


Figure 1.8 The one-dimensional potentials $V(z)$ in the conduction and valence band as might occur at a heterojunction (marked with a dashed line) between two dissimilar materials

In the above example, the one-dimensional potentials $V(z)$ representing the band discontinuities at the heterojunction would have the form shown in Fig. 1.8, noting that increasing hole energy in the valence band is measured downwards.

1.6 HETEROSTRUCTURES

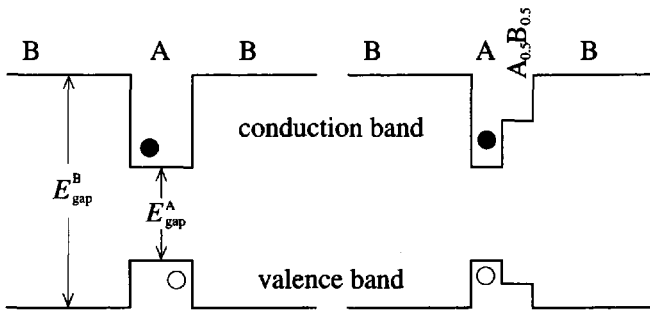


Figure 1.9 The one-dimensional potentials $V(z)$ in the conduction and valence bands for a typical single quantum well (left) and a stepped quantum well (right)

Heterostructures are formed from multiple heterojunctions, and thus a myriad of possibilities exist. If a thin layer of a narrower-bandgap material 'A' say, is sandwiched between two layers of a wider-bandgap material 'B', as illustrated in Fig. 1.9 (left) then they form a double heterojunction. If layer 'A' is sufficiently thin for *quantum properties* to be exhibited, then such a band alignment is called a *single quantum well*.

If any charge carriers exist in the system, whether thermally produced intrinsic or extrinsic as the result of doping, they will attempt to lower their energies. Hence in this example, any electrons (solid circles) or holes (open circles) will collect in the quantum well (see Fig. 1.9). Additional semiconductor layers can be included in the

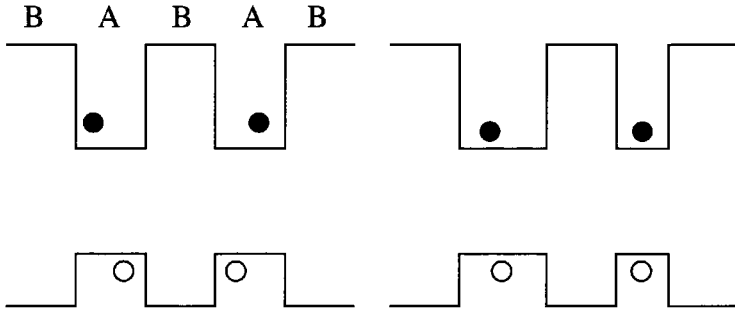


Figure 1.10 The one-dimensional potentials $V(z)$ in the conduction and valence band for typical *symmetric* (left) and *asymmetric* (right) double quantum wells

heterostructure, for example a *stepped* or *asymmetric* quantum well can be formed by the inclusion of an alloy between materials A and B, as shown in Fig. 1.9 (right).

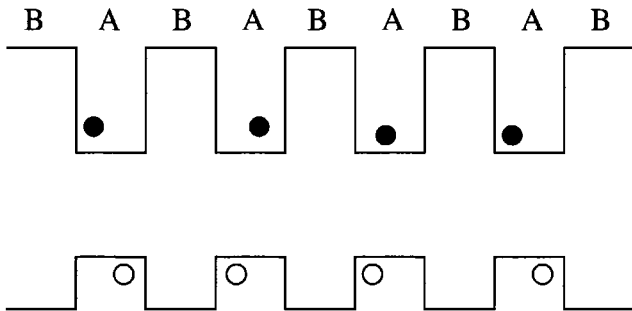


Figure 1.11 The one-dimensional potentials $V(z)$ in the conduction and valence band for a typical multiple quantum well or superlattice

Still more complex structures can be formed, such as symmetric or asymmetric *double quantum wells*, (see Fig. 1.10) and *multiple quantum wells* or *superlattices* (see Fig. 1.11). The difference between the latter is the extent of the interaction between the quantum wells; in particular, a multiple quantum well exhibits the properties of a collection of isolated single quantum wells, whereas in a superlattice the quantum wells do interact. The motivation behind introducing increasingly complicated structures is an attempt to tailor the electronic and optical properties of these materials for exploitation in devices. Perhaps the most complicated layer structure to date is the *chirped superlattice* active region of a mid-infrared laser [16].

All of the structures illustrated so far have been examples of Type-I systems. In this type, the bandgap of one material is nestled entirely within that of the wider-bandgap material. The consequence of this is that any electrons or holes fall into quantum wells which are *within* the same layer of material. Thus both types of charge carrier are

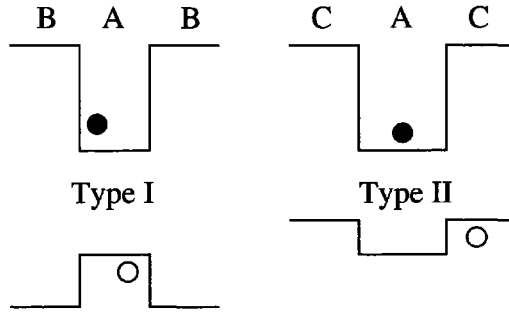


Figure 1.12 The one-dimensional potentials $V(z)$ in the conduction and valence bands for a typical Type-I single quantum well (left) compared to a Type-II system (right)

localised in the same region of space, which makes for efficient (fast) recombination. However other possibilities can exist, as illustrated in Fig. 1.12.

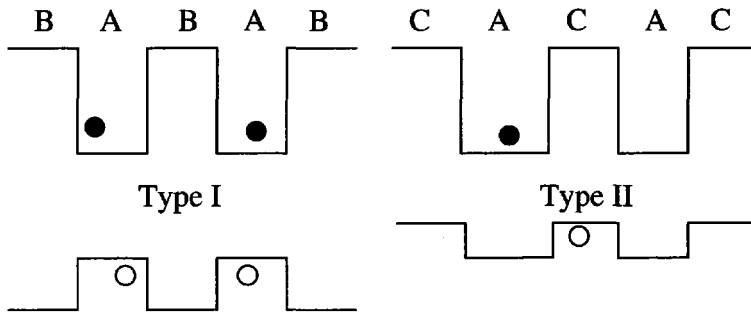


Figure 1.13 The one-dimensional potentials $V(z)$ in the conduction and valence bands for a typical Type-I superlattice (left) compared to a Type-II system (right)

In Type-II systems the bandgaps of the materials, say 'A' and 'C', are aligned such that the quantum wells formed in the conduction and valence bands are in different materials, as illustrated in Fig. 1.12 (right). This leads to the electrons and holes being confined in *different* layers of the semiconductor. The consequence of this is that the recombination times of electrons and holes are long.

1.7 THE ENVELOPE FUNCTION APPROXIMATION

Two important points have been argued:

1. The effective mass approximation is a valid description of bulk materials.

2. Heterojunctions between dissimilar materials, both of which can be well represented by the effective mass approximation, can be described by a material potential which derives from the difference in the bandgaps.

The logical extension to point 2 is that the crystal potential of multiple heterojunctions can also be described in this manner, as illustrated extensively in the previous section.

Once this is accepted, then the electronic structure *can* be represented by the simple one-dimensional Schrödinger equation that has been aspired to:

$$-\frac{\hbar^2}{2m^*} \frac{\partial^2}{\partial z^2} \psi(z) + V(z)\psi(z) = E\psi(z) \quad (1.27)$$

The *envelope function approximation* is the name given to the mathematical justification for this series of arguments (see for example works by Bastard [17, 18] and Burt [19, 20]). The name derives from the deduction that physical properties can be derived from the slowly varying envelope function, identified here as $\psi(z)$, rather than the total wave function $\psi(z)u(z)$ where the latter is rapidly varying on the scale of the crystal lattice. The validity of the envelope function approximation is still an active area of research [20]. With the line of reasoning used here, it is clear that the envelope function approximation *can* be thought of as an approximation on the material and *not* the quantum mechanics.

Some thought is enough to appreciate that the envelope function approximation will have limitations, and that these will occur for very thin layers of material. The materials *are* made of a collection of a large number of atomic potentials, so when a layer becomes thin, these individual potentials will become significant and the global average of representing the crystal potential by a constant will breakdown, for example [21]. However, for the majority of examples this approach works well; this will be demonstrated in later chapters, and, in particular, a detailed comparison with an alternative approach which does account fully for the microscopic crystal potential will be made in Chapter 12.

1.8 THE RECIPROCAL LATTICE

For later discussions the concept of the *reciprocal lattice* needs to be developed. It has already been shown that considering electron wave functions as plane waves ($e^{i\mathbf{k}\cdot\mathbf{r}}$), as found in a vacuum, but with a correction factor called the effective mass, is a useful method of approximating the electronic bandstructure. In general, such a wave will not have the periodicity of the crystal lattice; however, for certain wave vectors it will. Such a set of wave vectors \mathbf{G} are known as the *reciprocal lattice vectors* with the set of points mapped out by these primitives known as the *reciprocal lattice*.

If the set of vectors \mathbf{G} *did* have the periodicity of the lattice then this would imply that:

$$e^{i\mathbf{G}\cdot(\mathbf{r}+\mathbf{R})} = e^{i\mathbf{G}\cdot\mathbf{r}} \quad (1.28)$$

i.e. an electron with this wave vector \mathbf{G} would have a wave function equal at all points in real space separated by a Bravais lattice vector \mathbf{R} . Therefore:

$$e^{i\mathbf{G}\cdot\mathbf{r}}e^{i\mathbf{G}\cdot\mathbf{R}} = e^{i\mathbf{G}\cdot\mathbf{r}} \quad (1.29)$$

which implies that

$$\mathbf{G}\cdot\mathbf{R} = 2\pi n, \quad n \in \mathbb{Z} \quad (1.30)$$

Now learning from the form for the Bravais lattice vectors \mathbf{R} given earlier in equation (1.22), it might be expected that the reciprocal lattice vectors \mathbf{G} could be constructed in a similar manner from a set of three *primitive reciprocal lattice vectors*, i.e.

$$\mathbf{G} = \beta_1 \mathbf{b}_1 + \beta_2 \mathbf{b}_2 + \beta_3 \mathbf{b}_3, \quad \beta_1, \beta_2, \beta_3 \in \mathbb{Z} \quad (1.31)$$

With these choices then, the primitive reciprocal lattice vectors can be written as follows:

$$\mathbf{b}_1 = 2\pi \frac{\mathbf{a}_2 \times \mathbf{a}_3}{\mathbf{a}_1 \cdot (\mathbf{a}_2 \times \mathbf{a}_3)} \quad (1.32)$$

$$\mathbf{b}_2 = 2\pi \frac{\mathbf{a}_3 \times \mathbf{a}_1}{\mathbf{a}_1 \cdot (\mathbf{a}_2 \times \mathbf{a}_3)} \quad (1.33)$$

$$\mathbf{b}_3 = 2\pi \frac{\mathbf{a}_1 \times \mathbf{a}_2}{\mathbf{a}_1 \cdot (\mathbf{a}_2 \times \mathbf{a}_3)} \quad (1.34)$$

It is possible to verify that these forms do satisfy equation (1.30):

$$\mathbf{G}\cdot\mathbf{R} = (\beta_1 \mathbf{b}_1 + \beta_2 \mathbf{b}_2 + \beta_3 \mathbf{b}_3) \cdot (\alpha_1 \mathbf{a}_1 + \alpha_2 \mathbf{a}_2 + \alpha_3 \mathbf{a}_3) \quad (1.35)$$

Now \mathbf{b}_1 is perpendicular to both \mathbf{a}_2 and \mathbf{a}_3 , and so only the product of \mathbf{b}_1 with \mathbf{a}_1 is non-zero, and similarly for \mathbf{b}_2 and \mathbf{b}_3 ; hence:

$$\mathbf{G}\cdot\mathbf{R} = \beta_1 \alpha_1 \mathbf{b}_1 \cdot \mathbf{a}_1 + \beta_2 \alpha_2 \mathbf{b}_2 \cdot \mathbf{a}_2 + \beta_3 \alpha_3 \mathbf{b}_3 \cdot \mathbf{a}_3 \quad (1.36)$$

and in fact, the products $\mathbf{b}_i \cdot \mathbf{a}_i = 2\pi$; therefore:

$$\mathbf{G}\cdot\mathbf{R} = 2\pi (\beta_1 \alpha_1 + \beta_2 \alpha_2 + \beta_3 \alpha_3) \quad (1.37)$$

Clearly $\beta_1 \alpha_1 + \beta_2 \alpha_2 + \beta_3 \alpha_3$ is an integer, and hence equation (1.30) is satisfied.

Using the face-centred cubic lattice vectors defined in equation (1.21), then:

$$\mathbf{a}_1 \cdot (\mathbf{a}_2 \times \mathbf{a}_3) = \begin{vmatrix} a_{1x} & a_{1y} & a_{1z} \\ a_{2x} & a_{2y} & a_{2z} \\ a_{3x} & a_{3y} & a_{3z} \end{vmatrix} = \begin{vmatrix} 0 & \frac{A_0}{2} & \frac{A_0}{2} \\ \frac{A_0}{2} & 0 & \frac{A_0}{2} \\ \frac{A_0}{2} & \frac{A_0}{2} & 0 \end{vmatrix} \quad (1.38)$$

which gives:

$$\mathbf{a}_1 \cdot (\mathbf{a}_2 \times \mathbf{a}_3) = 0 \times \begin{vmatrix} 0 & \frac{A_0}{2} \\ \frac{A_0}{2} & 0 \end{vmatrix} - \frac{A_0}{2} \begin{vmatrix} \frac{A_0}{2} & \frac{A_0}{2} \\ \frac{A_0}{2} & 0 \end{vmatrix} + \frac{A_0}{2} \begin{vmatrix} \frac{A_0}{2} & 0 \\ \frac{A_0}{2} & \frac{A_0}{2} \end{vmatrix} \quad (1.39)$$

$$\therefore \mathbf{a}_1 \cdot (\mathbf{a}_2 \times \mathbf{a}_3) = 0 - \frac{A_0}{2} \left[0 - \left(\frac{A_0}{2} \right)^2 \right] + \frac{A_0}{2} \left[\left(\frac{A_0}{2} \right)^2 - 0 \right] \quad (1.40)$$

$$\therefore \mathbf{a}_1 \cdot (\mathbf{a}_2 \times \mathbf{a}_3) = 2 \left(\frac{A_0}{2} \right)^3 \quad (1.41)$$

Therefore, the first of the primitive reciprocal lattice vectors follows as:

$$\mathbf{b}_1 = 2\pi \left[2 \left(\frac{A_0}{2} \right)^3 \right]^{-1} \begin{vmatrix} \hat{\mathbf{i}} & \hat{\mathbf{j}} & \hat{\mathbf{k}} \\ \frac{A_0}{2} & 0 & \frac{A_0}{2} \\ \frac{A_0}{2} & \frac{A_0}{2} & 0 \end{vmatrix} \quad (1.42)$$

$$\begin{aligned} \therefore \mathbf{b}_1 &= 2\pi \frac{1}{2} \left(\frac{2}{A_0} \right)^3 \\ &\times \left\{ \left[0 - \left(\frac{A_0}{2} \right)^2 \right] \hat{\mathbf{i}} - \left[0 - \left(\frac{A_0}{2} \right)^2 \right] \hat{\mathbf{j}} + \left[\left(\frac{A_0}{2} \right)^2 - 0 \right] \hat{\mathbf{k}} \right\} \end{aligned} \quad (1.43)$$

$$\therefore \mathbf{b}_1 = \frac{2\pi}{A_0} (-\hat{\mathbf{i}} + \hat{\mathbf{j}} + \hat{\mathbf{k}}) \quad (1.44)$$

A similar calculation of the remaining primitive reciprocal lattice vectors \mathbf{b}_2 and \mathbf{b}_3 gives the complete set as follows:

$$\mathbf{b}_1 = \frac{2\pi}{A_0} (-\hat{\mathbf{i}} + \hat{\mathbf{j}} + \hat{\mathbf{k}}), \quad \mathbf{b}_2 = \frac{2\pi}{A_0} (\hat{\mathbf{i}} - \hat{\mathbf{j}} + \hat{\mathbf{k}}), \quad \mathbf{b}_3 = \frac{2\pi}{A_0} (\hat{\mathbf{i}} + \hat{\mathbf{j}} - \hat{\mathbf{k}}) \quad (1.45)$$

which are of course equivalent to the body-centred cubic Bravais lattice vectors (see reference [1], p. 68). Thus the reciprocal lattice constructed from the linear combinations:

$$\mathbf{G} = \beta_1 \mathbf{b}_1 + \beta_2 \mathbf{b}_2 + \beta_3 \mathbf{b}_3, \quad \beta_1, \beta_2, \beta_3 \in \mathbb{Z} \quad (1.46)$$

is a body-centred cubic lattice with lattice constant $4\pi/A_0$. Taking the face-centred cubic primitive reciprocal lattice vectors in equation (1.45), then:

$$\mathbf{G} = \frac{2\pi}{A_0} \left[\beta_1 (-\hat{\mathbf{i}} + \hat{\mathbf{j}} + \hat{\mathbf{k}}) + \beta_2 (\hat{\mathbf{i}} - \hat{\mathbf{j}} + \hat{\mathbf{k}}) + \beta_3 (\hat{\mathbf{i}} + \hat{\mathbf{j}} - \hat{\mathbf{k}}) \right] \quad (1.47)$$

$$\therefore \mathbf{G} = \frac{2\pi}{A_0} \left[(-\beta_1 + \beta_2 + \beta_3) \hat{\mathbf{i}} + (\beta_1 - \beta_2 + \beta_3) \hat{\mathbf{j}} + (\beta_1 + \beta_2 - \beta_3) \hat{\mathbf{k}} \right] \quad (1.48)$$

The specific reciprocal lattice vectors are therefore generated by taking different combinations of the integers β_1 , β_2 , and β_3 . This is illustrated in Table 1.1.

It was shown by von Laue that when waves in a periodic structure satisfied the following:

$$\mathbf{k} \cdot \hat{\mathbf{G}} = \frac{1}{2} |\mathbf{G}| \quad (1.49)$$

then diffraction would occur (see reference [1], p. 99). Thus the 'free' electron dispersion curves of earlier (Fig. 1.5), will be perturbed when the electron wave

Table 1.1 Generation of the reciprocal lattice vectors for the face-centred cubic crystal by the systematic selection of the integer coefficients β_1 , β_2 , and β_3

β_1	β_2	β_3	$\mathbf{G} (2\pi/A_0)$
0	0	0	(0, 0, 0)
1	0	0	($\bar{1}$, 1, 1)
0	1	0	(1, $\bar{1}$, 1)
0	0	1	(1, 1, $\bar{1}$)
-1	0	0	(1, $\bar{1}$, $\bar{1}$)
0	-1	0	($\bar{1}$, 1, $\bar{1}$)
0	0	-1	($\bar{1}$, $\bar{1}$, 1)
1	1	1	(1, 1, 1)
-1	-1	-1	($\bar{1}$, $\bar{1}$, $\bar{1}$)
1	1	0	(0, 0, 2)
1	0	1	(0, 2, 0)
0	1	1	(2, 0, 0)
-1	1	0	(2, $\bar{2}$, 2)
(etc.)			

vector satisfies equation (1.49). Along the [001] direction, the smallest reciprocal lattice vector \mathbf{G} is (0,0,2) (in units of $2\pi/A_0$). Substituting into equation (1.49) gives:

$$\mathbf{k}_\bullet \cdot (0\hat{\mathbf{i}} + 0\hat{\mathbf{j}} + \hat{\mathbf{k}}) = \frac{1}{2} \times 2 \times \frac{2\pi}{A_0} \quad (1.50)$$

This then implies the electron will be diffracted when:

$$\mathbf{k} = \frac{2\pi}{A_0} \hat{\mathbf{k}} \quad (1.51)$$

Fig. 1.14 illustrates the effect that such diffraction would have on the 'free-electron' curves. At wavevectors which satisfy von Laue's condition, the energy bands are disturbed and an energy gap opens. Such an improvement on the parabolic dispersion curves of earlier, is known as the *nearly free electron model*.

The space between the lowest wavevector solutions to von Laue's condition is called the first *Brillouin zone*. Note that the reciprocal lattice vectors in any particular direction span the Brillouin zone. As mentioned above a face-centred cubic lattice has a body-centred cubic reciprocal lattice, and thus the Brillouin zone is therefore a three-dimensional solid, which happens to be a 'truncated octahedron' (see, for example reference [1], p. 89). High-symmetry points around the Brillouin zone are often labelled for ease of reference, with the most important of these, for this work, being the $k = 0$ point, referred to as 'T', and the $\langle 001 \rangle$ zone edges, which are called the 'X' points.

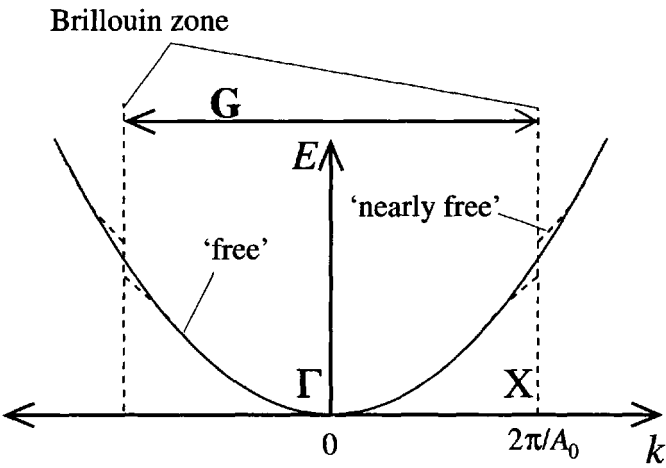


Figure 1.14 Comparison of the free and nearly free electron models

CHAPTER 2

SOLUTIONS TO SCHRÖDINGER'S EQUATION

2.1 THE INFINITE WELL

The infinitely deep one-dimensional potential well is the simplest confinement potential to treat in quantum mechanics. Virtually every introductory level text on quantum mechanics considers this system, but nonetheless it is worth visiting again as some of the standard assumptions often glossed over, do have important consequences for one-dimensional confinement potentials in general.

The time-independent Schrödinger equation summarises the wave mechanics analogy to Hamilton's formulation of classical mechanics [22], for *time-independent potentials*. In essence this states that the kinetic and potential energy components sum to the total energy; in wave mechanics, these quantities are the eigenvalues of linear operators, i.e.

$$T\psi + V\psi = E\psi \quad (2.1)$$

where the eigenfunction ψ describes the state of the system. Again in analogy with classical mechanics the kinetic energy operator for a particle of *constant* mass is given

by the following:

$$T = \frac{\mathcal{P}^2}{2m} \quad (2.2)$$

where \mathcal{P} is the usual quantum mechanical linear momentum operator:

$$\mathcal{P} = -i\hbar\nabla = -i\hbar\left(\frac{\partial}{\partial x}\mathbf{i} + \frac{\partial}{\partial y}\mathbf{j} + \frac{\partial}{\partial z}\mathbf{k}\right) \quad (2.3)$$

By using this form for the kinetic energy operator T , the Schrödinger equation then becomes:

$$-\frac{\hbar^2}{2m}\left(\frac{\partial^2}{\partial x^2} + \frac{\partial^2}{\partial y^2} + \frac{\partial^2}{\partial z^2}\right)\psi + V(x, y, z)\psi = E\psi \quad (2.4)$$

where the function $V(x, y, z)$ represents the potential energy of the system as a function of the spatial coordinates. Restricting this to the one-dimensional potential of interest here, then the Schrödinger equation for a particle of mass m in a potential well aligned along the z -axis (as in Fig. 2.1) would be:

$$-\frac{\hbar^2}{2m}\frac{\partial^2}{\partial z^2}\psi(z) + V(z)\psi(z) = E\psi(z) \quad (2.5)$$

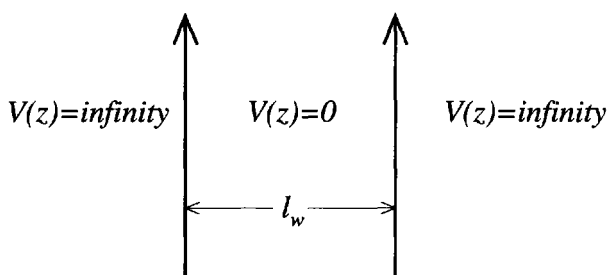


Figure 2.1 The one-dimensional infinite well confining potential

Outside of the well, $V(z) = \infty$, and hence the only possible solution is $\psi(z) = 0$, which in turn implies that all values of the energy E are allowed. Within the potential well, the Schrödinger equation simplifies to:

$$-\frac{\hbar^2}{2m}\frac{\partial^2}{\partial z^2}\psi(z) = E\psi(z) \quad (2.6)$$

which implies that the solution for ψ is a linear combination of the functions $f(z)$ which when differentiated twice give $-f(z)$. Hence try the solution:

$$\psi(z) = A \sin kz + B \cos kz \quad (2.7)$$

Substituting into equation (2.6) then gives:

$$\frac{\hbar^2 k^2}{2m} (A \sin kz + B \cos kz) = E (A \sin kz + B \cos kz) \quad (2.8)$$

$$\therefore \frac{\hbar^2 k^2}{2m} = E \quad (2.9)$$

Consideration of the boundary conditions will yield the, as yet unknown, constant k . With this aim, consider again the kinetic energy term for this system, i.e.

$$T = -\frac{\hbar^2}{2m} \frac{\partial^2}{\partial z^2} \psi(z) \quad (2.10)$$

which can be rewritten as

$$T = -\frac{\hbar^2}{2m} \frac{\partial}{\partial z} \left(\frac{\partial}{\partial z} \psi(z) \right) \quad (2.11)$$

The mathematical form of this implies that, *as a minimum*, $\psi(z)$ must be continuous. If it is not, then the first derivative will contain poles which must be avoided if the system is to have finite values for the kinetic energy. Given that $\psi(z)$ has already been deduced as zero outside of the well, then $\psi(z)$ within the well must be zero at both edges too.

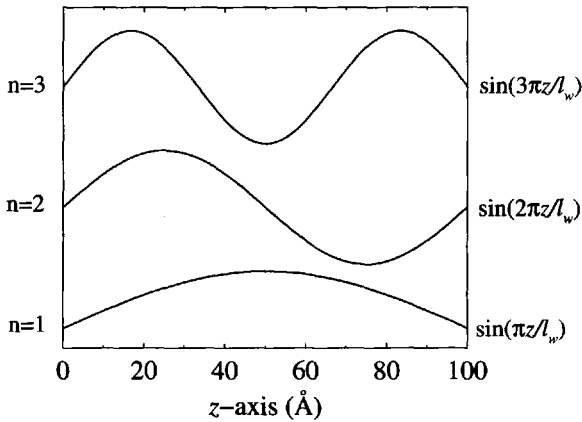


Figure 2.2 Solutions to the one-dimensional infinite well confining potential

If the origin is taken as the left hand edge of the well as in Fig. 2.2, then $\psi(z)$ as defined in equation (2.8) can contain no cosine terms, i.e. $B = 0$, and hence $\psi(z) = A \sin kz$. In addition, for $\psi(0) = \psi(l_w) = 0$:

$$k = \frac{\pi n}{l_w} \quad (2.12)$$

where n is an integer, representing a series of solutions. Substituting into equation (2.9), then the energy of the confined states is given by:

$$E_n = \frac{\hbar^2 \pi^2 n^2}{2m l_w^2} \quad (2.13)$$

The only remaining unknown is the constant factor A , which is deduced by considering the normalisation of the wave function; as $\psi^*(z)\psi(z)$ represents the probability of finding the particle at a point z , then as the particle must exist somewhere:

$$\int_0^{l_w} \psi^*(z)\psi(z) dz = 1 \quad (2.14)$$

which gives $A = \sqrt{(2/l_w)}$, and therefore

$$\psi_n(z) = \sqrt{\frac{2}{l_w}} \sin\left(\frac{\pi n z}{l_w}\right) \quad (2.15)$$

Under the effective mass and envelope function approximations, the energy of an electron or hole in a hypothetical infinitely deep semiconductor quantum well can be calculated by using the effective mass m^* for the particle mass m of equation (2.13).

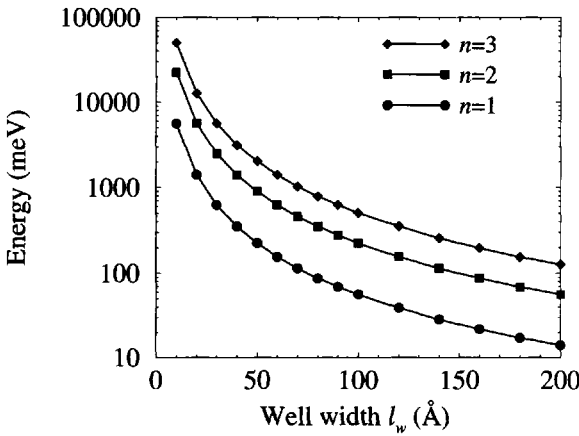


Figure 2.3 First three energy levels versus well width for an electron in a GaAs infinite potential well

Figure 2.3 displays the results of calculations of the lowest three energy states of an electron in a GaAs well of width l_w surrounded by hypothetical infinite barriers (for these and all material parameters see Appendix A). All three states show the same monotonic behaviour, with the energy decreasing as the well width increases.

The sine function solutions derived for this system are completely standard and found extensively in the literature. Although it should be noted that the arguments developed for setting the boundary conditions, i.e. $\psi(z)$ continuous, also implied that the first derivative should be continuous too, although use is never made of this second boundary condition. The limitations of solution imposed by this are avoided by saying

that not only is the potential infinite outside the well, but in addition the Schrödinger equation is not defined in these regions—a slight contradiction with the deduction of the first boundary condition. This point, i.e. that there is still ambiguity in the choice of boundary conditions for commonly accepted solutions, will be revisited later in this chapter.

2.2 IN-PLANE DISPERSION

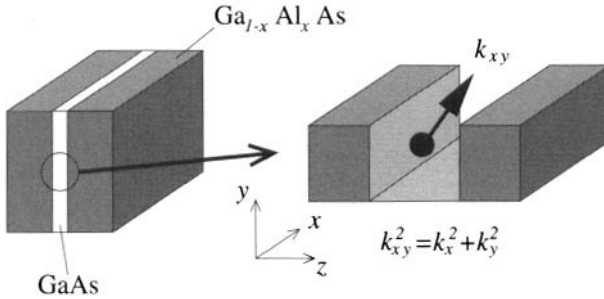


Figure 2.4 A GaAs/Ga_{1-x}Al_xAs layered structure and the in-plane motion of a charge carrier

If the one-dimensional potential $V(z)$ is constructed from alternating thin layers of dissimilar semiconductors, then the particle, whether it be an electron or a hole, can move in the plane of the layers (see Fig 2.4).

In this case, all of the terms of the kinetic energy operator are required, and hence the Schrödinger equation would be as follows:

$$-\frac{\hbar^2}{2m} \left(\frac{\partial^2}{\partial x^2} + \frac{\partial^2}{\partial y^2} + \frac{\partial^2}{\partial z^2} \right) \psi + V(z)\psi = E\psi \quad (2.16)$$

As the potential can be written as a sum of independent functions, i.e. $V = V(x) + V(y) + V(z)$, where it just happens in this case that $V(x) = V(y) = 0$, the eigenfunction of the system be written as a product:

$$\psi(x, y, z) = \psi_x(x)\psi_y(y)\psi_z(z) \quad (2.17)$$

Using this in the above Schrödinger equation then:

$$-\frac{\hbar^2}{2m} \left(\frac{\partial^2 \psi_x}{\partial x^2} \psi_y \psi_z + \frac{\partial^2 \psi_y}{\partial y^2} \psi_x \psi_z + \frac{\partial^2 \psi_z}{\partial z^2} \psi_x \psi_y \right) + V(z)\psi_x \psi_y \psi_z = E\psi_x \psi_y \psi_z \quad (2.18)$$

It is then possible to identify three distinct contributions to the total energy E , one from each of the perpendicular x -, y -, and z -axes, i.e. $E = E_x + E_y + E_z$. It is said

that the motions 'de-couple' giving an equation of motion for each of the axes:

$$-\frac{\hbar^2}{2m} \frac{\partial^2 \psi_x}{\partial x^2} \psi_y \psi_z = E_x \psi_x \psi_y \psi_z \quad (2.19)$$

$$-\frac{\hbar^2}{2m} \frac{\partial^2 \psi_y}{\partial y^2} \psi_x \psi_z = E_y \psi_x \psi_y \psi_z \quad (2.20)$$

$$-\frac{\hbar^2}{2m} \frac{\partial^2 \psi_z}{\partial z^2} \psi_x \psi_y + V(z) \psi_x \psi_y \psi_z = E_z \psi_x \psi_y \psi_z \quad (2.21)$$

Dividing throughout, then:

$$-\frac{\hbar^2}{2m} \frac{\partial^2 \psi_x}{\partial x^2} = E_x \psi_x \quad (2.22)$$

$$-\frac{\hbar^2}{2m} \frac{\partial^2 \psi_y}{\partial y^2} = E_y \psi_y \quad (2.23)$$

$$-\frac{\hbar^2}{2m} \frac{\partial^2 \psi_z}{\partial z^2} + V(z) \psi_z = E_z \psi_z \quad (2.24)$$

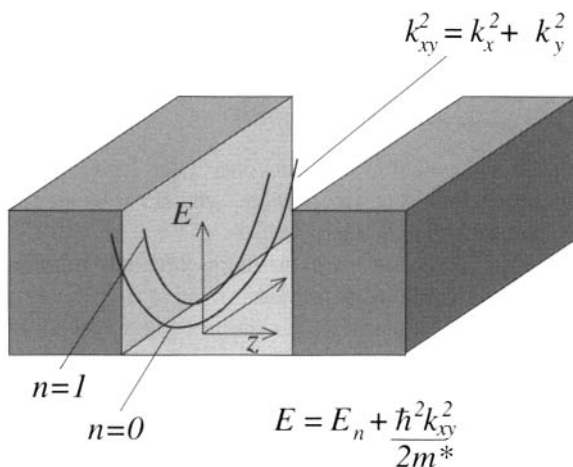


Figure 2.5 Schematic showing the in-plane ($k_{x,y}$) dispersion curves and the subband structure

The last component is identical to the one-dimensional Schrödinger equation for a confining potential $V(z)$ as discussed, for the particular case of an infinite well, in the last section. Consider the first and second components. Again, an eigenfunction f is sought which when differentiated twice returns $-f$; however, in this case it must be remembered that the solution will represent a moving particle. Thus the eigenfunction must reflect a current flow and have complex components, so try the standard travelling wave, $\exp(ik_x x)$. Then:

$$-\frac{\hbar^2}{2m} \frac{\partial^2}{\partial x^2} \exp(ik_x x) = E_x \exp(ik_x x) \quad (2.25)$$

$$\therefore \frac{\hbar^2 k_x^2}{2m} = E_x \quad (2.26)$$

which is clearly just the kinetic energy of a wave travelling along the x -axis. A similar equation follows for the y -axis, and hence the in-plane motion of a particle in a one-dimensional confining potential, but of infinite extent in the x - y plane can be summarised as:

$$\psi_{x,y}(x, y) = \frac{1}{A} \exp [i(k_x x + k_y y)] \quad \text{and} \quad E_{x,y} = \frac{\hbar^2 |\mathbf{k}_{x,y}|^2}{2m} \quad (2.27)$$

Therefore, while solutions of Schrödinger's equation along the axis of the one-dimensional potential produce discrete states of energy $E_z = E_n$, in the plane of a semiconductor quantum well there is a continuous range of allowed energies, as illustrated in Fig. 2.5. In bulk materials, such domains are called 'energy bands', while in quantum well systems these energy domains associated with confined levels are referred to as 'subbands'. Therefore the effect of the one-dimensional confining potential is to remove a degree of freedom, thus restricting the momentum of the charge carrier from three-dimensions to two. It is for this reason that the states within quantum well systems are generally referred to as two-dimensional.

Later in this text, quantum wires and dots will be considered which further restrict the motion of carriers in two and three dimensions respectively, thus giving rise to the terms one- and zero-dimensional states.

Summarising then, within a semiconductor quantum well system the total energy of an electron or hole, of mass m^* , with in-plane momentum $k_{x,y}$, is equal to $E_z + E_{x,y}$, which is given by:

$$E = E_n + \frac{\hbar^2 |\mathbf{k}_{x,y}|^2}{2m^*} \quad (2.28)$$

2.3 DENSITY OF STATES

Therefore the original confined states within the one-dimensional potential which could each hold two charge carriers of opposite spin, from the Pauli exclusion principle, broaden into subbands, thus allowing a continuous range of carrier momenta. In order to answer the question 'Given a particular number of electrons (or holes) within a subband, what is the distribution of their energy and momenta?', the first point that is required is a knowledge of the density of states, i.e. how many electrons can exist within a range of energies. In order to answer this point for the case of subbands in quantum wells, it is necessary first to understand this property in bulk crystals.

Following the idea behind Bloch's theorem (see reference [1] p. 133) that an eigenstate within a *bulk* semiconductor, which can be written as $\psi = (1/\Omega) \exp(i\mathbf{k} \cdot \mathbf{r})$, must display periodicity within the lattice, then if the unit cell is of side L :

$$\psi(x, y, z) = \psi(x + L, y + L, z + L) \quad (2.29)$$

$$\therefore \psi(x, y, z) = \frac{1}{\Omega} \exp \{i[k_x(x + L) + k_y(y + L) + k_z(z + L)]\} \quad (2.30)$$

$$\therefore \psi(x, y, z) = \frac{1}{\Omega} \exp [i(k_x x + k_y y + k_z z)] \exp [i(k_x L + k_y L + k_z L)] \quad (2.31)$$

Thus for the periodicity condition to be fulfilled, the second exponential term must be identical to 1, which implies that:

$$k_x = \frac{2\pi}{L} n_x \quad k_y = \frac{2\pi}{L} n_y \quad k_z = \frac{2\pi}{L} n_z \quad (2.32)$$

where n_x , n_y and n_z are integers. Each set of values of these three integers defines a distinct state, and hence the volume of k -space occupied by one state is $(2\pi/L)^3$. These states fill up with successively larger values of n_x , n_y and n_z , i.e. the lowest energy state has values (000), then permutations of (100), (110), etc., which gradually fill a sphere. At low temperatures, the sphere has a definite boundary between states that are all occupied followed by states that are unoccupied; the momentum of these states is called the *Fermi wave vector* and the equivalent energy is the *Fermi energy*. At higher temperatures, carriers near the edge of the sphere are often scattered to higher energy states, thus 'blurring' the boundary between occupied and unoccupied states. For a more detailed description see, for example, Ashcroft and Mermin [1].

Many of the interesting phenomena associated with semiconductors derive from the properties of electrons near the Fermi energy, as it is these electrons that are able to scatter into nearby states thus changing both their energy and momenta. In order to be able to progress with descriptions of, transport for example (later in this book), it is necessary to be able to describe the density of available states.

The density of states is defined as the number of states per energy per unit volume of *real* space:

$$\rho(E) = \frac{dN}{dE} \quad (2.33)$$

In k -space, the total number of states N is equal to the volume of the sphere of radius k , divided by the volume occupied by one state and divided again by the volume of real space, i.e.

$$N = 2 \frac{4\pi k^3}{3} \frac{1}{(2\pi/L)^3} \frac{1}{L^3} \quad (2.34)$$

$$\therefore N = 2 \frac{4\pi k^3}{3(2\pi)^3} \quad (2.35)$$

where the factor 2 has been introduced to allow for double occupancy of each state by the different carrier spins. Returning to the density of states, then:

$$\rho(E) = \frac{dN}{dE} = \frac{dN}{dk} \frac{dk}{dE} \quad (2.36)$$

Now equation (2.35) gives

$$\frac{dN}{dk} = 2 \frac{4\pi k^2}{(2\pi)^3} \quad (2.37)$$

In addition, the parabolic bands of effective mass theory give:

$$E = \frac{\hbar^2 k^2}{2m^*} \quad \therefore k = \left(\frac{2m^* E}{\hbar^2} \right)^{\frac{1}{2}} \quad (2.38)$$

$$\therefore \frac{dk}{dE} = \left(\frac{2m^*}{\hbar^2} \right)^{\frac{1}{2}} \frac{E^{-\frac{1}{2}}}{2} \quad (2.39)$$

Which finally gives the density of states in bulk as:

$$\rho(E) = \frac{1}{2\pi^2} \left(\frac{2m^*}{\hbar^2} \right)^{\frac{3}{2}} E^{\frac{1}{2}} \quad (2.40)$$

Thus the density of states within a band, and around a minimum where the energy can be represented as a parabolic function of momentum, is continual and proportional to the square root of the energy.

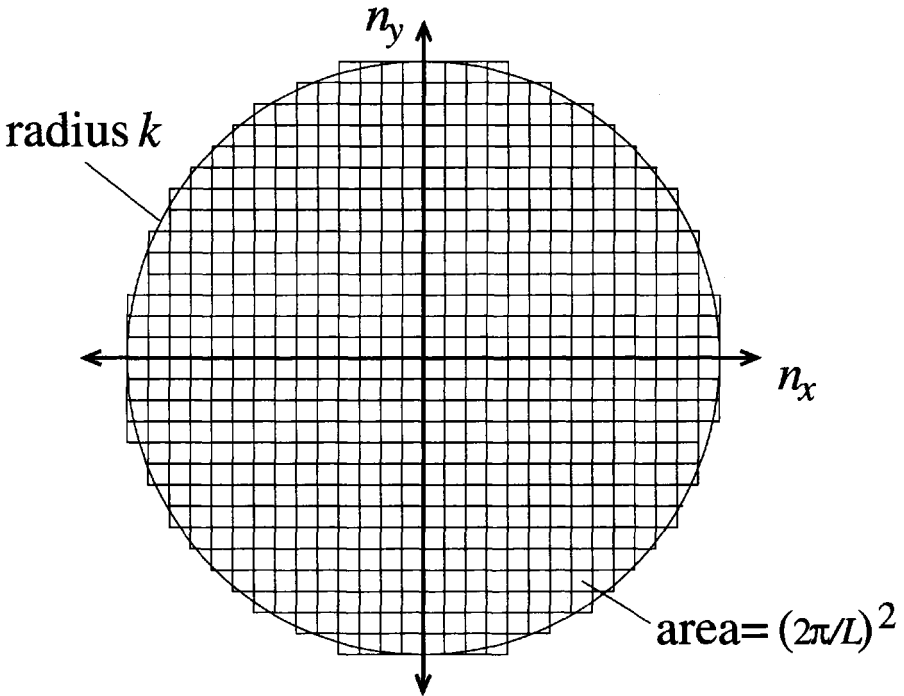


Figure 2.6 Illustration of filling the two-dimensional momenta states in a quantum well

The density of states in quantum well systems follows analogously; however this time, as there are only two degrees of freedom, successive states represented by values of n_x and n_y fill a circle in k -space, as illustrated in Fig. 2.6. Such a situation has

become known as a two-dimensional electron (or hole) gas (2DEG). Hence the total number of states per unit cross-sectional area is given by the spin degeneracy factor, multiplied by the area of the circle of radius k , divided by the area occupied by each state, i.e.

$$N^{2D} = 2 \pi k^2 \frac{1}{(2\pi/L)^2} \frac{1}{L^2} \quad (2.41)$$

$$\therefore N^{2D} = 2 \frac{\pi k^2}{(2\pi)^2} \quad (2.42)$$

$$\therefore \frac{dN^{2D}}{dk} = \frac{k}{\pi} \quad (2.43)$$

In analogy to the bulk three-dimensional (3D) case, define:

$$\rho^{2D}(E) = \frac{dN^{2D}}{dE} = \frac{dN^{2D}}{dk} \frac{dk}{dE} \quad (2.44)$$

As the in-plane dispersion curves are still described by parabolas, then reuse can be made of equation (2.39), as follows:

$$\rho^{2D}(E) = \frac{k}{\pi} \left(\frac{2m^*}{\hbar^2} \right)^{\frac{1}{2}} \frac{E^{-\frac{1}{2}}}{2} \quad (2.45)$$

By substituting for k in terms of the energy E , using equation (2.38) then finally the density of states for a single subband in a quantum well system is given by:

$$\rho^{2D}(E) = \frac{m^*}{\pi \hbar^2} \quad (2.46)$$

in agreement with Bastard [18] p. 12.

If there are many (n) confined states within the quantum well system then the density of states ρ^{2D} at any particular energy is the sum over all subbands *below* that point, which can be written succinctly as:

$$\rho^{2D}(E) = \sum_{i=1}^n \frac{m^*}{\pi \hbar^2} \Theta(E - E_i) \quad (2.47)$$

where Θ is the unit step function. Fig. 2.7 gives an example of the two-dimensional density of states for a particular quantum well showing the first three confined levels. Note that the steps are of equal height and occur at the subband minima—which are not equally spaced.

2.4 SUBBAND POPULATIONS

The total number of carriers within a subband is given by the integral of the product of the probability of occupation of a state and the density of states. Given that the

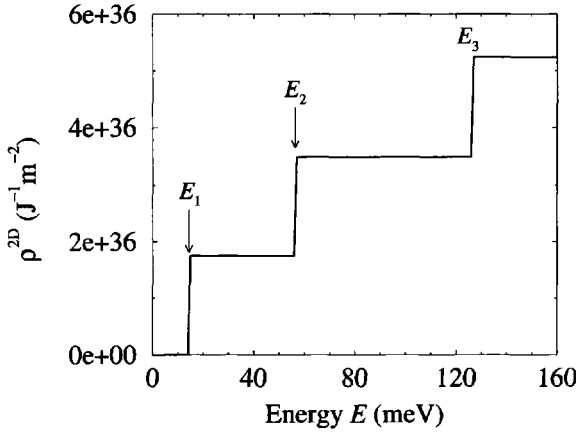


Figure 2.7 The density of states as a function of energy for a 200 Å GaAs quantum well surrounded by infinite barriers

carriers are fermions, then clearly the probability of occupation of a state is given by Fermi–Dirac statistics; hence:

$$N = \int_{\text{subband}} f^{\text{FD}}(E) \rho(E) dE \quad (2.48)$$

where the integral is over *all* of the energies of a given subband and, of course:

$$f^{\text{FD}}(E) = \frac{1}{\exp [(E - E_F)/kT] + 1} \quad (2.49)$$

Note that E_F is not the Fermi energy in the traditional sense [1]; it is a ‘quasi’ Fermi energy which describes the carrier population *within* a subband. For systems left to reach equilibrium, the temperature T can be assumed to be the lattice temperature; however this is not always the case. In many quantum well devices which are subject to excitation by electrical or optical means, the ‘electron temperature’ can be quite different from the lattice temperature, and furthermore the subband population could be non-equilibrium and not able to be described by Fermi–Dirac statistics. For now, however it is sufficient to discuss equilibrium electron populations and assume that the above equations are an adequate description.

Given a particular number of carriers within a quantum well, which can usually be deduced directly from the surrounding doping density, it is often desirable to be able to describe that distribution in terms of the quasi-Fermi energy E_F . With this aim substitute the two-dimensional density of states appropriate to a single subband from equation (2.46) into equation (2.48), then the carrier density, i.e. the number

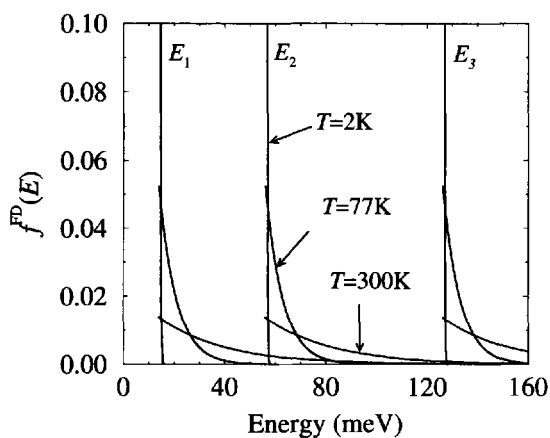


Figure 2.8 Effect of temperature on the distribution functions of the subband populations (all equal to $1 \times 10^{10} \text{ cm}^{-2}$) of the infinite quantum well of Fig. 2.7

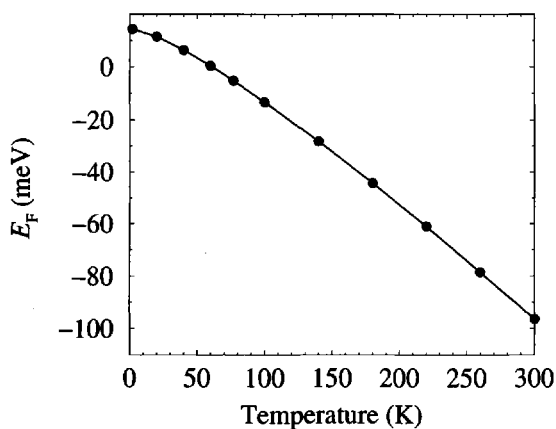


Figure 2.9 Effect of temperature on the quasi-Fermi energy describing the electron distribution of the ground state E_1

per unit area, is given by:

$$N = \int_{E_{\min}}^{E_{\max}} \frac{1}{\exp[(E - E_F)/kT] + 1} \frac{m^*}{\pi \hbar^2} dE \quad (2.50)$$

By putting:

$$E' = \frac{E - E_F}{kT}, \quad \text{and then} \quad dE' = \frac{dE}{kT} \quad (2.51)$$

equation (2.50) becomes:

$$N = \int_{(E_{\min} - E_F)/kT}^{(E_{\max} - E_F)/kT} \frac{kT}{\exp E' + 1} \frac{m^*}{\pi \hbar^2} dE' \quad (2.52)$$

which is a standard form (see for example, Gradshteyn and Ryzhik [23] equation 2.313.2, p. 112)

$$\int \frac{1}{1 + e^x} dx = x - \ln(1 + e^x) \quad (2.53)$$

Hence:

$$N = \frac{m^* kT}{\pi \hbar^2} \left[E' - \ln(1 + e^{E'}) \right]_{(E_{\min} - E_F)/kT}^{(E_{\max} - E_F)/kT} \quad (2.54)$$

Evaluation then gives:

$$N = \frac{m^* kT}{\pi \hbar^2} \left\{ \left[\frac{E_{\max} - E_F}{kT} - \ln(1 + e^{(E_{\max} - E_F)/kT}) \right] - \left[\frac{E_{\min} - E_F}{kT} - \ln(1 + e^{(E_{\min} - E_F)/kT}) \right] \right\} \quad (2.55)$$

The minimum of integration E_{\min} is taken as the subband minima and the maximum E_{\max} can either be taken as the top of the well, or even $E_F + 10kT$, say, with the latter being much more stable at lower temperatures. Given a total carrier density N , the quasi-Fermi energy E_F is the only unknown in equation (2.55) and can be found with standard techniques. For an example of such a method, see Section 2.5

Fig. 2.8 gives an example of the distribution functions $f^{\text{FD}}(E)$ for the first three confined levels within a 200 Å GaAs infinite quantum well. As the density of carriers, in this case electrons, have been taken as being equal and of value $1 \times 10^{10} \text{ cm}^{-2}$, then the distribution functions are all identical, but offset along the energy axis by the confinement energies. As mentioned above, at low temperatures the carriers tend to occupy the lowest available states, and hence the transition from states that are all occupied to those that are unoccupied is rapid—as illustrated by the 2 K data for all three subbands. As the temperature increases the distributions broaden and a range of energies exist in which the states are partially filled, as can be seen by the 77 and 300 K data. Physically this broadening occurs due mainly to the increase in electron-phonon scattering as the phonon population increases with temperature (more of this in Chapter 9). Fig. 2.9 displays the Fermi energy E_F as a function of temperature T for the ground state of energy $E_1 = 14.031 \text{ meV}$. At low temperatures, E_F is just above the confinement energy, since the electron density is fairly low ($1 \times 10^{10} \text{ cm}^{-2}$). As the temperature increases, E_F falls quite markedly and below the subband minima. If this seems counterintuitive, it must be remembered that E_F is a *quasi*-Fermi energy

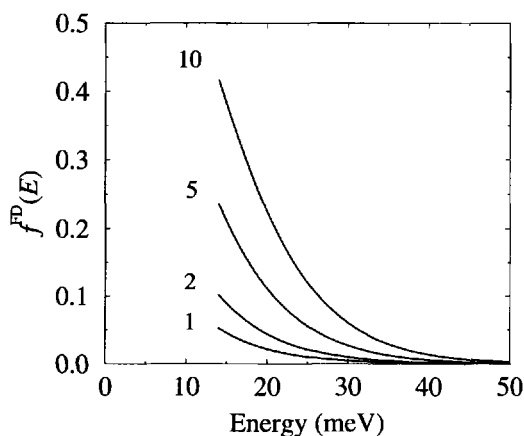


Figure 2.10 Effect of electron density, $N=1, 2, 5, 10 (\times 10^{10}) \text{ cm}^{-2}$, on the distribution function of the lowest subband of the infinite quantum well of Fig. 2.7

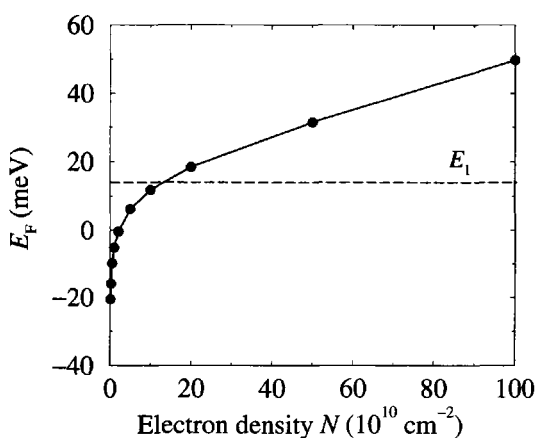


Figure 2.11 Effect of electron density on the quasi Fermi energy describing the distribution of the ground state E_1

whose only physical meaning is to describe the population within a subband—it is not the true Fermi energy of the complete system.

Fig. 2.10 displays the distribution functions for a range of carrier densities, for this same ground state and at a lattice temperature of 77 K. Although not obvious

from the mathematics, $f^{\text{FD}}(E)$ at any particular energy E appears to scale with N . The corresponding Fermi energy is illustrated in Fig. 2.11. Clearly, the Fermi energy starts below the subband minima at this mid-range temperature, as discussed above, and as expected increasing numbers of carriers in the subband increases the Fermi energy, i.e. the energy of the state whose probability of occupation is $1/2$.

2.5 FINITE WELL WITH CONSTANT MASS

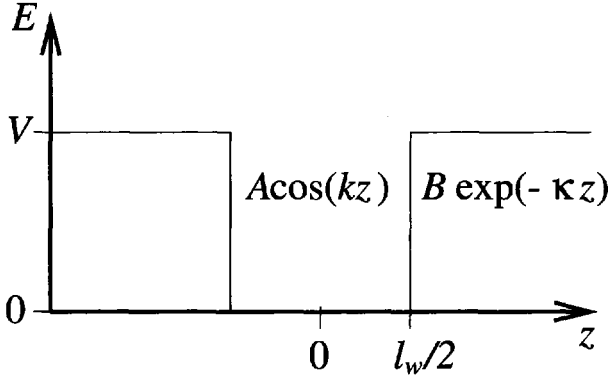


Figure 2.12 Solutions to the finite well potential

While the infinitely deep confining potential has served well as a platform for developing the physics of two-dimensional systems, more relevant to alternating layers of dissimilar semiconductors is the finite quantum well model, which under both the effective mass and envelope function approximations looks like Fig. 2.12. In particular, a layer of GaAs ‘sandwiched’ between two thick layers of $\text{Ga}_{1-x}\text{Al}_x\text{As}$ would form a type-I finite quantum well, where the conduction band has the appearance of Fig. 2.12, with the potential energy V representing the discontinuity in the conduction band edge between the materials.

Again taking the simplest starting case of a constant electron mass m^* throughout the dissimilar layers, and neglecting movement within the plane of the layers, then the standard Schrödinger equation can be written for each of the semiconductor layers as follows:

$$-\frac{\hbar^2}{2m^*} \frac{\partial^2}{\partial z^2} \psi(z) + V\psi(z) = E\psi(z), \quad z \leq -\frac{l_w}{2} \quad (2.56)$$

$$-\frac{\hbar^2}{2m^*} \frac{\partial^2}{\partial z^2} \psi(z) = E\psi(z), \quad -\frac{l_w}{2} \leq z \leq \frac{l_w}{2} \quad (2.57)$$

$$-\frac{\hbar^2}{2m^*} \frac{\partial^2}{\partial z^2} \psi(z) + V\psi(z) = E\psi(z), \quad +\frac{l_w}{2} \leq z \quad (2.58)$$

Considering solutions to the Schrödinger equation for the central well region, then as in the infinite well case, the general solution will be a sum of sine and cosine terms.

As the potential is symmetric, then the eigenstates will also have a definite symmetry, i.e. they will be either symmetric or antisymmetric. With the origin placed at the centre of the well, the symmetric (even parity) eigenstates will then be in cosine terms, while the antisymmetric (odd parity) states will be as sine waves.

For states confined to the well, the energy E must be less than the barrier height V , thus rearranging the Schrödinger equation for the right hand barrier:

$$\frac{\hbar^2}{2m^*} \frac{\partial^2}{\partial z^2} \psi(z) = (V - E) \psi(z) \quad (2.59)$$

Therefore, a function f is sought which when differentiated twice gives $+f$. The exponential function fits this description, therefore consider a sum of growing $\exp(+\kappa z)$ and decaying $\exp(-\kappa z)$ exponentials. In the right-hand barrier, z is positive, and hence as z increases the growing exponential will increase too and without limit. The probability interpretation of the wave function requires that:

$$\int_{\text{all space}} \psi^*(z) \psi(z) dz = 1 \quad (2.60)$$

which further demands that:

$$\psi(z) \rightarrow 0 \quad \text{and} \quad \frac{\partial}{\partial z} \psi(z) \rightarrow 0, \quad \text{as} \quad z \rightarrow \pm\infty \quad (2.61)$$

These boundary conditions for states confined in wells will be used again and again and will be referred to as the *standard boundary conditions*. Using this result, the growing exponential components must be rejected and the solutions are for the even parity states, which would follow as:

$$\psi(z) = B \exp(\kappa z), \quad z \leq -\frac{l_w}{2} \quad (2.62)$$

$$\psi(z) = A \cos(kz), \quad -\frac{l_w}{2} \leq z \leq \frac{l_w}{2} \quad (2.63)$$

$$\psi(z) = B \exp(-\kappa z), \quad \frac{l_w}{2} \leq z \quad (2.64)$$

Note for later that these wave functions are real, and that the eigenfunctions of this confined system carry no current and hence are referred to as stationary states. Using these trial forms of the wave function in their corresponding Schrödinger equations, gives the, as yet unknown constants:

$$k = \frac{\sqrt{2m^*E}}{\hbar}, \quad \text{and} \quad \kappa = \frac{\sqrt{2m^*(V-E)}}{\hbar} \quad (2.65)$$

In order to proceed it is necessary to impose boundary conditions. Recalling the constant mass kinetic energy operator employed in equations (2.56)-(2.58), then in order to avoid infinite kinetic energies:

$$\text{both } \psi(z) \quad \text{and} \quad \frac{\partial}{\partial z} \psi(z) \quad \text{must be continuous}$$

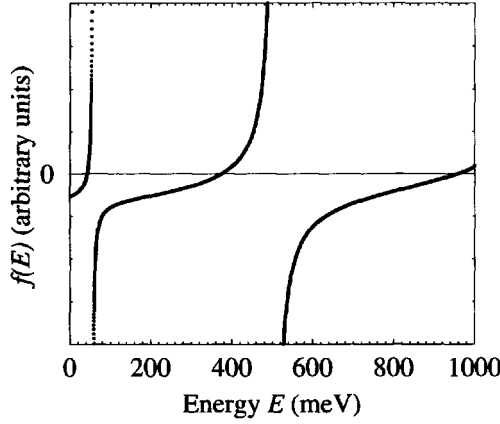


Figure 2.13 Illustration of $f(E)$ as a function of E for the even-parity solutions; where $l_w=100 \text{ \AA}$, $m^*=0.067 m_0$ and $V=1000 \text{ meV}$

Consider the interface at $z = +l_w/2$; by equating ψ in the well and the barrier:

$$A \cos\left(\frac{kl_w}{2}\right) = B \exp\left(-\frac{\kappa l_w}{2}\right) \quad (2.66)$$

and equating the derivatives gives:

$$-kA \sin\left(\frac{kl_w}{2}\right) = -\kappa B \exp\left(-\frac{\kappa l_w}{2}\right) \quad (2.67)$$

Dividing equation (2.66) by equation (2.67) then gives:

$$-\frac{1}{k} \cot\left(\frac{kl_w}{2}\right) = -\frac{1}{\kappa} \quad (2.68)$$

$$\therefore k \tan\left(\frac{kl_w}{2}\right) - \kappa = 0 \quad (2.69)$$

Odd parity states would require the choice of wave function in the well region as a sine wave, and hence equation (2.63) would become $\psi = A \sin(kz)$; following through the same analysis as above gives the equation to be solved for the odd parity eigenenergies as:

$$k \cot\left(\frac{kl_w}{2}\right) + \kappa = 0 \quad (2.70)$$

Remembering that both k and κ are functions of the energy E , then equations (2.69) and (2.70) are also functions of E only. There are many ways of solving such single-variable equations, and for this particular case the literature often talks of 'graphical

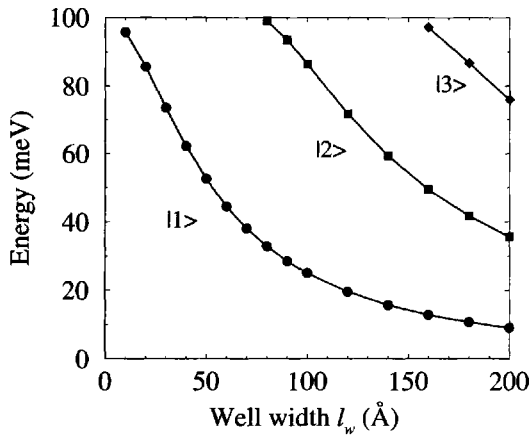


Figure 2.14 Energy levels in a GaAs single quantum well with constant effective mass $m^*=0.067m_0$ and $V=100$ meV

methods' [3,7,18]. While it is interesting to view the functional form of the equations, such methods are time consuming and inefficient and wouldn't be employed in the repetitive solution of many quantum wells. Computationally it is much more effective just to treat equations (2.69) and (2.70) with standard techniques, such as Newton-Raphson iteration. In this technique, if $E^{(n)}$ is a first guess to the solution of $f(E) = 0$, then a better estimate is given by:

$$E^{(n+1)} = E^{(n)} - \frac{f(E^{(n)})}{f'(E^{(n)})} \quad (2.71)$$

The new estimate $E^{(n+1)}$ is then used to generate a second approximation to the solution $E^{(n+2)}$, and so on, until the successive estimates converge to a required accuracy.

In order to provide all of the required information to implement the solution, all that remains is to deduce $f'(E)$. For the even parity states:

$$f(E) = k \tan\left(\frac{kl_w}{2}\right) - \kappa \quad (2.72)$$

and so therefore:

$$\frac{df}{dE} = \frac{dk}{dE} \tan\left(\frac{kl_w}{2}\right) + k \sec^2\left(\frac{kl_w}{2}\right) \times \frac{l_w}{2} \frac{dk}{dE} - \frac{d\kappa}{dE} \quad (2.73)$$

where

$$\frac{dk}{dE} = \frac{\sqrt{2m^*}}{2\sqrt{E\hbar}}; \quad \frac{d\kappa}{dE} = -\frac{\sqrt{2m^*}}{2\sqrt{V-E\hbar}} \quad (2.74)$$

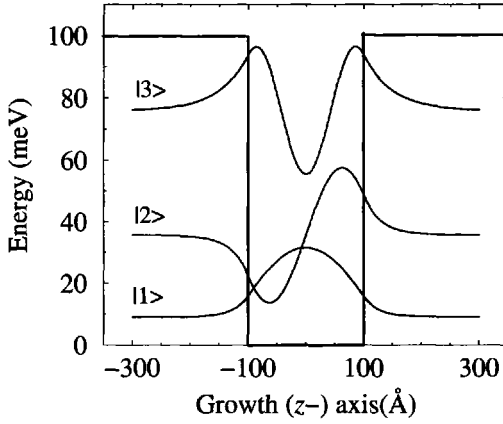


Figure 2.15 Eigenfunctions $\psi(z)$ for the first three energy levels of the 200 Å GaAs well of Fig. 2.14

For the odd-parity states:

$$f(E) = k \cot\left(\frac{kl_w}{2}\right) + \kappa \quad (2.75)$$

$$\therefore \frac{df}{dE} = \frac{dk}{dE} \cot\left(\frac{kl_w}{2}\right) - k \csc^2\left(\frac{kl_w}{2}\right) \times \frac{l_w}{2} \frac{dk}{dE} + \frac{d\kappa}{dE} \quad (2.76)$$

Figure 2.13 illustrates the even parity $f(E)$. In practice, the finite number of discontinuities can lead to solutions being missed if the first guess to the solution $E^{(0)}$ is not close to the true solution. In order to circumvent this computational problem, $f(E)$ is calculated at discrete points along the E axis, separated by an energy thought to be smaller than the minimum separation between adjacent states (generally 1 meV), when $f(E)$ changes sign; then the Newton–Raphson is then implemented to obtain the solution accurately.

Figures 2.14 and 2.15 summarise the application of the method to a GaAs single quantum well, surrounded by barrier of height 100 meV, with the same effective mass. Clearly, as the well width increases, then the energy levels all decrease, with the presence of excited states being also apparent at the larger well widths. The eigenstates are labelled according to their principle quantum number (energy order). The even and odd parities of the states within the well can be seen in Fig. 2.15.

2.6 EFFECTIVE MASS MISMATCH AT HETEROJUNCTIONS

Quantum wells are only fabricated by forming heterojunctions between *different* semiconductors. From an electronic viewpoint, the semiconductors are different because they have different band structures. The difference in perhaps the most fundamental property of a semiconductor, i.e. the band gap, (and its alignment) is accounted for by specifying a band offset, which has been labelled V . Of course, there are many other properties which are also different, such as the dielectric constant, the lattice constant and, what is considered the next most important quantity, the effective mass. It is generally accepted that the calculation of *static* energy levels within quantum wells should account for the variation in the effective mass across the heterojunction.

This problem has been continuously addressed in the literature [17, 24, 25] since the earliest work of Conley *et al.* [26] and BenDaniel and Duke [27], who derived the boundary conditions on solutions of the *envelope functions* as:

$$\text{both } \psi(z) \text{ and } \frac{1}{m^*} \frac{\partial}{\partial z} \psi(z) \text{ continuous} \quad (2.77)$$

by considering electron transport across a heterojunction. These boundary conditions have become known as the BenDaniel–Duke boundary conditions.

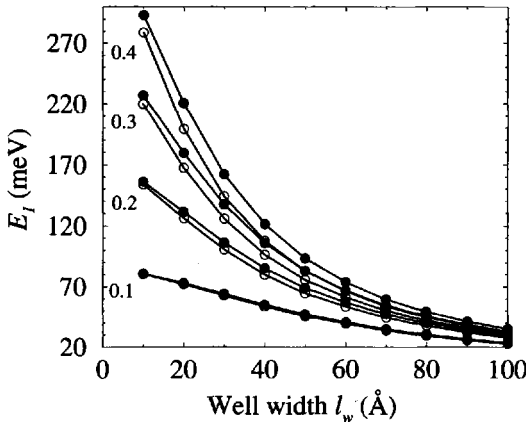


Figure 2.16 Electron ground state energy E_1 as a function of the width l_w of a GaAs well surrounded by $\text{Ga}_{1-x}\text{Al}_x\text{As}$ barriers, calculated for both the constant mass-model (closed circles) and different barrier masses (open circles) and for a range of barrier alloy concentrations x ($= 0.1, 0.2, 0.3, 0.4$)

Therefore applying this extension to the finite well of the previous section would require the Schrödinger equation to be specified in each region as follows:

$$-\frac{\hbar^2}{2m_b^*} \frac{\partial^2}{\partial z^2} \psi(z) + V\psi(z) = E\psi(z), \quad z \leq -\frac{l_w}{2} \quad (2.78)$$

$$-\frac{\hbar^2}{2m_w^*} \frac{\partial^2}{\partial z^2} \psi(z) = E\psi(z), \quad -\frac{l_w}{2} \leq z \leq \frac{l_w}{2} \quad (2.79)$$

$$-\frac{\hbar^2}{2m_b^*} \frac{\partial^2}{\partial z^2} \psi(z) + V\psi(z) = E\psi(z), \quad +\frac{l_w}{2} \leq z \quad (2.80)$$

with the additional restraint of the matching conditions of equation (2.77).

The solutions follow as previously for the constant-mass case, in equations (2.62), (2.63) and (2.64), but now k and κ contain different effective masses:

$$k = \frac{\sqrt{2m_w^* E}}{\hbar} \quad \kappa = \frac{\sqrt{2m_b^* (V - E)}}{\hbar} \quad (2.81)$$

The method of solution is almost identical: equating the envelope functions at the interface $z = +l_w/2$

$$A \cos\left(\frac{kl_w}{2}\right) = B \exp\left(-\frac{\kappa l_w}{2}\right) \quad (2.82)$$

and equating $1/m^*$ times the derivative gives

$$-\frac{kA}{m_w^*} \sin\left(\frac{kl_w}{2}\right) = -\frac{\kappa B}{m_b^*} \exp\left(-\frac{\kappa l_w}{2}\right) \quad (2.83)$$

Dividing equation (2.83) by equation (2.82) then:

$$f(E) = \frac{k}{m_w^*} \tan\left(\frac{kl_w}{2}\right) - \frac{\kappa}{m_b^*} = 0 \quad (2.84)$$

and similarly for the odd parity solutions, i.e.

$$f(E) = \frac{k}{m_w^*} \cot\left(\frac{kl_w}{2}\right) + \frac{\kappa}{m_b^*} = 0 \quad (2.85)$$

and obviously, equating the effective masses gives the original constant-mass equations. Again for numerical solution, the derivatives are required, i.e. for even parity:

$$\frac{df}{dE} = \frac{1}{m_w^*} \frac{dk}{dE} \tan\left(\frac{kl_w}{2}\right) + \frac{k}{m_w^*} \sec^2\left(\frac{kl_w}{2}\right) \times \frac{l_w}{2} \frac{dk}{dE} - \frac{1}{m_b^*} \frac{d\kappa}{dE} \quad (2.86)$$

and for odd parity

$$\frac{df}{dE} = \frac{1}{m_w^*} \frac{dk}{dE} \cot\left(\frac{kl_w}{2}\right) - \frac{k}{m_w^*} \csc^2\left(\frac{kl_w}{2}\right) \times \frac{l_w}{2} \frac{dk}{dE} + \frac{1}{m_b^*} \frac{d\kappa}{dE} \quad (2.87)$$

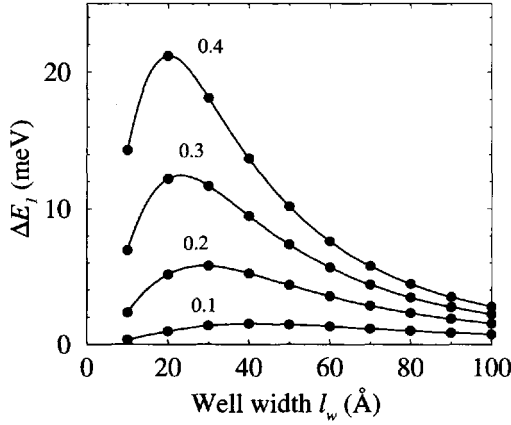


Figure 2.17 Energy difference $\Delta E_1 = E_1(m^*=\text{constant}) - E_1(m^*(z))$ for the structures shown in Fig. 2.16

where

$$\frac{dk}{dE} = \frac{\sqrt{2m_w^*}}{2\sqrt{E}\hbar} \quad \frac{d\kappa}{dE} = -\frac{\sqrt{2m_b^*}}{2\sqrt{V-E}\hbar} \quad (2.88)$$

Figure 2.16 compares the electron ground-state energy calculated with a constant GaAs mass, with the energy calculated with a material-dependent mass, for a single GaAs quantum well surrounded by $\text{Ga}_{1-x}\text{Al}_x\text{As}$ barriers. All of the calculated ground state energies E_1 decrease with increasing well width and increase with increasing Al fraction in the barriers (barrier height $V \propto x$). The effective mass in $\text{Ga}_{1-x}\text{Al}_x\text{As}$ is greater than in GaAs, hence the variable mass calculations give energies less than the constant-mass model for *all* systems considered here (see later).

Fig. 2.17 displays the calculated ground state energy difference between the two models, $\Delta E_1 = E_1(m^*=\text{constant}) - E_1(m^*(z))$. Clearly, and as would be expected, the larger the difference in the effective masses between the materials, the larger the difference in ground-state energies.

2.7 THE INFINITE BARRIER HEIGHT AND MASS LIMITS

It is interesting to take theoretical models to certain limits as a means of verifying, or otherwise, their behaviour with what might be expected intuitively. Fig. 2.18 illustrates that, in the limit of large barrier heights V , the finite well model recovers the result of the infinite well model, which is what would be hoped for, thus increasing confidence in the derivation.

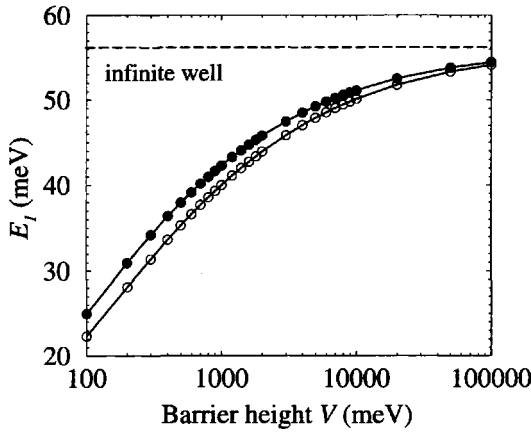


Figure 2.18 Electron ground-state energy E_1 as a function of barrier height V , for a 100 Å GaAs finite well with constant mass (closed circles) and different barrier mass (fixed at mass in $\text{Ga}_{0.6}\text{Al}_{0.4}\text{As}$ (open circles))

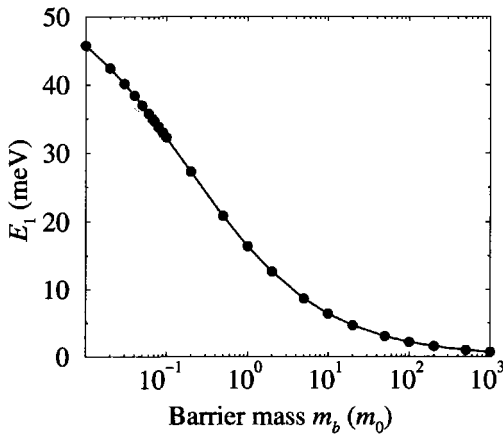


Figure 2.19 Electron ground-state energy E_1 as a function of the mass in the barrier, for a 100 Å GaAs well with a barrier height fixed at that for $\text{Ga}_{0.6}\text{Al}_{0.4}\text{As}$

As has been found in the literature, Fig. 2.19 illustrates the results of allowing the barrier mass to increase without limit, while keeping all other parameters constant. The tendency for the ground-state energy to tend towards zero and the unusual

looking wave functions of Fig. 2.20 have been well documented [18] and are a direct consequence of the second boundary condition, ψ'/m^* .

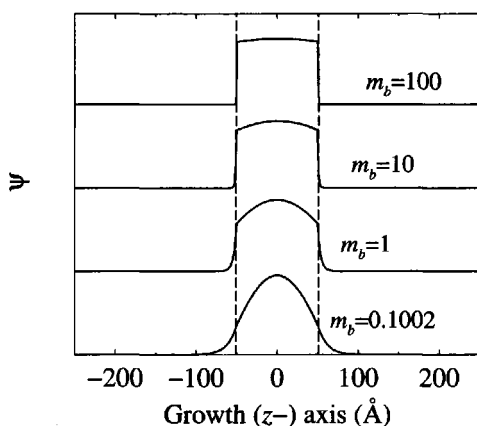


Figure 2.20 Electron ground-state wave functions for several barrier masses, as given, for a 100 Å GaAs well with a barrier height fixed at that for $\text{Ga}_{0.6}\text{Al}_{0.4}\text{As}$

It is worthwhile considering this limit still further. Fig. 2.21 reproduces the results of Fig. 2.19 but for a variety of well widths. Clearly, the ground-state energy tends to zero for all well widths:

$$\lim_{m_b \rightarrow \infty} E_1 = 0, \quad \forall l_w \quad (2.89)$$

which at first sight appears to violate Heisenberg's Uncertainty Principle, in that the ground state energy *can* be forced to zero for infinitesimal well widths, and thus the error in the measurement of position and momentum can be made arbitrarily small. This was an argument advanced by Hagston *et al.* [28]; however direct evaluation of the variance in the position and momentum of these states will show that the uncertainty relationships are not violated. At the moment, the techniques for such a calculation have not been covered; hence such discussions will be returned to later in Section 3.15.

2.8 HERMITICITY AND THE KINETIC ENERGY OPERATOR

The changes in the boundary conditions at a heterojunction are only necessary under theoretical models which parameterise physical quantities in terms of a variable mass. This is only encountered when applying the effective mass approximation to semiconductor heterostructures. Theories of semiconductor heterostructures do exist

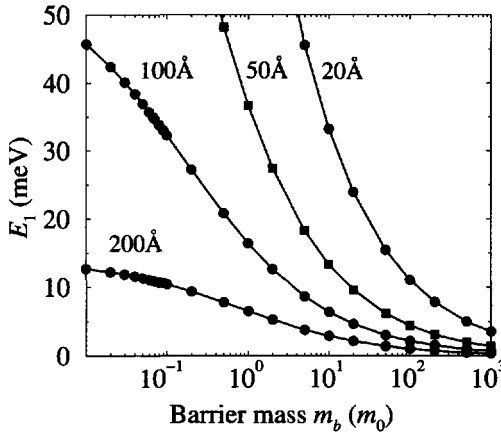


Figure 2.21 Electron ground state energy E_1 as a function of the mass in the barrier, for a variety of GaAs well widths, the barrier height is fixed at that for $\text{Ga}_{0.6}\text{Al}_{0.4}\text{As}$

which do not make this imposition (parameterisation)—within them all of the electron states, within both the conduction and valence band can be derived from the *constant* free electron mass. Therefore the kinetic energy operator remains

$$\mathcal{T} = -\frac{\hbar^2}{2m_0} \nabla^2 \quad (2.90)$$

and hence the boundary conditions are set as continuity in both wave function and derivative—note *wave function*, and not *envelope function*. Such methods are the subject of later chapters, for example, Chapter 11.

However within the auspices of this very successful model of semiconductor band structure, allowance *has* to be made for the additional complication of variable mass. As discussed in Section 2.6, this can be achieved with the original Schrödinger equation applied to each region, taken together with boundary conditions for matching solutions at the heterojunctions between the regions of different effective mass.

However, fundamentally both of these conditions can be incorporated within one Hamiltonian which is valid throughout all space at once. Its construction derives from the cornerstones of quantum mechanics—all physical observables can be represented by linear Hermitian operators [12]. With this aim consider the classical (non-relativistic) kinetic energy:

$$T = \frac{p^2}{2m} \quad (2.91)$$

Quantum mechanically, the linear kinetic energy operator for a variable mass $m^*(z)$ say, is formed by replacing the classical quantities with the corresponding linear

operators, giving:

$$T = \frac{P_z^2}{m^*(z)} \quad (2.92)$$

Inspection of equation (2.92) reveals that there is still ambiguity as to the *order* by which the operators will act upon the eigenvector. To resolve this, guidance is sought from a property of Hermitian operators, and that is they are equal to their own Hermitian conjugate (see Schiff [29] p. 151), i.e.

$$T = T^\dagger \quad (2.93)$$

In addition the Hermitian conjugate of a product is given by $(ABC)^\dagger = C^\dagger B^\dagger A^\dagger$. Applying this to the effective mass kinetic energy operator then:

$$T = P_z \frac{1}{m^*(z)} P_z \quad (2.94)$$

$$\therefore T = -\frac{\hbar^2}{2} \frac{\partial}{\partial z} \frac{1}{m^*(z)} \frac{\partial}{\partial z} \quad (2.95)$$

which would give a Schrödinger equation of the form:

$$-\frac{\hbar^2}{2} \frac{\partial}{\partial z} \frac{1}{m^*(z)} \frac{\partial}{\partial z} \psi(z) + V(z)\psi(z) = E\psi(z) \quad (2.96)$$

which in order to avoid differentiating discontinuous functions and producing infinities, clearly demands the boundary conditions of equation 2.77, i.e.,

$$\text{both } \psi(z) \quad \text{and} \quad \frac{1}{m^*} \frac{\partial}{\partial z} \psi(z) \quad \text{are continuous}$$

2.9 ALTERNATIVE KINETIC ENERGY OPERATORS

The Hermiticity requirements above have been used to derive a kinetic energy operator and hence a single Hamiltonian which is valid throughout all space, and reproduces the accepted boundary conditions on the envelope functions of heterostructures. This form is, however, not unique. In particular, any combination of linear momentum operators and effective mass of the form:

$$T = m^{*\alpha}(z) P_z m^{*\beta}(z) P_z m^{*\alpha}(z), \quad \text{where} \quad 2\alpha + \beta = -1 \quad (2.97)$$

will also satisfy these requirements [30,31].

The accepted choice is $\alpha = 0$, $\beta = -1$, but clearly there are infinitely many solutions. Morrow [32] and Hagston *et al.* [28] have independently forwarded $\alpha = -1/2$, $\beta=0$, giving

$$T = \frac{1}{\sqrt{m^*(z)}} P_z P_z \frac{1}{\sqrt{m^*(z)}} \quad (2.98)$$

which can be written succinctly as:

$$T = Q^\dagger Q, \quad \text{where} \quad Q = P_z \frac{1}{\sqrt{m^*(z)}} \quad (2.99)$$

Transforming the wave function $\psi(z)$ into $\sqrt{m^*(z)}\phi(z)$ and substituting into the Schrödinger equation gives:

$$-\frac{\hbar^2}{2} \frac{1}{\sqrt{m^*(z)}} \frac{\partial^2}{\partial z^2} \frac{1}{\sqrt{m^*(z)}} \sqrt{m^*(z)}\phi(z) + V(z)\sqrt{m^*(z)}\phi(z) = E\sqrt{m^*(z)}\phi(z) \quad (2.100)$$

$$\therefore -\frac{\hbar^2}{2m^*(z)} \frac{\partial^2}{\partial z^2} \phi(z) + V(z)\phi(z) = E\phi(z) \quad (2.101)$$

which implies that $\phi(z)$ is an eigenfunction of this transformed Hamiltonian, but with the same eigenvalue E . This new entity, $\phi(z)$, satisfies the boundary conditions:

$$\text{both } \phi(z) \quad \text{and} \quad \frac{\partial}{\partial z} \phi(z) \quad \text{are continuous}$$

Therefore, discontinuous changes in the effective mass $m^*(z)$ at abrupt heterojunctions would give a discontinuity in the wave function ψ , a view which has been suggested elsewhere [33].

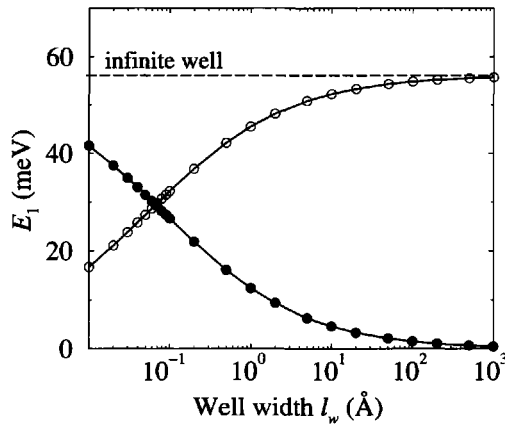


Figure 2.22 Effect of the two different kinetic energy operators on the ground-state energy of a 100 Å GaAs well surrounded by $\text{Ga}_{0.8}\text{Al}_{0.2}\text{As}$ barriers: $T = \frac{1}{m} P^2$ (open circles); $T = \frac{P_\perp}{m} P$ (closed circles)

Fig. 2.22 compares the consequences of such an alternative kinetic energy operator on the ground state energy E_1 of a finite quantum well, with that of the accepted

operator. In the limit of large barrier mass, the energy given by the new operator tends towards that of the infinitely deep quantum well. This appears intuitively satisfying, given that the wave function penetration into the barrier, $\exp(-\kappa z)$, has a similar dependence on the barrier mass m_b as on the height V , i.e.

$$\kappa = \frac{\sqrt{2m_b^*(V-E)}}{\hbar} \quad (2.102)$$

Thus mathematically speaking, increasing m_b would appear to produce the same effect as increasing V .

Galbraith and Duggan [25] have compared the results of similar calculations with experimental photoluminescence excitation measurements and have concluded that for their single-quantum-well samples the accepted form of $T = \mathcal{P} \frac{1}{m} \mathcal{P}$ fits the data better. There is no doubt that this is not the last word on the form of the kinetic energy operator, and the associated boundary conditions, on the effective mass envelope functions which follow. However, for now it is best to conform to the commonly agreed standard and employ $T = \mathcal{P} \frac{1}{m} \mathcal{P}$ throughout the remainder of this book.

2.10 EXTENSION TO MULTIPLE-WELL SYSTEMS

For a whole variety of reasons semiconductor physicists and electronic device engineers need to design and fabricate heterostructures more complex than the single quantum well [7]. The above techniques of solving Schrödinger's equation in each semiconductor layer separately and deducing the unknown coefficients by implementing boundary conditions can be applied to these multilayered systems. In particular, consider the asymmetric double quantum well of Fig. 2.23.

$$\begin{array}{c}
 V \\
 \hline
 \begin{array}{c}
 A \exp(+\kappa z) \\
 + \\
 B \exp(-\kappa z)
 \end{array}
 \left| \begin{array}{c}
 C \sin(kz) \\
 + \\
 D \cos(kz)
 \end{array} \right|
 \begin{array}{c}
 F \exp(+\kappa z) \\
 + \\
 G \exp(-\kappa z)
 \end{array}
 \left| \begin{array}{c}
 H \sin(kz) \\
 + \\
 I \cos(kz)
 \end{array} \right|
 \begin{array}{c}
 J \exp(+\kappa z) \\
 + \\
 K \exp(-\kappa z)
 \end{array} \\
 \hline
 \begin{array}{cccc}
 z=0 & & a & & b & & c
 \end{array} \\
 \hline
 \begin{array}{cc}
 \frac{\hbar^2 k^2}{2m_w^*} = E & \quad \quad \quad & \frac{\hbar^2 \kappa^2}{2m_b^*} = V-E
 \end{array}
 \end{array}$$

Figure 2.23 Solutions to Schrödinger's equation in a double quantum well

Choosing both wells to be of the same depth, and all of the barriers to be of the same height V , allows the simplification that k and κ are constant throughout the structure; the general solution to the Schrödinger equation in each region are as given in the figure. Proceeding as before, then matching envelope functions and the derivative divided by the mass at the interfaces, then considering $z = 0$ first:

$$A + B = D \quad (2.103)$$

$$\frac{1}{m_b^*} (\kappa A - \kappa B) = \frac{1}{m_w^*} kC \quad (2.104)$$

with $z = a$ then:

$$C \sin ka + D \cos ka = F \exp(+\kappa a) + G \exp(-\kappa a) \quad (2.105)$$

$$\frac{1}{m_w^*} (kC \cos ka - kD \sin ka) = \frac{1}{m_b^*} [\kappa F \exp(+\kappa a) - \kappa G \exp(-\kappa a)] \quad (2.106)$$

$z = b$:

$$F \exp(+\kappa b) + G \exp(-\kappa b) = H \sin kb + I \cos kb \quad (2.107)$$

$$\frac{1}{m_b^*} [\kappa F \exp(+\kappa b) - \kappa G \exp(-\kappa b)] = \frac{1}{m_w^*} (kH \cos kb - kI \sin kb) \quad (2.108)$$

$z = c$:

$$H \sin kc + I \cos kc = J \exp(+\kappa c) + K \exp(-\kappa c) \quad (2.109)$$

$$\frac{1}{m_w^*} (kH \cos kc - kI \sin kc) = \frac{1}{m_b^*} [\kappa J \exp(+\kappa c) - \kappa K \exp(-\kappa c)] \quad (2.110)$$

which can be rewritten more neatly in matrix form as:

$$\begin{pmatrix} 1 & 1 \\ \frac{1}{m_b^*} \kappa & -\frac{1}{m_b^*} \kappa \end{pmatrix} \begin{pmatrix} A \\ B \end{pmatrix} = \begin{pmatrix} 0 & 1 \\ \frac{1}{m_w^*} k & 0 \end{pmatrix} \begin{pmatrix} C \\ D \end{pmatrix} \quad (2.111)$$

$$\begin{aligned} & \begin{pmatrix} \sin ka & \cos ka \\ \frac{1}{m_w^*} k \cos ka & -\frac{1}{m_w^*} k \sin ka \end{pmatrix} \begin{pmatrix} C \\ D \end{pmatrix} \\ &= \begin{pmatrix} \exp(+\kappa a) & \exp(-\kappa a) \\ \frac{1}{m_b^*} \kappa \exp(+\kappa a) & -\frac{1}{m_b^*} \kappa \exp(-\kappa a) \end{pmatrix} \begin{pmatrix} F \\ G \end{pmatrix} \end{aligned} \quad (2.112)$$

$$\begin{aligned} & \begin{pmatrix} \exp(+\kappa b) & \exp(-\kappa b) \\ \frac{1}{m_b^*} \kappa \exp(+\kappa b) & -\frac{1}{m_b^*} \kappa \exp(-\kappa b) \end{pmatrix} \begin{pmatrix} F \\ G \end{pmatrix} \\ &= \begin{pmatrix} \sin kb & \cos kb \\ \frac{1}{m_w^*} k \cos kb & -\frac{1}{m_w^*} k \sin kb \end{pmatrix} \begin{pmatrix} H \\ I \end{pmatrix} \end{aligned} \quad (2.113)$$

$$\begin{aligned} & \begin{pmatrix} \sin kc & \cos kc \\ \frac{1}{m_w^*} k \cos kc & -\frac{1}{m_w^*} k \sin kc \end{pmatrix} \begin{pmatrix} H \\ I \end{pmatrix} \\ &= \begin{pmatrix} \exp(+\kappa c) & \exp(-\kappa c) \\ \frac{1}{m_b^*} \kappa \exp(+\kappa c) & -\frac{1}{m_b^*} \kappa \exp(-\kappa c) \end{pmatrix} \begin{pmatrix} J \\ K \end{pmatrix} \end{aligned} \quad (2.114)$$

Labelling the 2×2 matrix for the left-hand side of the n th interface as \mathbf{M}_{2n-1} and the corresponding matrix for the right hand side of the interface as \mathbf{M}_{2n} , $n=1, 2, 3$, etc., then the above matrix equations would become:

$$\mathbf{M}_1 \begin{pmatrix} A \\ B \end{pmatrix} = \mathbf{M}_2 \begin{pmatrix} C \\ D \end{pmatrix} \quad (2.115)$$

$$\mathbf{M}_3 \begin{pmatrix} C \\ D \end{pmatrix} = \mathbf{M}_4 \begin{pmatrix} F \\ G \end{pmatrix} \quad (2.116)$$

$$\mathbf{M}_5 \begin{pmatrix} F \\ G \end{pmatrix} = \mathbf{M}_6 \begin{pmatrix} H \\ I \end{pmatrix} \quad (2.117)$$

$$\mathbf{M}_7 \begin{pmatrix} H \\ I \end{pmatrix} = \mathbf{M}_8 \begin{pmatrix} J \\ K \end{pmatrix} \quad (2.118)$$

Now equation (2.115) gives:

$$\begin{pmatrix} A \\ B \end{pmatrix} = \mathbf{M}_1^{-1} \mathbf{M}_2 \begin{pmatrix} C \\ D \end{pmatrix} \quad (2.119)$$

and equation (2.116) gives:

$$\begin{pmatrix} C \\ D \end{pmatrix} = \mathbf{M}_3^{-1} \mathbf{M}_4 \begin{pmatrix} F \\ G \end{pmatrix} \quad (2.120)$$

$$\therefore \begin{pmatrix} A \\ B \end{pmatrix} = \mathbf{M}_1^{-1} \mathbf{M}_2 \mathbf{M}_3^{-1} \mathbf{M}_4 \begin{pmatrix} F \\ G \end{pmatrix} \quad (2.121)$$

and eventually:

$$\begin{pmatrix} A \\ B \end{pmatrix} = \mathbf{M}_1^{-1} \mathbf{M}_2 \mathbf{M}_3^{-1} \mathbf{M}_4 \mathbf{M}_5^{-1} \mathbf{M}_6 \mathbf{M}_7^{-1} \mathbf{M}_8 \begin{pmatrix} J \\ K \end{pmatrix} \quad (2.122)$$

The product of the eight 2×2 matrices is still a 2×2 matrix; thus writing:

$$\begin{pmatrix} A \\ B \end{pmatrix} = \mathcal{M} \begin{pmatrix} J \\ K \end{pmatrix} \quad (2.123)$$

then

$$A = \mathcal{M}_{11}J + \mathcal{M}_{12}K \quad (2.124)$$

$$B = \mathcal{M}_{21}J + \mathcal{M}_{22}K \quad (2.125)$$

Again the probability interpretation of the wave function implies that the wave function must tend towards zero into the outer barriers, i.e. the coefficients of the growing exponentials must be zero. *In this case*, with the origin at the 1st interface (see Fig. 2.23), then this implies that $B = 0$ and $J = 0$, and hence the second of the above equations would imply that $\mathcal{M}_{22} = 0$. As all of the elements of \mathcal{M} are functions of both k and κ , which are both in turn functions of the energy E , then an energy is sought which satisfies:

$$\mathcal{M}_{22}(E) = 0 \quad (2.126)$$

which can be found by standard numerical procedures, as discussed in Section 2.5. This approach and variations upon it are often referred to as the *Transfer matrix technique*. Once the energy is known, the coefficients A to K follow simply and the envelope wave function can be deduced.

2.11 THE ASYMMETRIC SINGLE QUANTUM WELL

The above system has illustrated a method of solution for a system in which the confinement energy E was always less than the barrier height V . While this is often true, a class of quantum well structures exist in which there is more than one barrier height (or well depth); Fig. 2.24 illustrates one such system.

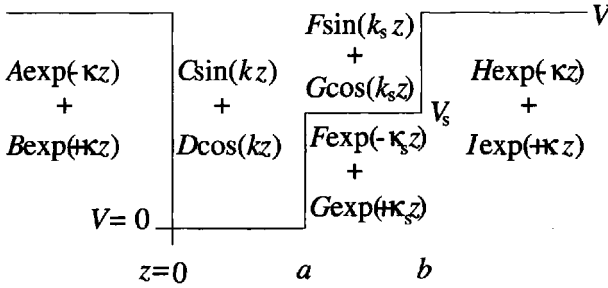


Figure 2.24 Solutions to Schrödinger's equation in a stepped asymmetric quantum well

A confined state could exist which has an energy below or above that of the step height V_s , hence the functional form of the solution to Schrödinger's equation in that region is dependent upon the energy E . This is illustrated schematically in Fig. 2.24. For states above the step, the wave function has the form of a normal well state, but with a k value different from that of the well region. For states of energy below the step potential V_s , the wave function resembles a barrier state. This causes difficulties computationally, as two transfer matrices have to be determined. A better approach is to write the wave function in *every* region (r) as a linear combination of travelling waves, i.e.

$$\psi_r = A' \exp(ik_r z) + B' \exp(-ik_r z) \quad (2.127)$$

where the coefficients A' and B' are now allowed to be complex and the wave vector k is as before:

$$k = \sqrt{\frac{2m^*}{\hbar^2}(E - V_r)} \quad (2.128)$$

Forming the transfer matrix from this standard form for the wave function and applying the boundary condition of decaying exponentials at both ends of the well structure, allows the method of solution as before. For now it is worthwhile confirming that ψ_r can take both the usual well- and barrier-state forms.

Consider regions where $E > V$:

$$\psi_r = A'(\cos k_r z + i \sin k_r z) + B'(\cos k_r z - i \sin k_r z) \quad (2.129)$$

$$\therefore \psi_r = (A' + B') \cos k_r z + i(A' - B') \sin k_r z \quad (2.130)$$

or by collecting real and imaginary components, then:

$$\psi_r = [\Re(A' + B') \cos k_r z - \Im(A' - B') \sin k_r z] \quad (2.131)$$

$$+ i [\Im(A' + B') \cos k_r z + \Re(A' - B') \sin k_r z] \quad (2.132)$$

Now eigenstates confined within potential wells are *stationary states*, which means that they have no time dependence and carry no current, i.e. their wave functions are real. Thus it would be expected that the solution would naturally yield such states, and in fact for the imaginary component of ψ_r to be zero, then:

$$\Im(A' + B') = 0 \quad \text{and} \quad \Re(A' - B') = 0 \quad (2.133)$$

which are simultaneously satisfied if:

$$B' = A'^* \quad (2.134)$$

This implies that in regions where the energy of the state of interest is greater than the potential, the solution will then give the coefficients as a complex conjugate pair in order to ensure a real wave function.

In regions where $E < V$, then:

$$k = i\sqrt{\frac{2m^*}{\hbar^2}(V_r - E)} = i\kappa \quad (2.135)$$

where κ is the usual decay constant as defined in equation (2.81). Then the general form for a barrier state is generated, i.e.

$$\psi_r = A' \exp(-\kappa z) + B' \exp(+\kappa z) \quad (2.136)$$

As mentioned above, application of the boundary conditions of $\psi \rightarrow 0$ as $z \rightarrow \pm\infty$, would force the choice that B' was zero and A' was real. In other regions where $E < V$, e.g. the step in Fig. 2.24 or the central barrier of Fig. 2.23, then both A' and B' should naturally arise as real.

2.12 ADDITION OF AN ELECTRIC FIELD

As illustrated in Fig. 2.25, the effect of an electric field along the growth (z -) axis is to add a linear potential, which for an electron of charge $-e$ in the conduction band will be written as $-eFz$. For small electric fields, the effect of this electric field on the confined energy levels within a quantum well can be approximated by first-order perturbation theory.

If the perturbing potential is written as V' then the change in the ground-state energy level is given, for example, by Schiff [29], p. 246:

$$\Delta E^{(1)} = \langle \psi_1 | V' | \psi_1 \rangle \quad (2.137)$$

with the electric field along the growth axis as the perturbation; then this translates to

$$\Delta E^{(1)} = \int_{-\infty}^{+\infty} \psi_1^*(z)(-eFz)\psi_1(z) dz \quad (2.138)$$

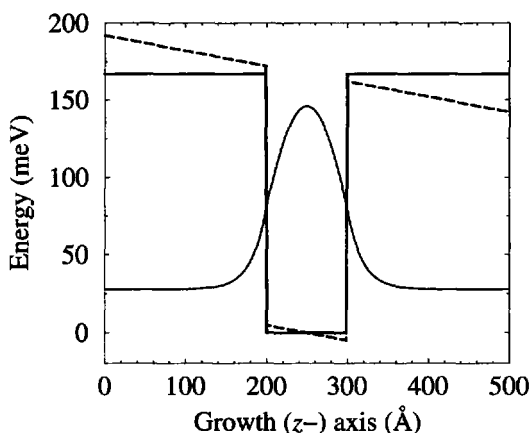


Figure 2.25 The ground state wave function of the zero-field potential (solid line) and the corresponding potential at a field of 10 kVcm^{-1} (dashed line)

The ground state of the *symmetric* quantum well of Fig. 2.25 is an even function, and hence the integrand of equation (2.138) is odd. Therefore, evaluation of the integral gives zero, i.e. to first-order the addition of a *small* electric field has no effect on the *ground-state* energy level. When applying the same logic to the first excited state, although ψ_2 is odd the integrand will still be an odd function, which again results in no change. There is currently much interest in the literature in breaking the symmetry of quantum wells in order to induce a first-order response in the ground state energy to an electric field [34–36].

Larger electric fields would require a more accurate evaluation of the perturbation as can be given by second-order perturbation theory—see Schiff [29], p. 247:

$$\Delta E^{(2)} = \sum_{m=2}^{\infty} \frac{|\langle \psi_m | V' | \psi_1 \rangle|^2}{E_m - E_1} \quad (2.139)$$

which in this case translates to:

$$\Delta E^{(2)} = \sum_{m=2}^{\infty} \frac{|\int_{-\infty}^{+\infty} \psi_m^*(z)(-eFz)\psi_1(z) dz|^2}{E_m - E_1} \quad (2.140)$$

where the sum is over *all* excited states, including those with an energy $E_m > V$, i.e. the so called ‘continuum’ states. As the electric field F does not have a z -dependence, then clearly the change in the energy to second-order $\Delta E^{(2)}$ is proportional to F^2 . In addition, a charged particle prefers to move to areas of lower potential, i.e. the electron within the quantum well of Fig. 2.25 moves to the right hand side of the

well, thus lowering its total energy, so indeed $\Delta E^{(2)} \propto -F^2$. This suppression of the confined energy level by an electric field is known as the 'quantum confined Stark effect' [37] and is commonly observed experimentally in heterostructures [35].

While such a perturbative treatise may be adequate for single quantum wells, its accuracy for complicated, perhaps multiple well systems and at high fields, is questionable. In order to improve upon this, a full (non-perturbative) solution to the heterostructure and field potentials is required. Within the envelope function and effective mass approximations, the Schrödinger equation within a particular material (i.e. region of constant mass) and with an applied electric field of strength F , is:

$$-\frac{\hbar^2}{2m^*} \frac{\partial^2 \psi}{\partial z^2} + [V(z) - eFz] \psi = E\psi \quad (2.141)$$

This new equation does not have the standard solution of linear combinations of trigonometric and exponential functions, and therefore a different approach is required. First, rearrange equation (2.141) to give

$$\frac{\partial^2 \psi}{\partial z^2} - \frac{2m^*}{\hbar^2} [V(z) - eFz - E] \psi = 0 \quad (2.142)$$

Then by making the substitutions:

$$\alpha = \frac{2m^*}{\hbar^2} [V(z) - E] \quad (2.143)$$

and

$$\beta = \frac{2m^*}{\hbar^2} (-eF) \quad (2.144)$$

the Schrödinger equation becomes:

$$\frac{\partial^2 \psi}{\partial z^2} - (\alpha + \beta z) \psi = 0 \quad (2.145)$$

Consider the further substitution:

$$z' = \frac{\alpha + \beta z}{\gamma} \quad (2.146)$$

where γ is an, as yet, unknown constant. Then, since z' is a first-order linear function of z :

$$\frac{\partial^2 \psi}{\partial z^2} = \frac{\partial^2 \psi}{\partial z'^2} \times \left(\frac{\partial z'}{\partial z} \right)^2 \quad (2.147)$$

$$\text{i.e.} \quad \frac{\partial^2 \psi}{\partial z^2} = \frac{\beta^2}{\gamma^2} \frac{\partial^2 \psi}{\partial z'^2} \quad (2.148)$$

Substituting into equation (2.145) then gives:

$$\frac{\beta^2}{\gamma^2} \frac{\partial^2 \psi}{\partial z'^2} - \gamma z' \psi = 0 \quad (2.149)$$

If γ^3 is set equal to β^2 , then the full transformation is:

$$z' = \left(\frac{2m^*}{\hbar^2} \right)^{\frac{1}{3}} \left[\frac{V(z) - E}{(eF)^{\frac{2}{3}}} - (eF)^{\frac{1}{3}} z \right] \quad (2.150)$$

and the Schrödinger equation can be written as:

$$\frac{\partial^2 \psi}{\partial z'^2} - z' \psi = 0 \quad (2.151)$$

The reason for such a procedure is that equation (2.151) has a standard solution which is a linear combination of Airy functions (see Abramowitz and Stegun [38], p. 446):

$$\psi(z') = A \text{Ai}(z') + B \text{Bi}(z') \quad (2.152)$$

The full solution proceeds by matching the wave functions at the heterojunctions according to the BenDaniel–Duke boundary conditions as before.

This is a standard procedure and many examples of its use can be found in the literature [39]. It is included along with the transfer matrix method of the previous sections for completeness. However, it can be appreciated that within these methods the study of a new quantum well system requires quite a large investment in developing new computational solutions. While a general computer code can be written to solve all semiconductor heterostructures, another one has to be written to solve all of these systems again with an electric field. The possibility of adding piezoelectric fields [40], which tilt the bands like an electric field, but with the additional possibility of being in the other direction, would require another computer code, and so on. While we continue in this vain for the remainder of this chapter, in the next chapter a numerical solution is introduced, which although occasionally has a few limitations, is still very powerful because of its *versatility*.

2.13 THE INFINITE SUPERLATTICE

Finite superlattices (or multiple quantum wells) can be treated with the transfer matrix technique, as described above for a double quantum well (a two period finite superlattice). Such approaches yield a discrete step of states due to the confinement along the growth axis of the semiconductor multilayer. However, when many identical quantum wells are stacked within the same semiconductor layer the electrons and holes see a periodic potential, which *can* appear to be infinite in exactly the same way as bulk crystals. When this occurs, the electron and hole wave functions are no longer localised but are of infinite extent and equally likely to be in any of the quantum wells. They are said to occupy ‘Bloch states’.

As described in Section 2.11, and following Jaros [5], p. 220, the envelope function within the well region can be written as:

$$\psi_w = A \exp(ik_w z) + B \exp(-ik_w z) \quad (2.153)$$

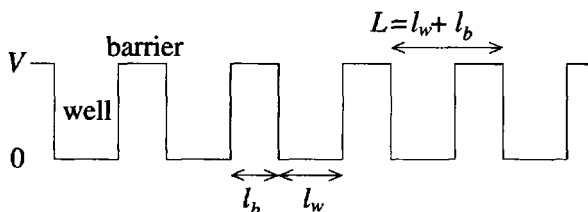


Figure 2.26 Band-edge potential profile of a superlattice of well and barrier widths, l_w and l_b respectively

and in the barrier region as:

$$\psi_b = C \exp(ik_b z) + D \exp(-ik_b z) \quad (2.154)$$

where:

$$k_w = \sqrt{\frac{2m_w^*}{\hbar^2} E} \quad k_b = \sqrt{\frac{2m_b^*}{\hbar^2} (E - V)} \quad (2.155)$$

and it is assumed (for now) that $E > V$, so that k_b is real. Clearly, and as mentioned above, if the superlattice is infinite then a particle is equally likely to be found in any well, which means that its wave function must be periodic with the lattice, i.e.

$$\psi(z) = \psi(z + L) \quad (2.156)$$

For a travelling-wave state of the form, $\exp(ikz)$, then:

$$\psi(z + L) = \exp[ik(z + L)] = \exp(ikz) \exp(ikL) \quad (2.157)$$

$$\text{i.e.} \quad \psi(z + L) = \psi(z) \exp(ikL) \quad (2.158)$$

By using this *periodic* form of the first BenDaniel–Duke boundary conditions (ψ continuous) at $z = L$, then:

$$\psi_w(L) = \psi_w(0) \exp(ikL) = \psi_b(L) \quad (2.159)$$

Using the wave functions of equations (2.153) and (2.154), then:

$$(A + B) \exp(ikL) = C \exp(ik_b L) + D \exp(-ik_b L) \quad (2.160)$$

By employing the same periodicity condition in the second of the BenDaniel–Duke boundary conditions (ψ/m), also at $z = L$, then:

$$\frac{ik_w}{m_w^*} (A - B) \exp(ikL) = \frac{ik_b}{m_b^*} [C \exp(ik_b L) - D \exp(-ik_b L)] \quad (2.161)$$

Then using the BenDaniel–Duke boundary conditions at $z = l_w$:

$$A \exp(ik_w l_w) + B \exp(-ik_w l_w) = C \exp(ik_b l_w) + D \exp(-ik_b l_w) \quad (2.162)$$

and

$$\frac{ik_w}{m_w^*} [A \exp(ik_w l_w) - B \exp(-ik_w l_w)] = \frac{ik_b}{m_b^*} [C \exp(ik_b l_w) - D \exp(-ik_b l_w)] \quad (2.163)$$

Equations (2.160), (2.161), (2.162) and (2.163), can be rewritten as:

$$\begin{pmatrix} \exp(ikL) & \exp(ikL) & -\exp(ik_b L) & -\exp(-ik_b L) \\ \frac{k_w}{m_w^*} \exp(ikL) & -\frac{k_w}{m_w^*} \exp(ikL) & -\frac{k_b}{m_b^*} \exp(ik_b L) & \frac{k_b}{m_b^*} \exp(-ik_b L) \\ \exp(ik_w l_w) & \exp(-ik_w l_w) & -\exp(ik_b l_w) & -\exp(-ik_b l_w) \\ \frac{k_w}{m_w^*} \exp(ik_w l_w) & -\frac{k_w}{m_w^*} \exp(-ik_w l_w) & -\frac{k_b}{m_b^*} \exp(ik_b l_w) & \frac{k_b}{m_b^*} \exp(-ik_b l_w) \end{pmatrix} \times \begin{pmatrix} A \\ B \\ C \\ D \end{pmatrix} = 0 \quad (2.164)$$

which implies that, for a solution other than the trivial $A = B = C = D = 0$, the determinant of the 4×4 must be zero, i.e.

$$\begin{vmatrix} \exp(ikL) & \exp(ikL) & -\exp(ik_b L) & -\exp(-ik_b L) \\ \frac{k_w}{m_w^*} \exp(ikL) & -\frac{k_w}{m_w^*} \exp(ikL) & -\frac{k_b}{m_b^*} \exp(ik_b L) & \frac{k_b}{m_b^*} \exp(-ik_b L) \\ \exp(ik_w l_w) & \exp(-ik_w l_w) & -\exp(ik_b l_w) & -\exp(-ik_b l_w) \\ \frac{k_w}{m_w^*} \exp(ik_w l_w) & -\frac{k_w}{m_w^*} \exp(-ik_w l_w) & -\frac{k_b}{m_b^*} \exp(ik_b l_w) & \frac{k_b}{m_b^*} \exp(-ik_b l_w) \end{vmatrix} = 0 \quad (2.165)$$

Proceeding one step would give:

$$\begin{aligned} \exp(ikL) & \begin{vmatrix} -\frac{k_w}{m_w^*} \exp(ikL) & -\frac{k_b}{m_b^*} \exp(ik_b L) & \frac{k_b}{m_b^*} \exp(-ik_b L) \\ \exp(-ik_w l_w) & -\exp(ik_b l_w) & -\exp(-ik_b l_w) \\ -\frac{k_w}{m_w^*} \exp(-ik_w l_w) & -\frac{k_b}{m_b^*} \exp(ik_b l_w) & \frac{k_b}{m_b^*} \exp(-ik_b l_w) \end{vmatrix} \\ -\exp(ikL) & \begin{vmatrix} \frac{k_w}{m_w^*} \exp(ikL) & -\frac{k_b}{m_b^*} \exp(ik_b L) & \frac{k_b}{m_b^*} \exp(-ik_b L) \\ \exp(ik_w l_w) & -\exp(ik_b l_w) & -\exp(-ik_b l_w) \\ \frac{k_w}{m_w^*} \exp(ik_w l_w) & -\frac{k_b}{m_b^*} \exp(ik_b l_w) & \frac{k_b}{m_b^*} \exp(-ik_b l_w) \end{vmatrix} \\ -\exp(ik_b L) & \begin{vmatrix} \frac{k_w}{m_w^*} \exp(ikL) & -\frac{k_w}{m_w^*} \exp(ikL) & \frac{k_b}{m_b^*} \exp(-ik_b L) \\ \exp(ik_w l_w) & \exp(-ik_w l_w) & -\exp(-ik_b l_w) \\ \frac{k_w}{m_w^*} \exp(ik_w l_w) & -\frac{k_w}{m_w^*} \exp(-ik_w l_w) & \frac{k_b}{m_b^*} \exp(-ik_b l_w) \end{vmatrix} \end{aligned}$$

$$\exp(-ik_b L) \begin{vmatrix} \frac{k_w}{m_w^*} \exp(ikL) & -\frac{k_w}{m_w^*} \exp(ikL) & -\frac{k_b}{m_b^*} \exp(-ik_b L) \\ \exp(ik_w l_w) & \exp(-ik_w l_w) & -\exp(ik_b l_w) \\ \frac{k_w}{m_w^*} \exp(ik_w l_w) & -\frac{k_w}{m_w^*} \exp(-ik_w l_w) & -\frac{k_b}{m_b^*} \exp(ik_b l_w) \end{vmatrix}$$

which the reader can see becomes very tedious indeed. Fortunately use can be made of the constant mass case of Jaros [5], and generalised to account for $m_w \neq m_b$ as here, giving:

$$\cos(k_w l_w) \cos(k_b l_b) - \sin(k_w l_w) \sin(k_b l_b) \left(\frac{m_b^2 k_w^2 + m_w^2 k_b^2}{2m_w m_b k_w k_b} \right) = \cos(kL) \quad (2.166)$$

If $E < V$, then following the arguments in Section 2.11, k_b becomes $i\kappa$, where:

$$\kappa = \sqrt{\frac{2m_b^*}{\hbar^2} (V - E)} \quad (2.167)$$

Substituting for k_b into equation (2.166), and using the following identities:

$$\cos(i\kappa l_b) \equiv \cosh(\kappa l_b) \quad \text{and} \quad \sin(i\kappa l_b) \equiv i \sinh(\kappa l_b) \quad (2.168)$$

then:

$$\cos(k_w l_w) \cosh(\kappa l_b) - \sin(k_w l_w) \sinh(\kappa l_b) \left(\frac{m_b^2 k_w^2 - m_w^2 \kappa^2}{2m_w m_b k_w \kappa} \right) = \cos(kL) \quad (2.169)$$

The superlattice dispersion curves, i.e. the energy E of a particle as a function of its wave vector k , are obtained by solving equations (2.166) and (2.169). This is accomplished by using the same methods as in Section 2.5, i.e. equations (2.166) and (2.169) are rewritten in the form $f(E, k) = 0$, and solved for chosen values of k . Again, a Newton–Raphson iteration is efficient; however, in order to avoid having to deduce $f'(E, k)$, a finite difference expansion can be employed:

$$f'(E, k) \approx \frac{f(E + \delta E, k) - f(E - \delta E, k)}{2\delta E} \quad (2.170)$$

Hence the Newton–Raphson iterative equation becomes:

$$E^{(n+1)} = E^{(n)} - \frac{2f(E^{(n)}) \delta E}{f(E^{(n)} + \delta E) - f(E^{(n)} - \delta E)} \quad (2.171)$$

where δE is smaller than adjacent solutions for E ; typically 10^{-3} meV is a good choice. This convenient method of approximating derivatives with nearby values of the function will be returned to in the next chapter and dealt with there in much more detail.

Fig. 2.27 shows the lowest-energy solutions of a 40 Å GaAs/40 Å Ga_{0.6}Al_{0.4}As superlattice, for a range of k values. It can be seen that the energy E is a periodic function of k with period $2\pi/L$. The wave vector k was introduced

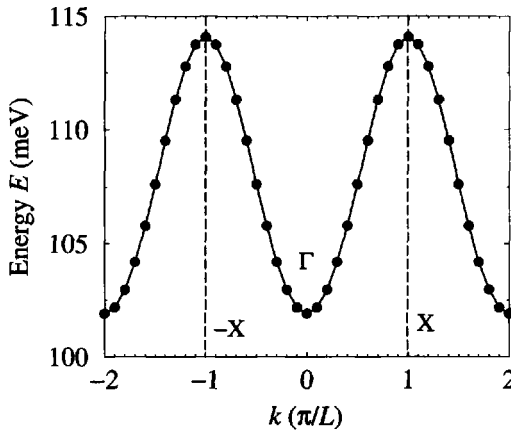


Figure 2.27 Lowest-energy solutions of the Kronig–Penney model of a 40 Å GaAs/40 Å $\text{Ga}_{0.6}\text{Al}_{0.4}\text{As}$ superlattice, for a range of electron momenta along the growth (z -) axis

earlier to describe the wave function as a travelling wave; clearly it can be identified as the momentum of a charge carrier along the growth (z -) axis of an *infinite* superlattice. Using Fig. 2.27, it can be seen that a definite minimum exists when the particle is at rest, i.e. $k=0$. As the carrier momentum increases, its energy increases too and reaches a maximum at $k = \pm\pi/L$. Thus the carrier within the superlattice occupies a continuous range of energies with a maxima and minima; this domain is the superlattice analogy to the energy bands of a crystal and is referred to as a *miniband*.

In addition the symmetry points of the E – k curves are labelled as in bulk crystals, where $k=0$ is called Γ and $k = \pi/L$ is referred to as X. This domain in k -space has become known as the *superlattice Brillouin zone* and as in the bulk, is a convenient way to express the relationship between energy and momenta.

Just as single quantum wells can have more than one confined state, superlattices can have more than one miniband. Fig. 2.28 displays the two lowest energy solutions of the longer period 60 Å GaAs/60 Å $\text{Ga}_{0.6}\text{Al}_{0.4}\text{As}$ superlattice; note on this scale that the lowest energy miniband looks almost flat. Clearly, the second miniband has the same periodicity as the first, although it should be noted that its minimum in energy occurs at the edge of the superlattice Brillouin zone and not at the centre. This effect is exploited in the recently developed ‘inter-miniband laser’ [41–43], which is a very promising high-power source of mid-infrared radiation. This device will be discussed in more detail later, i.e. in the chapter on electron scattering.

The structural dependence of the miniband is illustrated in Fig. 2.29 for a series of superlattices with equal well and barrier widths. The two solid curves display the

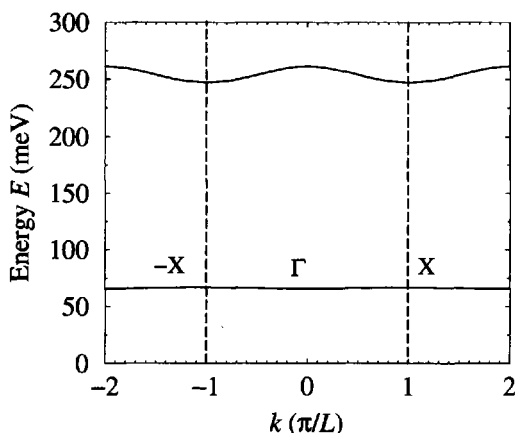


Figure 2.28 Two lowest-energy solutions of an 60 Å GaAs/60 Å Ga_{0.6}Al_{0.4}As superlattice

energy as a function of l_w at Γ and X . It can be seen that these energies fall with increasing well width, as does the miniband width, defined as follows:

$$\text{miniband width} = |E(X) - E(\Gamma)| \quad (2.172)$$

where the modulus bars have been included to account for minibands such as the second, for which $E(X) < E(\Gamma)$. Interestingly, the top and bottom of the miniband straddle the energy of the corresponding single quantum well (given by the dashed line). As the wells become more isolated and further apart, the miniband narrows and the energies tend towards that of the single quantum well. This represents the cross-over from a superlattice to a multiple quantum well—a collection of non-interacting identical wells.

As the miniband width changes with structure, the curvature of the E - k curve changes too. It is clear from Chapter 1 that the effective mass of a particle subject to an energy-momentum dependence of $E(k)$ can be derived from:

$$m^* = \frac{\hbar^2}{\partial^2 E / \partial k^2} \quad (2.173)$$

Therefore, when an electron or hole is moving along the growth axis of a superlattice, the energy-momentum relationship is such that they appear to have a different effective mass from the bulk. Fig. 2.30 plots this new effective mass at Γ for the range of structures in Fig. 2.29.

It can be seen from the figure that at very short periods the electron's miniband structure produces a new effective mass which is considerably smaller than the bulk value ($0.067 m_0$). Thus electrons within this miniband will have a much greater

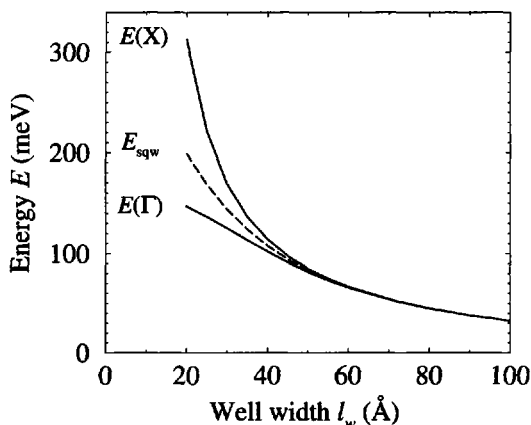


Figure 2.29 Energy at $k = 0(\Gamma)$ and $k = \pi/L(X)$, showing the lower and upper energy limits of the superlattice minibands, compared with the ground-state energy of the single quantum well of the same width

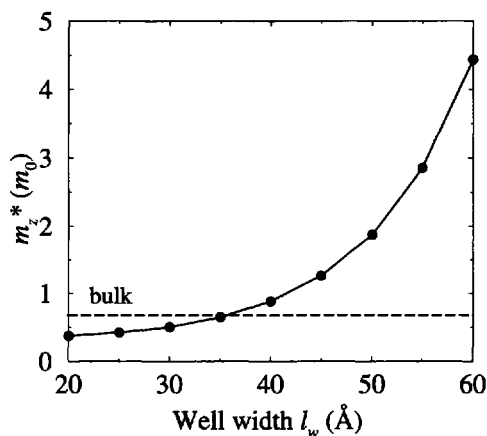


Figure 2.30 The new effective mass for an electron moving in the superlattice miniband

response to an electric field than in bulk, just as from Newton's second law, ' $F = ma$ ', a lower m requires a smaller force in order to produce the same acceleration. In strict semiconductor language, this is quantified in terms of the carrier mobility, which is

given by:

$$\mu = \frac{e\tau_{\text{coll.}}}{m^*} \quad (2.174)$$

where $\tau_{\text{coll.}}$ is the time between collisions, see [1], p. 601, the reciprocal of which is usually referred to as the total scattering rate for all mechanisms, via phonons, ionised impurities, other carriers, etc. Clearly, a reduced effective mass within a miniband gives an increased mobility μ for transport within that miniband, *provided* that all other scattering terms remain the same.

In this series of calculations, attention has been focussed on superlattices with a simple single quantum-well unit cell, and indeed with equal well and barrier widths. If these restrictions are lifted, then there are an almost infinite number of possible superlattices that can be constructed, thus giving the opportunity to engineer, even design, the effective mass to suit the device.

2.14 THE SINGLE BARRIER

Hitherto, only semiconductor layered systems have been considered which form quantum wells, i.e. produce semiconductor layers which trap or confine the electrons along one axis. The 'opposite' of such structures also exist and are known as *barrier structures*. If a layer of a larger-band-gap material, e.g. $\text{Ga}_{1-x}\text{Al}_x\text{As}$, is sandwiched between layers of a narrower-band-gap material, e.g. GaAs, then a potential barrier *can* result which repels carriers.

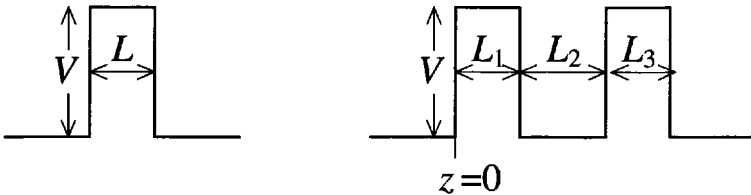


Figure 2.31 Single- and double-barrier structures

Fig. 2.31 illustrates such a single barrier structure. Electrons placed within this layered system simply collect in the GaAs outer (lower energy) regions and there are no quantum-confined energy states. These systems do, however, exhibit quantum behaviour when an electric field is applied perpendicular to the layers (along the growth (z -) axis). Electrons (or holes) arising from doping are accelerated and impinge upon the barrier and even when they have an energy E which is less than the potential energy height V of the barrier, they have a finite probability of passing through the barrier and appearing on the other side. This phenomenon is called *quantum mechanical tunnelling* or often just *tunnelling*. The bizarre nature of quantum mechanical tunnelling can be illustrated with a classical analogy, i.e. one would be pretty disturbed, if when kicking a football against a brick wall, it went straight through! Yet this is exactly what happens with *some* electrons and holes when

they meet a potential barrier within a crystal. Tunnelling is an everyday phenomena which occurs in a range of semiconductor devices some of which appear in consumer electronics products, see, for example, Sze [44].

One way of quantifying the proportion of electrons that tunnel through a barrier is in terms of the *transmission coefficient* which is defined as the probability that any single electron impinging on a barrier structure will tunnel and contribute to the current flow through the barrier. Ferry has produced a comprehensive analysis of the transmission coefficient for a single-barrier structure (see [45], p. 60). Suffice here to quote Ferry's result in ([45], equation 3.12), for a constant effective mass across the structure, i.e. the transmission coefficient at an energy E for a barrier of width L and height V is given by:

$$T(E) = \frac{1}{1 + \left(\frac{k^2 + \kappa^2}{2k\kappa}\right)^2 \sinh^2(\kappa L)}, \quad \text{for} \quad E < V \quad (2.175)$$

where as usual:

$$k = \frac{\sqrt{2m^*E}}{\hbar} \quad \text{and} \quad \kappa = \frac{\sqrt{2m^*(V-E)}}{\hbar} \quad (2.176)$$

For values of the carrier energy E greater than the barrier height V , $\kappa \rightarrow ik'$ (as in Section 2.13), and hence:

$$T(E) = \frac{1}{1 + \left(\frac{k^2 - k'^2}{2kk'}\right)^2 \sin^2(k'L)} \quad \text{for} \quad E > V \quad (2.177)$$

where

$$k = \frac{\sqrt{2m^*E}}{\hbar} \quad \text{and} \quad k' = \frac{\sqrt{2m^*(E-V)}}{\hbar} \quad (2.178)$$

The mathematics show, that for $E > V$, the transmission coefficient would be expected to oscillate, with resonances when the sine term is zero. These occur when:

$$k'L = n\pi \quad (2.179)$$

which is implied when:

$$E = \frac{(n\hbar\pi)^2}{2m^*L^2} + V \quad (2.180)$$

The squared dependence implies that the resonances when $T = 1$ occur at larger and larger intervals in E , which can be clearly seen in Figs 2.32 and 2.33.

Fig. 2.32 displays the transmission coefficient as a function of the energy E and for a range of barrier widths L . For this range of calculations the barrier height V was fixed at 100 meV, and below this energy the thinner the barrier, then the higher the probability of tunnelling. For $E > V$, the situation is more complex due to the oscillatory nature of T . The trend, however, as highlighted by the curves, is that the

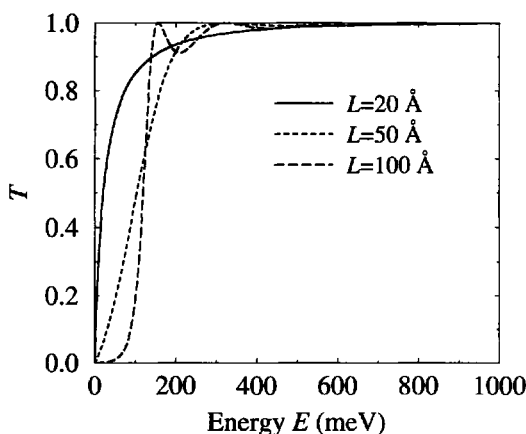


Figure 2.32 Transmission coefficient as a function of the energy through a single barrier for different barrier widths

thicker the barrier, the closer the first resonance ($T = 1$) is to the top of the barrier. This can be understood from equation (2.180):

$$\lim_{L \rightarrow \infty} E_{\text{resonance}}^{n=1} = \lim_{L \rightarrow \infty} \frac{(\hbar\pi)^2}{2m^*L^2} + V = V \quad (2.181)$$

Conversely, for a fixed L and a variable V , the first resonance occurs at the same point above the barrier height; this is clearly illustrated in Fig. 2.33.

2.15 THE DOUBLE BARRIER

If two barriers are placed a reasonably small distance apart (in the same crystal, perhaps a few nm) then the system is known as a *double barrier* (see Fig. 2.31), and has quite different transmission properties to the single barrier. Datta [46], p. 33, has deduced the transmission $T(E)$ dependence for the restricted case of symmetric barriers, while Ferry [45], p. 66, has considered asymmetric barriers. In this formalism, allowance will be made for differing barrier widths as well as the discontinuous change in the effective mass between well and barrier materials. Thus, with the aim of deducing the new $T(E)$, consider the solutions to Schrödinger's equation within

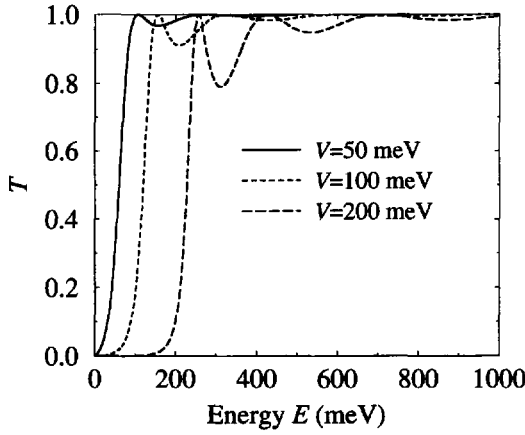


Figure 2.33 Transmission coefficient as a function of the energy through a single 100 Å barrier for different barrier heights

each region for $E < V$:

$$\psi(z) = A \exp(ikz) + B \exp(-ikz), \quad z < I_1 \quad (2.182)$$

$$\psi(z) = C \exp(\kappa z) + D \exp(-\kappa z), \quad I_1 < z < I_2 \quad (2.183)$$

$$\psi(z) = F \exp(ikz) + G \exp(-ikz), \quad I_2 < z < I_3 \quad (2.184)$$

$$\psi(z) = H \exp(\kappa z) + J \exp(-\kappa z), \quad I_3 < z < I_4 \quad (2.185)$$

$$\psi(z) = K \exp(ikz) + L \exp(-ikz), \quad I_4 < z \quad (2.186)$$

where k and κ have their usual forms as given in equation (2.176) and the positions of the interfaces have been labelled I_1 , I_2 , I_3 , and I_4 , respectively. Using the standard BenDaniel-Duke boundary conditions at each interface gives the following

$z = I_1 = 0$:

$$A + B = C + D \quad (2.187)$$

$$\frac{1}{m_w}(ikA - ikB) = \frac{1}{m_b}(\kappa C - \kappa D) \quad (2.188)$$

$z = I_2 = L_1$:

$$C \exp(\kappa I_2) + D \exp(-\kappa I_2) = F \exp(ikI_2) + G \exp(-ikI_2) \quad (2.189)$$

$$\frac{1}{m_b}[\kappa C \exp(\kappa I_2) - \kappa D \exp(-\kappa I_2)] = \frac{1}{m_w}[ikF \exp(ikI_2) - ikG \exp(-ikI_2)] \quad (2.190)$$

$$z = I_3 = L_1 + L_2:$$

$$F \exp(ikI_3) + G \exp(-ikI_3) = H \exp(\kappa I_3) + J \exp(-\kappa I_3) \quad (2.191)$$

$$\frac{1}{m_w} [ikF \exp(ikI_3) - ikG \exp(-ikI_3)] = \frac{1}{m_b} [\kappa H \exp(\kappa I_3) - \kappa J \exp(-\kappa I_3)] \quad (2.192)$$

$$z = I_4 = L_1 + L_2 + L_3:$$

$$H \exp(\kappa I_4) + J \exp(-\kappa I_4) = K \exp(ikI_4) + L \exp(-ikI_4) \quad (2.193)$$

$$\frac{1}{m_b} [\kappa H \exp(\kappa I_4) - \kappa J \exp(-\kappa I_4)] = \frac{1}{m_w} [ikK \exp(ikI_4) - ikL \exp(-ikI_4)] \quad (2.194)$$

The method of solution is the transfer matrix technique as before, writing the above equations in matrix form:

$$\mathbf{M}_1 \begin{pmatrix} A \\ B \end{pmatrix} = \mathbf{M}_2 \begin{pmatrix} C \\ D \end{pmatrix} \quad (2.195)$$

$$\mathbf{M}_3 \begin{pmatrix} C \\ D \end{pmatrix} = \mathbf{M}_4 \begin{pmatrix} F \\ G \end{pmatrix} \quad (2.196)$$

$$\mathbf{M}_5 \begin{pmatrix} F \\ G \end{pmatrix} = \mathbf{M}_6 \begin{pmatrix} H \\ J \end{pmatrix} \quad (2.197)$$

$$\mathbf{M}_7 \begin{pmatrix} H \\ J \end{pmatrix} = \mathbf{M}_8 \begin{pmatrix} K \\ L \end{pmatrix} \quad (2.198)$$

Then, as before, the coefficients of the outer regions can be linked by forming the transfer matrix, i.e.

$$\begin{pmatrix} A \\ B \end{pmatrix} = \mathbf{M}_1^{-1} \mathbf{M}_2 \mathbf{M}_3^{-1} \mathbf{M}_4 \mathbf{M}_5^{-1} \mathbf{M}_6 \mathbf{M}_7^{-1} \mathbf{M}_8 \begin{pmatrix} K \\ L \end{pmatrix} \quad (2.199)$$

Clearly, this 2×2 matrix equation still has four unknowns and can't be solved—it is at this point that additional boundary conditions have to be imposed from physical intuition. Whereas before, the standard boundary conditions, i.e. $\psi(z) \rightarrow 0$ as $z \rightarrow \pm\infty$, were used to solve for the confined states within quantum wells, in these barrier structures these are not appropriate since the travelling waves in the outer layers can have infinite extent. The standard procedure is to assume, quite correctly, that all of the charge carriers approach the double barrier from the same side, as would occur when as part of a biased device, as illustrated schematically in Fig. 2.34. Furthermore, if it is assumed that there are no further heterojunctions to the right of the structure, then no further reflections can occur and the wave function beyond the structure can only have a travelling wave component moving to the right, i.e. the coefficient L must be zero.

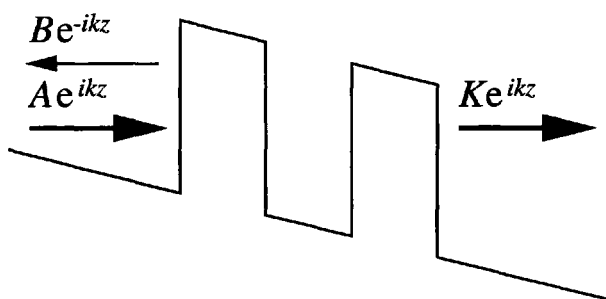


Figure 2.34 The wave function after imposition of the boundary conditions

Therefore, if the 2×2 matrix of equation (2.199) is written as \mathcal{M} , then:

$$\begin{pmatrix} A \\ B \end{pmatrix} = \mathcal{M} \begin{pmatrix} K \\ 0 \end{pmatrix} \quad (2.200)$$

$$\therefore A = \mathcal{M}_{11} K \quad (2.201)$$

and the ratio of transmitted to incident current, i.e. the transmission coefficient, is simply:

$$T(E) = \frac{K^* K}{A^* A} = \frac{1}{\mathcal{M}_{11}^* \mathcal{M}_{11}} \quad (2.202)$$

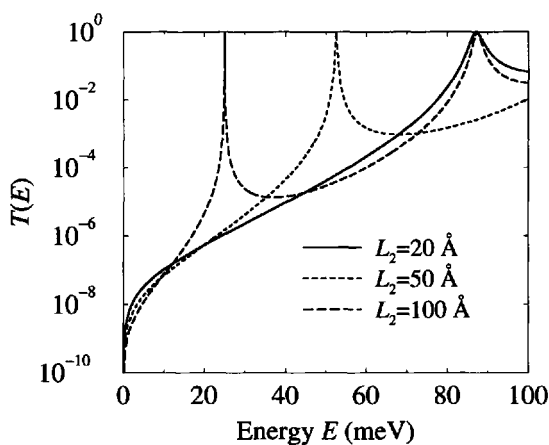


Figure 2.35 Transmission coefficient as a function of the energy through a double barrier of width 100 \AA and height 100 meV , separated by a distance L_2

Fig. 2.35 gives an example of the form of $T(E)$ for barriers of height 100 meV and width 100 \AA as a function of the distance L_2 between them. The effective masses

were both taken to be equal to the bulk Γ valley electron mass of $0.067 m_0$. It can be seen from the figure that the curves contain Dirac δ -functions at certain energies E below the potential barrier height V . This is quite unlike the single barrier case. At these *resonance* energies, the double-barrier system appears transparent and has a transmission coefficient of 1. The wave functions of these states are localised between the barriers and are often referred to as *quasi-bound states* since they resemble the bound states of quantum well structures. However, they are not stationary states in that electrons or holes in such states will eventually scatter into the lower energy states outside of the barriers.

The effect of an increasing barrier height V is shown in Fig. 2.36. It can be seen that, away from a resonance, an increasing barrier height leads, as would be expected, to a decrease in the transmission coefficient T . The classical explanation for this would be, 'it is harder for the electrons to tunnel through higher barriers'. The *resonance energies* increase with increasing barrier height due to confinement effects, and the appearance of the second resonance at higher energies is a reflection on the existence of a second quasi bound state.

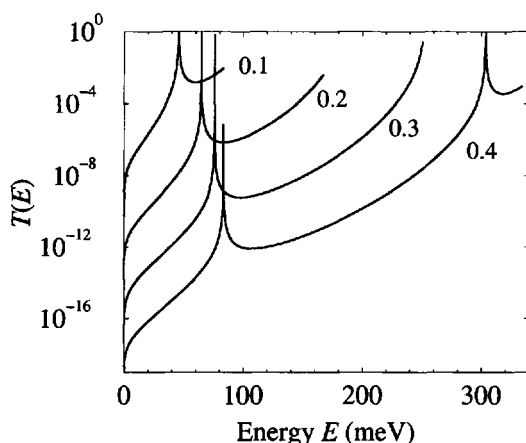


Figure 2.36 Transmission coefficient as a function of the energy through a 100 \AA $\text{Ga}_{1-x}\text{Al}_x\text{As}$ / 50 \AA GaAs / 100 \AA $\text{Ga}_{1-x}\text{Al}_x\text{As}$ double barrier, for $x=0.1, 0.2, 0.3$, and 0.4

While the transmission coefficient represents a very important parameterisation of the properties of a double-barrier structure, it itself is not a measurable quantity. In fact, the properties of such two-terminal electronic devices are generally inferred (or summarised) from their current–voltage characteristics (I – V curve).

When an electric field is placed across such a double barrier structure, any charge carriers present in the semiconductor, intrinsic or extrinsic, constitute a current which approaches the left-hand barrier. These charge carriers have a distribution of energy

and momenta, often a Fermi-Dirac distribution. Those carriers that are of the same energy as the resonance are able to pass right through the double barrier without hindrance—a phenomenon which has become known as *resonant tunnelling*. As the applied electric field (applied voltage) is increased, the number of carriers with the resonance energy increases and peaks as the Fermi level of the semiconductor to the left of the first barrier is brought into alignment. Therefore the current gradually increases. At higher fields, the current falls away and a period of negative differential resistance ensues [47, 48]. Such a current–voltage characteristic has been exploited in high frequency circuits, thus bringing the *resonant tunnelling diode* to prominence as a very useful electronic device [49].

There are various models of the current–voltage properties of different levels of complexity, the simplest of which would probably be to return to the idea that the current at any particular field would be equal to the number of carriers that tunnellled through the structure. This in turn would be the probability of a particular carrier tunnelling, multiplied by the number of carriers at that energy, i.e.

$$I \propto \int_{\text{band}} T(E) f^{\text{FD}}(E) \rho^{3\text{D}}(E) dE \quad (2.203)$$

As the carriers approaching the barrier structure are in a bulk band, then the integral is over their energies, and the Fermi–Dirac distribution function and density of states have the bulk (3D) forms.

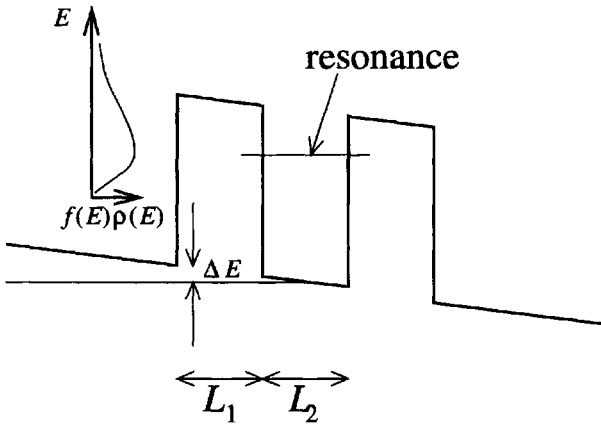


Figure 2.37 Simple model of current flow through a double barrier structure

Figure 2.37 outlines this model. The electric field dependence is introduced via f^{FD} and $\rho^{3\text{D}}$. As the electric field is increased, the bottom of the bulk band is increased in energy relative to the centre of the double-barrier structure by an amount $\Delta E = eF(L_1 + L_2/2)$. Hence, the reference energy (the band minimum) will increase, and by using equation (2.40), the density of states at some energy E , measured from the

band minimum at the centre of the well, would become

$$\rho(E) = \frac{1}{2\pi^2} \left(\frac{2m^*}{\hbar^2} \right)^{\frac{3}{2}} (E - \Delta E)^{\frac{1}{2}} \Theta(E - \Delta E) \quad (2.204)$$

where the unit step function ensures that the energy of the carriers impinging on the left hand barrier E is greater than ΔE . At the bottom of the band, $E = \Delta E$, and $\rho(E) = 0$. In addition, the Fermi energy of the bulk carriers increases by the same amount, i.e. from equation (2.49):

$$f^{\text{FD}}(E) = \frac{1}{\exp \{ [E - (E_F + \Delta E)] / kT \} + 1} \quad (2.205)$$

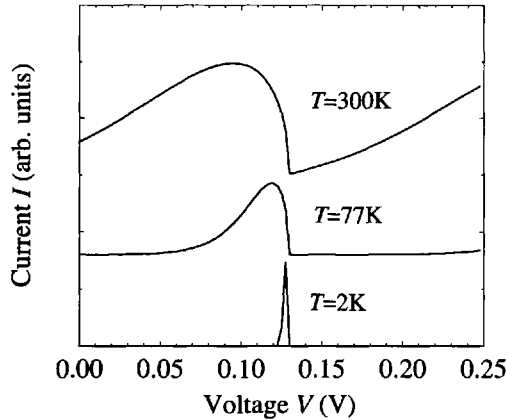


Figure 2.38 Current–voltage curve for the 100 \AA $\text{Ga}_{0.8}\text{Al}_{0.2}\text{As}/50 \text{ \AA}$ $\text{GaAs}/100 \text{ \AA}$ $\text{Ga}_{0.8}\text{Al}_{0.2}\text{As}$ of earlier, obtained at different temperatures

Figure 2.38 illustrates the results obtained by using this model. The current is obtained in arbitrary units and the voltage V has been defined simply as the potential difference across the structure at a particular field, i.e. $V = F(L_1 + L_2 + L_3)$. The I – V curves have been plotted for several temperatures for the $x = 0.2$ structure of Fig. 2.36. The single resonance within this system shows itself as a single peak in the current. Clearly, the current peak broadens as the temperature increases, with this being a direct result of the broadening of the carrier distribution (the number of carriers in a given energy range), given by the product $f^{\text{FD}}(E)\rho(E)$ appearing in the integral for the current, and illustrated schematically in Fig. 2.37. At low temperatures, the carriers occupy a small region of energy space around the band minima. As the field is increased, current only begins to flow when this narrow distribution is brought into

line with the resonance energy, and hence there is a narrow current peak. As the temperature increases, the carrier distribution broadens, therefore there is a greater range of applied voltages that give some degree of alignment of the carriers with the resonance energy. The peak occurs when the peak of the distribution is aligned with the resonance energy. At voltages above this, the number of carriers available for tunnelling decreases, and hence the current also decreases.

The non-zero current at zero field for $T = 300$ K is a consequence of this simplistic model. It arises because even at zero field, there is a finite number of carriers in the broadened distribution which are aligned with the resonance energy and are therefore able to tunnel.

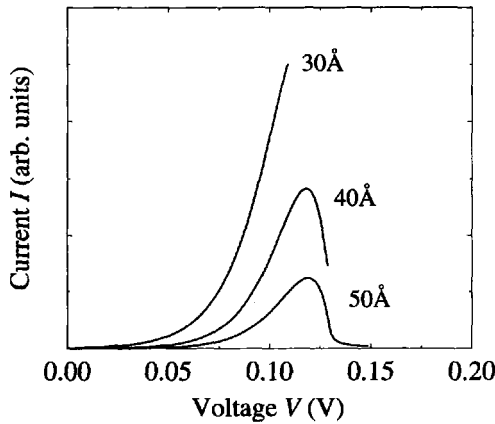


Figure 2.39 Current–voltage curves obtained at 77 K for a variety of barrier widths the structure of Fig. 2.38

Fig. 2.39 displays the results of the same simple model as a function of barrier width for the same structure of Fig. 2.38, but this time at the fixed temperature of 77 K. Again, as noted before, thinner barriers allow the electrons to tunnel more easily and hence give a higher current. An interesting physical point can be inferred from this data. The incomplete nature of the peaks in the 40 and 30 Å data is an indication that a significant fraction of the electrons are passing over the top of the barriers. While the current–voltage model advanced in this section can account for such a situation, the transmission coefficient versus energy data is terminated at the top of the barrier, and so only the fraction of electrons below this are included in the integration for the current in equation (2.203). It is left as an exercise for the reader to generalise the transmission coefficient theory to account for the case when $E > V$, and therefore improve the current–voltage model. This can be achieved by using a similar approach to that of Section 2.13.

The above discussion has been a simple introduction to the modelling of I - V curves for barrier structures, but nonetheless it shows some of the features of real devices. For a much more complete and in depth study see, for example, Mizuta and Tanoue [49].

2.16 EXTENSION TO INCLUDE ELECTRIC FIELD

An obvious improvement to the above model would be to account for the changes in the transmission coefficient as a function of the applied electric field. By using the substitution as before (equation (2.150)), i.e.

$$z' = \left(\frac{2m^*}{\hbar^2} \right)^{\frac{1}{3}} \left[\frac{V(z) - E}{(eF)^{\frac{2}{3}}} - (eF)^{\frac{1}{3}} z \right] \quad (2.206)$$

the solution in each region can then be written as a linear combination of Airy functions, just as for the general electric field case of equation (2.152), i.e.

$$\psi(z) = AAi(z') + BBi(z'), \quad z < I_1 \quad (2.207)$$

$$\psi(z) = CAi(z') + DBi(z'), \quad I_1 < z < I_2 \quad (2.208)$$

$$\psi(z) = FAi(z') + GBi(z'), \quad I_2 < z < I_3 \quad (2.209)$$

$$\psi(z) = H Ai(z') + J Bi(z'), \quad I_3 < z < I_4 \quad (2.210)$$

$$\psi(z) = K Ai(z') + L Bi(z'), \quad I_4 < z \quad (2.211)$$

while Airy functions *can* be difficult to work with numerically, an immediate advantage over the zero-field case is that this solution is valid for both $E < V$ and $E > V$, and hence generalisation to this form produces two benefits. The method of solution is analogous to the zero-field case, in that application of the BenDaniel-Duke boundary conditions yields two equations for each interface, which in this case gives eight equations. The unknown coefficients, A and B , are linked to K and L as before by forming the transfer matrix, and are solved by imposition of a boundary condition, which is again a travelling wave in the direction of $+z$ to the right of the barrier structure. It is left to the interested reader to follow through such a derivation. A very general implementation for multiple barrier structures has been reported in the literature by Vatannia and Gildenblat [39].

2.17 MAGNETIC FIELDS AND LANDAU QUANTISATION

If a magnetic field is applied externally to a *non-magnetic* semiconductor heterostructure then the constant effective mass Hamiltonian (familiar from equation (2.5)):

$$\mathcal{H} = -\frac{\hbar^2}{2m^*} \frac{\partial^2}{\partial z^2} + V(z) \quad (2.212)$$

which can be written:

$$\mathcal{H} = \frac{p^2}{2m^*} + V(z) \quad (2.213)$$

becomes [50–53]:

$$\mathcal{H} = \frac{1}{2m^*} (\mathcal{P} + e\mathbf{A})^2 \mp \frac{1}{2} g^*(z) \mu_B B + V(z) \quad (2.214)$$

where the kinetic energy operator becomes modified by the magnetic field vector potential \mathbf{A} , and the second term produces a splitting, known as the ‘gyromagnetic spin splitting’ between the spin-up (– sign) and spin-down (+ sign) electrons. g^* is known as the ‘Landau factor’ and is really a function of z as it depends on the material, however it is generally assumed to be constant and approximately 2 for conduction band electrons, μ_B is the Bohr magneton and B is the magnitude of the magnetic flux density which is assumed aligned along the growth (z -) axis.

Although the heterostructure potential $V(z)$ remains one-dimensional, the vector potential means that the wave functions are not necessarily one-dimensional so the Schrödinger equation must be written:

$$\left[\frac{1}{2m^*} (\mathcal{P} + e\mathbf{A})^2 \mp \frac{1}{2} g^*(z) \mu_B B + V(z) \right] \psi(x, y, z) = E \psi(x, y, z) \quad (2.215)$$

The magnetic field produces a parabolic potential along one of the in-plane axes, the x -axis say, leaving the particle free to move (with a wave vector k_y , say) along the other axis. The standard approach is to employ the Landau gauge $\mathbf{A} = Bx\mathbf{e}_y$, then following the notation of Savić *et al.* [54] the wave function can be written in the separable form $\psi(x, y, z) = \psi_x(x)\psi_y(y)\psi_z(z)$, i.e.

$$\psi(x, y, z) = \frac{1}{\sqrt{L_y}} \psi_j(x - X_{k_y}) e^{ik_y y} \psi(z) \quad (2.216)$$

where L_y is a normalisation constant (the length of the structure along the y -axis, $X_{k_y} = -k_y l_B^2$, where $l_B = \sqrt{\hbar/eB}$ is the Landau length, j is an index over the $\psi_j(x - X_{k_y})$ harmonic oscillator solutions of the parabolic potential and $\psi(z)$ is the usual one-dimensional envelope function of the heterostructure potential without the magnetic field. The harmonic oscillator solutions will be generated numerically in Section 3.5, however they can also be expressed analytically as:

$$\psi_j(x - X_{k_y}) = \frac{1}{\pi^{1/4} \sqrt{2^j j! l_B}} \exp\left(-\frac{(x - X_{k_y})^2}{2l_B^2}\right) H_j\left(\frac{x - X_{k_y}}{l_B}\right) \quad (2.217)$$

where H_j is the j th Hermite polynomial, see Liboff [55] page 200 for a detailed description of the analytical solutions of the harmonic oscillator.

Taking g^* as a constant equal to 2 and the Bohr magneton as $9.274 \times 10^{-24} \text{ JT}^{-1}$ then it can be seen that even in a relatively high magnetic field of 10 T, the difference in energy between the spin-up and spin-down electrons is:

$$g^* \mu_B B \approx 2 \times 9.274 \times 10^{-24} \text{ JT}^{-1} \times 10 \text{ T} \approx 1 \text{ meV} \quad (2.218)$$

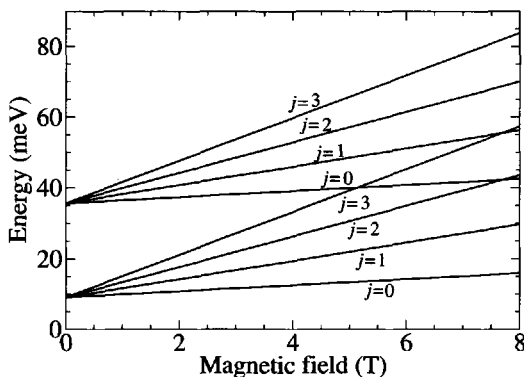


Figure 2.40 The magnetic field induced Landau levels of a 200 Å GaAs quantum well surrounded by $\text{Ga}_{1-x}\text{Al}_x\text{As}$ barriers of height 100 meV (same structure as Fig. 2.15), with constant effective mass of $0.067m_0$. Note only the first 4 Landau levels associated with each quantum well subband are shown for clarity.

which on the scales of typical quantum well systems and typical carrier densities is relatively small and can generally be ignored. Thus the total energy therefore can be considered to be composed of just two components: the usual energy eigenvalue E_n associated with the n th state of the quantum well without a magnetic field and the harmonic oscillator type energy associated with the in-plane cyclotron motion $(j + \frac{1}{2})\hbar\omega_c$, i.e.

$$E_{n,j} = E_n + \left(j + \frac{1}{2}\right) \hbar\omega_c \quad (2.219)$$

where the 'cyclotron frequency' is the usual one from bulk semiconductors [1,2,56]:

$$\omega_c = \frac{eB}{m^*} \quad (2.220)$$

Fig. 2.40 shows the Landau level splitting induced by the application of an external magnetic field on a 200 Å GaAs quantum well surrounded by $\text{Ga}_{1-x}\text{Al}_x\text{As}$ barriers, assuming a constant electron effective mass of $0.067m_0$. The figure illustrates the lowest four Landau levels for each of the two lowest energy confined states within the quantum well. A more accurate calculation could include the variation in effective mass of the carrier between the various layers; this would be done by calculating the weighted mean or expectation value of the effective mass according to the probability of finding the carrier in each of the semiconductor layers.

2.18 IN SUMMARY

The consideration of simple layered semiconductor heterostructures with analytical forms for the solutions to Schrödinger's equation has allowed exploration and discovery of the properties of two-dimensional systems. Such models are invaluable and allow a whole range of physical observables of experimental layer structures and electronic devices to be explained. However, implementing these methods computationally *can* be tedious, in that a computer program, written to calculate the energy levels of a single quantum well, has to be rewritten for a double quantum well. In addition, a different program would be needed to solve a triangular well, and furthermore some potential profiles, which can be fabricated, such as diffused quantum wells, have no analytical solution at all. While such continuously varying structures can be expressed with flat step potentials for each monolayer [57], the treatment of an electric field in this manner is questionable.

In the next chapter, a simple, but very general *numerical* solution to Schrödinger's equation will be derived, which will overcome all of these difficulties.

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CHAPTER 3

NUMERICAL SOLUTIONS

3.1 SHOOTING METHOD

As a starting point, consider a general, but simple (constant-mass) form for the time-independent Schrödinger equation, the analytical solutions of which have been extensively studied in the previous chapter:

$$-\frac{\hbar^2}{2m^*} \frac{\partial^2}{\partial z^2} \psi(z) + V(z)\psi(z) = E\psi(z) \quad (3.1)$$

where the one-dimensional potential $V(z)$ will remain undefined and again $\psi(z)$ is the wave function representing the particle of interest *while under the effective mass and envelope function approximations*. This, of course, can be written as follows:

$$-\frac{\hbar^2}{2m^*} \frac{\partial^2}{\partial z^2} \psi(z) + [V(z) - E] \psi(z) = 0 \quad (3.2)$$

The problem now is to find a numerical method for the solution of both the energy eigenvalues E and the eigenfunctions $\psi(z)$ for *any* $V(z)$.

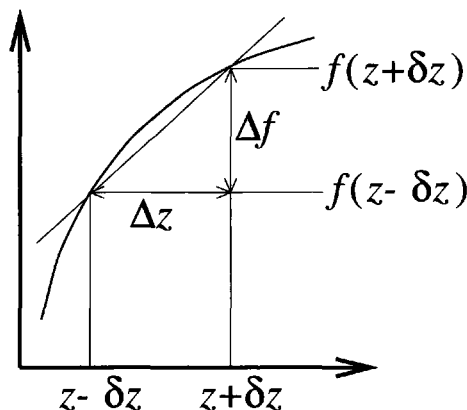


Figure 3.1 The first derivative of a function

With this aim, consider expanding the second-order derivative in terms of finite differences. For example, as in Fig. 3.1, the first derivative of any function is defined as:

$$\lim_{\Delta z \rightarrow 0} \frac{\Delta f}{\Delta z} = \frac{df}{dz} \quad (3.3)$$

It suits the purpose here to retain the approximate form, i.e.

$$\frac{df}{dz} \approx \frac{\Delta f}{\Delta z} = \frac{f(z + \delta z) - f(z - \delta z)}{2\delta z} \quad (3.4)$$

Hence, the second derivative follows as:

$$\frac{d^2 f}{dz^2} \approx \frac{\left. \frac{df}{dz} \right|_{z+\delta z} - \left. \frac{df}{dz} \right|_{z-\delta z}}{2\delta z} \quad (3.5)$$

By using the finite difference forms in equation (3.4) for the first derivatives, then:

$$\frac{d^2 f}{dz^2} \approx \frac{\left[\frac{f(z+2\delta z) - f(z)}{2\delta z} \right] - \left[\frac{f(z) - f(z-2\delta z)}{2\delta z} \right]}{2\delta z} \quad (3.6)$$

$$\therefore \frac{d^2 f}{dz^2} \approx \frac{f(z+2\delta z) - 2f(z) + f(z-2\delta z)}{(2\delta z)^2} \quad (3.7)$$

As δz is an, as yet, undefined small step length along the z -axis, and as it only appears in equation (3.7) with the factor 2, then this finite difference representation of the second derivative can be simplified slightly by substituting δz for $2\delta z$, i.e.

$$\frac{d^2 f}{dz^2} \approx \frac{f(z + \delta z) - 2f(z) + f(z - \delta z)}{(\delta z)^2} \quad (3.8)$$

Using this form for the second derivative in the original Schrödinger equation and taking the step length δz as sufficiently small that the approximation is good, i.e. drop the ' \approx ' in favour of '=', then:

$$-\frac{\hbar^2}{2m^*} \left[\frac{\psi(z + \delta z) - 2\psi(z) + \psi(z - \delta z)}{(\delta z)^2} \right] + [V(z) - E] \psi(z) = 0 \quad (3.9)$$

$$\therefore \psi(z + \delta z) - 2\psi(z) + \psi(z - \delta z) = \frac{2m^*}{\hbar^2} (\delta z)^2 [V(z) - E] \psi(z) \quad (3.10)$$

which can finally be written as:

$$\psi(z + \delta z) = \left[\frac{2m^*}{\hbar^2} (\delta z)^2 (V(z) - E) + 2 \right] \psi(z) - \psi(z - \delta z) \quad (3.11)$$

Equation (3.11) implies that if the wave function is known at the two points $(z - \delta z)$ and z , then the value of the wave function at $(z + \delta z)$ can be calculated for any energy E . This iterative equation forms the basis of a standard method of solving differential equations numerically, and is known as the *shooting method* [58].

Using two known values of the wave function $\psi(z - \delta z)$ and $\psi(z)$, a third value, i.e. $\psi(z + \delta z)$, can be predicted. Using this new point $\psi(z + \delta z)$, together with $\psi(z)$ and by making the transformation $z + \delta z \rightarrow z$, a fourth point, $\psi(z + 2\delta z)$, can be calculated, and so on. Hence the complete wave function can be deduced for any particular energy. The solutions for stationary states have wave functions which satisfy the standard boundary conditions, i.e.

$$\psi(z) \rightarrow 0 \quad \text{and} \quad \frac{\partial}{\partial z} \psi(z) \rightarrow 0, \quad \text{as} \quad z \rightarrow \pm\infty \quad (3.12)$$

The first two values of the wave function necessary to start off the procedure can be deduced by using simple symmetry arguments, namely that if the potential $V(z)$ is symmetric, then the eigenstates must be either symmetric (even parity) or anti-symmetric (odd parity). If the state of interest has odd parity, e.g. the first excited state of a symmetric quantum well, as illustrated in Fig. 3.2, then the wave function at the centre of the well (call this the origin $z = 0$ for now) must be zero. Correspondingly, a small displacement along the growth (z -) direction must yield a finite value for the wave function. The actual magnitude is not relevant since the energy eigenvalues of the linear Schrödinger equation are unchanged if the wave function is scaled by any given number. Hence in this case, the following starting conditions could be chosen:

$$\psi(0) = 0; \quad \psi(\delta z) = 1 \quad (3.13)$$

Given this, it remains to find the eigenvalue energy E . As stated above, the value of E corresponding to a stationary state, or more specifically in this case a confined state within a well potential, is that value which produces a wave function conforming to the standard boundary conditions. As E is an unknown in equation (3.11), then ψ is really a function of both position z and energy E , since given the starting conditions

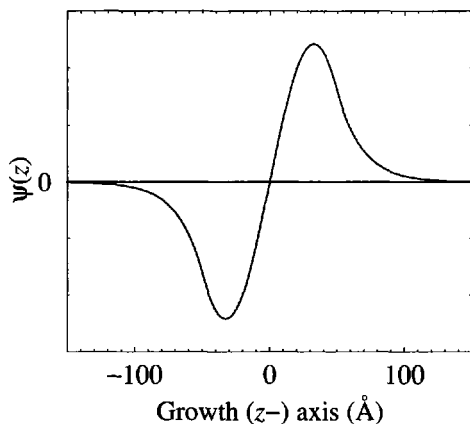


Figure 3.2 An odd-parity wave function—the first excited state of a symmetric quantum well

a wave function can be generated for any E —although it will not always tend to zero at infinity and be a stationary state. Thus the wave function should be written as $\psi(z, E)$, in which case solutions are sought to the equation:

$$\psi(\infty, E) = 0 \quad (3.14)$$

which can be found by using standard techniques such as the Newton–Raphson iteration (see, for example, Section 2.5).

If the eigenstate of interest has even parity, e.g. the ground state of a symmetric quantum well, as shown in Fig. 3.3, then new starting conditions must be deduced. In particular, as the value of the wave function at the origin is non-zero, then $\psi(0) = 1$ can be chosen. In addition, as the wave function is symmetric then $\psi(-\delta z) = \psi(+\delta z)$; substituting both of these expressions into equation (3.11) then gives:

$$\psi(+\delta z) = \left\{ \frac{2m^*}{\hbar^2} (\delta z)^2 [V(0) - E] + 2 \right\} \times 1 - \psi(+\delta z) \quad (3.15)$$

$$\therefore \psi(+\delta z) = \frac{1}{2} \left\{ \frac{2m^*}{\hbar^2} (\delta z)^2 [V(0) - E] + 2 \right\} \quad (3.16)$$

3.2 GENERALISED INITIAL CONDITIONS

While such initial (starting) conditions for symmetric potentials are very useful, they are restrictive in that there are many systems which do not have symmetric potentials, with perhaps the most obvious being the quantum well with an electric field applied

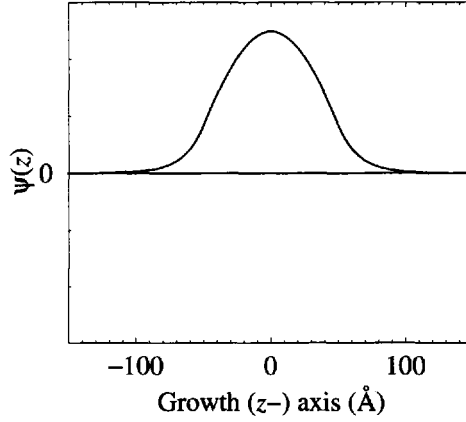


Figure 3.3 An even-parity wave function—the first excited state of a symmetric quantum well

(see Section 2.12). With the aim of deducing more general starting conditions for the iterative equation (equation 3.11): Note that at the moment the potentials of interest are all confining potentials, and hence as stated above many times, all wave functions satisfy the standard boundary conditions:

$$\psi(z) \rightarrow 0 \quad \text{and} \quad \frac{\partial}{\partial z} \psi(z) \rightarrow 0, \quad \text{as } z \rightarrow \pm\infty \quad (3.17)$$

In addition, learning from the analytical solutions of the previous chapter, the wave function decays exponentially into the end barriers (see Fig. 3.4). The decay constant κ is also known as follows:

$$\kappa = \frac{\sqrt{2m^*(V - E)}}{\hbar} \quad (3.18)$$

Hence: upon choosing the first value of the wave function $\psi(z - \delta z) = 1$, then

$$\psi(z - \delta z) = \psi(z) \exp(-\kappa|\delta z|) \quad (3.19)$$

Therefore, given the starting value $\psi(z - \delta z)$, the next value $\psi(z)$ can be calculated, thus implementing the boundary condition of exponential growth.

In practice, however, these boundary conditions *can* be unreliable, not because of the starting conditions they impose, but because of the boundary conditions that must be sought at the other end of the potential structure. The mathematics imply that energies must be sought for which ψ tends to zero; however, the original choice of $\psi = 1$ at the start of the potential structure immediately implies an asymmetry. Generally, this asymmetry is very small, but however, not always. In order to correct

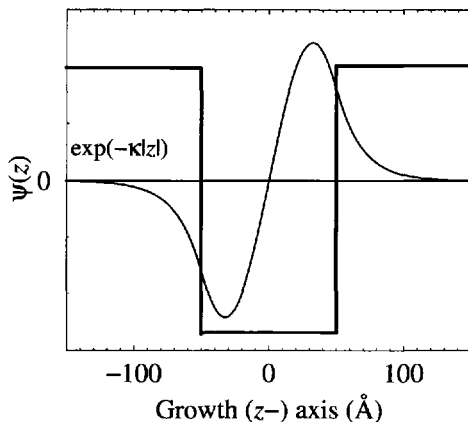


Figure 3.4 Exponential decay of the wave function into the end barrier

this asymmetry, the, at first sight bizarre, starting conditions are chosen as:

$$\psi(z - \delta z) = 0 \quad \text{and} \quad \psi(z) = 1 \quad (3.20)$$

Such starting conditions for the shooting method solution of the Schrödinger equation are important because of their generality. They are applicable to all potential profiles, whether symmetric, or not, whether the outer barrier is flat, or not, and whether the eigenstate of interest is symmetric, anti-symmetric, or without definite parity.

These conditions can be partially justified mathematically. As multiplying an eigenstate (wave function ψ) by a constant doesn't affect the eigenvalue (the energy E), then if the first wave function point, $\psi(z - \delta z)$, was taken as a small, but finite value, say $\delta\psi$, then the second starting point, $\psi(z)$, could be given any value larger than $\delta\psi$, say $N\delta\psi$, where N is a large number, while still giving exponential growth and without changing the energy eigenvalue. The third value of the wave function simply follows from the shooting equation (equation 3.11) as follows:

$$\psi(z + \delta z) = \left\{ \frac{2m^*}{\hbar^2} (\delta z)^2 [V(z) - E] + 2 \right\} N\delta\psi - \delta\psi \quad (3.21)$$

$$\therefore \psi(z + \delta z) = \left(\left\{ \frac{2m^*}{\hbar^2} (\delta z)^2 [V(z) - E] + 2 \right\} N - 1 \right) \delta\psi \quad (3.22)$$

These new starting conditions of 0 and 1 merely represent the limit of large N .

While the above only offers partial justification, full vindication will be given by the results of the convergence tests in the following sections.

3.3 PRACTICAL IMPLEMENTATION OF THE SHOOTING METHOD

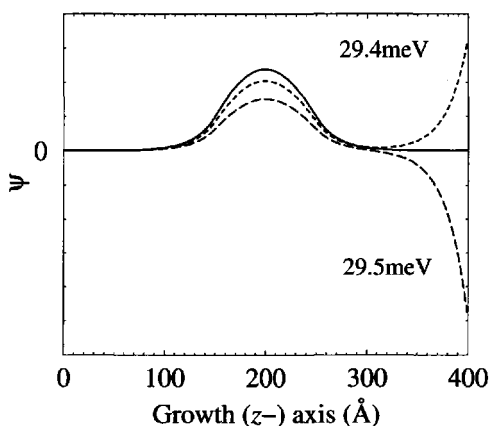


Figure 3.5 Numerically obtained wave functions above and below the true solution at $E = 29.43$ meV, for an electron in a 150 \AA $\text{Ga}_{0.8}\text{As}_{0.2}\text{As}$ / 100 \AA GaAs / 150 \AA $\text{Ga}_{0.8}\text{As}_{0.2}\text{As}$ single quantum well

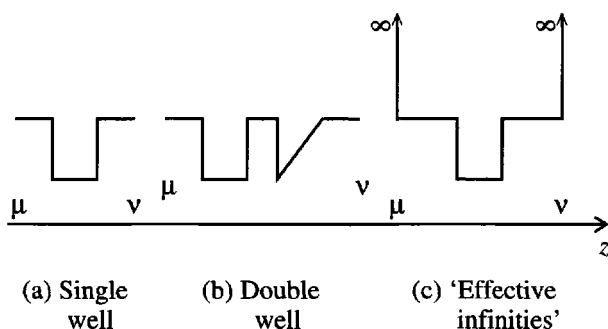


Figure 3.6 Schematic illustration of the extent of the potential $V(z)$, where $\mu < z < \nu$

Fig. 3.5 illustrates the practical implementation of the shooting method. The energy is varied systematically until the wave function switches from diverging to $+\infty$ to $-\infty$; clearly an energy exists between these values for which the wave function will tend smoothly to zero. A Newton–Raphson iteration is then begun to converge on the true solution, in this case 29.43 meV.

In practice, the wave function iteration is begun a finite distance to the left of the first well in the potential and is halted a finite distance to the right of the last well

(see Fig. 3.6). The potential, which can be merely a list of numbers specifying $V(z)$ at regular intervals along the z -axis, defines the step length, the extent of the wave functions, and the effective infinities. The latter are the points by which the wave function is considered to have converged towards zero at both limits of the potential profile. Seeking solutions based on this criterion is, in effect, equivalent to applying infinite potentials a sufficient distance into the outer barriers of our finite structure, as shown in Fig. 3.6(c). The positions of these effective infinities, ' μ ' and ' ν ', which are the lower and upper limits of the z -domain, should be chosen to be of sufficient extent so as not to affect the eigenvalues, defining this mathematically:

$$\lim_{\mu \rightarrow -\infty} E = E_n, \quad \text{and} \quad \lim_{\nu \rightarrow +\infty} E = E_n \quad (3.23)$$

where E_n are the set of true eigenenergies.

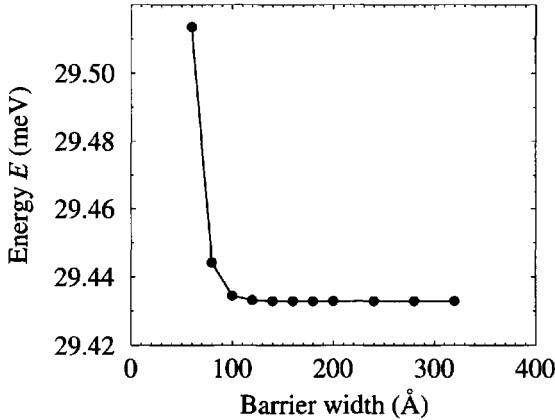


Figure 3.7 Effect of barrier width (effective infinities, μ and ν) on energy

In Fig. 3.7 the left- and right-hand barriers are varied in thickness to illustrate the effect on the solution for the energy produced by this numerical method, for the single quantum well of above. Clearly, the energy converges to a constant, which is the true stationary state—in this case, the ground state E_1 . Thus if the energy is the only motivation for the calculation, any barrier width beyond, say 150 Å, will suffice. Although note this can vary depending on the barrier height, well width and carrier effective mass.

If the wave function is also desired, e.g. to be used as an input for calculating another property, such as the exciton binding energy or an electron–phonon scattering rate, then greater care has to be taken. While very large outer barrier layers can be chosen to ensure convergence of the eigenenergy, this can have a detrimental effect on the wave function, as can be seen in Fig. 3.8. At too narrow a barrier width,

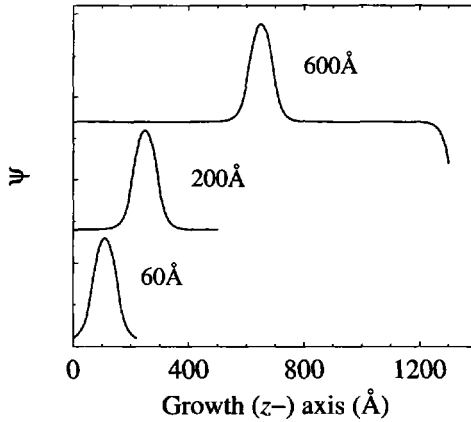


Figure 3.8 Effect of barrier width (effective infinities, μ and ν) on the solution to the wave function

e.g. the 60 Å case, although the energy may be returned to within 1 meV, the wave function doesn't satisfy the second of the standard boundary conditions, i.e. $\psi' = 0$. With a 200 Å barrier, which corresponds to just after the energy has converged, the wave function is as expected and satisfies both the standard boundary conditions—this represents the optimal barrier width. However at very large barrier widths, such as 600 Å as illustrated, the wave function is beginning to diverge and as stated above, although the energy obtained from such a solution is reliable, the wave function cannot be used to calculate other properties.

The energy E can always be obtained to far higher accuracy than would be required for comparison of say, electron confinement energies with photoluminescence data. Computer codes typically assign 16-figure accuracy to a double precision number, and the shooting method can be used to determine E to 12 of these, for example. However, even at this high accuracy the wave function at the end of an iteration, i.e., the value of $\psi(\infty, E) = \psi(\nu, E)$, cannot be made equal to zero for large barrier widths, simply because E cannot be stored to enough significant figures. If 32-figure accuracy could be used in the computing, then the wave function could be returned realistically for larger barrier widths.

The wave functions obtained from this numerical method are not normalised, i.e. they do not satisfy:

$$\int_{\text{all space}} \psi^*(z)\psi(z) \, dz = 1 \quad (3.24)$$

This can easily be achieved with the following transformation:

$$\psi(z) \rightarrow \frac{\psi(z)}{\sqrt{\int_{\text{all space}} \psi^*(z)\psi(z) dz}} \quad (3.25)$$

This numerical solution to a still simplistic Schrödinger equation is of use as it enables comparison with a number of curved potentials that have analytical solutions, such as the parabolic and the Pöschl–Teller potentials.

3.4 HETEROJUNCTION BOUNDARY CONDITIONS

In the previous chapter it was shown that the conditions for matching solutions at interfaces between dissimilar materials were fixed as soon as the Hamiltonian was decided. This is also reflected here within the shooting method. Consider again the shooting equation for the constant-mass Hamiltonian ($\frac{1}{m^*}\mathcal{P}_z\mathcal{P}_z$) (equation (3.11)):

$$\psi(z + \delta z) = \left\{ \frac{2m^*}{\hbar^2}(\delta z)^2 [V(z) - E] + 2 \right\} \psi(z) - \psi(z - \delta z) \quad (3.26)$$

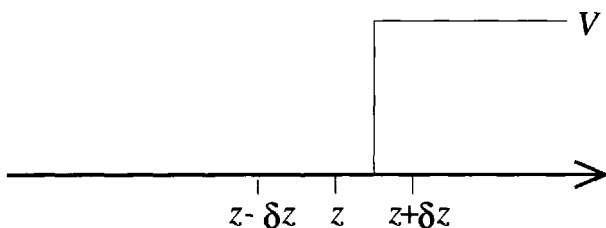


Figure 3.9 Calculating a new wave function point across a discontinuity in the potential

Consider also that the two known points of the wave function, i.e. $\psi(z - \delta z)$ and $\psi(z)$, lie within a quantum well with $V(z) = 0$, but the new point to be calculated lies across a heterojunction in a barrier (as in Fig. 3.9) then:

$$\psi(z + \delta z) = \left[\frac{2m^*}{\hbar^2}(\delta z)^2 (-E) + 2 \right] \psi(z) - \psi(z - \delta z) \quad (3.27)$$

Then in the limit of decreasing step length δz , i.e. eliminating terms in $(\delta z)^2$:

$$\psi(z + \delta z) = 2\psi(z) - \psi(z - \delta z) \quad (3.28)$$

$$\therefore \psi(z + \delta z) - \psi(z) = \psi(z) - \psi(z - \delta z) \quad (3.29)$$

which implies continuity of the *derivative* across a heterojunction, as expected for this constant effective mass Hamiltonian.

3.5 THE PARABOLIC POTENTIAL WELL

The parabolic potential well is a good testing ground for the accuracy of the numerical method introduced, in that it allows progression from the standard step potentials that have been dealt with over and over again, to a curved profile, but still with analytical solutions to compare with.

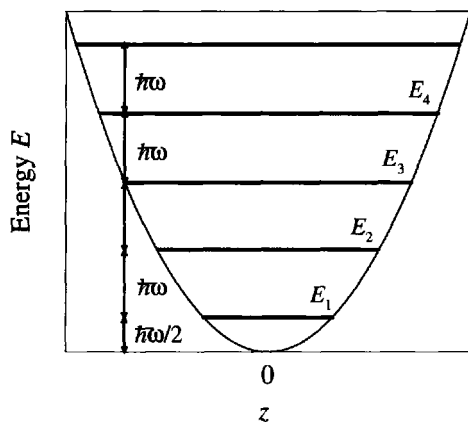


Figure 3.10 The quantum mechanical harmonic oscillator

The parabolic potential well is a direct analogy to the harmonic oscillator of classical mechanics in that the potential is, of course, proportional to the square of the displacement, i.e. following Eisberg [3]:

$$V(z) = C \frac{z^2}{2} \quad (3.30)$$

where C is a constant. If a particle of mass m is displaced from the equilibrium position by an amount Δz , then the restoring force is proportional to $-\partial V/\partial z = -Cz$, thus causing oscillations of amplitude Δz and angular frequency:

$$\omega = \sqrt{\frac{C}{m}} \quad (3.31)$$

The behaviour of a quantum particle, such as an electron, within such a potential is quite different. The Schrödinger equation for the constant-mass case would then become:

$$-\frac{\hbar^2}{2m} \frac{\partial^2}{\partial z^2} \psi + C \frac{z^2}{2} \psi = E \psi \quad (3.32)$$

Substituting ω for C , then:

$$-\frac{\hbar^2}{2m^*} \frac{\partial^2}{\partial z^2} \psi + m^* \omega^2 \frac{z^2}{2} \psi = E \psi \quad (3.33)$$

Table 3.1 Convergence test of the shooting method solution for a parabolic quantum well; the energy eigenvalues E_n for the lowest 10 confined states—note that the value for E_1 implies $\hbar\omega = 871.876$ meV

n	E_n (meV)	$E_n/(2E_1)$
1	435.938	0.500
2	1307.809	1.500
3	2179.671	2.500
4	3051.522	3.500
5	3923.341	4.500
6	4794.990	5.500
7	5665.755	6.498
8	6532.518	7.492
9	7382.747	8.468
10	8156.309	9.355

The method of solution of this eigenvalue problem has been covered many times in standard texts, for example [3, 45] or alternatively [12]. It is suffice here to quote the result, i.e. the energy levels are quantised and given by

$$E_{n+1} = \left(n + \frac{1}{2} \right) \hbar\omega \quad (3.34)$$

where the subscript $(n+1)$ has been introduced in order for the notation to agree with that developed so far, i.e. the ground state being labelled '1'.

Figure 3.10 illustrates the solution; the ground state has an energy $\hbar\omega/2$ even at the absolute zero of temperature, thus coining the term the *zero point energy*. Above this the equal-spaced energy steps form a ladder, which has prompted suggestions for exploitation in non-linear optics [59].

Focusing on parabolic quantum wells within semiconductor multilayers, then it has to be acknowledged, that as only finite potentials are available through the standard band offsets, the potential profile will resemble that shown in Fig. 3.11, i.e. a *finite parabolic quantum well*, rather than that without limit as shown in Fig. 3.10.

Taking the central well to be of width a and the outer barriers to have widths b , the parabolic quantum well can then be specified by the variation in the alloy component x along the z -axis, say in $\text{Ga}_{1-x}\text{Al}_x\text{As}$. In particular, if x is allowed to vary from a minimum value x_{\min} to a maximum x_{\max} , then the well profile is given by the

following quadratic relationship:

$$x(z) = x_{\min} + \frac{[z - (b - a/2)]^2 (x_{\max} - x_{\min})}{(a/2)^2} \quad (3.35)$$

Table 3.1 displays the energies of the lowest ten levels within a $\text{Ga}_{1-x}\text{Al}_x\text{As}$ parabolic quantum well. The layer thicknesses have been taken as $a = b = 100$ Å, and the effective mass has been taken as a constant, i.e. $0.067 m_0$; however, the maximum alloy concentration, x_{\max} , has been taken artificially as high as '10', in order to produce a large number of confined levels. As $E_1 = \hbar\omega/2$, then the third column of data displays the 'half-integer' $(n + \frac{1}{2})$. Clearly, the agreement with the analytical theory is exact for the lower states, although as the levels approach the top of the barrier, they experience the finiteness of the potential, and hence some discrepancy with the infinite parabola solution arises.

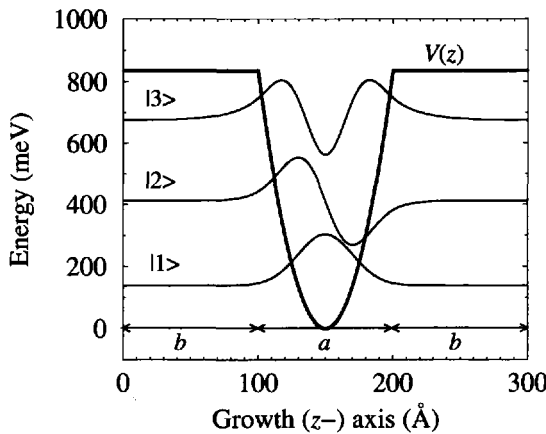


Figure 3.11 The three confined eigenstates of a finite parabolic quantum well

The energy step $\hbar\omega$ between the levels is a useful quantity for describing the properties of the well, even though ω doesn't represent the angular frequency of oscillation. With this in mind, consider the conversion of the alloy concentration of equation (3.35) into a potential profile, (see Appendix A); the conduction band profile of $\text{Ga}_{1-x}\text{Al}_x\text{As}$ would then become:

$$V(z) = \Delta V_{\text{CB}} x(z) (E_g^{\text{AlAs}} - E_g^{\text{GaAs}}) \quad (3.36)$$

Recalling that $\partial V/\partial z = Cz$, then:

$$C = \frac{1}{z} \frac{\partial V}{\partial z} = \frac{1}{z} \Delta V_{\text{CB}} (E_g^{\text{AlAs}} - E_g^{\text{GaAs}}) \frac{\partial}{\partial z} x \quad (3.37)$$

Ignoring the origin shift, $b = a/2$:

$$C = \frac{1}{z} \Delta V_{CB} (E_g^{\text{AlAs}} - E_g^{\text{GaAs}}) \frac{8z(x_{\max} - x_{\min})}{a^2} \quad (3.38)$$

$$\therefore C = \Delta V_{CB} (E_g^{\text{AlAs}} - E_g^{\text{GaAs}}) \frac{8(x_{\max} - x_{\min})}{a^2} \quad (3.39)$$

Then, ω follows from equation (3.31) as:

$$\omega = \sqrt{8\Delta V_{CB} (E_g^{\text{AlAs}} - E_g^{\text{GaAs}}) (x_{\max} - x_{\min})} \frac{1}{a\sqrt{m}} \quad (3.40)$$

By using the values corresponding to the data in Table 3.1, i.e. $V_{CB} = 0.67$, $E_g^{\text{AlAs}} - E_g^{\text{GaAs}} = 1247$ meV, $x_{\max} = 10$, $x_{\min} = 0$, $a = 100$ Å and $m = 0.067 m_0$ then $\hbar\omega = 871.879$ meV, which is in excellent agreement with the numerical result as displayed in the table.

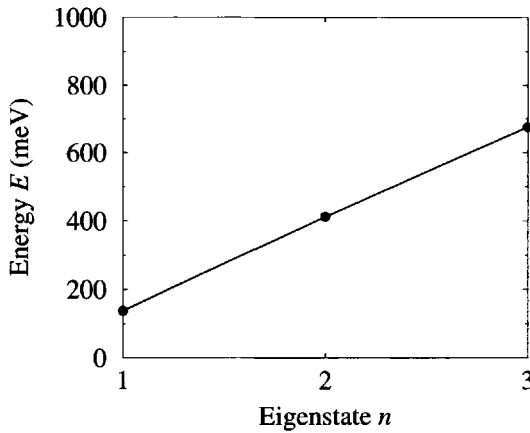


Figure 3.12 Eigenenergies for a GaAs/AlAs parabolic quantum well

Figures 3.11 and 3.12 display the wave functions and energies, respectively, of the three confined states of the more realistic parabolic quantum well, with $x_{\max}=1$. The wave functions resemble those of the single quantum well and the energy levels are to all intents and purposes equally spaced.

In summary, it has been demonstrated that the simple shooting-method solution of the Schrödinger equation, taken together with the stated starting conditions, can produce energy levels in exact agreement with the analytical values for the curved potential of the parabolic quantum well. In addition, equation (3.40), summarizes the behaviour of parabolic wells within semiconductor multilayers. The energy level spacing is inversely proportional to both the well width and the square root of the mass.

3.6 THE PÖSCHL–TELLER POTENTIAL HOLE

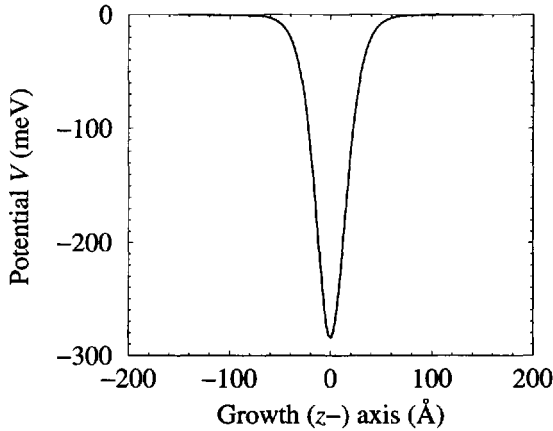


Figure 3.13 Pöschl–Teller potential hole; $\alpha = 0.05$ and $\lambda = 2.0$

The modified Pöschl–Teller potential hole [60], an example of which is shown in Fig. 3.13, is important in that it resembles the profile of a diffused quantum well, but has the advantage of analytical solutions. Therefore as is currently the aim, it will serve well for validation of the shooting method *and* the choice of starting conditions described in Section 3.1.

The potential is given by the following [61]:

$$V(z) = -\frac{\hbar^2}{2m^*} \alpha^2 \frac{\lambda(\lambda-1)}{\cosh^2 \alpha z} \quad (3.41)$$

where α is known as the width parameter and λ as the depth parameter (see Fig. 3.14). Clearly, values of $0 < \lambda < 1$ give potential barriers, $\lambda = 1$ gives a flat band and the area of interest here, namely potential wells, are given by values of $\lambda > 1$.

The eigenvalues of the resulting Schrödinger equation are given by [60]:

$$E_n = -\frac{\hbar^2 \alpha^2}{2m^*} (\lambda - 1 - n)^2 \quad (3.42)$$

Table 3.2 compares the analytical solutions given by equation (3.42) for the two lowest energy confined states of the Pöschl–Teller potential holes of Fig. 3.14, with that obtained by the numerical shooting method. The agreement is very good for all values of λ and for both the ground state, of energy E_1 , and the first excited state, of energy E_2 . However, for the largest value of λ used, which does represent a very deep well, of a magnitude not found in semiconductor heterostructures, the discrepancy

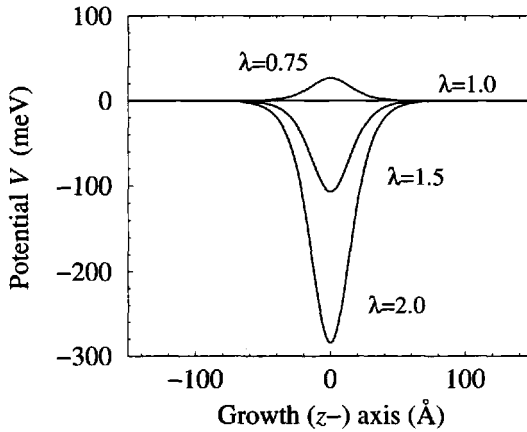


Figure 3.14 Pöschl–Teller potentials as a function of the depth parameter λ for a fixed width parameter α of 0.05

Table 3.2 Comparison of the numerical solution with the analytical solution of the Pöschl–Teller potential with a step length δz of 1 Å

λ	<u>Analytical solution</u>		<u>Numerical solution</u>	
	E_1 (meV)	E_2 (meV)	E_1 (meV)	E_2 (meV)
1.5	-35.541154	—	-35.442724	—
2.0	-142.164614	—	-142.178093	—
5.0	-2274.633827	-1279.481528	-2274.940451	-1280.518108
10.0	-11515.333749	-9098.535308	-11517.014128	-9105.712331

between the solutions has increased to 2 and 7 meV for the ground and first excited states respectively. This discrepancy can be removed by increasing the accuracy of the numerical solution.

3.7 CONVERGENCE TESTS

This is an appropriate point to perform the final convergence test, i.e. the final test of numerical accuracy of the shooting method solution of Schrödinger's equation, as derived in Section 3.1. As discovered in the previous section, while the agreement between the analytical solution of the Pöschl–Teller potential and the numerical solution is generally very good, at times, for certain parameter values, discrepancies can

occur. This is often a reflection of the choice of the computational values, for example, and probably most importantly, the resolution in the definition of the potential profile, i.e. the step length δz . The user has complete control over this value, and hence an improved comparison with analytical solutions (should they exist) can be achieved, but as always, at the inconvenience of increased computational time.

Table 3.3 Comparison of the numerical solution with the analytical solution of the Pöschl–Teller potential as a function of the number (N) of points per Å

N (Å ⁻¹)	Analytical solution		Numerical solution	
	E_1 (meV)	E_2 (meV)	E_1 (meV)	E_2 (meV)
1	-2274.633827	-1279.481528	-2274.940451	-1280.518108
2	-2274.633827	-1279.481528	-2274.710431	-1279.740343
5	-2274.633827	-1279.481528	-2274.646081	-1279.522923
10	-2274.633827	-1279.481528	-2274.636891	-1279.491876
20	-2274.633827	-1279.481528	-2274.634593	-1279.484115
50	-2274.633827	-1279.481528	-2274.633950	-1279.481942
100	-2274.633827	-1279.481528	-2274.633858	-1279.481631

Table 3.3 compares the analytical solutions corresponding to the $\lambda = 5$ potential hole of Table 3.2 with the corresponding numerical solution as a function of the number of potential points $V(z)$ per Å along the growth axis. The step length δz in Å is just $1/N$. It can be seen that as the number of points increases the discrepancy between the data decreases (obviously the analytical solutions are not a function of N). At the upper limit of the data in the table, the solutions agree to 7 or 8 significant figures. This is well beyond the accuracy of spectroscopic data, which in the highest quality of semiconductor heterostructure samples, might yield transition energies to 0.1 meV, e.g. [62], a precision which is reached for $N = 1$.

In summary, for the majority of applications, a step length δz of 1 Å is adequate.

3.8 EXTENSION TO VARIABLE EFFECTIVE MASS

Hitherto, the numerical solution has focused on the constant-mass Schrödinger equation, and this has served the purposes of development and application well. However in the real world, just as in the analytical solutions before, the Schrödinger equation of real interest accounts for the possibility of the effective mass varying between dissimilar semiconductor layers.

With the aim of generalising the numerical solution for this situation, consider the Schrödinger equation (equation (2.96)), i.e.

$$-\frac{\hbar^2}{2} \frac{\partial}{\partial z} \frac{1}{m^*(z)} \frac{\partial}{\partial z} \psi(z) + V(z)\psi(z) = E\psi(z) \quad (3.43)$$

which can be written:

$$\frac{\partial}{\partial z} \frac{1}{m^*(z)} \frac{\partial}{\partial z} \psi(z) = \frac{2}{\hbar^2} [V(z) - E] \psi(z) \quad (3.44)$$

The variable mass kinetic energy operator can be expanded directly to give:

$$\frac{\partial}{\partial z} \left[\frac{1}{m^*(z)} \right] \frac{\partial}{\partial z} \psi(z) + \frac{1}{m^*(z)} \frac{\partial^2}{\partial z^2} \psi(z) = \frac{2}{\hbar^2} [V(z) - E] \psi(z) \quad (3.45)$$

and then:

$$-\frac{1}{[m^*(z)]^2} \frac{\partial}{\partial z} m^*(z) \frac{\partial}{\partial z} \psi(z) + \frac{1}{m^*(z)} \frac{\partial^2}{\partial z^2} \psi(z) = \frac{2}{\hbar^2} [V(z) - E] \psi(z) \quad (3.46)$$

However, shooting equations derived from this point by expanding the derivatives in terms of finite differences have lead to significant computational inaccuracies in systems with a large discontinuous change in the effective mass $m^*(z)$, as occurs, e.g. in GaAs/AlAs quantum wells. The source of the inaccuracy is thought to arise from the δ -function nature of $\partial m^*(z)/\partial z$.

A more robust scheme can be derived by expanding the left-hand derivative, $\partial/\partial z$, in equation (3.44), first giving the following:

$$\frac{\frac{1}{m^*(z+\delta z)} \frac{\partial \psi(z)}{\partial z} \Big|_{z+\delta z} - \frac{1}{m^*(z-\delta z)} \frac{\partial \psi(z)}{\partial z} \Big|_{z-\delta z}}{2\delta z} = \frac{2}{\hbar^2} [V(z) - E] \psi(z) \quad (3.47)$$

$$\therefore \frac{1}{m^*(z+\delta z)} \frac{\partial \psi}{\partial z} \Big|_{z+\delta z} - \frac{1}{m^*(z-\delta z)} \frac{\partial \psi}{\partial z} \Big|_{z-\delta z} = \frac{2(2\delta z)}{\hbar^2} [V(z) - E] \psi(z) \quad (3.48)$$

Recalling the centred finite difference expansion for the first derivative, i.e.

$$\frac{\partial f}{\partial z} \Big|_z = \frac{f(z+\delta z) - f(z-\delta z)}{2\delta z} \quad (3.49)$$

Then:

$$\begin{aligned} \frac{1}{m^*(z+\delta z)} \left[\frac{\psi(z+2\delta z) - \psi(z)}{2\delta z} \right] - \frac{1}{m^*(z-\delta z)} \left[\frac{\psi(z) - \psi(z-2\delta z)}{2\delta z} \right] \\ = \frac{2(2\delta z)}{\hbar^2} [V(z) - E] \psi(z) \end{aligned} \quad (3.50)$$

$$\begin{aligned} \therefore \left[\frac{\psi(z+2\delta z) - \psi(z)}{m^*(z+\delta z)} \right] - \left[\frac{\psi(z) - \psi(z-2\delta z)}{m^*(z-\delta z)} \right] \\ = \frac{2(2\delta z)^2}{\hbar^2} [V(z) - E] \psi(z) \end{aligned} \quad (3.51)$$

Table 3.4 Comparison of the numerical solution with the analytical solution for a single quantum well with differing effective masses in the well and barrier

Well width (Å)	Analytical solution		Numerical solution	
	E_1 (meV)	E_2 (meV)	E_1 (meV)	E_2 (meV)
20	126.227914	—	126.204335	—
40	80.111376	—	80.087722	—
60	53.276432	166.522007	53.260451	167.766634
80	37.619825	137.330295	37.609351	137.308742
100	27.884814	106.557890	27.877769	106.535858
120	21.463972	83.606781	21.459068	83.589149
160	13.820474	54.647134	13.817861	54.636614
200	9.629394	38.282383	9.627854	38.275914

By gathering terms in $\psi(z)$ on the right hand side, then:

$$\begin{aligned} & \frac{\psi(z + 2\delta z)}{m^*(z + \delta z)} + \frac{\psi(z - 2\delta z)}{m^*(z - \delta z)} \\ &= \left\{ \frac{2(2\delta z)^2}{\hbar^2} [V(z) - E] + \frac{1}{m^*(z + \delta z)} + \frac{1}{m^*(z - \delta z)} \right\} \psi(z) \end{aligned} \quad (3.52)$$

Making the transformation, $2\delta z \rightarrow \delta z$, then gives:

$$\begin{aligned} \frac{\psi(z + \delta z)}{m^*(z + \delta z/2)} &= \left\{ \frac{2(\delta z)^2}{\hbar^2} [V(z) - E] + \frac{1}{m^*(z + \delta z/2)} + \frac{1}{m^*(z - \delta z/2)} \right\} \psi(z) \\ &\quad - \frac{\psi(z - \delta z)}{m^*(z - \delta z/2)} \end{aligned} \quad (3.53)$$

which is the variable effective-mass shooting equation, and is solved according to the boundary conditions (as in Section 3.1). The effective mass m^* can be found at the intermediate points, $z \pm \delta z/2$, by taking the mean of the two neighbouring points at z and $z \pm \delta z$. Clearly, equation 3.53 collapses back to the original form in equation (3.11) when m^* is constant.

Table 3.4 compares the ground-state and first excited-state energy levels, E_1 and E_2 , respectively, calculated with this extended shooting equation, with the analytical solution from Section 2.6, for a GaAs quantum well surrounded by $\text{Ga}_{0.8}\text{Al}_{0.2}\text{As}$ barriers. In this series of calculations, the step length δz was taken as 1 Å and it can be seen from the data in the table that the agreement is very good for both the ground-state energy E_1 and the first excited-state energy E_2 across the range of well widths. The discrepancy between the solutions of the two methods is largest for the excited state of the narrower wells, at which point it is of the order of 1 meV. For

Table 3.5 Comparison of the numerical solution with the analytical solution for a single GaAs quantum well surrounded by $\text{Ga}_{0.25}\text{Al}_{0.75}\text{As}$ barriers, with differing effective masses in the well and barrier

Well width (\AA)	Analytical solution		Numerical solution	
	E_1 (meV)	E_2 (meV)	E_1 (meV)	E_2 (meV)
20	270.764969	—	270.357809	—
40	129.450671	512.393113	129.247261	512.047324
60	75.774672	307.023049	75.669905	306.736857
80	49.774624	201.176843	49.715086	200.986873
100	35.204788	141.851288	35.168092	141.724429
120	26.217761	105.411306	26.193670	105.323933
160	16.158835	64.812071	16.146909	64.766509
200	10.949827	43.869233	10.943102	43.842871

the wider wells, the discrepancy reduces to less than 0.1 meV. Such accuracies are entirely acceptable when modelling, e.g. experimental spectroscopic data.

As this shooting equation for a variable effective mass will be used widely, it is worthwhile performing a few more convergence tests in order to increase confidence in its applicability. In particular, Table 3.5 repeats the calculations of the previous table but with a much higher barrier Al concentration. The effect of this is twofold, i.e. there is an increased difference in the potential between the well and barrier, but more importantly for this present section, there is an increased difference in the effective masses. The discrepancies between the analytical solution and the numerical solution are of a similar order as before, for this step length δz of 1 \AA .

Table 3.6 Comparison of the numerical solution with the analytical solution for a 20 \AA single GaAs quantum well surrounded by $\text{Ga}_{0.25}\text{Al}_{0.75}\text{As}$ barriers, as a function of the number (N) of points per \AA in the mesh—note the step length $\delta z = 1/N$ (\AA)

N (\AA^{-1})	Analytical solution E_1 (meV)	Numerical solution E_1 (meV)
2	270.764969	270.663106
4	270.764969	270.739499
6	270.764969	270.753649
8	270.764969	270.758601
10	270.764969	270.760894
12	270.764969	270.762139

However, as before, the agreement between the numerical solution and that obtained from the analytical form can be improved by increasing the computational accuracy of the shooting method, i.e. increasing the number of points per Å (decreasing δz). This is highlighted in Table 3.6, which compares the solutions from the two methods for a decreasing step length δz .

When using the form for the variable effective mass shooting equation in equation (3.53), the BenDaniel–Duke boundary conditions are ‘hard-wired’ in. Thus it is unnecessary to repeat the analysis described in Section 3.4 in order to recover them.

3.9 THE DOUBLE QUANTUM WELL

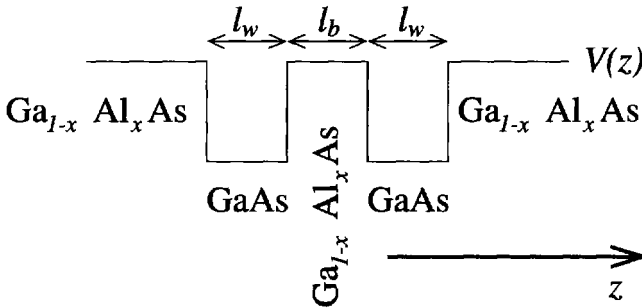


Figure 3.15 The band profile of symmetric GaAs/Ga_{1-x}Al_xAs double quantum well

The convergence tests of the numerical solution to Schrödinger's equation have provided the confidence to apply it to systems for which the analytical solutions have not (in this book!) been developed or do not exist.

The simplest example, of the former would be the symmetric double quantum well of Fig. 3.15. The potential function $V(z)$ required for the numerical solution is simply the (in this case) conduction band edge as given in the figure.

Figure 3.16 displays the results of calculations of the lowest two energy states as a function of the central barrier width for a double quantum well with the Al barrier concentration $x = 0.2$ and a fixed well width $l_w = 60$ Å. When the wells are separated by a large distance, the interaction between the eigenstates localised within each well is very small and the wells behave as two independent single quantum wells. However, as illustrated in Fig. 3.16, as the central barrier is decreased, the energy levels interact, with one being forced to higher energies and the other to lower energies. This is directly analogous to the formation of a pair of bonding and anti-bonding orbitals when two hydrogen atoms are brought together to form a hydrogen molecule. In that case, as here, the wave function of the lower (bonding) state is a sum of the wave functions of the separate atoms (wells), and the higher (anti-bonding)

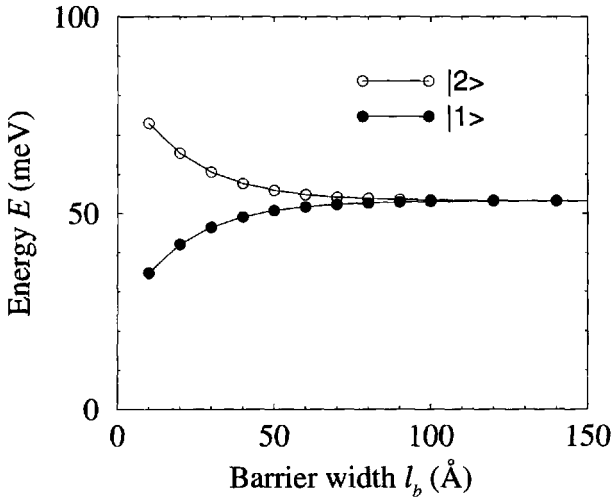


Figure 3.16 The confinement energies of the lowest two states of a symmetric double quantum well as a function of the central barrier width

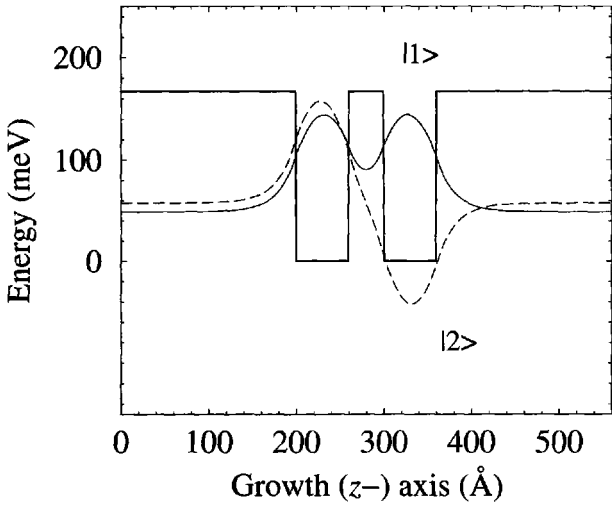


Figure 3.17 The wave functions of the lowest two energy levels of the symmetric double quantum well with a central barrier width of 40 Å

state is a difference in wave functions, i.e.

$$\Psi_{\text{bonding}} = \frac{1}{\sqrt{2}} (\psi_1 + \psi_2); \quad \Psi_{\text{anti-bonding}} = \frac{1}{\sqrt{2}} (\psi_1 - \psi_2) \quad (3.54)$$

In the case of hydrogen, both electrons move into the lower energy orbital, thus reducing the total energy of the two-hydrogen-atom system to form a chemically bound hydrogen molecule. In the case of the double quantum well here, a similar situation occurs with the electron spins aligning in an 'anti-parallel' arrangement in order to satisfy the Pauli exclusion principle.

Figure 3.17 displays the wave functions of the double quantum well with a central barrier width of 40 Å. Clearly, they form a symmetric and anti-symmetric pair, with the former being of lower energy, as discussed for the hydrogen molecule as above.

3.10 MULTIPLE QUANTUM WELLS AND FINITE SUPERLATTICES

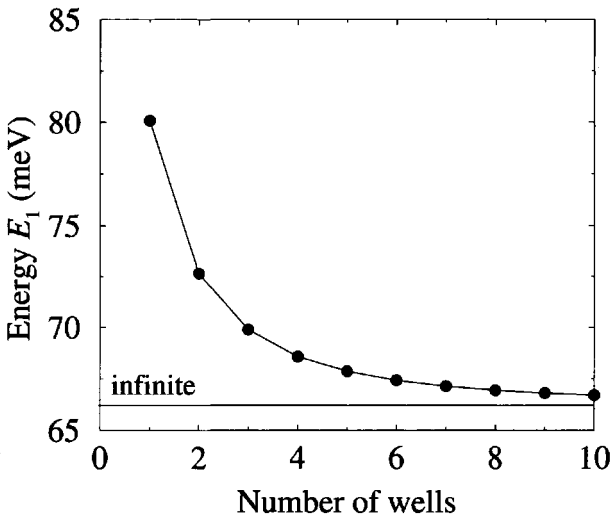


Figure 3.18 The ground-state energy of a N -well heterostructure

Real layered structures can only contain a finite number of quantum wells, so infinite superlattices don't actually exist. Thus, the Kronig-Penney model derived in the previous chapter has to be used carefully; i.e. it will give the energy and dispersion relationships of carriers *near the centre of a many-period quantum well system*.

One way of discovering exactly how many quantum wells are required before a finite structure resembles an infinite would be to look at the ground-state energy as a function of the number of periods. Fig. 3.18 does exactly this for N repeats of a 40 Å GaAs/40 Å Ga_{0.8}Al_{0.2}As unit cell capped with 200 Å Ga_{0.8}Al_{0.2}As barriers. By the time there are 10 wells, the ground state energy is within 1 meV of that given by the Krönig-Penney model of an infinite structure.

While 10 wells are sufficient from an energy perspective for this particular finite system to resemble the infinite case, the picture given by the wave function is quite

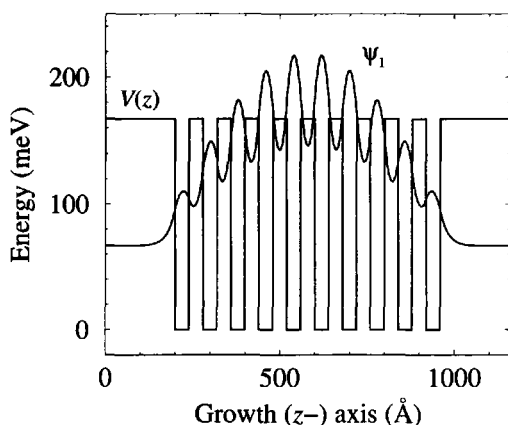


Figure 3.19 The ground-state wave function of a finite superlattice

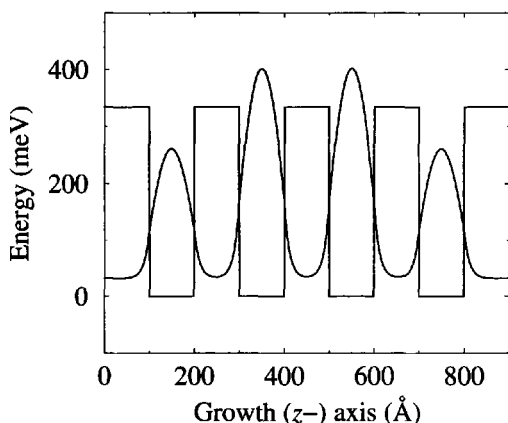


Figure 3.20 The ground-state wave function of a multiple quantum well

different, as illustrated in Fig. 3.19. Clearly, this system is a superlattice in that the wave function ‘associated’ with each well has a significant overlap with that of the adjacent well. In addition, the finiteness of the structure leads to the feature that the electron is not equally likely to be in any of the wells. Compare this to the wave function for the heterostructure based on a 100 Å GaAs/100 Å $\text{Ga}_{0.6}\text{Al}_{0.4}\text{As}$ unit cell, as in Fig. 3.20, which is clearly a multiple quantum well in that the wave

function reaches zero between the wells. It should be noted, however, that the effect of finiteness is *still* apparent in that the probability of the electron being in any of the quantum wells is not equal.

3.11 ADDITION OF ELECTRIC FIELD

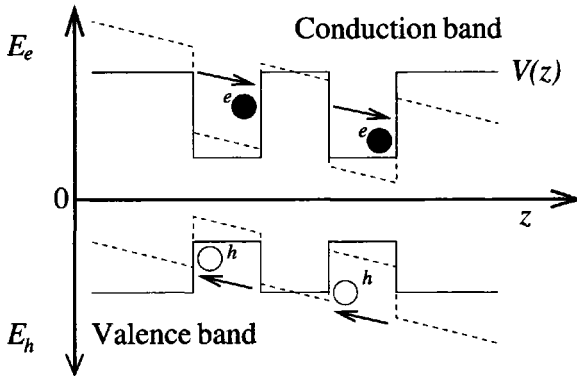


Figure 3.21 Schematic illustration of the tilting of the conduction and valence bands under the influence of an electric field

Unlike the previous analytical forms for the solution of Schrödinger's equation (described in Chapter 2), the numerical solution which includes the effect of an electric field on a heterostructure is simplicity itself. The potential energy is simply added to the potential term within the appropriate shooting equation (equations (3.11) or (3.53)), i.e.

$$V(z) \rightarrow V(z) + qF(z - z_0) \quad (3.55)$$

where for an electron, $q = -e$ and for a hole, $q = +e$. The position z_0 represents the origin of the field, often chosen to be the centre of the well. Fig. 3.21 shows the effect of an electric field on both the conduction- and valence-band potentials; note that increasing hole energies are measured downwards. Thus, any electrons in the double well are pulled to the right-hand side, while holes are pulled to the left, thus producing space charge or a polarisation of the carriers.

3.12 QUANTUM CONFINED STARK EFFECT

Section 2.12 demonstrated through perturbation theory the well known phenomenon of the quantum-confined Stark effect. Fig. 3.22 illustrates this energy level suppression as a function of field for the ground state of a single quantum well.

In order to validate the perturbation theory approach of earlier, the curve represents a parabolic fit to the calculated data which are displayed as points on the figure. It

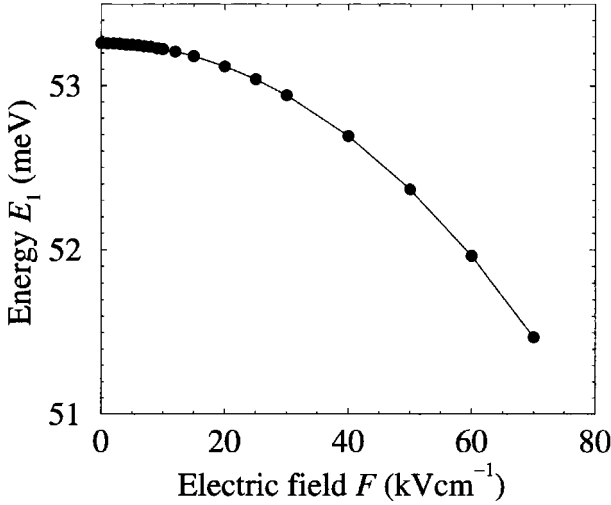


Figure 3.22 The effect of an electric field on the ground state energy E_1 of a 200 Å Ga_{0.8}Al_{0.2}As/60 Å GaAs/200 Å Ga_{0.8}Al_{0.2}As single quantum well

was found that the ground state energy over the range of fields investigated, could be represented very well by the parabola:

$$E_1(F) = E_1(0) - 0.00036F^2 \quad (3.56)$$

where $E_1(0)$ refers to the ground-state energy (53.310 meV) at zero field, and the electric field strength F is in units of kVcm $^{-1}$.

3.13 FIELD-INDUCED ANTI-CROSSINGS

As a further example of electric-field-induced phenomena and the versatility of the numerical solution, consider a double quantum well, as shown in Fig. 3.15 but with the left-hand well wider than the right. This asymmetry breaks the degeneracy of the two confined states and results in the lower energy state being more localised in the wider left hand well and the higher energy state being associated with the narrower right hand well.

On the addition of an electric field, the left hand side of the structure increases in potential, while the right hand side decreases, and thus the energy levels are brought closer together. A priori, it is not clear what happens in such a situation. However, the system can be solved numerically for all field points in order to reveal the particulars of the quantum mechanics. Fig. 3.23 displays the results of such a calculation.

The two energy levels *are* brought closer together, and as the field increases their separation $\Delta E (= E_2 - E_1)$, which is initially decreasing linearly, reaches a minimum

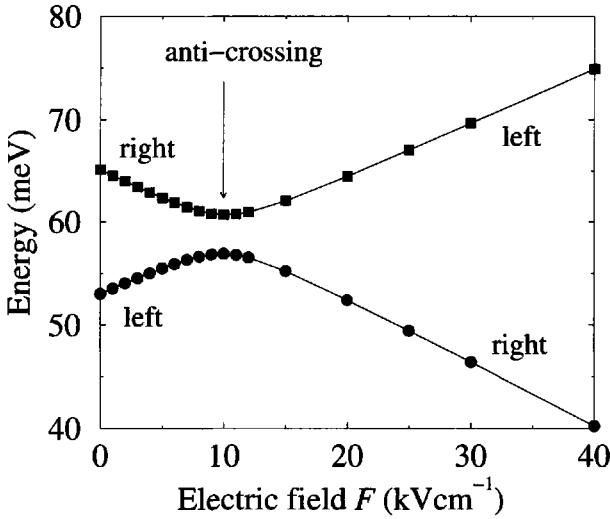


Figure 3.23 The two lowest energy confined states of a 200 \AA $\text{Ga}_{0.8}\text{Al}_{0.2}\text{As}/60 \text{ \AA}$ $\text{GaAs}/60 \text{ \AA}$ $\text{Ga}_{0.8}\text{Al}_{0.2}\text{As}/50 \text{ \AA}$ $\text{GaAs}/200 \text{ \AA}$ $\text{Ga}_{0.8}\text{Al}_{0.2}\text{As}$ double quantum well as a function of the applied electric field

value and then increases again. Within the linear sections, ΔE is merely equal to the difference in field potentials between the centres of the wells, in this case:

$$\Delta E = eF\Delta z = eF(30 \text{ \AA} + 30 \text{ \AA} + 25 \text{ \AA}) \quad (3.57)$$

At the minimum separation, the wave functions reveal a change in character—the lowest energy state switches from being localised in the left-hand well to the right-hand well, and vice versa for the higher energy level. This phenomenon is referred to as an *anti-crossing*.

3.14 SYMMETRY AND SELECTION RULES

Overlap integrals can be a very quick way of understanding or explaining complex interactions and are often instrumental in determining selection rules for carrier scattering events. The simplest overlap integral is defined as follows:

$$O = \langle \psi_n | \psi_{n'} \rangle = \int_{-\infty}^{+\infty} \psi_n^* \psi_{n'} dz \quad (3.58)$$

If ψ_n and $\psi_{n'}$ are eigenstates of the same Hamiltonian, whether they be wave or envelope functions, then they are orthonormal, i.e.

$$\langle \psi_n | \psi_{n'} \rangle = \delta_{nn'} \quad (3.59)$$

Hence, any scattering event which involves simply O will follow this selection rule. As will be encountered later in Chapter 9, the relative strength of an *interband* electron-hole recombination (both excitonic and free-carrier) is proportional to O , even though one of the eigenstates corresponds to a conduction-band electron and the other to a valence-band hole; under the envelope function approximation, they are eigenstates of different Hamiltonians. For symmetric potentials, such as a single quantum well, the states have exact parity, and thus O is non-zero, so corresponding to an allowed transition, for $\Delta n=0, 2$, etc.

Electron-electron scattering will also be discussed in Chapter 9, this is a scattering event between two initial states, resulting in two new states. The rate therefore contains an overlap between all four wave functions, as follows:

$$\text{rate} \propto \int_{-\infty}^{+\infty} \psi_i^* \psi_j^* \psi_f \psi_g \, dz \quad (3.60)$$

Again for electrons in symmetric potentials, the wave functions have definite parity, and thus for a two subband system, the $|2\rangle|2\rangle \rightarrow |1\rangle|1\rangle$ transition is allowed, while the $|2\rangle|2\rangle \rightarrow |2\rangle|1\rangle$ event is not.

Occasionally it is worthwhile to define another form for the overlap integral, i.e. the overlap integral of the moduli:

$$O_{\text{mod}} = \langle |\psi_n| | |\psi_{n'}| \rangle = \int_{-\infty}^{+\infty} |\psi_n^*| |\psi_{n'}| \, dz \quad (3.61)$$

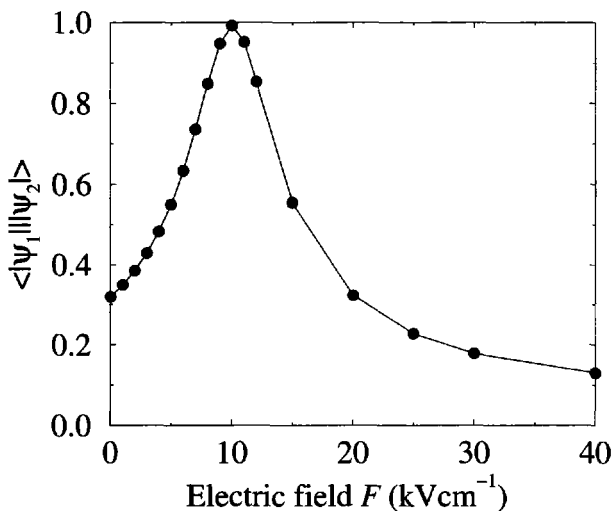


Figure 3.24 The overlap integral of the modulus of the wave functions undergoing the anti-crossing in Fig. 3.23

For example, Fig. 3.24 displays the overlap integral of the wave function moduli for the double quantum well of the previous section. As discussed there, the eigenstates undergo an anti-crossing at a field of 10 kVcm^{-1} . Initially, the wave functions are localised in separate wells, and then as the field is increased the energy levels are brought into alignment and the wave functions 'spread out' over both wells. As both states are simultaneously eigenstates of the same Hamiltonian, O remains zero; however, the nature of the anti-crossing is reflected in the peak of O_{mod} .

3.15 THE HEISENBERG UNCERTAINTY PRINCIPLE

This is an opportune moment to digress slightly and use the techniques now available to explore a fundamental building block of quantum mechanics, namely Heisenberg's Uncertainty Principle. In particular, the aspect of interest in this work will be 'Do the eigenstates of one-dimensional finite potentials always obey the principle and how close to the limit can they be taken?'

Mathematically, the Uncertainty Principle is given by the following [63]:

$$\Delta x_i \Delta p_j \gtrsim \frac{\hbar}{2} \delta_{ij} \quad (3.62)$$

where the δ function implies that for an uncertainty in the determination of the particle's position along the growth (z -) axis, there is no physical limit to the accuracy of any determination in the particle's in-plane momentum.

Restricting interest to the direction of confinement then the uncertainty relationship simplifies to:

$$\Delta z \Delta p_z \gtrsim \frac{\hbar}{2} \quad (3.63)$$

where

$$(\Delta z)^2 = \langle z^2 \rangle - \langle z \rangle^2 \quad \text{and} \quad (\Delta p_z)^2 = \langle p_z^2 \rangle - \langle p_z \rangle^2 \quad (3.64)$$

In the previous chapter, Section 2.7 discussed the relevance of Heisenberg's Uncertainty relationship with respect to solutions to Schrödinger's equation in the limit of infinite barrier mass. At that point, the techniques had not been developed—the analytical forms for the wave functions do not lend themselves well to the calculation of Δz and Δp_z . Now, however, numerical solutions are at hand and the calculation can proceed.

The individual components contributing to Δz and Δp_z are expectation values of the state of interest, calculated in the usual way, i.e.

$$\langle z \rangle = \int_{-\infty}^{+\infty} \psi^*(z) z \psi(z) dz \quad (3.65)$$

$$\langle z^2 \rangle = \int_{-\infty}^{+\infty} \psi^*(z) z^2 \psi(z) dz \quad (3.66)$$

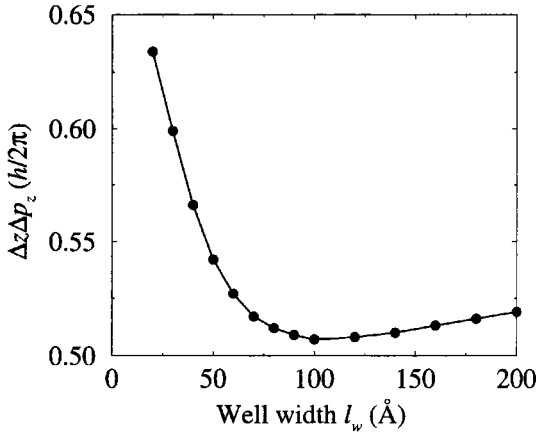


Figure 3.25 The product of the uncertainties in the position and momentum for an electron in a GaAs single quantum well surrounded by barriers of 100 meV height and with constant mass, as a function of the well width

$$\langle p_z \rangle = -i\hbar \int_{-\infty}^{+\infty} \psi^*(z) \frac{\partial}{\partial z} \psi(z) dz \quad (3.67)$$

$$\langle p_z^2 \rangle = -\hbar^2 \int_{-\infty}^{+\infty} \psi^*(z) \frac{\partial^2}{\partial z^2} \psi(z) dz \quad (3.68)$$

Whereas analytical calculations of $\langle p_z \rangle$ and $\langle p_z^2 \rangle$ are tedious, with the numerical formalism and the familiarity built up with finite difference expansions, both quantities follow simply as:

$$\langle p_z \rangle = -i\hbar \int_{-\infty}^{+\infty} \psi^*(z) \frac{\psi(z + \delta z) - \psi(z - \delta z)}{2\delta z} dz \quad (3.69)$$

and

$$\langle p_z^2 \rangle = -\hbar^2 \int_{-\infty}^{+\infty} \psi^*(z) \frac{\psi(z + \delta z) - 2\psi(z) + \psi(z - \delta z)}{(\delta z)^2} dz \quad (3.70)$$

Figure 3.25 plots $\Delta z \Delta p_z$ for a typical single quantum well, as a function of the well width. The form of the curve is very interesting in that it is a non-monotonic function with a minimum at around $l_w = 100$ Å. This can be understood intuitively in terms of the spatial coordinate—as the well width is reduced from large values, Δz , which might be considered as the well width, decreases. However, at the same time the confinement energy increases and at narrow well widths lies just below the top of

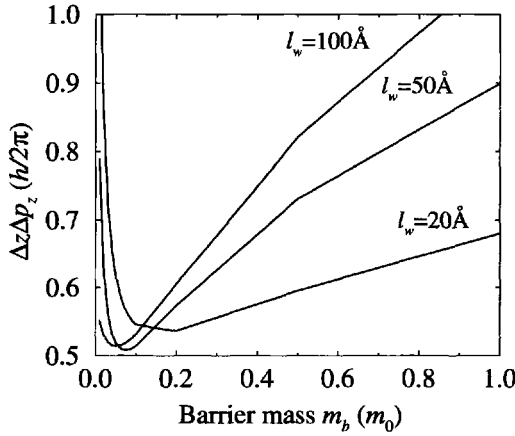


Figure 3.26 The product of the uncertainties in the position and momentum for an electron in a GaAs single quantum well surrounded by $\text{Ga}_{0.6}\text{Al}_{0.4}\text{As}$ barriers, as a function of barrier mass and for various well widths

the well. At this point, the wave function ‘spills over’ the barriers, thus leading to an increase in Δz .

Interestingly, the minimum in $\Delta z \Delta p_z$ is just above the $\hbar/2$ limit given by Heisenberg’s Uncertainty Principle. This is perhaps even more dramatically illustrated by the curves shown in Fig. 3.26, which plots $\Delta z \Delta p_z$ versus the barrier mass for a variety of well widths. These data correspond to the single quantum well shown in Fig. 2.21, which at the point of introduction was thought to imply a violation of Heisenberg’s Uncertainty Principle, but now, however, the calculations reveal this not to be the case.

The non-monotonic behaviour as a function of well width is again apparent from the ordering of the curves for any particular mass. In addition, a non-monotonic behaviour as a function of the electron mass in the barrier can be seen. The minimum value of $\Delta z \Delta p_z$ on Fig. 3.26 occurs for $l_w = 50 \text{ \AA}$, as would be expected from earlier. This occurs for a barrier mass which is very close to the well mass of $0.067 m_0$ and is equal to $0.508 \hbar$, which is very close, but still larger than $\hbar/2$.

3.16 EXTENSION TO INCLUDE BAND NON-PARABOLICITY

For semiconductor heterostructures with relatively low barrier heights and low carrier densities, the electrons cluster around the subband minima and their energy is in turn reasonably close to the bulk conduction band minima, i.e. within a couple of hundred meV, compared to a bandgap of the order of 1.5 eV. In this region, the band minima,

both the bulk conduction band and the in-plane subband, can be described by a parabolic E - \mathbf{k} curve, i.e. in the usual form:

$$E = \frac{\hbar^2 |\mathbf{k}|^2}{2m^*} \quad (3.71)$$

where \mathbf{k} can be read as a three-dimensional vector for the bulk, or as a two-dimensional in-plane vector $|\mathbf{k}_{x,y}|$ for a subband.

However, in situations where the electrons are forced up to higher energies, by either large barrier heights and narrow wells, or in the case of very high carrier densities, equation (3.71) becomes more approximate. This is even more important for holes in the valence band. The approximation can be improved upon by adding addition terms into the polynomial expansion for the energy. For example, no matter how complex the band structure along a particular direction, it clearly has inversion symmetry and hence can always be represented by an expansion in even powers of k , if sufficient terms are included, i.e. the energy E can always be described by:

$$E = a_0 k^0 + a_2 k^2 + a_4 k^4 + a_6 k^6 + a_8 k^8 + \dots = \sum_{i=0}^{\infty} a_{2i} k^{2i} \quad (3.72)$$

Usually when discussing single band models, as has been focused on entirely so far, the energy origin is set at the bottom of the band of interest, and thus $a_0 = 0$. In addition, truncating the series at k^2 , gives $a_2 = \hbar^2/(2m^*)$.

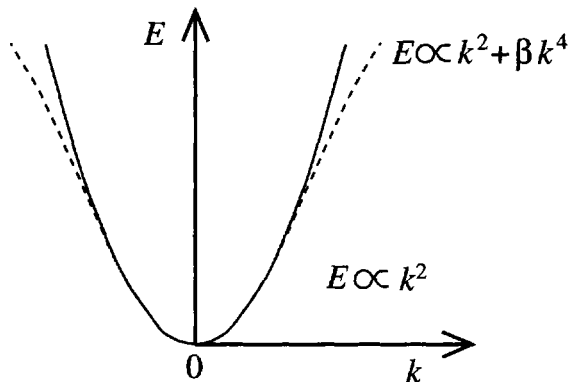


Figure 3.27 Non-parabolicity of the bulk and subband structure

The next best approximation is to include terms in k^4 , hence accounting for band non-parabolicity, as displayed in Fig. 3.27, with:

$$E = a_2 k^2 + a_4 k^4 = \frac{\hbar^2}{2m^*} (k^2 + \beta k^4) \quad (3.73)$$

where

$$\beta = a_2 \frac{2m^*}{\hbar^2} \quad (3.74)$$

Remembering the basic definition of effective mass, i.e.

$$m^* = \hbar^2 \left(\frac{\partial^2 E}{\partial k^2} \right)^{-1} \quad (3.75)$$

then clearly m^* remains a function of k , unlike the case of parabolic bands where it is a constant. As it is a function of k , then it is also a function of energy, and indeed band non-parabolicity is accounted for by allowing the effective mass to have an energy dependence [64]:

$$m^*(E) = m^*(0)[1 + \alpha(E - V)] \quad (3.76)$$

where V is the barrier height and the parameter α is given by [65, 66]:

$$\alpha = \left[1 - \frac{m^*(0)}{m_0} \right]^2 / E_g \quad (3.77)$$

where E_g is the semiconductor bandgap. With this new form of the effective mass, the Schrödinger equation then becomes:

$$-\frac{\hbar^2}{2} \frac{\partial}{\partial z} \frac{1}{m^*(z, E)} \frac{\partial}{\partial z} \psi(z) + V(z)\psi(z) = E\psi(z) \quad (3.78)$$

which can be solved by using the iterative shooting equation (equation (3.53)) as before, but with the additional feature of adjusting the effective mass for each energy E .

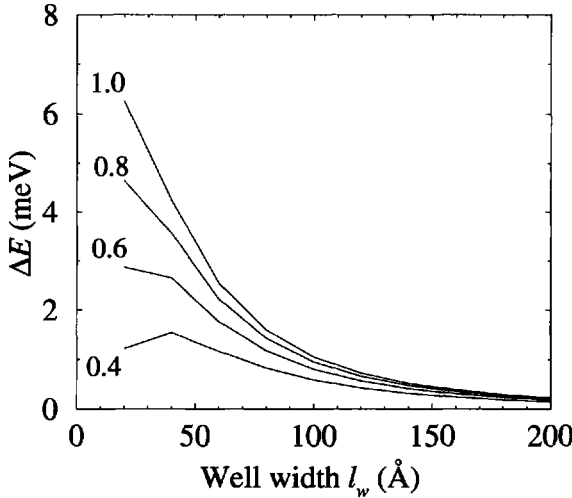


Figure 3.28 The difference in ground-state energy of an electron at the subband minima in a GaAs single quantum well, with and without non-parabolicity, $\Delta E = E_1(\text{with}) - E_1(\text{without})$, for a range of $\text{Ga}_{1-x}\text{Al}_x\text{As}$ barrier concentrations x

Fig. 3.28 displays the effect of the inclusion of non-parabolicity on the ground-state energy of a single GaAs quantum well surrounded by $\text{Ga}_{1-x}\text{Al}_x\text{As}$ barriers. This figure plots ΔE , i.e. the energy calculated including non-parabolicity minus the energy which does not include non-parabolicity. Clearly, non-parabolicity increases the energy calculated, and the increase is (generally) larger for an increasing barrier height and a decreasing well width. The latter two points reflect the expected conclusions formed earlier, i.e. that as the electron (or hole) is forced up the E - k curve, the parabolic description of the band becomes more approximate.

The barrier material, $\text{Ga}_{1-x}\text{Al}_x\text{As}$, becomes indirect as x increases beyond 0.42, thus producing an extra complication. Many real layered structures are grown with concentrations below this value, and therefore the majority of examples in this present book do conform to this criterion. Given this, the *maximum* effect of band non-parabolicity is largely given by the bottom curve of Fig. 3.28, i.e. 2 meV. The chapters that follow are centred around the development of techniques and methods and the illustration of generic physics which is applicable across material systems, so band non-parabolicity will not be employed in future examples of calculations based on the GaAs/ $\text{Ga}_{1-x}\text{Al}_x\text{As}$ material system.

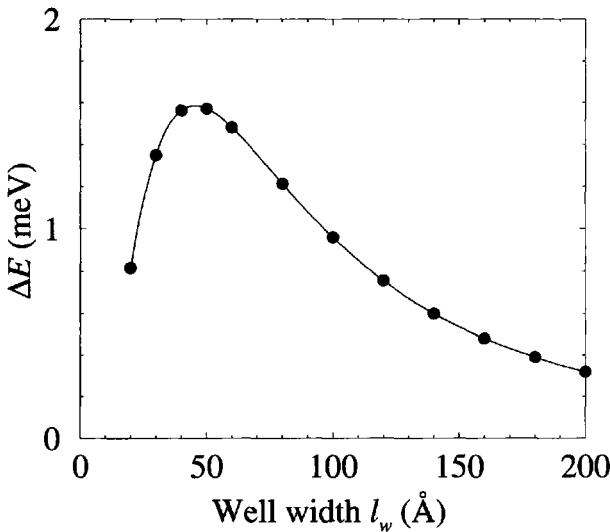


Figure 3.29 The difference in ground-state energy of an electron in a single $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}/\text{In}_{0.52}\text{Al}_{0.48}\text{As}$ quantum well, with and without non-parabolicity, $\Delta E = E_1(\text{with}) - E_1(\text{without})$

One material in which this effect is thought to be more important is the quaternary $\text{In}_{1-x-y}\text{Al}_x\text{Ga}_y\text{As}$ system, which has become an important material system for inter-subband lasers (more of this later in Chapter 9) because of its large conduction band offset. In particular, the $\text{In}_{0.52}\text{Al}_{0.48}\text{As}$ and $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}$ alloys are lattice-matched to the readily available InP substrates (see [15] p. 196).

In fact, Fig. 3.29 demonstrates that the effect is relatively small, but nevertheless significant, at narrower well widths. Indeed, energy level spacings in intersubband lasers are designed in relation to phonon energies which are typically a few 10s of meV, and hence it is debatable whether a 2 meV effect should be included. The effect is larger when the $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}$ wells are surrounded by strained AlAs barriers, so providing even more offset, and this material system has been the subject of several recent papers [64, 67, 68].

3.17 POISSON'S EQUATION

All of the techniques are now in place to be able to solve the Schrödinger equation for any heterostructure for which the band-edge potential profile defining the structure is known. However, all of the theoretical methods and examples described so far have concentrated solely on solving systems for a single charge carrier. In many devices such models would be inadequate as large numbers of charge carriers, e.g. electrons, can be present in the conduction band. In order to decide whether or not typical carrier densities would give rise to a *significant* additional potential on top of the usual band-edge potential terms (which will be labelled specifically as V_{CB} or V_{VB}), it then becomes necessary to solve the electrostatics describing the system.

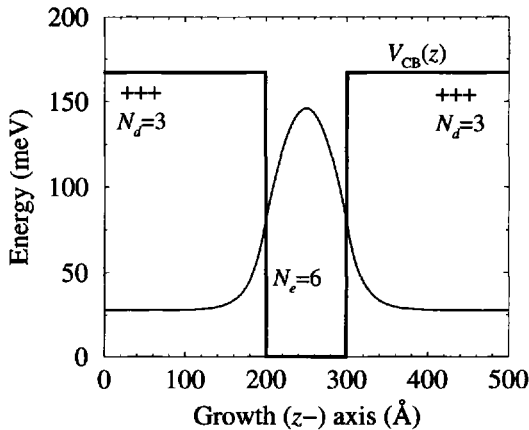


Figure 3.30 A modulation-doped single quantum well

When considering the case of an n -type material, then (although obvious) it is worth stating that the number of 'free' electrons in the conduction band is equal to the number of positively charged ionised donors in the heterostructure. In the example shown in Fig. 3.30 this equates to six donors becoming ionised and supplying six electrons into the quantum well—thus the system maintains charge neutrality. Fig. 3.30 is a

modulation-doped system in that the doping is located in a position where the free carriers it produces will become spatially separated from the ions.

The additional potential term $V_\rho(z)$ arising from this, or any other charge distribution ρ , can be expressed by using Poisson's equation:

$$\nabla^2 V_\rho = -\frac{\rho}{\epsilon} \quad (3.79)$$

where ϵ is the permittivity of the material, i.e. $\epsilon = \epsilon_r \epsilon_0$. The solution is generally obtained via the electric field strength \mathbf{E} ; recalling that:

$$\mathbf{E} = -\nabla V \quad (3.80)$$

the potential then would follow in the usual way [69]:

$$V_\rho(\mathbf{r}) = - \int_{-\infty}^{\mathbf{r}} \mathbf{E} \cdot d\mathbf{r} \quad (3.81)$$

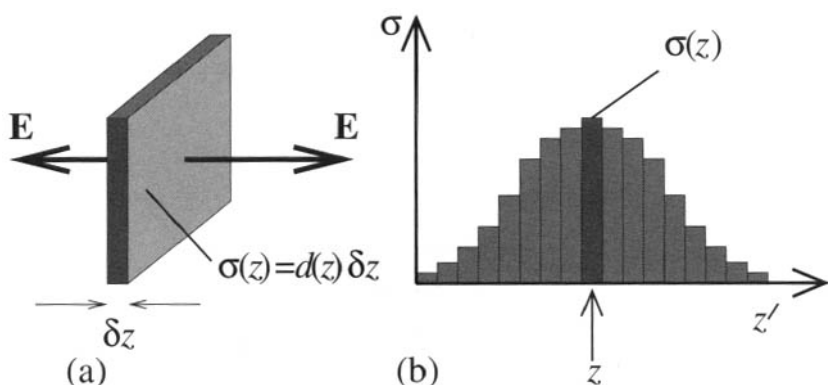


Figure 3.31 Electric field strength from an infinite plane of charge of volume density $d(z)$ and thickness δz

Given that the potential profiles, $V_{CB}(z)$ for example, are one-dimensional, then they will also produce a one-dimensional charge distribution. In addition, remembering that the quantum wells are assumed infinite in the x - y plane then any charge density $\rho(z)$ can be thought of as an infinite plane, i.e. a *sheet*, with areal charge density $\sigma(z)$ and thickness δz , as shown in Fig. 3.31(a). Such an infinite plane of charge produces an electric field perpendicular to it, and with a strength:

$$\mathbf{E} = \frac{\sigma}{2\epsilon} \quad (3.82)$$

Note that as the sheet is infinite in the plane, then the field strength is constant *for all distances from the plane*. The total electric field strength due to many of these planes

of charge, as shown in Fig. 3.31(b), is then just the sum of the individual contributions as follows:

$$\mathbf{E}(z) = \sum_{z'=-\infty}^{\infty} \frac{\sigma(z')}{2\epsilon} \text{sign}(z - z') \quad (3.83)$$

where the function sign is defined as

$$\text{sign}(z) = +1, \quad z \geq 0; \quad \text{sign}(z) = -1, \quad z < 0 \quad (3.84)$$

and has been introduced to account for the vector nature of \mathbf{E} , i.e. if a single sheet of charge is at a position z' , then for $z > z'$, $\mathbf{E}(z) = +\sigma/(2\epsilon)$, whereas for $z < z'$, $\mathbf{E}(z) = -\sigma/(2\epsilon)$. Note further that it is only the charge neutrality, i.e. there are as many ionised donors (or acceptors) in the system as there are electrons (or holes), or expressed mathematically:

$$\sum_{z=-\infty}^{+\infty} \sigma(z) = 0 \quad (3.85)$$

which ensures that the electric field, and hence the potential, go to zero at large distances from the charge distribution. For the case of a doped semiconductor, there would be two contributions to the charge density $\sigma(z)$, where the first would be the ionised impurities and the second the free charge carriers themselves. While the former would be known from the doping density in each semiconductor layer, as defined at growth time, the latter would be calculated from the probability distributions of the carriers in the heterostructure. Thus if $d(z)$ defines the volume density of the dopants at position z , where the planes are separated by the usual step length δz , then the total number of carriers, per unit cross-sectional area, introduced into the heterostructure is given by:

$$N = \int_{-\infty}^{+\infty} d(z) \, dz \quad (3.86)$$

The net charge density in any of the planes follows as

$$\sigma(z) = q [N\psi^*(z)\psi(z) - d(z)] \delta z \quad (3.87)$$

where q is the charge on the extrinsic carriers. The step length δz selects the proportion of the carriers that are within that 'slab' and converts the volume density of dopant, $d(z)$ into an areal density.

If the charge carriers are distributed over more than one subband, then the contribution to the charge density $\sigma(z)$ would have to be summed over the relevant subbands, i.e.

$$\sigma(z) = q \left(\sum_{i=1}^n N_i \psi_i^*(z) \psi_i(z) - d(z) \right) \delta z \quad (3.88)$$

where $\sum_{i=1}^n N_i = N$.

Fig. 3.32 shows the areal charge density along the growth axis for a 100 Å GaAs well, n -type doped to $2 \times 10^{18} \text{cm}^{-3}$, surrounded by 200 Å undoped $\text{Ga}_{0.8}\text{Al}_{0.2}\text{As}$ barriers. The ionised donors yield a constant contribution to σ within the well of $d(z)\delta z =$

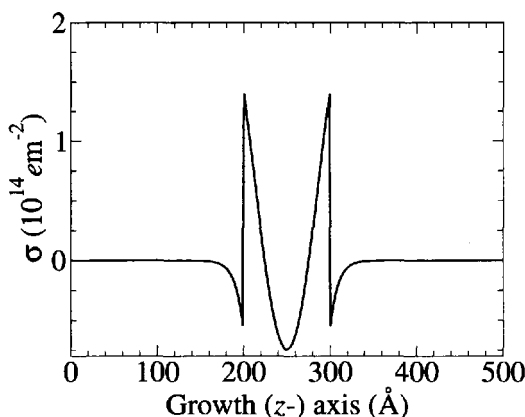


Figure 3.32 Areal charge density σ for a 100 Å GaAs well, n -type doped to $2 \times 10^{18} \text{ cm}^{-3}$, surrounded by undoped $\text{Ga}_{0.8}\text{Al}_{0.2}\text{As}$ barriers

$2 \times 10^{24} \text{ m}^{-3} \times 1 \text{ Å} = +2 \times 10^{14} \text{ m}^{-2}$, in each of the 1 Å thick slabs. Hence, the total number N of electrons in the quantum well is $100 \times 2 \times 10^{14} \text{ m}^{-2} = 2 \times 10^{12} \text{ cm}^{-2}$. By assuming that the electrons introduced by such doping all occupy the ground state of the quantum well, then the curve on top of the ionised impurity background clearly resembles $-\psi^* \psi$, as expected from the mathematics. The discontinuities in σ occur at the edges of the doping profiles and are of magnitude $2 \times 10^{14} \text{ m}^{-2}$, again as expected.

There are a number of points to note about Fig. 3.33, which plots the electric field strength E due to the charge distribution (as defined in equation (3.83)) along the growth axis of the heterostructure. First, the field does reach zero at either end of the structure, which implies charge neutrality. In addition, the zero field point at the centre of the structure reflects the symmetry of the charge distribution. The electric field strength itself is not an observable, merely an intermediate quantity which *can* be useful to plot from time to time; the quantity which is significant is, of course, the potential due to this charge distribution. Fig. 3.34 plots the potential as calculated from equation (3.81), as usual defining the origin, in this case for the potential, at the effective infinity μ at the left-hand edge of the barrier–well–barrier structure.

Again, the symmetry of the original heterostructure and doping profiles are reflected in the symmetric potential. The potential is positive at the centre of the well since the system under consideration consists of electrons in the conduction band, so any test charge used to probe the potential is also an electron which would be repelled by the existing charge. The carrier density in this single quantum well is reasonably high at $2 \times 10^{12} \text{ cm}^{-2}$, and this produces a potential of up to 4 meV; while this is

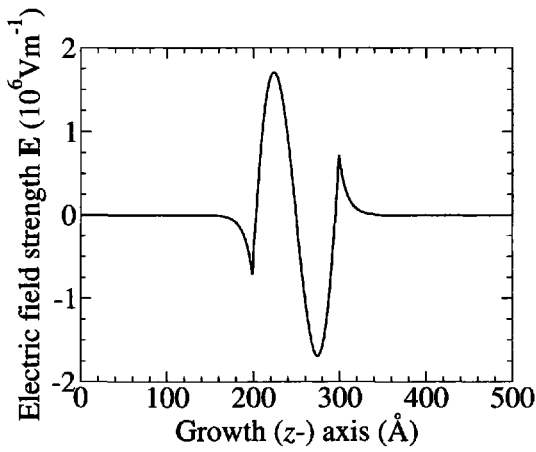


Figure 3.33 The electric field strength E due to the charge distribution shown in Fig. 3.32

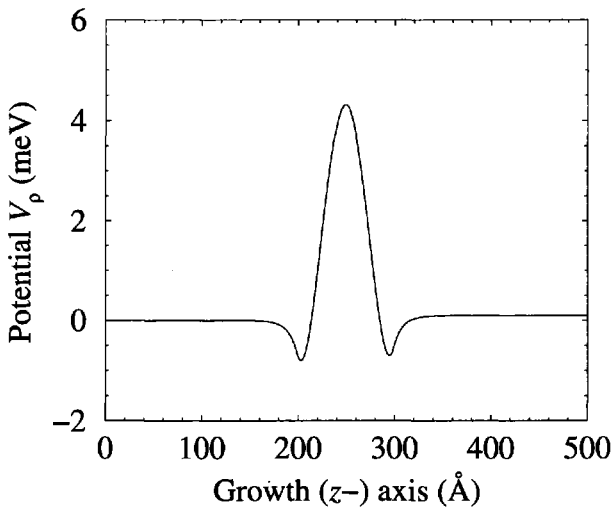


Figure 3.34 The potential due to the ionised donor/electron charge distribution

small compared to the conduction band offset, which is usually of the order of one or two hundred meV or more, it could still have a measurable effect on the energy eigenvalues of the quantum well.

3.18 SELF-CONSISTENT SCHRÖDINGER–POISSON SOLUTION

The energy eigenvalues are calculated by considering the introduction of a further test electron into the system and incorporating the potential due to the carrier density already present into the standard Schrödinger equation, i.e. the potential term $V(z)$ in equations (2.96) or (3.43) becomes:

$$V(z) \rightarrow V_{CB}(z) + V_{\rho}(z) \quad (3.89)$$

where V_{CB} represents the band edge potential at zero doping and the potential due to the non-zero number of carriers, i.e. the charge density ρ , is represented by the function V_{ρ} .

The numerical shooting method, described in detail earlier in this chapter, can be used without alteration to solve for this new potential, which will thus yield new energies and wave functions. The latter is an important point since the potential due to the charge distribution is itself dependent on the wave functions. Therefore, it is necessary to form a closed loop solving Schrödinger's equation, calculating the potential due to the resulting charge distribution, adding it to the original band-edge potential, solving Schrödinger's equation again, and so on—a process which is illustrated schematically in Fig. 3.35.

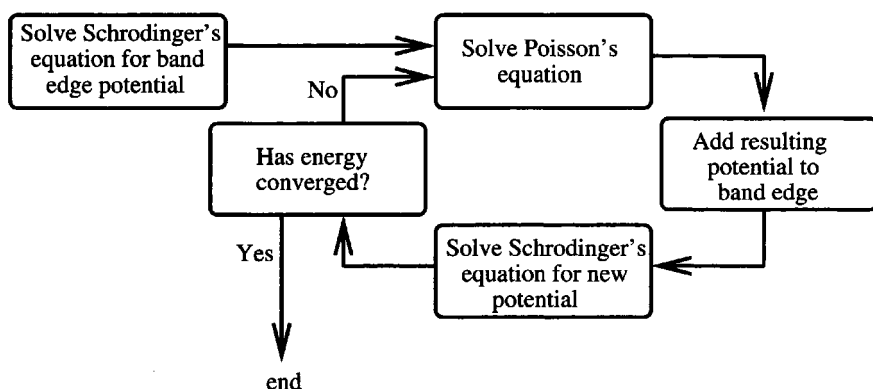


Figure 3.35 Block diagram illustrating the process of self-consistent iteration

The process is repeated until the energy eigenvalues converge; at this point the wave functions are simultaneously solutions to both Schrödinger's and Poisson's equations—the solutions are described as *self-consistent*, rather like Hartree's approach to solving many electron atoms (see e.g. reference [3] p. 396).

Figure 3.36 shows the result of adding the potential due to the charge distribution V_{ρ} , as displayed in Fig. 3.34, to the original band-edge potential V_{CB} for the single quantum well of the previous section. The perturbation, even at this relatively high carrier density of $2 \times 10^{12} \text{cm}^{-2}$, is rather small compared to the barrier height, for instance. Nonetheless it is important to calculate the effect of this perturbation on

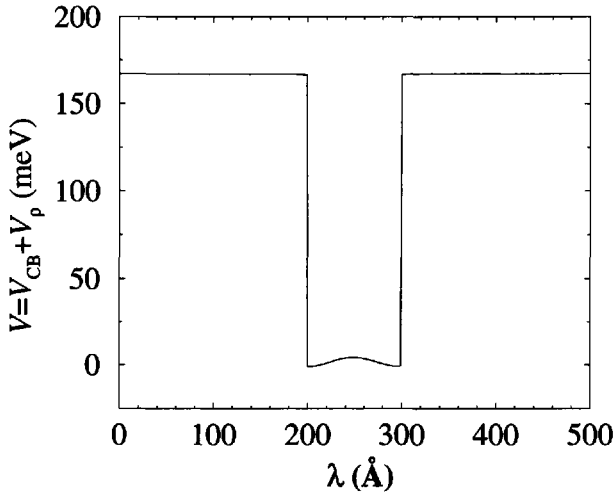


Figure 3.36 The sum of the band-edge potential V_{CB} and Poisson's potential V_p for the single quantum well of Fig. 3.32

the electron energy levels by continuing with the iterative process and looking for convergence of the resulting energy solutions. This process is illustrated in Fig. 3.37.

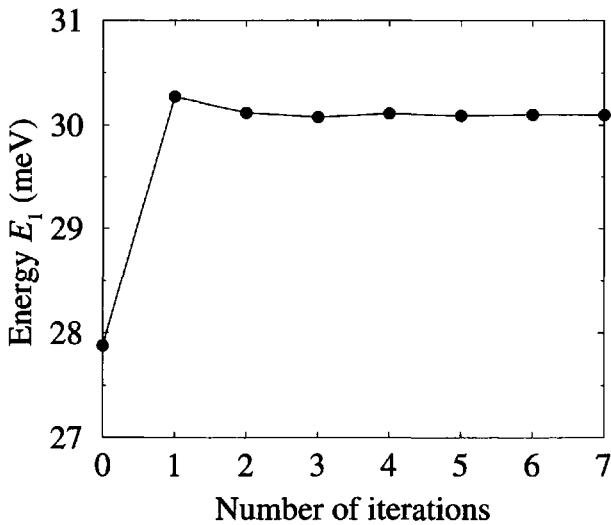


Figure 3.37 The ground-state energy of the single quantum well of Fig. 3.32 as a function of the number of iterative steps

The first iterative loop produces the majority of the change in the energy level from the single carrier system to the doped system. Subsequent iterations produce only minor refinements to the energy level, which has certainly converged by the eighth loop. Altogether the ground state energy has changed by about 2 meV when accounting for this finite charge density.

3.19 COMPUTATIONAL IMPLEMENTATION

As this work is about how to actually implement a computational method to provide quantitative predictions, *in addition to* deriving the theoretical equations in the first place, it is important to be aware that the self-consistent Schrödinger–Poisson solution presented here, can at times diverge.

The doping density in the last series of examples in the previous two sections was chosen such that it produced a noticeable perturbation to the total potential profile, as illustrated in Fig. 3.36. In fact, this total carrier density of $2 \times 10^{12} \text{cm}^{-2}$ is fairly high *when contained within just one quantum well*. The majority of calculations later in this text, on active regions of unipolar lasers for example, will have carrier densities much lower than this. However, in order to illustrate the intricacies of the computational method, consider now an even higher doping density, i.e. double that of the previous.

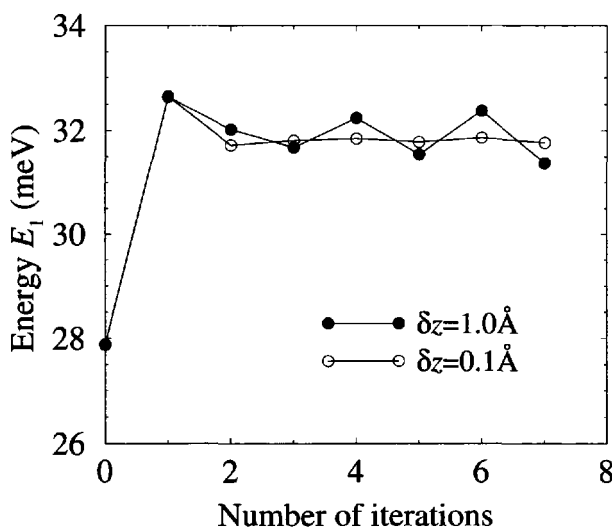


Figure 3.38 The ground state energy of the single quantum well of Fig. 3.32, but doped to $4 \times 10^{18} \text{cm}^{-3}$ in this case, thus giving a total carrier density of $4 \times 10^{12} \text{cm}^{-2}$

The first energy calculated in the process is that of the ground state for an undoped system, which is obviously the same as before. However, the higher doping density

leads to different Poisson potentials, and hence subsequent iterations *do* yield different energy levels. Although just one iterative step is sufficient to yield a result to within 1 meV of the self-consistent solution, the calculations with the same step length ($\delta z = 1 \text{ \AA}$) as before, diverge as the iteration proceeds. This is a manifestation of the computational implementation—the smallest asymmetry in the potential profile is exaggerated with each iterative loop. Fig. 3.38 also illustrates the solution, which is merely to increase the computational accuracy by way of reducing the step length δz in the original discretization of the potential. Asymmetric potentials, such as step quantum wells, or quantum wells with electric fields applied, are naturally more resistant to such effects.

3.20 MODULATION DOPING

Although mention has been made of quantum well systems in which doping in the barriers leads to a spatial separation of the ions and charge carriers, i.e. which collect in a quantum well, quantitative calculations presented thus far have not considered these modulation-doped systems. Fig. 3.39 shows the band-edge potential, V_{CB} , and the self-consistent potential, $V_{CB} + V_p$, for a system of the type illustrated in Fig. 3.30, i.e. an undoped single quantum well surrounded by doped barriers; with the full layer definition thus being: 100 \AA $\text{Ga}_{0.8}\text{Al}_{0.2}\text{As}$ doped n -type to $2 \times 10^{17} \text{ cm}^{-3}$; 100 \AA GaAs undoped; 100 \AA $\text{Ga}_{0.8}\text{Al}_{0.2}\text{As}$ doped n -type to $2 \times 10^{17} \text{ cm}^{-3}$.

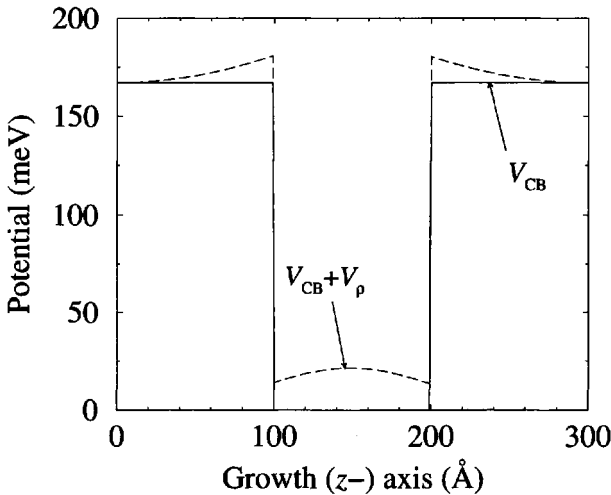


Figure 3.39 The band-edge potential and the self-consistent potential of a modulation-doped single quantum well

The electrons introduced into the system are physically separated from the ionised donors, so therefore instead of an ion/charge carrier plasma, the mobile charge in this

case is often referred to as a *two-dimensional electron gas*. The physical separation leads to a reduction in the ionised impurity scattering and hence increased electron mobilities for in-plane (x - y) transport, a feature which is exploited in *High-Electron-Mobility Transistors* (HEMTs).

3.21 THE HIGH-ELECTRON-MOBILITY TRANSISTOR

The high-electron-mobility transistor (HEMT), the heterostructure field-effect transistor (HFET), the modulated-doped field-effect transistor (MODFET), or even the two-dimensional electron-gas field-effect transistor (TEGFET), are all names which refer to a transistor designed to exploit the high in-plane (x - y) mobility which arises when a (usually single) heterojunction is modulation-doped. In essence, a high-bandgap material such as $\text{Ga}_{1-x}\text{Al}_x\text{As}$ is doped n -type, and upon ionisation of the donors the electrons move to the lower energy levels available in the narrower-bandgap material, such as GaAs—this situation is very similar to the single quantum well of the previous section, but with just one barrier layer in this case.

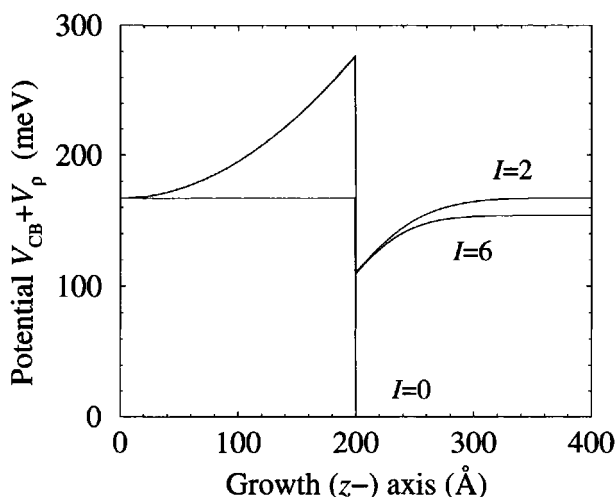


Figure 3.40 The evolution of the self-consistent potential $V_{\text{CB}} + V_p$, with the number of iterations I , for a single modulation doped heterojunction

Figure 3.40 shows the evolution of the self-consistent potential, $V_{\text{CB}} + V_p$, with each iterative step. After the first solution of Schrödinger's equation, the potential V consists entirely of the band-edge potential V_{CB} and clearly defines the single heterojunction. However, as the iteration proceeds the charge carriers are drawn towards the interface and form a triangular well potential—a process often referred to as *band bending*. This is vividly illustrated by Fig. 3.41, which shows the corresponding evolution of the electron wave functions.

This section has merely ‘scratched the surface’ of transistor modelling, and has been used to illustrate the applicability of quantum and electrostatic theory to everyday electronic devices, as well as the more esoteric semiconductor structures concentrated on in this work. For more information on heterojunction-based transistors see Kelly [7], Ch. 16, or more specialised works, such as [70–73].

3.22 BAND FILLING

One way of measuring the energy levels of quantum well systems is via the use of spectroscopy, which involves illuminating the semiconductor with light from a laser. If the photon energy is greater than the bandgap then, in the bulk material, electrons are excited from the valence-band states to the conduction-band, which depending in more detail on the excitation energy, can produce pairs of free-electrons and free-holes, or excitons (see Chapter 6). In a quantum well system, the photons must have additional energy in order to overcome the electron- and hole- ground-state confinement energies as well.

In a doped system, e.g. n -type, many electrons already exist in the conduction band, and thus spectroscopic measurements will observe the self-consistent solutions, $E_{e1}^{s.c.}$ and $E_{h1}^{s.c.}$ for the confinement energies.

However, in addition there is another effect which may or may not be large enough to observe. Thus far, it has been assumed that there are an unlimited number of states available to electrons (or holes) when they are introduced as extrinsic carriers; note

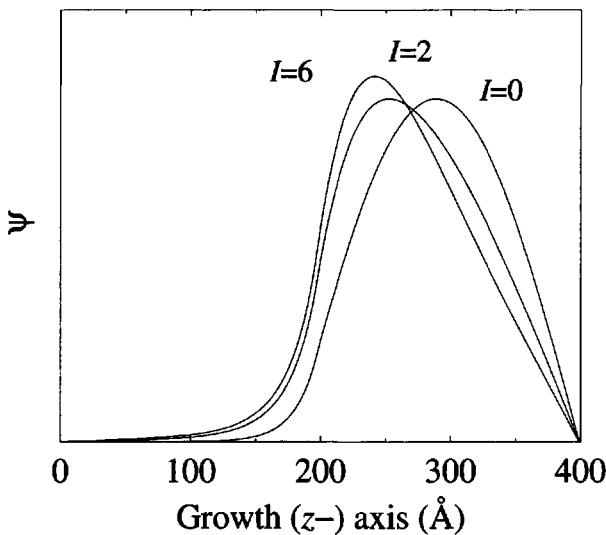


Figure 3.41 The evolution of the self-consistent wave functions, with the number of iterations I , for a single modulation doped heterojunction

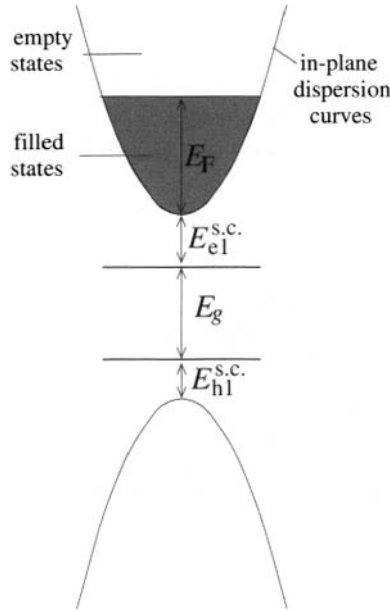


Figure 3.42 Schematic representation of the increase in band edge due to band filling

however (see Chapter 2) that the density of states has been derived for two-dimensional systems. Thus excitation can only occur when the photon energy is sufficiently large to excite an electron into an *empty* state, see Fig. 3.42. At low temperatures, the Fermi edge defines a sharp boundary between the filled states $f^{\text{FD}}(E) = 1$ and the empty states $f^{\text{FD}}(E) = 0$, and hence the *minimum* excitation energy must overcome this additional component, i.e.

$$E = E_g + E_{\text{el}}^{\text{s.c.}} + E_{\text{hl}}^{\text{s.c.}} + E_F \quad (3.90)$$

This phenomenon of an apparently increased ‘band edge’ to absorption is known as *band filling*. This can also occur in undoped systems under very high excitation intensities which produce large numbers of both electrons and holes, which in turn can fill all of the available states and thus increase the minimum required photon energy for absorption. It can be accounted for under the present scheme by simply calculating the quasi-Fermi energy, E_F , as described in Section 2.4.

CHAPTER 4

DIFFUSION

4.1 INTRODUCTION

Any substance will attempt to diffuse from an area where it is present in high concentrations to an area of low concentration. For example, obvious though it seems, if the curve $x(z)$ in Fig. 4.1 represented the concentration of water in a trough then the water would fall very rapidly from the region of high concentration to the region of low! Crude though it may seem, this is an example of diffusion. It occurs also for gaseous systems, e.g. smoke gradually disperses in an enclosed room; however the point of interest in this present work is diffusion of material species in solids.

In the context of semiconductors, and in particular semiconductor heterostructures, it is clear that diffusion of *material species* could be important as their very nature derives from discontinuous changes in materials. Fig. 4.1 could therefore represent a dopant, in either a bulk semiconductor, i.e. a *homojunction*, or an alloy component, e.g. Al at a GaAs/Ga_{1-x}Al_xAs *heterojunction*.

Diffusion at such boundaries is a strong possibility, particularly during the elevated temperatures often used during growth, or the lower but more prolonged heating that may occur during normal device operation. Any movement of material, e.g. Al at a

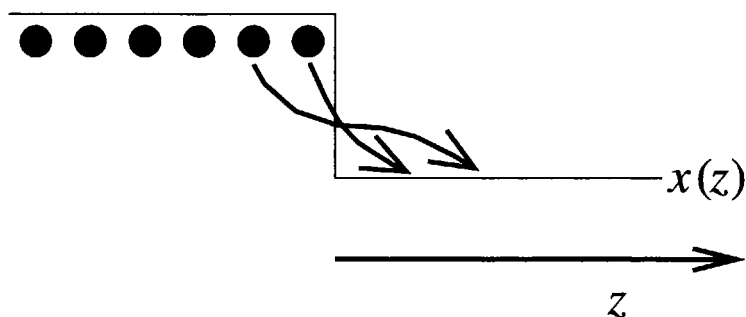


Figure 4.1 Simple illustration of diffusion; diffusant moves from areas of high concentration to low

GaAs/Ga_{1-x}Al_xAs heterojunction will ‘blur’ the interface, i.e. the change from one material to the other will occur via a range of intermediary alloys. Such a process is known as *interface mixing* and is represented schematically in Fig. 4.2. The change in profile of the junction will inevitably alter the electronic properties of the system, which will in turn affect the device characteristics and therefore the operating lifetime. The motivation behind modelling diffusion is to understand such time dependency of the properties of quantum well systems, in order to be able to control or prevent it, design a device in which its impact will be minimised, or predict the lifetime of a device. Besides these detrimental effects, controlled diffusion is sometimes initiated deliberately in order to tailor the optical and electronic properties of quantum wells systems [59]. For a comprehensive treatment of utilizing quantum well interdiffusion for photonics see the review by Li [74].

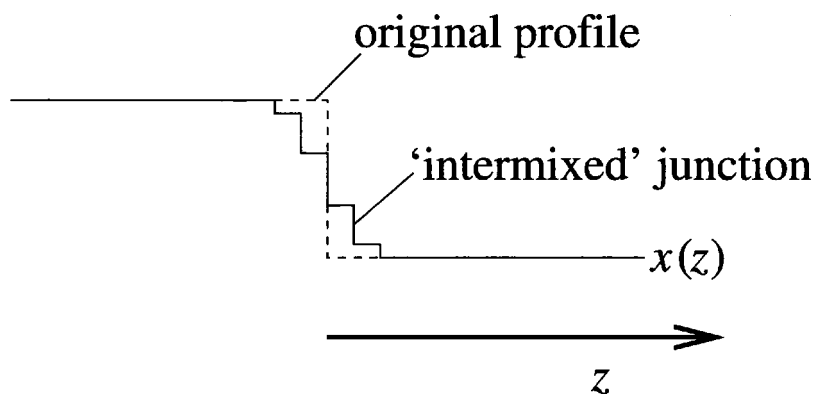


Figure 4.2 An intermixed heterojunction; note that the total amount of diffusant represented by the area under the $x(z)$ curve, is constant

Diffusion was first put on a quantitative basis by Fick in 1855 [75] who derived two laws. The first of these stated that the steady state flux through a plane is proportional to, but in the opposite direction to, the concentration gradient, i.e.

$$\text{flux} = -\mathcal{D}\nabla c \quad (4.1)$$

while the second showed that the time dependency of the concentration is given by the following:

$$\frac{\partial c}{\partial t} = -\nabla \cdot \text{flux} \quad (4.2)$$

$$\therefore \frac{\partial c}{\partial t} = \nabla \cdot (\mathcal{D}\nabla c) \quad (4.3)$$

which in one dimension would become:

$$\frac{\partial c}{\partial t} = \frac{\partial}{\partial z} \mathcal{D} \frac{\partial c}{\partial z} \quad (4.4)$$

For a simple introduction to these laws see p.66 of the text by Callister [76].

Many different particular solutions of Fick's first and second laws can be found, with perhaps the most common one being the error-function solution for diffusion, described by a constant \mathcal{D} , at the interface between two semi-infinite slabs. Such a model is commonly employed to model diffusion at a semiconductor heterojunction (see, e.g. [59, 77–80]).

Crank [81] and Shewmon [82] offer several solutions based upon numerical methods and involving complex initial diffusant profiles and variable diffusion coefficients. However, as with Tuck [83], each situation is considered separately and a new numerical solution is sought. It would be advantageous if a general, probably numerical, method could be developed which was applicable to all of these situations and also new ones.

Such a method would attempt to model diffusion processes, such as those mentioned above, in terms of an average *diffusion coefficient* which may be a constant or a simple function of the material properties. The mechanisms by which the diffusion proceeds, e.g. vacancy assisted [84], are not of interest. Atomistic models of diffusion *do* exist and are generally centred around a Monte Carlo simulation of the movement of individual atoms within the crystal (see, e.g. [85]).

For the purpose of this present work, solutions are demonstrated within semiconductor heterostructures constructed of an alloy $A_{1-x}B_xC$, where the system can be most generally represented by the z -dependence of the alloy component, i.e. $x = x(z)$. The methods developed are equally applicable to a dopant distribution whose profile may be $c(z)$, as labelled above.

4.2 THEORY

As mentioned above, the *most general* one-dimensional diffusion equation for a diffusant distribution represented by $x(z)$ is given by Fick's second law for non-steady

state diffusion [81]:

$$\frac{\partial x}{\partial t} = \frac{\partial}{\partial z} \left(\mathcal{D} \frac{\partial x}{\partial z} \right) \quad (4.5)$$

where t is the time and the diffusion coefficient \mathcal{D} could have temporal t , spatial z and concentration x dependencies, i.e. $\mathcal{D} = \mathcal{D}(x, z, t)$. Given this, then the derivative with respect to z operates on both factors, resulting in the following:

$$\frac{\partial x}{\partial t} = \frac{\partial \mathcal{D}}{\partial z} \frac{\partial x}{\partial z} + \mathcal{D} \frac{\partial^2 x}{\partial z^2} \quad (4.6)$$

which is a second-order ($\partial^2/\partial z^2$) non-linear ($(\partial/\partial z)^2$) differential equation. In any given problem it is likely that two of the following unknowns will be known (!):

- The initial diffusant profile, $x(z, t = 0)$;
- The final diffusant profile, $x(z, t)$;
- The diffusion coefficient, $\mathcal{D} = \mathcal{D}(x, z, t)$.

The problem will be to deduce the third unknown. This could manifest itself in several ways:

- a. given the initial diffusant profile and the diffusion coefficient, predict the diffusant profile a certain time into the future;
- b. given the initial and final diffusant profiles, calculate the diffusion coefficient;
- c. given the final diffusant profile and the diffusion coefficient, calculate the initial diffusant profile.

Knowing the versatility achieved by the *numerical* shooting method solution to Schrödinger's equation as discussed in the previous chapter, then clearly a numerical solution would again be favourable. Learning from the benefits of expanding the derivatives in the Schrödinger equation with finite differences, this would then appear to offer a promising way forward.

Recall the finite difference approximations to first and second derivatives, i.e.

$$\frac{\partial f}{\partial z} \approx \frac{f(z + \delta z) - f(z - \delta z)}{2\delta z} \quad (4.7)$$

and

$$\frac{\partial^2 f}{\partial z^2} \approx \frac{f(z + \delta z) - 2f(z) + f(z - \delta z)}{(\delta z)^2} \quad (4.8)$$

Then equation (4.6) can be expanded to give:

$$\frac{x(z, t + \delta t) - x(z, t)}{\delta t} = \left[\frac{\mathcal{D}(x, z + \delta z, t) - \mathcal{D}(x, z - \delta z, t)}{2\delta z} \right]$$

$$\times \left[\frac{x(z + \delta z, t) - x(z - \delta z, t)}{2\delta z} \right] \\ + \mathcal{D}(x, z, t) \left[\frac{x(z + \delta z, t) - 2x(z, t) + x(z - \delta z, t)}{(\delta z)^2} \right] \quad (4.9)$$

Notice that the derivative with respect to time has been written as:

$$\frac{x(z, t + \delta t) - x(z, t)}{\delta t}, \quad \text{rather than} \quad \frac{x(z, t + \delta t) - x(z, t - \delta t)}{2\delta t} \quad (4.10)$$

as would be expected from the expansion in equation (4.7). In fact, in this case where this is the only time derivative, the two are equivalent.

Assuming the most common class of problem, as highlighted above, as point (a), namely that the function $x(z, t)$ is known when $t = 0$, i.e. it is simply the initial profile of the diffusant, and the diffusion coefficient \mathcal{D} is fully prescribed, then it is apparent from equation (4.9) that the concentration x at any point z can be calculated a short time interval δt into the future, provided that the concentration x is known at small spatial steps δz either side of z . This approach to the solution of the differential equation is known as a *numerical simulation*. It is not a mathematical solution, but rather a computational scheme which has been derived to mirror the physical process.

It has already been mentioned that the diffusion coefficient \mathcal{D} could be a function of x , z , and t , with the form of \mathcal{D} being used to define the class of diffusion problem, e.g.

- i. $\mathcal{D} = D_0$, a constant, for simple diffusion problems.
- ii. $\mathcal{D} = D(x)$, a function of the concentration as encountered in non-linear diffusion problems [86]. Note, that as $x = x(z)$ then \mathcal{D} is intrinsically a function of position too.
- iii. $\mathcal{D} = D(z)$, a function of position only, as could occur in ion implantation problems [87]. Here the diffusion coefficient could be linearly dependent on the concentration of vacancies for example, where the latter itself is depth dependent.
- iv. $\mathcal{D} = D(t)$, a function of time, as could occur during the annealing of radiation damage. For example, ion implantation can produce vacancies which aid diffusion [88, 89]. During an anneal, the vacancy concentration decreases as the lattice is repaired, which in turn alters the diffusion coefficient.

4.3 BOUNDARY CONDITIONS

Thus, given the initial diffusant profile and a fully prescribed diffusion coefficient, everything is in place for predicting the profile of the diffusant at any time in the future, except for the conditions at the ends of the system. These can not be calculated with the iterative equation (equation (4.9)), as the equation requires points which lie outside the z -domain.

For diffusion from an infinite source, it may be appropriate to fix the diffusant concentration x at the two end points, e.g. $x(z = 0, t) = x(z = 0, t = 0)$. Alternatively, the concentrations x at the limits of the z -domain could be set equal to the adjacent points which can be deduced from equation (4.9). Physically this defines the semiconductor structure as a closed system, with the total amount of diffusant remaining the same. It is these latter 'closed system' boundary conditions which will be employed exclusively in the following examples.

4.4 CONVERGENCE TESTS

Figure 4.3 shows the result of allowing the diffusant profile in Fig. 4.1 to evolve to equilibrium, using the closed-system boundary conditions as described above. Clearly the 'closed' nature of the system can be seen—the total amount of diffusant remains the same, and ultimately as would be expected for the 'water step', the concentration reaches a constant value. In the case of water, this could be looked upon as minimising the potential energy.

If the diffusion process can be described by a constant diffusion coefficient, $\mathcal{D} = D_0$, then the general diffusion equation, equation (4.5), and its equivalent computation form, (equation (4.9)) simplifies to the following:

$$\frac{\partial x}{\partial t} = D_0 \frac{\partial^2 x}{\partial z^2} \quad (4.11)$$

which as mentioned above has error-function solutions for the case of diffusion at the interface ($z=0$) of a semi-infinite slab of concentration x_0 [81], i.e.

$$x(z) = \frac{x_0}{2} \operatorname{erfc} \frac{z}{2\sqrt{D_0 t}} \quad (4.12)$$

where erfc is the complementary error function see reference [38], p. 295. This technique can be imposed on multiple heterojunctions by linearly superposing solutions [77]. Fig. 4.4 compares just such an error function solution* with the numerical solution for a 200 Å single GaAs quantum well surrounded by 200 Å $\text{Ga}_{0.9}\text{Al}_{0.1}\text{As}$ barriers after 100 s of diffusion described by a constant coefficient $D_0 = 10 \text{ Å}^2 \text{ s}^{-1}$. Clearly, the numerical method advocated here exactly reproduces the analytical solution.

*Thanks are expressed here to T. Stirner for providing the error function data

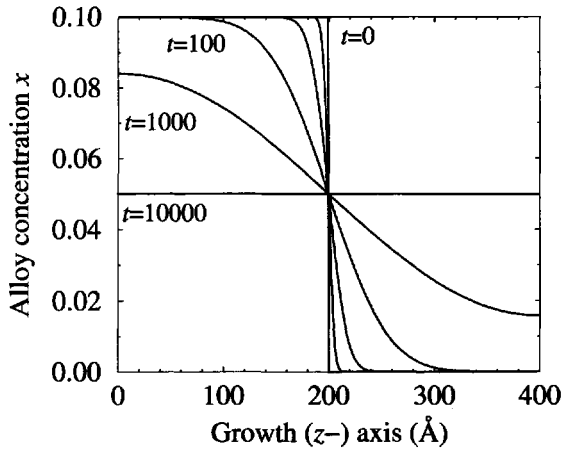


Figure 4.3 Time evolution of adjacent finite slabs of $\text{Ga}_{0.9}\text{Al}_{0.1}\text{As}/\text{GaAs}$ with a constant diffusion coefficient of $D = D_0 = 10 \text{ \AA}^2 \text{ s}^{-1}$

As with any numerical method, convergence tests must be performed. In the present case, these are necessary in order to determine that the results are independent of the intervals δz and δt . For the characteristic dimensions of interest in the following problems, namely nanostructures (~ 100 – 1000 \AA) and macroscopic annealing times (~ 100 – 1000 s), $\delta z = 1 \text{ \AA}$ and $\delta t = 0.01 \text{ s}$ were found to be satisfactory.

From a computational viewpoint, the technique is useful since, in general, it is anticipated that the numerical complexity scales linearly with the physical complexity. In actuality, it has been found that for the series of problems discussed below, the number of spatial points required for the calculation is independent of the variability in the initial concentration profile. Increasingly larger microstructures, i.e. superlattices with more and more periods, do require more points (i.e. δz remains constant), but the computational effort required is proportional to the number of points. For a given structure, but with an increasingly complex dependence of the diffusion coefficient, e.g. D having a stronger functional dependence on the concentration x , then smaller time intervals δt are required. Again, however, halving the time interval merely doubles the necessary computational effort.

Having therefore established a numerical method for simulating diffusion and after validating it by comparison to the accepted analytical form for a special case, it is now an appropriate time to demonstrate the solution's versatility by visiting, in turn, the examples (i–iv) listed in Section 4.2.

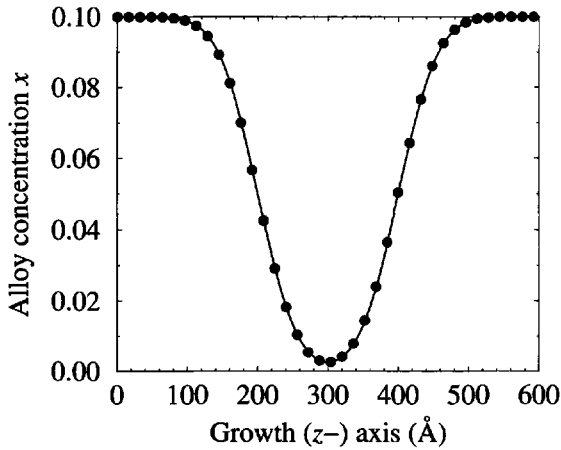


Figure 4.4 Comparison between the numerical solution (solid symbols) and the standard error-function solution (continuous line) for the concentration profile $x(z)$ of a 200 \AA $\text{Ga}_{0.9}\text{Al}_{0.1}\text{As}/200 \text{ \AA}$ $\text{GaAs}/200 \text{ \AA}$ $\text{Ga}_{0.9}\text{Al}_{0.1}\text{As}$ single quantum well, after 100 s of diffusion described by a constant diffusion coefficient $D_0 = 10 \text{ \AA}^2\text{s}^{-1}$

4.5 CONSTANT DIFFUSION COEFFICIENTS

The case of a constant diffusion coefficient has already been touched upon in the last section, as its analytical solutions represent an important test for the numerical solution forwarded here. Such a numerical solution can be used to illustrate two specific points for this situation.

In particular, the error function solutions given by equation (4.12) imply *universality*, in that they are dependent upon the value of the product $D_0 t$ only, and not the individual values themselves. Therefore, a system with a high diffusion coefficient can give the same diffused profile after a short time as a system with a low diffusion coefficient which is allowed to diffuse for longer periods.

The continuous lines in Fig. 4.5 display the results of diffusing a single quantum well for 100 (bottom) and 1000 s (top curve) with $D_0 = 10 \text{ \AA}^2\text{s}^{-1}$. In comparison with this, the solid symbols are the results of diffusing the same initial system for just 10 s but with $D_0 = 100$ (circles) and $1000 \text{ \AA}^2\text{s}^{-1}$ (squares). Clearly the numerical solution reproduces the universality as well—a point which is not at all obvious from the numerical form of the diffusion equation in equation (4.9).

Fig. 4.6 displays a succession of diffusant profiles for the same single quantum well as before, again with $D_0 = 10 \text{ \AA}^2\text{s}^{-1}$, where the evolution of the profile at longer diffusing times is clear. Therefore given a semiconductor heterostructure and an annealing time, it is apparent that the diffused profile can be predicted, *provided that*

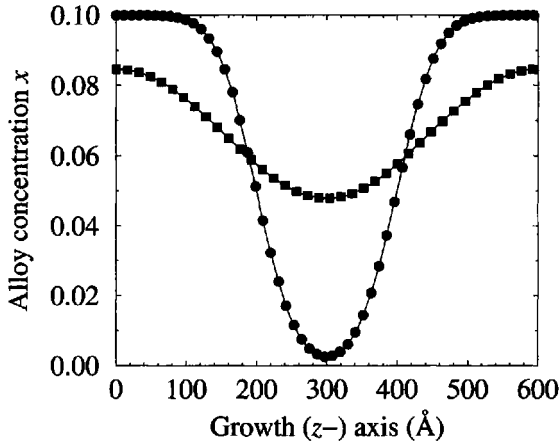


Figure 4.5 Diffusion profiles for a 200 Å $\text{Ga}_{0.9}\text{Al}_{0.1}\text{As}/200 \text{ Å GaAs}/200 \text{ Å Ga}_{0.9}\text{Al}_{0.1}\text{As}$ single quantum well, with constant diffusion coefficient D_0 , and $D_0t=1000 \text{ Å}^2\text{s}^{-1}$ (continuous line plus solid circles) and $D_0t=10000 \text{ Å}^2\text{s}^{-1}$ (continuous line plus solid squares)

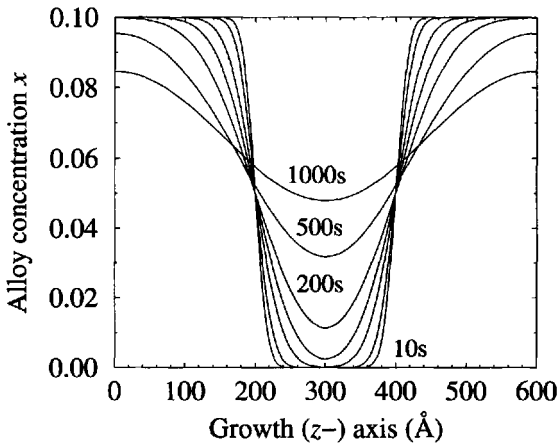


Figure 4.6 Diffusion profiles for the single quantum well of Fig. 4.5 with constant diffusion coefficient $D_0=10 \text{ Å}^2\text{s}^{-1}$, and diffusing times $t=10, 20, 50, 100, 200, 500$, and 1000 s

the diffusion coefficient characterising the process is known from other experiments. The direct determination of the diffused profile on such short length scales is difficult (although not impossible) from traditional techniques used for bulk analysis, such as *Secondary Ion Mass Spectroscopy* (SIMS) (see reference [7], Section 2.4). However, alternative spectroscopic techniques can be used to infer the extent of diffusion by its effect on other observables. In particular, as mentioned in Chapter 3, photolumi-

nescence and photoluminescence excitation can be used to measure the excitation energy of electrons across the band gap. As the electron and hole confinement energies, E_e and E_h , are dependent upon the band-edge profiles of the conduction- and valence-band edges, respectively, the excitation energy ($E = E_g + E_e + E_h$, ignoring the exciton binding energy), is then also dependent upon the amount of diffusion.

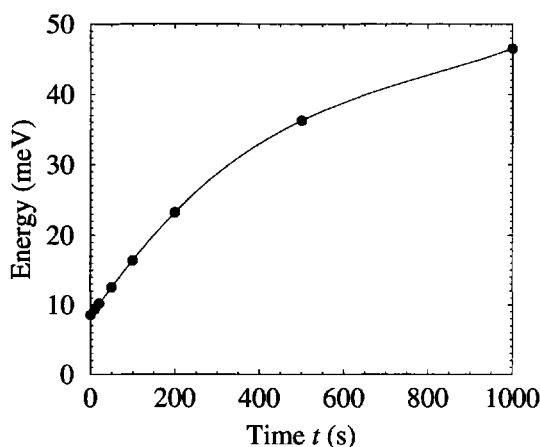


Figure 4.7 The corresponding ground state electron energies for the diffused profiles given in Fig. 4.6

The curve in Fig. 4.7 illustrates the change in the electron energy corresponding to the diffusion process of Fig. 4.6. The magnitude of the variation is substantial and easily detectable by experiment [78–80, 90–92]. For a theoretical interpretation of such experiments, *including* the change in exciton binding energy, see [77].

4.6 CONCENTRATION DEPENDENT DIFFUSION COEFFICIENT

There are many examples for which the diffusion coefficient of a species is a function of the concentration of that species itself [83, 93, 94].

The work by Tuck is perhaps the most interesting as it makes a detailed comparison with experiment. Tuck considered the diffusant of a dopant, Zn in bulk GaAs, a well studied system in which the ‘substitutional-interstitial’ mechanism is thought to dominate [95]. In this system the dopant (Zn) sits at two different positions within the crystal lattice, i.e. as the name implies both substitutionally and at interstitial sites. Atoms on the substitutional sites, which are in the majority, diffuse very slowly; however, the minority of interstitial atoms diffuse so readily that they dominate the evolution of the dopant profile with time.

The diffusion coefficient describing the process is concentration (c) dependent, and is given mathematically as:

$$D = kc^n \quad (4.13)$$

hence the diffusion equation becomes:

$$\frac{\partial c}{\partial t} = \frac{\partial}{\partial z} \left(k c^n \frac{\partial c}{\partial z} \right) \quad (4.14)$$

The constant k was deduced from experiment, and the index n , for this system, was found to be 2. Tuck considered several situations and implemented solutions which agreed very well with experiment. In this work, the numerical simulation can be implemented directly.

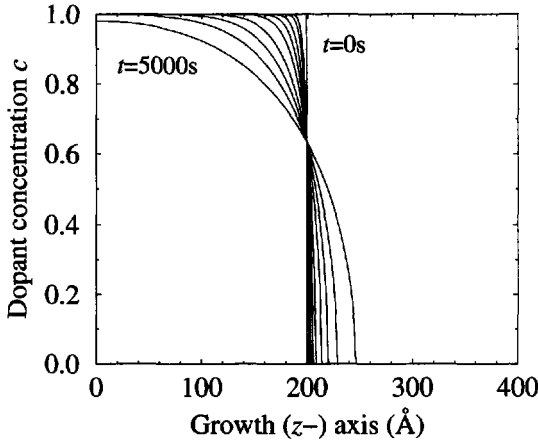


Figure 4.8 Concentration dependent ($\mathcal{D} = kc^2$) diffusion

Figure 4.8 presents examples of the concentration profiles that such a concentration-dependent diffusion coefficient yields. The initial concentration c was just taken to be unity and k was taken as $1 \text{ Å}^2\text{s}^{-1}$. Note that unlike the case of the constant diffusion coefficient, the curves do not all pass through the midpoint of the initial profile height, as shown in Fig. 4.3. Most importantly, the curves are of the same form as those given in Tuck [83] and hence agree with experiment.

4.7 DEPTH DEPENDENT DIFFUSION COEFFICIENT

As mentioned above, diffusion is becoming widely used as a post-growth method for fine tuning the structural, and hence electronic and optical properties of semiconductor heterostructures. Ion implantation (see Kelly [7], Section 3.4 for an introduction) is one method of stimulating diffusion; it is controllable, reproducible and its spatial resolution can be used to pattern a semiconductor wafer. Ion implantation can also be used as a way of introducing new species into a crystal, e.g. optically active rare earth ions into a host semiconductor [96]. However, the focus of interest here is in ion implantation as a means of enhancing the diffusion of the material species already

present [89,90,92]. The implantation itself produces lattice damage, displacing atoms from their equilibrium positions and thus creating interstitial atoms and vacancies. As in the previous example, the interstitial atoms can diffuse very readily and in addition the presence of vacancies provides a route for the diffusion of substitutional species.

The diffusion coefficient describes the average speed by which atoms diffuse, and clearly the more vacancies and interstitials in a given region of material, then the faster diffusion will proceed. Hence, in order to attempt to quantify the diffusion coefficient, it is necessary to have a knowledge of the amount of lattice damage that an implantation produces; this can be calculated with a Monte Carlo simulation of the atoms impinging on the crystal planes. Such computer simulations, known as T.R.I.M. codes, are well understood and documented (see for example [97]). An example of the output from such a simulation is given in Fig. 4.9.

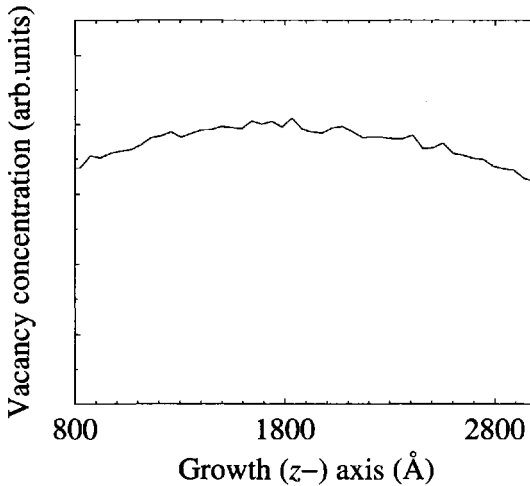


Figure 4.9 Typical depth dependency of vacancy concentration given by T.R.I.M. code data for 200 keV Ar^{2+} ions impinging on a [001] CdTe surface

The effect of ‘vacancy-assisted’ diffusion from such a profile has been calculated for a $\text{CdTe}/\text{Cd}_{1-x}\text{Mn}_x\text{Te}$ multiple quantum well [87,98]. In this work, the depth dependence will be exaggerated merely to produce a more interesting variation in the diffusion. The depth dependence of the vacancy concentration could actually be described *empirically* by a Gaussian distribution of the form:

$$\rho^{\text{vacancy}} = \rho_0^{\text{vacancy}} \exp \left[-\frac{(z - z_0)^2}{2\sigma^2} \right] \quad (4.15)$$

where z_0 is the depth of the maximum, e.g. $z_0 \sim 1800 \text{ Å}$ in Fig. 4.9, and, of course, the standard deviation σ is related to the width of the distribution.

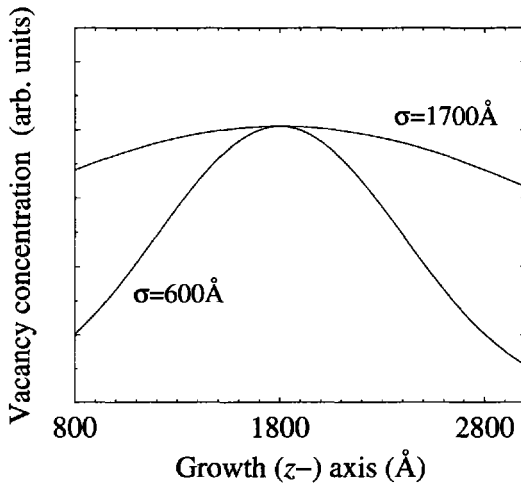


Figure 4.10 Gaussian fit to the T.R.I.M. code data of Fig. 4.9 together with an exaggerated depth dependence

Figure 4.10 displays the Gaussian fit to the T.R.I.M. data of Fig. 4.9, described by the parameter $\sigma=1700 \text{ Å}$, together with the modified data for the purpose of this example, given by $\sigma=600 \text{ Å}$.

In real systems where intensive investigations for a particular semiconductor multilayer with a particular ion implantation dosage are carried out, it would be necessary to relate the diffusion coefficient to the absolute vacancy concentration; however, in these present demonstrator examples of the numerical solution to the diffusion equation, it suffices to say let \mathcal{D} be proportional to ρ^{vacancy} , and furthermore, choose the constant of proportionality to be $10 \text{ Å}^2 \text{ s}^{-1}$, i.e.

$$\mathcal{D} = 10 \exp \left[-\frac{(z - z_0)^2}{2\sigma^2} \right] \text{Å}^2 \text{s}^{-1} \quad (4.16)$$

Given this, Fig. 4.11 displays both the initial and final alloy concentration profile after 200 s of diffusion described by equation (4.16) for a generic $150 \text{ Å AC}/50 \text{ Å A}_{1-x}\text{B}_x\text{C}$ superlattice/multiple quantum well. As expected, the central wells have diffused considerably more than those near the edges where the diffusion coefficient is lower. The exaggerated z -dependence of \mathcal{D} has fulfilled its goal in this illustration of producing a much clearer depth dependence than previously published for a realistic system (see for example [98]).

Photoluminescence measurements on diffused systems such as those shown in Fig. 4.11 would exhibit a broadened emission line, as the photogenerated carriers in the central wells would have a different energy to those in the outer wells. This

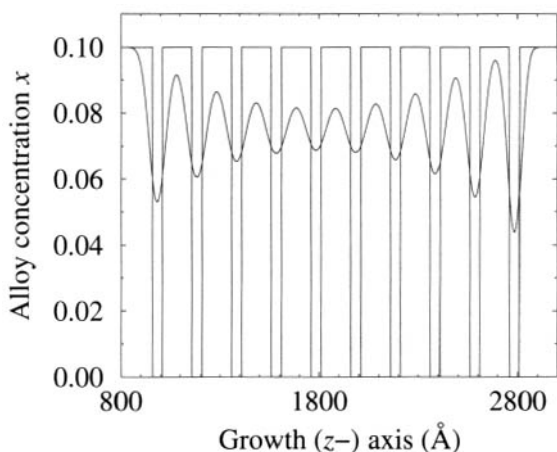


Figure 4.11 Alloy concentration profile x after 200 s of diffusion described by the depth dependent coefficient of equation (4.16)

phenomenon of line broadening in a diffused superlattice has been observed by Elman *et al.* [90] and subsequently modelled theoretically [98].

Alternatively, the structure, including the depth dependence of the diffusion, can be mapped directly by using *double-crystal X-ray diffraction* (DCXRD) [99], a technique which has showed itself to be very valuable in monitoring the progress of interface intermixing [89].

4.8 TIME DEPENDENT DIFFUSION COEFFICIENT

Vacancy-enhanced diffusion will continue for as long as the vacancies are present, but one way to control their lifetime and ‘freeze’ the diffusant profile is to anneal out the radiation damage. The annealing process thermally activates the interstitials back into vacancies and so restores order to the crystal. This subsequent annealing process *could* be described by a time-dependent diffusion coefficient, perhaps of the form:

$$\mathcal{D} = \mathcal{D}(z) \exp\left(-\frac{t}{\tau}\right) \quad (4.17)$$

thus presenting the opportunity to complete the examples of the functional dependencies listed in Section 4.2. $\mathcal{D}(z)$ is the initial (time-independent) depth-dependent diffusion coefficient due to the vacancy distribution. Using the form in equation (4.16), the diffusion coefficient would then be fully specified by:

$$\mathcal{D} = 10 \exp\left[-\frac{(z - z_0)^2}{2\sigma^2}\right] \exp\left(-\frac{t}{\tau}\right) \text{Å}^2 \text{s}^{-1} \quad (4.18)$$

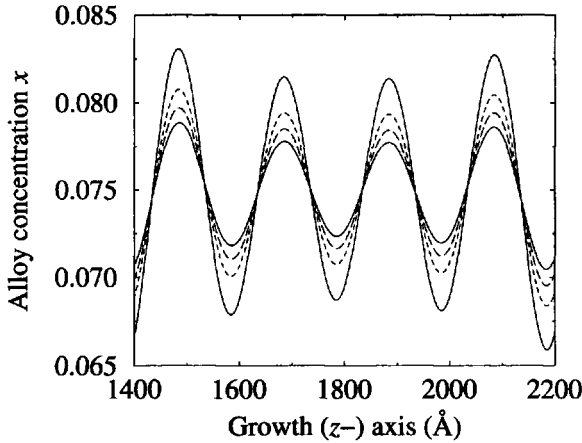


Figure 4.12 Annealing out radiation damage—an example of time-dependent diffusion, with annealing times of 0, 50, 100, and 200 s giving the concentration profiles of *decreasing* amplitude

Making use of the ion-implantation-enhanced diffusion profile of Fig. 4.11 as a starting point, Fig. 4.12 shows the results of simulating annealing out the lattice damage with a time-dependent diffusion coefficient of the form shown in equation (4.18). In this case, the decay time τ of the vacancy concentration was taken to be 100 s, and so the curves represent the points at which the vacancy concentration is a fraction 1, $e^{-\frac{1}{2}}$, e^{-1} , and e^{-2} of its original value. Clearly the curves are converging to a point which represents the region at which the ion-implantation-enhanced diffusion has been frozen.

4.9 δ -DOPED QUANTUM WELLS

In the previous sections, examples have been given of simulating diffusion for all of the various forms of diffusion coefficient that can exist. It is clear that the computational method can be extended to include combinations of all three dependencies, and indeed in the final example the diffusion coefficient had both depth and time dependency. It now serves a purpose to follow through an example of a diffusion problem which is of direct relevance to semiconductor heterostructures, and to calculate the subsequent effects on an observable, in this case, the quantum-confinement energies.

Contemporary epitaxial growth techniques, such as molecular beam epitaxy (MBE) and chemical beam epitaxy (CBE) etc., allow for the possibility of growing very thin layers of semiconductor material. Another possibility, is the potential for these techniques to lay down very thin layers of dopant atoms. Fig. 4.13(a) represents bulk doping of a layer, as used in HEMTs and the majority of semiconductor heterostructure devices. Fig. 4.13(b) represents the dopant profile for a single δ -layer in a quantum

well, although of course, it could be in a barrier or at an interface. Such thin layers of dopant are called *delta-layers*, as the dopant profile resembles the Dirac δ -function, and the whole process is known as δ -doping; for a comprehensive treatise of this technique see Schubert [100].

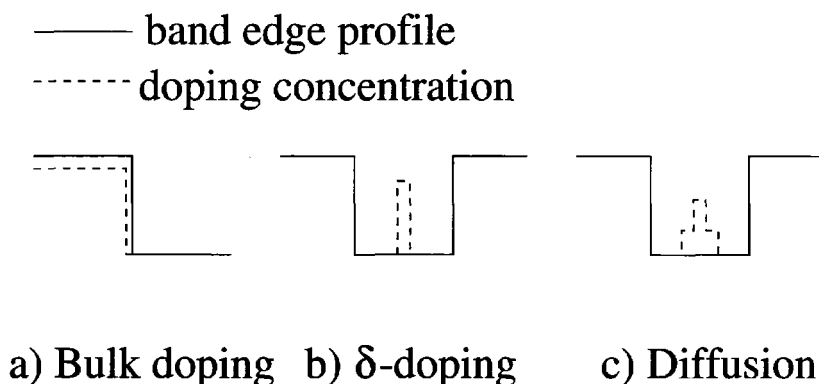


Figure 4.13 Doping and δ -doping

One of the main problems affecting δ -doped layers is diffusion of the dopant which can often occur during the high temperatures employed during growth. A δ -layer designed to extend over just one monolayer can in fact spread over several layers. This would affect the electronic and optical properties of the device, in particular the electron- or hole-confinement energies, and scattering from the distribution of ionised impurities would be different. In this section, concentration will be focused on the first of these points, as the techniques to model such a system have already been covered in this work.

Figure 4.14 displays the dopant profile of a single-monolayer-thick δ -layer when subject to diffusion described by a constant coefficient $D_0 = 1 \text{ \AA}^2 \text{ s}^{-1}$. Clearly, the total amount of dopant remains constant, and hence as the profile broadens, the height of the peak decreases. If the dopant were *p*-type, e.g. Beryllium (Be) in GaAs, then at elevated temperatures the acceptors would be ionised and donate holes which would become localised in the quantum well. By using the methods outlined in Chapter 3, the Coulomb potential due to this negative acceptor/positive hole distribution can be calculated and incorporated in the self-consistent solution of Schrödinger's and Poisson's equations.

Assuming that all of the holes occupy the lowest heavy-hole subband in the quantum well, then Fig. 4.15 shows the self-consistent potential V_p due to the charge distribution for various diffusion times. Clearly, as the δ -doped layer broadens, then the depth of the potential decreases. Interestingly, the potential seems to tend towards zero; this is because the diffused acceptor distribution begins to resemble the self-consistent hole wave functions thus leading to an almost complete cancellation of the positive and negative potentials.

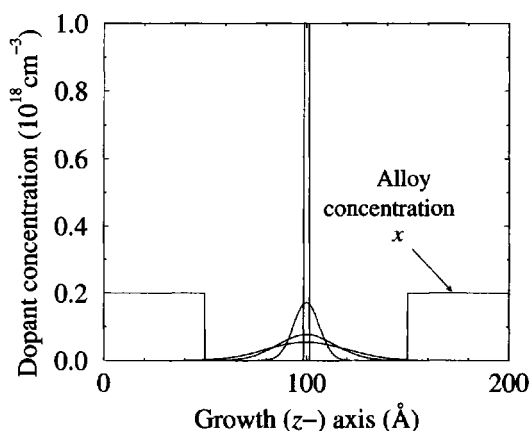


Figure 4.14 Dopant profile after 0, 20, 100 and 200 s (in order of decreasing maximum height) of diffusion described by a constant diffusion coefficient of $1 \text{ Å}^2\text{s}^{-1}$, together with the alloy concentration x of the $\text{Ga}_{1-x}\text{Al}_x\text{As}/\text{GaAs}/\text{Ga}_{1-x}\text{Al}_x\text{As}$ single quantum well

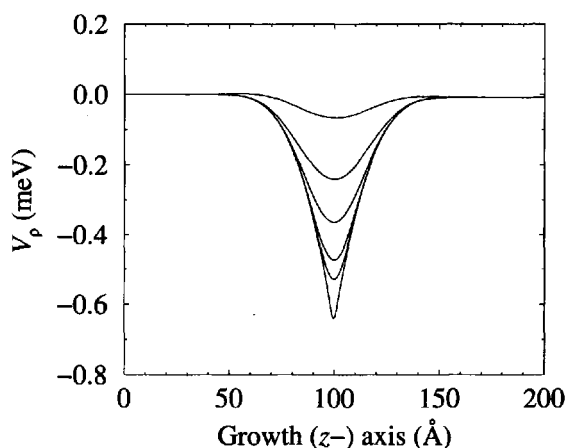


Figure 4.15 The self-consistent potential V_p due to the acceptor/hole distribution in the δ -doped quantum well of Fig. 4.14, for diffusion times, in order of decreasing depth, of 0, 10, 20, 50, 100, and 200 s

Figure 4.16 displays the self-consistent heavy-hole energy as a function of the diffusion time. At this relatively low doping density, the effect of diffusion on the heavy-hole energy is small; nonetheless, this example serves the purpose of demonstrating the applicability of the diffusion equation solution to systems of interest for devices, while the inclusion of the self-consistent Schrödinger's and Poisson's solu-

tion, shows the power of combining such techniques together. Note that as a constant diffusion coefficient was employed, then all of the curves are universal in $D_0 t$.

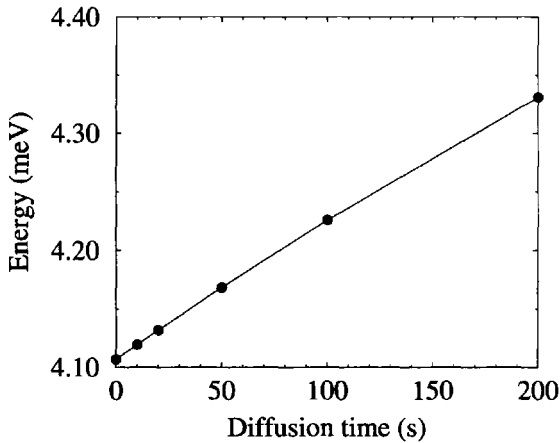


Figure 4.16 The self-consistent heavy-hole energy as a function of the diffusion time

4.10 EXTENSION TO HIGHER DIMENSIONS

The one-dimensional diffusion equation can be generalised to higher dimensions [81], i.e.

$$\frac{\partial x}{\partial t} = \nabla \cdot (\mathcal{D} \nabla x) \quad (4.19)$$

This can again be expanded in terms of finite differences to give an iterative equation in analogy with equation (4.9). This time, however, the concentration at a point for a time interval δt into the future, is dependent on the diffusion coefficient \mathcal{D} and the concentration x , which may both be functions of all of the spatial coordinates. Therefore, an iteration needs to be performed for each spatial dimension. Following this, an important check must be made that the resulting new concentration is independent of the order in which the iterations were performed.

CHAPTER 5

IMPURITIES

5.1 DONORS AND ACCEPTORS IN BULK MATERIAL

The ability to introduce impurities directly into the lattice was the most important technological advance in the development of semiconductors as electronic materials. This allows their electronic properties to be tailored to suit the engineer's needs. Atoms that are introduced with an additional electron more than that required to form the chemical bonds with neighbouring atoms are easily ionised, thus donating electrons to the crystal. Such atoms are known as donors. Alternatively, impurity atoms can be incorporated into the lattice which are an electron short of that needed to form the chemical bonds. These atoms are able to accept electron from nearby bonds, thus generating an empty state in the valence band. Atoms of this type are known as acceptors. This *hole* is then mobile within the lattice and can contribute to the conductivity. Doping different regions of the same lattice with donors and acceptors leads to a *p-n* junction, which was the basis of the first transistor—the bipolar.

The interest here is not with the transport properties of doped semiconductors and their exploitation as the electronic materials of commercial devices such as diodes and transistors, as this has been covered very successfully by many other authors, see

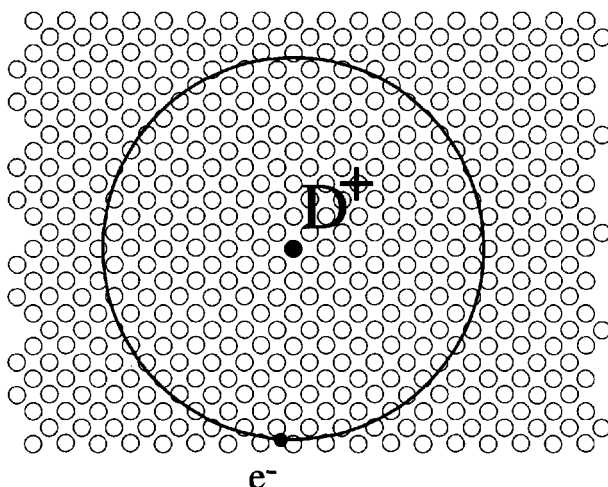


Figure 5.1 Schematic representation of a donor in a bulk semiconductor

for example, Sze [101]. The interest here *is* in the properties of impurity states within semiconductor heterostructures. These states have a bearing on the electronic and optical properties of these materials, and again it is the evaluation of the fundamentals governing their behaviour that is the primary concern.

Figure 5.1 shows a schematic representation of a neutral (occupied) donor in a bulk semiconductor. The regular lattice represents a $\{100\}$ plane. An example of a donor in silicon is the group V element phosphorus. In a compound semiconductor such as gallium arsenide or cadmium telluride, the dopant would occupy a cation site. Thus the atoms of Fig. 5.1 would be gallium or cadmium respectively. In GaAs, a typical donor is silicon, where the latter is forced to sit on the Ga site by growth under an As overpressure. The group IV element Si has one electron more than the group III element Ga and it donates this to the lattice. Note that a Si atom occupying an As site would be an acceptor.

It can be seen from the figure that the unionised donor resembles the electron–proton pair constituting a hydrogen atom; the physics of this is well understood and was initially described with a semi-classical model by Bohr [4] and reinforced later with full quantum mechanical descriptions [3, 29].

The binding energy of this neutral donor–electron pair is given by the following:

$$E_{D^0} = -\frac{m_e^* e^4}{32\pi^2 \hbar^2 \epsilon_r^2 \epsilon_0^2} \quad (5.1)$$

and the Bohr radius follows as:

$$\lambda = \frac{4\pi\epsilon_r\epsilon_0\hbar^2}{m_e^* e^2} \quad (5.2)$$

In this case the mass of the donor atom is much larger than the electron mass and hence the simpler 'infinite nucleus mass' variant can be employed. Later, in the chapter dealing with excitons (Chapter 6), this hydrogenic model will again be employed; however, on that occasion the hole mass is of the same order of magnitude as the electron mass and therefore cannot be ignored. Note that the only material parameters needed are the electron effective mass and the permittivity of the host material, and thus the neutral donor binding energy should be independent of the impurity atom. Experimental data reproduced by Sze [44] substantiate this result, for the majority of cases in GaAs.

Taking typical values for GaAs, e.g. the Γ valley electron effective mass $m_e^*=0.067m_0$ and the static dielectric constant $\epsilon_r=13.18$ [14], then:

$$E_{D^0} = -5.3 \text{ meV} \quad \text{and} \quad \lambda = 103 \text{ \AA}$$

which is very close to the measured value of 5.8 meV [44]. Correspondingly for CdTe, $m_e^*=0.096m_0$ and $\epsilon_r=10.6$, and therefore:

$$E_{D^0} = -11.7 \text{ meV} \quad \text{and} \quad \lambda = 58 \text{ \AA}$$

The Bohr radii produced by these simple calculations represent an important result. The lattice constant of GaAs is 5.65 Å, and hence 103 Å represents around 18 unit cells along any radius from the donor to the electron orbit. Or, alternatively, each face-centred-cubic unit cell contains four lattice points, or eight atoms, and has a volume of $A_0^3=(5.65 \text{ \AA})^3$, and hence the volume occupied by one atom is $A_0^3/8$. Thus the number of atoms within the spherical electron orbit of radius λ is given by:

$$\text{number of atoms} = \frac{4\pi\lambda^3/3}{A_0^3/8} \quad (5.3)$$

which for GaAs gives, 203,000 atoms. This seems like quite a large number and hence provides justification for using the bulk value of the permittivity ϵ_r . For systems with a much smaller Bohr radius, which might contain substantially fewer atoms, it would be apparent that the electromagnetic properties of the crystal are quite different from the bulk and care must be taken in choosing the value for the permittivity. A recent work [102], has shown by careful comparison of exciton binding calculations with detailed experimental work, that the permittivity required to produce agreement lies between the static, ϵ_s , and infinite, ϵ_∞ , frequency values. The binding energies themselves are small relative to the bandgap and, obviously, negative, which implies that they lie just below the conduction-band edge.

At low temperatures, there are few lattice vibrations (phonons) and hence the electrons remain bound to the donors. However, as the temperature is increased the number of phonons within the lattice increases and the donors can become ionised, thus liberating electrons into the conduction band of the crystal. The occupancy (proportion of ionised donors) can be represented by statistics [1, 2], but this is not of particular concern here as most experiments are performed at liquid helium temperatures where all of the donors can be considered occupied, or at room temperature where they can be considered ionised.

At first sight, the direct analogy of acceptors in *p*-type material would appear to be described by the above equations, but with the hole mass replacing the electron mass and the binding energy now referring to a hole bound to a negatively charged acceptor. The hydrogenic model for a hole bound to an acceptor would give

$$E_{A^0} = -\frac{m_h^* e^4}{32\pi^2 \hbar^2 \epsilon_r^2 \epsilon_0^2} \quad \text{and} \quad \lambda = \frac{4\pi \epsilon_r \epsilon_0 \hbar^2}{m_h^* e^2}$$

Taking the typical values for GaAs of $m_{hh}=0.62 m_0$ and $\epsilon_r=13.18$ again, then:

$$E_{A^0} = -49.0 \text{ meV} \quad \text{and} \quad \lambda = 11 \text{ \AA}$$

Unlike the donor case, this doesn't agree well with experimentally measured values [44]. In fact, in practice the acceptor state is much more complex for a number of reasons. First, often the valence band at the Γ minimum consists of two degenerate states (the light- and heavy-holes), and thus it is not clear which effective mass should be employed, or indeed whether the hole is a mixture of both light- and heavy-hole states. Although, however, this degeneracy is usually broken within a quantum well, due to the differing effect of the confining potential on the effective masses. Secondly, the much larger heavy-hole mass has the effect of producing a much smaller Bohr radius λ , i.e. the hole orbits much more closely to the central Coulombic potential, as can be seen from the calculated radius above. Consequently the approximation that the bulk relative permittivity ϵ describes the electromagnetic response of the lattice is questionable, although a technique for accounting for this problem will be introduced later, in Section 5.6. The problem of point defects will be revisited much later in Chapter 11 and dealt with by a microscopic model which will take these effects into account. For now concentration will be focused on donors.

5.2 BINDING ENERGY IN A HETEROSTRUCTURE

When a donor is placed within a quantum well structure, the situation is considerably more complex than in the bulk, due to two additional degrees of freedom. First, the binding energy depends upon the confining potential due to the quantum well structure. In its simplest form this would be the well width, schematically represented in Fig. 5.2. Beyond this, however, lies the possibility of a more complex heterostructure. The donor binding energy will be different, e.g. in a double quantum well, a superlattice and a diffused quantum well; in fact, it is just as sensitive to structure as a lone electron is.

Secondly, the donor binding energy and wave function are also a function of the donor position *within* the heterostructure. The binding energy is, of course, different for a donor at the centre of a quantum well than it is for a donor at the edge of a well, as illustrated in Fig. 5.3.

Any theoretical study of the properties of neutral donors in a heterostructure necessitates solving the standard Schrödinger equation for the particular structure and

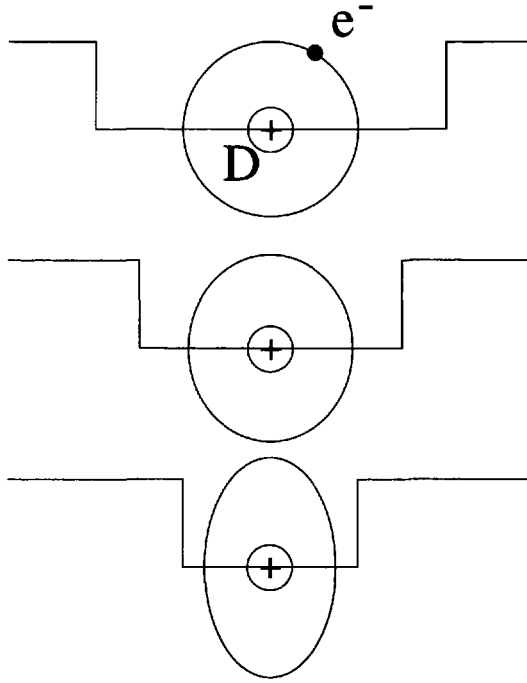


Figure 5.2 Schematic representation of the effect of quantum well width on the donor wave function

with the inclusion of the additional Coulombic term representing the donor potential. Within the envelope function and effective mass approximations, the Hamiltonian for an electron confined in a heterostructure and in the presence of a shallow donor is merely the standard Hamiltonian of earlier, plus an additional term due to the Coulombic interaction as follows:

$$\mathcal{H} = -\frac{\hbar^2}{2m^*} \nabla^2 + V(z) - \frac{e^2}{4\pi\epsilon r'} \quad (5.4)$$

In the interests of generality, the potential $V(z)$ describing the conduction-band-edge potential of the heterostructure will remain unspecified. The displacement between the electron and the donor is given by:

$$r'^2 = x^2 + y^2 + (z - r_d)^2 = x^2 + y^2 + z'^2 \quad (5.5)$$

where r_d is the position of the donor along the growth (z) direction; note the origin of the x - y plane has been defined on the donor atom for convenience. Note also, that for now the effective mass m^* has been assumed to be constant, which leads to simplifications in the analysis. The increased localisation of the electron within the well ensures that this approximation has little effect on the final numerical values. The extension to include the effects of a varying effective mass will be discussed later.

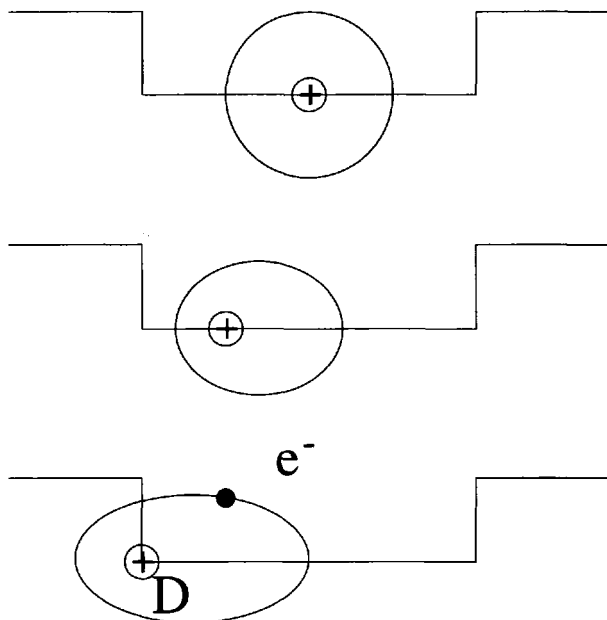


Figure 5.3 Schematic representation of the effect of donor position on the wave function

Methods of solution of the Schrödinger equation have centred around two basic approaches. The first of these involves expanding the electron wave function as a linear combination of Gaussian functions [103, 104]. While this technique has been successful in calculating the properties of donors in simple quantum well structures, the generalisation to more complex structures, including graded gap materials and systems where piezo electric fields are present, is non-trivial.

The other category of approach is based on the *variational principle*. In this method, a trial wave function is chosen whose functional form may contain one or more unknown parameters. These parameters are varied systematically and the expectation value of the energy calculated for each set. The variational principle [3, 29] states that *the lowest energy obtained is the closest approximation to the true state of the system*.

The success of variational approaches centres around the general choice of the trial wave function. A common choice is a product of two terms [105], i.e.

$$\Psi = \psi(z)e^{-\frac{r'}{\lambda}} \quad (5.6)$$

where r' is the electron-donor separation and λ is a variational parameter; thus the second factor is a simple hydrogenic wave function. The function $\psi(z)$ is the uncorrelated eigenfunction, i.e. the straightforward wave function calculated in Chapters 2 and 3, of the electron in the quantum well *without* the donor [106, 107].

The latter restriction can be removed and a much more general choice of the donor wave function Ψ can be made. The motivation being generality, the problem is recast in a form which is suitable for numerical solution as described in Chapter 3, thus making it applicable to any quantum well structure, be it a double quantum well, a diffused quantum well, a graded gap quantum well, etc. With this aim, the trial wave function to be employed here is influenced by the above, but for now will be written as follows:

$$\Psi = \chi(z)\xi(x, y, z - r_d) \quad (5.7)$$

The one-dimensional envelope function $\chi(z)$ is yet to be determined and the function $\xi(x, y, z - r_d)$ is expected to be a hydrogenic type factor which is a function of the spatial coordinates and one or more variational parameters. It can be seen that forming the Schrödinger equation with the Hamiltonian of equation (5.4) and the trial wave function Ψ of equation (5.7) will lead to a term $\nabla^2\Psi$; it is worthwhile to derive this now. Consider:

$$\frac{\partial^2\Psi}{\partial x^2} = \frac{\partial}{\partial x} \frac{\partial}{\partial x} [\chi(z)\xi(x, y, z - r_d)] \quad (5.8)$$

$$\therefore \frac{\partial^2\Psi}{\partial x^2} = \frac{\partial}{\partial x} \left[\chi(z) \frac{\partial \xi}{\partial x} \right] = \chi \frac{\partial^2 \xi}{\partial x^2} \quad (5.9)$$

and similarly for y . Consider now

$$\frac{\partial^2\Psi}{\partial z^2} = \frac{\partial}{\partial z} \left(\frac{\partial \chi}{\partial z} \xi + \chi \frac{\partial \xi}{\partial z} \right) \quad (5.10)$$

$$\therefore \frac{\partial^2\Psi}{\partial z^2} = \frac{\partial^2 \chi}{\partial z^2} \xi + 2 \frac{\partial \chi}{\partial z} \frac{\partial \xi}{\partial z} + \chi \frac{\partial^2 \xi}{\partial z^2} \quad (5.11)$$

Hence

$$\nabla^2\Psi = (\nabla_z^2 \chi) \xi + 2 \nabla_z \chi \nabla_z \xi + \chi \nabla^2 \xi \quad (5.12)$$

where the subscript z implies derivatives with respect to z only. Then, forming the Schrödinger equation with the Hamiltonian of equation (5.4) gives the following:

$$\begin{aligned} -\frac{\hbar^2}{2m^*} \{ [\nabla_z^2 \chi(z)] \xi + 2 \nabla_z \chi(z) \nabla_z \xi + \chi(z) \nabla^2 \xi \} - \frac{e^2}{4\pi\epsilon r'} \chi(z) \xi + V(z) \chi(z) \xi \\ = E \chi(z) \xi \end{aligned} \quad (5.13)$$

where E is the total energy of the system. The donor binding energy E_{D0} is equal to the difference between this and the standard confinement energy of the electron in the heterostructure without the donor present, i.e. using the notation of Chapter 2, the ground state energy for the electron in the well without the donor would be E_1 , and hence

$$E_{D0} = E - E_1 \quad (5.14)$$

In order to proceed further, it is necessary to be more specific about the form of the hydrogenic factor ξ . In a free hydrogen atom the ground state wave function would be spherically symmetric and given by:

$$e^{-\frac{r}{\lambda}}, \quad \text{where} \quad r^2 = x^2 + y^2 + z^2 \quad (5.15)$$

where λ is known as the Bohr radius. Indeed this is the form also used for donors in bulk materials. However, in heterostructures there is a loss of translational symmetry along the growth (z -) axis and hence it cannot be assumed that the spatial co-ordinate is spherically symmetric. For now, the hydrogenic term can be taken as:

$$\xi(x, y, z - r_d) = e^{-\frac{r''}{\lambda}} \quad (5.16)$$

where the spatial coordinate r'' is, of course, a function of x , y and $(z - r_d)$. Again, λ is referred to as the Bohr radius, but now it will be employed as a variational parameter in order to minimise the total energy of the system. Substituting for ξ into equation (5.13) gives:

$$\begin{aligned} -\frac{\hbar^2}{2m^*} \left\{ [\nabla_z^2 \chi(z)] e^{-\frac{r''}{\lambda}} + 2\nabla_z \chi(z) \nabla_z e^{-\frac{r''}{\lambda}} + \chi(z) \nabla^2 e^{-\frac{r''}{\lambda}} \right\} \\ - \frac{e^2}{4\pi\epsilon r'} \chi(z) e^{-\frac{r''}{\lambda}} + V(z) \chi(z) e^{-\frac{r''}{\lambda}} = E \chi(z) e^{-\frac{r''}{\lambda}} \end{aligned} \quad (5.17)$$

Multiplying by $e^{-\frac{r''}{\lambda}}$ and integrating over the x - y plane, leads to an equation of the form:

$$\begin{aligned} -\frac{\hbar^2}{2m^*} [\nabla_z^2 \chi(z) I_1 + 2\nabla_z \chi(z) I_2 + \chi(z) I_3] \\ - \frac{e^2}{4\pi\epsilon} \chi(z) I_4 + V(z) \chi(z) I_1 = E \chi(z) I_1 \end{aligned} \quad (5.18)$$

where the integrals I_j , ($j = 1, 2, 3, 4$) are defined as follows:

$$I_1 = \int_0^\infty \int_0^\infty e^{-\frac{2r''}{\lambda}} dx dy \quad (5.19)$$

$$I_2 = \int_0^\infty \int_0^\infty e^{-\frac{r''}{\lambda}} \nabla_z e^{-\frac{r''}{\lambda}} dx dy \quad (5.20)$$

$$I_3 = \int_0^\infty \int_0^\infty e^{-\frac{r''}{\lambda}} \nabla^2 e^{-\frac{r''}{\lambda}} dx dy \quad (5.21)$$

$$I_4 = \int_0^\infty \int_0^\infty \frac{e^{-\frac{2r''}{\lambda}}}{r'} dx dy \quad (5.22)$$

Returning to the Schrödinger equation (equation (5.18)) then:

$$\nabla_z^2 \chi(z) I_1 + 2\nabla_z \chi(z) I_2 + \chi(z) \left\{ I_3 + \frac{2m^*}{\hbar^2} \frac{e^2}{4\pi\epsilon} I_4 - \frac{2m^*}{\hbar^2} [V(z) - E] I_1 \right\} = 0 \quad (5.23)$$

Remembering that the integrals I_1 , I_2 , I_3 , and I_4 are just real numbers then equation (5.23) is just a linear second-order differential equation, very much like the one met in Chapter 3. The energy E can be solved for any choice of λ by expanding the derivatives in finite differences and forming an iterative shooting algorithm as before.

First though for simplicity sake, it is better to simplify equation (5.23), and rewrite it as:

$$\alpha \nabla_z^2 \chi(z) + \beta \nabla_z \chi(z) + \gamma \chi(z) = 0 \quad (5.24)$$

Then, by using the finite difference expansions:

$$\nabla_z \chi(z) = \frac{\chi(z + \delta z) - \chi(z - \delta z)}{2\delta z} \quad (5.25)$$

and

$$\nabla_z^2 \chi(z) = \frac{\chi(z + \delta z) - 2\chi(z) + \chi(z - \delta z)}{(\delta z)^2} \quad (5.26)$$

equation (5.24) becomes

$$\frac{\chi(z + \delta z) - 2\chi(z) + \chi(z - \delta z)}{(\delta z)^2} + \frac{\beta}{\alpha} \left[\frac{\chi(z + \delta z) - \chi(z - \delta z)}{2\delta z} \right] + \frac{\gamma}{\alpha} \chi(z) = 0 \quad (5.27)$$

Finally, gathering the terms in $\chi(z + \delta z)$ on the left-hand side gives:

$$\left(1 + \frac{\beta}{2\alpha} \delta z\right) \chi(z + \delta z) = \left(-1 + \frac{\beta}{2\alpha} \delta z\right) \chi(z - \delta z) + \left[2 - (\delta z)^2 \frac{\gamma}{\alpha}\right] \chi(z) \quad (5.28)$$

which is an iterative shooting equation and can be solved subject to the standard boundary conditions, $\chi(z) \rightarrow \infty$ as $z \rightarrow \pm\infty$. In practice, this involves choosing exponential-growth starting conditions deep inside the quantum barrier and then progressively calculating $\chi(z)$ at points along the z -axis across the structure. The energy E is varied until the function $\chi(z)$ satisfies the standard boundary condition, i.e. it tends to zero at the other end of the structure. For more details, see again Chapter 3.

The variational aspect of the calculation arises from the value of the unknown constant λ . This is varied systematically with the aim of minimising the total energy. The system coordinates are therefore represented by the minimum energy and the corresponding value of the Bohr radius λ .

The derivation thus far is independent of the choice of the form of the hydrogenic exponential term, $\exp(-r''/\lambda)$, and this summarizes the versatility of this approach. In the following sections, the various choices that can be made for this 'orbital' term will be investigated. In addition, the power of this method is represented by the generality of the conduction-band-edge potential $V(z)$. No assumptions have been made relating to the form of this function and hence the method is applicable to all forms of $V(z)$, i.e. the formalism can be applied to any heterostructure.

5.3 TWO-DIMENSIONAL TRIAL WAVE FUNCTION

The simplest choice that can be made for the hydrogenic factor of the trial wave function for an electron confined in a quantum well structure in the presence of a positively charged donor, is one which is dependent only upon an in-plane co-ordinate, say, for example, as follows:

$$r'' = \sqrt{x^2 + y^2} = r_{\perp} \quad (5.29)$$

Therefore:

$$\xi = \exp\left(-\frac{\sqrt{x^2 + y^2}}{\lambda}\right) = \exp\left(-\frac{r_{\perp}}{\lambda}\right) \quad (5.30)$$

This choice replaces the spherical symmetry perhaps expected, a priori, with circular symmetry and is often referred to as the two-dimensional (2D) form. These are an unfortunate choice of words as in fact ξ has cylindrical symmetry, and hence is infinitely extended along the z -axis. This is another example of the word 'dimension' being interchanged with 'coordinate'.

Given this specific form for the spatial coordinate r'' , then it is possible to calculate the integrals I_j explicitly, and hence all of the terms in the Schrödinger equation, (equation (5.23)) and the corresponding iterative shooting equation (5.28) will be known.

Consider first:

$$I_1 = \int_0^{\infty} \int_0^{\infty} e^{-\frac{2r_{\perp}}{\lambda}} dx dy \quad (5.31)$$

In this case, and indeed all that follows, it makes sense to exploit the circular symmetry by transforming the integral from the Cartesian coordinates, x and y , to plane polar coordinates r_{\perp} and θ , i.e.

$$I_1 = \int_0^{2\pi} \int_0^{\infty} e^{-\frac{2r_{\perp}}{\lambda}} r_{\perp} dr_{\perp} d\theta \quad (5.32)$$

As the integrand, and all that follow, have no angular dependences then the θ integral is trivial, i.e.

$$I_1 = 2\pi \int_0^{\infty} e^{-\frac{2r_{\perp}}{\lambda}} r_{\perp} dr_{\perp} \quad (5.33)$$

This can now be solved with integration by parts, remembering the standard formula:

$$\int_a^b u \frac{dv}{dx} dx = [uv]_a^b - \int_a^b v \frac{du}{dx} dx \quad (5.34)$$

and that the independent variable is r_{\perp} , i.e. $x = r_{\perp}$. By choosing:

$$u = r_{\perp} \quad \text{and} \quad \frac{dv}{dr_{\perp}} = e^{-\frac{2r_{\perp}}{\lambda}} \quad (5.35)$$

then:

$$\frac{du}{dr_{\perp}} = 1 \quad \text{and} \quad v = -\frac{\lambda}{2} e^{-\frac{2r_{\perp}}{\lambda}} \quad (5.36)$$

Substituting into equation (5.34) gives:

$$I_1 = 2\pi \left\{ \left[-\frac{\lambda r_{\perp}}{2} e^{-\frac{2r_{\perp}}{\lambda}} \right]_0^{\infty} - \int_0^{\infty} -\frac{\lambda}{2} e^{-\frac{2r_{\perp}}{\lambda}} dr_{\perp} \right\} \quad (5.37)$$

$$\therefore I_1 = 2\pi \left[\left(-\frac{\lambda r_{\perp}}{2} - \frac{\lambda^2}{4} \right) e^{-\frac{2r_{\perp}}{\lambda}} \right]_0^{\infty} \quad (5.38)$$

$$\therefore I_1 = 2\pi \frac{\lambda^2}{4} \quad (5.39)$$

Next, consider I_2 as defined in equation (5.20)

$$I_2 = \int_0^{\infty} \int_0^{\infty} e^{-\frac{r_{\perp}}{\lambda}} \nabla_z e^{-\frac{r_{\perp}}{\lambda}} dx dy \quad (5.40)$$

Immediately it can be seen, that with this particular choice of $r'' = r_{\perp}$, the derivative with respect to z is zero, i.e.

$$\nabla_z e^{-\frac{r_{\perp}}{\lambda}} = 0 \quad (5.41)$$

$$\therefore I_2 = 0 \quad (5.42)$$

Now, consider the third integral, as specified in equation (5.21):

$$I_3 = \int_0^{\infty} \int_0^{\infty} e^{-\frac{r_{\perp}}{\lambda}} \nabla^2 e^{-\frac{r_{\perp}}{\lambda}} dx dy \quad (5.43)$$

First it is necessary to calculate the terms arising from the ∇^2 , so consider:

$$\frac{\partial}{\partial x} e^{-\frac{r_{\perp}}{\lambda}} = \frac{\partial}{\partial r_{\perp}} e^{-\frac{r_{\perp}}{\lambda}} \frac{\partial r_{\perp}}{\partial x} = -\frac{1}{\lambda} e^{-\frac{r_{\perp}}{\lambda}} \frac{\partial}{\partial x} \sqrt{x^2 + y^2} = -\frac{x}{\lambda r_{\perp}} e^{-\frac{r_{\perp}}{\lambda}} \quad (5.44)$$

and then:

$$\frac{\partial^2}{\partial x^2} e^{-\frac{r_{\perp}}{\lambda}} = \frac{\partial}{\partial x} \left(-\frac{x}{\lambda r_{\perp}} e^{-\frac{r_{\perp}}{\lambda}} \right) \quad (5.45)$$

$$\therefore \frac{\partial^2}{\partial x^2} e^{-\frac{r_{\perp}}{\lambda}} = -\frac{1}{\lambda r_{\perp}} e^{-\frac{r_{\perp}}{\lambda}} - \frac{x}{\lambda} e^{-\frac{r_{\perp}}{\lambda}} \frac{\partial}{\partial x} \left(\frac{1}{r_{\perp}} \right) - \frac{x}{\lambda r_{\perp}} \frac{\partial}{\partial x} e^{-\frac{r_{\perp}}{\lambda}} \quad (5.46)$$

which gives:

$$\frac{\partial^2}{\partial x^2} e^{-\frac{r_{\perp}}{\lambda}} = \left(-\frac{1}{\lambda r_{\perp}} + \frac{x^2}{\lambda r_{\perp}^3} + \frac{x^2}{\lambda^2 r_{\perp}^2} \right) e^{-\frac{r_{\perp}}{\lambda}} \quad (5.47)$$

and similarly for the y -direction. Again there are no terms arising from differentiation with respect to z . Adding the two contributing terms together gives:

$$\nabla^2 e^{-\frac{r_{\perp}}{\lambda}} = \left(\frac{\partial^2}{\partial x^2} + \frac{\partial^2}{\partial y^2} \right) e^{-\frac{r_{\perp}}{\lambda}} = \left(-\frac{1}{\lambda r_{\perp}} + \frac{1}{\lambda^2} \right) e^{-\frac{r_{\perp}}{\lambda}} \quad (5.48)$$

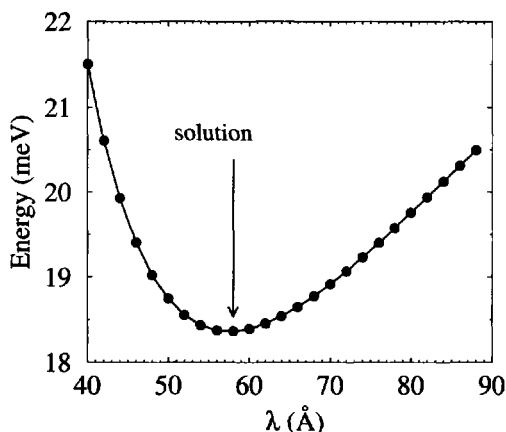


Figure 5.4 Total energy E of the electron as a function of the variational parameter λ , as an illustration of the variational principle

Hence the integral I_3 , as in equation (5.43), with the usual transformation to plane polar coordinates becomes:

$$I_3 = 2\pi \int_0^\infty e^{-\frac{r_\perp}{\lambda}} \left(-\frac{1}{\lambda r_\perp} + \frac{1}{\lambda^2} \right) e^{-\frac{r_\perp}{\lambda}} r_\perp dr_\perp \quad (5.49)$$

By using equations (5.33) and (5.39), obtain:

$$I_3 = 2\pi \left(\int_0^\infty -\frac{1}{\lambda r_\perp} e^{-\frac{2r_\perp}{\lambda}} r_\perp dr_\perp + \frac{1}{\lambda^2} I_1 \right) \quad (5.50)$$

$$\therefore I_3 = 2\pi \left\{ \left[-\frac{1}{\lambda} e^{-\frac{2r_\perp}{\lambda}} \times -\frac{\lambda}{2} \right]_0^\infty + \frac{1}{\lambda^2} I_1 \right\} \quad (5.51)$$

which gives:

$$I_3 = 2\pi \left(-\frac{1}{4} \right) \quad (5.52)$$

Finally, for the two-dimensional case, consider the only remaining integral:

$$I_4 = 2\pi \int_0^\infty \frac{e^{-\frac{2r_\perp}{\lambda}}}{r'} r_\perp dr_\perp \quad (5.53)$$

Consider, say for example, the substitution $r'^2 = r_\perp^2 + (z - r_d)^2 = r_\perp^2 + z'^2$. In this case the limits of integration change as follows. When $r_\perp = 0$, $r' = |z'|$ and when

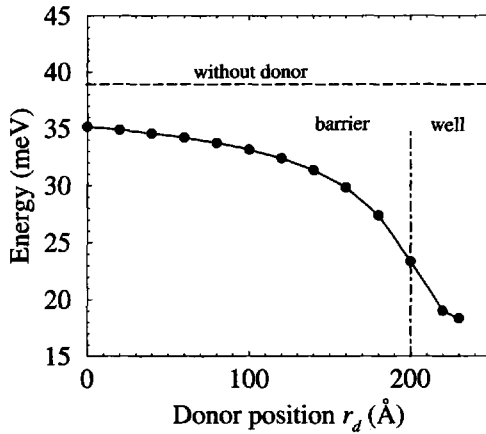


Figure 5.5 Total energy E of the electron in a 60 Å CdTe quantum well surrounded by $\text{Cd}_{0.9}\text{Mn}_{0.1}\text{Te}$ barriers in the presence of a donor at position r_d

$r_{\perp} = \infty$, $r' = \infty$, hence:

$$I_4 = 2\pi \int_{|z'|}^{\infty} \frac{e^{-\frac{2\sqrt{r'^2 - z'^2}}{\lambda}}}{r'} r' dr' \quad (5.54)$$

Note that the lower limit for r' is $|z'|$ and not just z' —this is an important point! This change of limit will be used repeatedly in this analysis. The modulus bars are present as r' is an absolute distance; it is not a vector quantity, and is therefore always positive.

Now this integration cannot be performed analytically and it is always prudent to avoid evaluating definite integrals numerically where the upper limit is infinity. After all exactly what value of infinity would one use? The form for I_4 can be manipulated further however, into a more manageable integral between 0 and 1, which can then be evaluated accurately and with confidence. The mathematical trick* introduced here will be used again and again in this present chapter and also later in Chapter 6.

$$\text{Put } r' = |z'| \cosh \theta \quad \text{then } dr' = |z'| \sinh \theta \, d\theta$$

and the limits become zero and infinity, which for now seems to be contrary to what was intended. Hence:

$$I_4 = 2\pi \int_0^{\infty} \exp\left(-\frac{2\sqrt{z'^2 \cosh^2 \theta - z'^2}}{\lambda}\right) |z'| \sinh \theta \, d\theta \quad (5.55)$$

*Credit to Winston Hagston for this

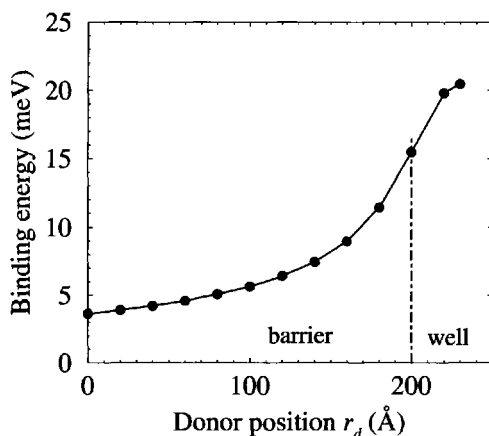


Figure 5.6 Neutral donor binding energy E_{D^0} in the 60 Å CdTe quantum well as before, as a function of the donor position r_d

$$\therefore I_4 = 2\pi \int_0^\infty \exp\left(-\frac{2|z'|\sinh\theta}{\lambda}\right) |z'| \sinh\theta \, d\theta \quad (5.56)$$

Using the substitution $w = e^{-\theta}$, then when $\theta = 0$, $w = 1$, and when $\theta = \infty$, $w = 0$. Also, remembering:

$$\sinh\theta = \frac{1}{2}(e^\theta - e^{-\theta}), \quad \text{then} \quad \sinh\theta = \frac{1}{2}\left(\frac{1}{w} - w\right) \quad (5.57)$$

equation (5.56) therefore becomes:

$$I_4 = 2\pi \int_1^0 \exp\left[-\frac{2|z'|\frac{1}{2}\left(\frac{1}{w} - w\right)}{\lambda}\right] |z'|\frac{1}{2}\left(\frac{1}{w} - w\right) \left(-\frac{dw}{w}\right) \quad (5.58)$$

which gives, finally:

$$I_4 = 2\pi \int_0^1 \exp\left[-\frac{|z'|\left(\frac{1}{w} - w\right)}{\lambda}\right] |z'|\frac{1-w^2}{2w^2} \, dw \quad (5.59)$$

Figure 5.4 demonstrates the principle behind the variational calculation, using the above analysis, by employing a 60 Å CdTe well surrounded by 200 Å $\text{Cd}_{0.9}\text{Mn}_{0.1}\text{Te}$ barriers for illustration. The graph plots the total energy E as a function of the variational parameter λ , for an electron in the presence of a donor at the centre of the well. The system assumes the lowest energy state possible, and hence the electron moves to a Bohr orbit of radius $\lambda=58$ Å, with energy $E=18.366$ meV.

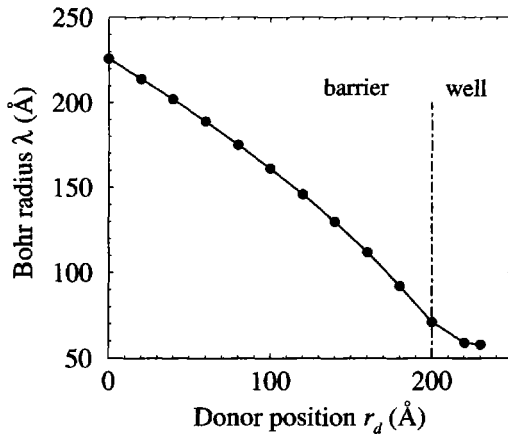


Figure 5.7 Bohr radius λ as a function of the donor position r_d across the 60 Å CdTe quantum well of above

Fig. 5.5 shows the results of calculations for a range of donor positions across the same 60 Å CdTe quantum well. The $\text{Cd}_{1-x}\text{Mn}_x\text{Te}$ system has been chosen to illustrate the method merely because the binding energy is larger, as calculated earlier, so it seems a bit more interesting than the more common $\text{Ga}_{1-x}\text{Al}_x\text{As}$ system. In fact, $\text{Cd}_{1-x}\text{Mn}_x\text{Te}$ is quite similar to $\text{Ga}_{1-x}\text{Al}_x\text{As}$; i.e. it is a direct material, with a similar bandgap, and CdTe forms type I quantum wells with a valence band offset similar to that of $\text{GaAs}/\text{Ga}_{1-x}\text{Al}_x\text{As}$. None of the results obtained are particularly features of the choice of material, and re-emphasizing the aim of this present book, it is the theory and computational methods and the generic deductions that are important, and *not the absolute values* of the calculations. As mentioned already, all of the necessary tools are provided for the reader to quite quickly repeat the calculations for the material system of interest to them. Therefore, returning to Fig. 5.5, as probably expected the electron energy E is lowest (and therefore the donor binding energy E_{D^0} highest) for a donor at the centre of the well. It would be expected a priori that the electron energy would return to the value without the donor present, as the donor is moved further and further away from the well, i.e.

$$\lim_{r_d \rightarrow \infty} E = E_1, \quad \text{or} \quad \lim_{r_d \rightarrow \infty} E_{D^0} = 0 \quad (5.60)$$

This second point is highlighted by the plot of the magnitude of the binding energy, again as a function of donor position, in Fig. 5.6. For completeness, the corresponding plot of the variational parameter λ is displayed in Fig. 5.7. Clearly there is a correlation between λ and E_{D^0} , with the larger E_{D^0} giving the smaller λ ; physically this implies that the binding energy increases as the radius of the electron orbit is decreased.

In order to produce the data plotted in Fig. 5.8, the electron ground-state energies E_1 were calculated without the donor present, and with the donor present E , for a donor fixed at the centre of the well, as a function of well width. This figure illustrates two points that are both important convergence tests for this analysis and for the analysis given in Chapter 3. First, as the well width increases then the electron ground-state energy without the donor present, i.e. E_1 , decreases monotonically and tends towards zero. Secondly, and perhaps more importantly, and relevant to this section, as the well width increases and the contribution to the electron energy due to quantum confinement decreases, the total energy for the electron tends towards the binding energy of the neutral donor in bulk (11.7 meV, as calculated earlier).

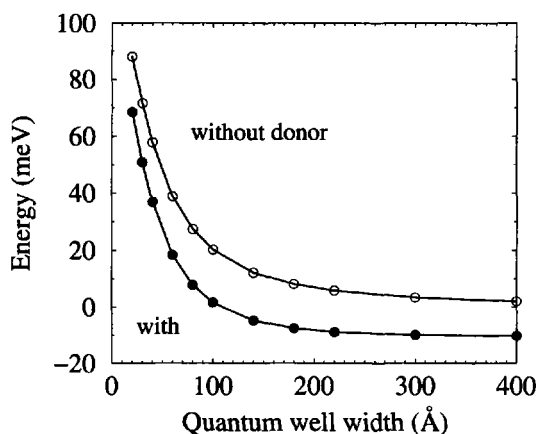


Figure 5.8 Energy of an electron in a quantum well, both with and without a donor at the centre of the well, as a function of the well width

This latter result is perhaps represented more clearly in Fig. 5.9. The binding energy is a non-monotonic function of the well width and peaks at a relatively small width value. This is a very similar result to that of the more complex case of exciton binding energies which will appear again in Chapter 6; in the latter case, such non-monotonic behaviour has been observed experimentally. The non-monotonic behaviour in both cases is merely due to the probability of the electron being within the well. For very narrow wells, the increasing confinement energy, as seen in Fig. 5.8, pushes the electron closer to the top of the well, eventually forcing it to 'spill over the top', thus leading to decreases in the binding energy. Returning to the earlier point, as the well width increases then the binding energy of the donor tends towards its bulk value, as previously hoped.

These results show that the theoretical approach is sound and, when implemented computationally, the model assumes simple limits according to sensible physical thinking. Therefore, confidence can be placed in this method. However, this form

for the wave function would probably never be used in real calculations. The reason being, that while it appears to be the simplest choice of trial wave function, and although it does appear to follow the expected limits, a three-dimensional trial wave function, as discussed in the next section, supersedes it. Calculations will show that a spherically symmetric wave function gives lower variational energies, which are therefore considered to be better approximations, and in addition, the computational implementation of the mathematics is more efficient than in this, the two-dimensional case.

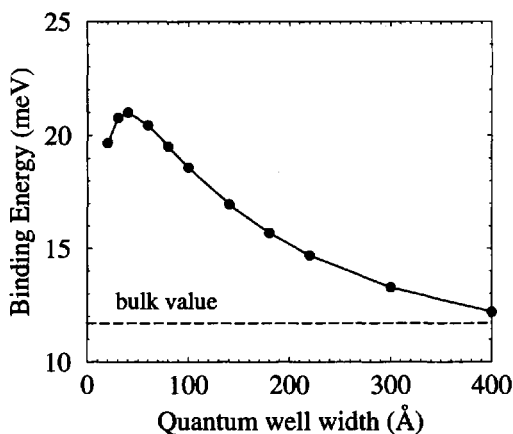


Figure 5.9 The neutral donor binding energy E_{D0} for a donor at the centre of a well, as a function of the well width

5.4 THREE-DIMENSIONAL TRIAL WAVE FUNCTION

Driven by knowledge of bulk material and the spherical symmetry of the hydrogenic wave function, the next obvious choice of the spatial coordinate r'' is given by:

$$r'' = \sqrt{x^2 + y^2 + (z - r_d)^2} \quad (5.61)$$

which is just the electron-donor separation, defined earlier as r' . Again, the investment in the general formalism developed in Section 5.2 pays dividends in that the full requirements for the calculation merely require an evaluation of the integrals I_j , ($j = 1, 2, 3, 4$) but with the new form for r'' .

Consider first I_1 as originally defined in equation (5.19). Using the new form for $r'' = r'$ and switching to plane polar coordinates gives:

$$I_1 = \int_0^\infty \int_0^\infty e^{-\frac{2r''}{\lambda}} dx dy = \int_0^\infty e^{-\frac{2r'}{\lambda}} 2\pi r_\perp dr_\perp \quad (5.62)$$

Now:

$$r'^2 = r_{\perp}^2 + z'^2 \quad (5.63)$$

and therefore $r' dr' = r_{\perp} dr_{\perp}$. Preparing for the substitution r' for r_{\perp} , then when $r_{\perp} = 0$, $r' = |z'|$, and when $r_{\perp} = \infty$, $r' = \infty$, and hence:

$$I_1 = \int_{|z'|}^{\infty} e^{-\frac{2r'}{\lambda}} 2\pi r' dr' \quad (5.64)$$

which can be integrated by parts in the same manner as in Section 5.3, with the following choices:

$$u = r' \quad \text{and} \quad \frac{dv}{dr'} = e^{-\frac{2r'}{\lambda}} \quad (5.65)$$

which gives:

$$\frac{du}{dr'} = 1 \quad \text{and} \quad v = -\frac{\lambda}{2} e^{-\frac{2r'}{\lambda}} \quad (5.66)$$

Thus, applying the above to equation (5.64) gives:

$$I_1 = 2\pi \left\{ \left[-\frac{\lambda r'}{2} e^{-\frac{2r'}{\lambda}} \right]_{|z'|}^{\infty} - \int_{|z'|}^{\infty} -\frac{\lambda}{2} e^{-\frac{2r'}{\lambda}} dr' \right\} \quad (5.67)$$

$$\therefore I_1 = 2\pi \left[-\frac{\lambda r'}{2} e^{-\frac{2r'}{\lambda}} - \frac{\lambda^2}{4} e^{-\frac{2r'}{\lambda}} \right]_{|z'|}^{\infty} \quad (5.68)$$

which after evaluation gives:

$$I_1 = 2\pi \left(\frac{\lambda |z'|}{2} + \frac{\lambda^2}{4} \right) e^{-\frac{2|z'|}{\lambda}} \quad (5.69)$$

Turning attention to the second integral, and using the particular expression for r'' and plane polar coordinates, then:

$$I_2 = \int_0^{\infty} \int_0^{\infty} e^{-\frac{r''}{\lambda}} \nabla_z e^{-\frac{r''}{\lambda}} dx dy = \int_0^{\infty} e^{-\frac{r'}{\lambda}} \nabla_z e^{-\frac{r'}{\lambda}} 2\pi r_{\perp} dr_{\perp} \quad (5.70)$$

$$\therefore I_2 = 2\pi \int_0^{\infty} e^{-\frac{r'}{\lambda}} \frac{\partial}{\partial r'} e^{-\frac{r'}{\lambda}} \frac{\partial r'}{\partial z'} r_{\perp} dr_{\perp} \quad (5.71)$$

Now:

$$r' = \sqrt{x^2 + y^2 + (z - r_d)^2} = \sqrt{x^2 + y^2 + z'^2} \quad (5.72)$$

and therefore:

$$\frac{\partial r'}{\partial z'} = \frac{1}{2\sqrt{x^2 + y^2 + z'^2}} \times 2z' = \frac{z'}{r'} \quad (5.73)$$

Note here the z' and *not* $|z'|$. Using this result in equation (5.71) then:

$$I_2 = 2\pi \int_0^{\infty} e^{-\frac{r'}{\lambda}} \left(-\frac{z'}{r' \lambda} \right) e^{-\frac{r'}{\lambda}} r_{\perp} dr_{\perp} \quad (5.74)$$

but $r'^2 = r_{\perp}^2 + z'^2$, and therefore $r' dr' = r_{\perp} dr_{\perp}$. Using this substitution and changing the limits of integration, then obtain:

$$I_2 = 2\pi \int_{|z'|}^{\infty} \left(-\frac{z'}{r'\lambda} \right) e^{-\frac{2r'}{\lambda}} r' dr' \quad (5.75)$$

$$\therefore I_2 = 2\pi \left[\frac{z'}{2} e^{-\frac{2r'}{\lambda}} \right]_{|z'|}^{\infty} \quad (5.76)$$

and finally:

$$I_2 = 2\pi \left(-\frac{z'}{2} e^{-\frac{2|z'|}{\lambda}} \right) \quad (5.77)$$

Writing I_3 in plane polar coordinates with the three-dimensional form for r'' , then:

$$I_3 = 2\pi \int_0^{\infty} e^{-\frac{r'}{\lambda}} \nabla^2 e^{-\frac{r'}{\lambda}} r_{\perp} dr_{\perp} \quad (5.78)$$

In order to proceed, it is necessary to evaluate the differential. Consider:

$$\nabla_x e^{-\frac{r'}{\lambda}} = \frac{\partial}{\partial r'} e^{-\frac{r'}{\lambda}} \frac{\partial r'}{\partial x} \quad (5.79)$$

which, because of the isotropy of the exponential term in this case, yields, in the same manner as above in equation (5.73):

$$\nabla_x e^{-\frac{r'}{\lambda}} = -\frac{x}{r'\lambda} e^{-\frac{r'}{\lambda}} \quad (5.80)$$

Differentiating again:

$$\nabla_x^2 e^{-\frac{r'}{\lambda}} = -\frac{1}{r'\lambda} e^{-\frac{r'}{\lambda}} - \frac{x}{\lambda} \times -\frac{1}{r'^2} \frac{\partial r'}{\partial x} e^{-\frac{r'}{\lambda}} - \frac{x}{r'\lambda} \times -\frac{1}{\lambda} e^{-\frac{r'}{\lambda}} \frac{\partial r'}{\partial x} \quad (5.81)$$

$$\therefore \nabla_x^2 e^{-\frac{r'}{\lambda}} = \left(-\frac{1}{r'\lambda} + \frac{x^2}{r'^3\lambda} + \frac{x^2}{r'^2\lambda^2} \right) e^{-\frac{r'}{\lambda}} \quad (5.82)$$

and similarly for $\nabla_y^2 e^{-\frac{r'}{\lambda}}$. The same is also true for $\nabla_z^2 e^{-\frac{r'}{\lambda}}$ and can be followed through by noting, however, that:

$$\nabla_z^2 e^{-\frac{r'}{\lambda}} = \nabla_z^2 e^{-\frac{r'}{\lambda}} \times \left(\frac{\partial z'}{\partial z} \right)^2, \quad \text{where} \quad \frac{\partial z'}{\partial z} = 1 \quad (5.83)$$

Gathering all of the terms together, obtain:

$$\nabla^2 e^{-\frac{r'}{\lambda}} = (\nabla_x^2 + \nabla_y^2 + \nabla_z^2) e^{-\frac{r'}{\lambda}} = \left(-\frac{2}{r'\lambda} + \frac{1}{\lambda^2} \right) e^{-\frac{r'}{\lambda}} \quad (5.84)$$

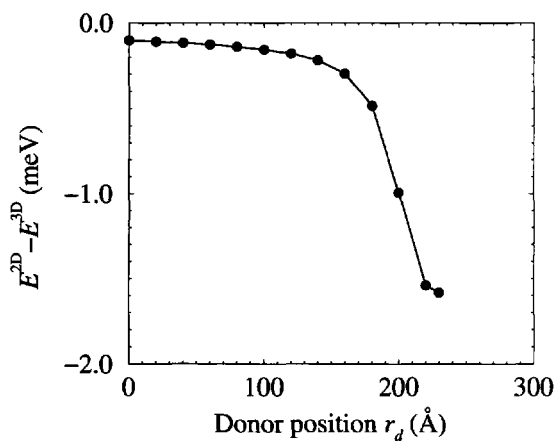


Figure 5.10 Difference in total energy for the two- and three-dimensional cases, as a function of donor position r_d across the 60 Å CdTe well

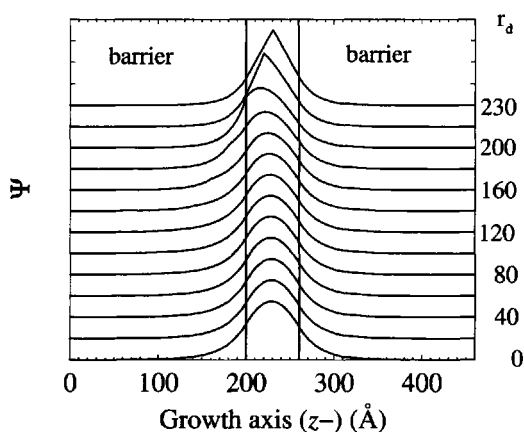


Figure 5.11 The electron wave function Ψ , as in equation 5.7, for the donor positions given on the right hand axis

Using this form in equation (5.78) then:

$$I_3 = 2\pi \int_0^\infty e^{-\frac{r'}{\lambda}} \left(-\frac{2}{r'\lambda} + \frac{1}{\lambda^2} \right) e^{-\frac{r'}{\lambda}} r_\perp dr_\perp \quad (5.85)$$

which, using equation (5.64) gives the following:

$$I_3 = \frac{I_1}{\lambda^2} + 2\pi \int_0^\infty \left(-\frac{2}{r'\lambda} \right) e^{-\frac{2r'}{\lambda}} r_\perp dr_\perp \quad (5.86)$$

Again, substituting r' for r_\perp , then gives:

$$I_3 = \frac{I_1}{\lambda^2} + 2\pi \left(-\frac{2}{\lambda} \right) \int_{|z'|}^\infty \frac{e^{-\frac{2r'}{\lambda}}}{r'} r' dr' \quad (5.87)$$

$$\therefore I_3 = \frac{I_1}{\lambda^2} + 2\pi \left(-\frac{2}{\lambda} \right) \left[-\frac{\lambda}{2} e^{-\frac{2r'}{\lambda}} \right]_{|z'|}^\infty \quad (5.88)$$

Using the final form for I_1 , as in equation (5.69), then:

$$I_3 = 2\pi \left(\frac{|z'|}{2\lambda} - \frac{3}{4} \right) e^{-\frac{2|z'|}{\lambda}} \quad (5.89)$$

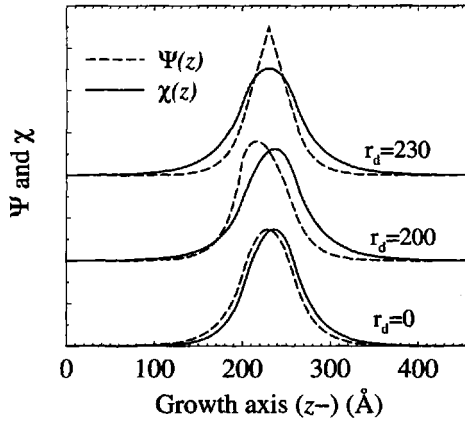


Figure 5.12 Comparison between the total electron wave function $\Psi(z)$ and the numerically determined envelope $\chi(z)$ for three different donor positions

Finally, for the three-dimensional case, I_4 becomes:

$$I_4 = 2\pi \int_0^\infty \frac{e^{-\frac{2r'}{\lambda}}}{r'} r_\perp dr_\perp \quad (5.90)$$

which, on changing the variable to r' , becomes trivial, i.e.

$$\therefore I_4 = 2\pi \int_{|z'|}^\infty \frac{e^{-\frac{2r'}{\lambda}}}{r'} r' dr' \quad (5.91)$$

and therefore:

$$I_4 = 2\pi \left(\frac{\lambda}{2} e^{-\frac{2|z'|}{\lambda}} \right) \quad (5.92)$$

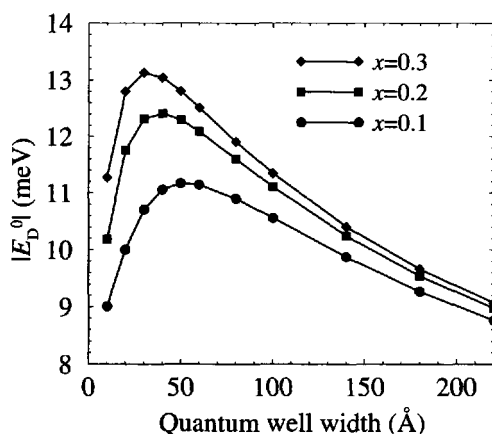


Figure 5.13 Magnitude of the donor binding energy E_{D0} in GaAs/Ga_{1-x}Al_xAs as a function of quantum well width, for a variety of barrier alloys x

Looking upon these results from a computational viewpoint, it can then be seen that this 3D trial wave function has an immediate advantage over the previous, seemingly simpler, 2D case, namely that all of the integrals I_1 , I_2 , I_3 and I_4 have analytical expressions. Indeed, evaluation of the integrals and the minimisation of the energy is computationally much less demanding than previously.

Figure 5.10 shows the change in the total energy E of the electron between the previous 2D trial wave function and the more complex 3D case presented in this section. It is clear that the energy E is lower for *all* donor positions across this (a typical) quantum well. The difference between the two trial wave functions is smallest when the separation between the donor and the electron is larger, i.e. when the donor is deep in the barrier ($r_d = 0$ Å) and the electron is, as always, centred in the well ($z = 230$ Å). Thus it might be concluded that the 2D wave function is a reasonable approximation when the donor is in the barrier, although as the graph shows, for donors in the well, considerably lower energies can be obtained by using a spherical hydrogenic term. Recalling the variational principle, then the lower energies obtained imply that the 3D approximation to the wave function is a more accurate representation than the 2D case. When coupled together with the computational advantage, as mentioned above, then the argument in favour of the 3D trial wave function is clear.

Figure 5.11 displays the total wave function, $\Psi = \chi(z)e^{-\frac{r'}{\lambda}}$, for the range of donor positions across the quantum well. It can be seen that the wave function Ψ

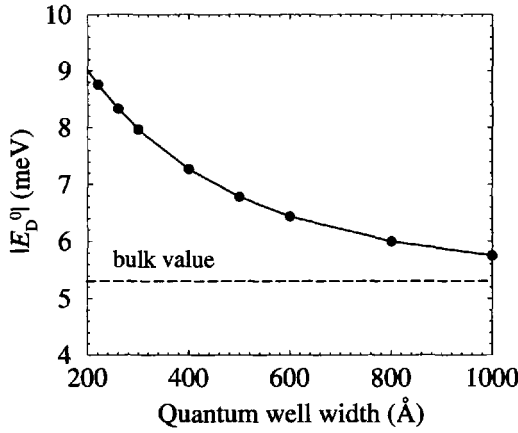


Figure 5.14 Magnitude of the donor binding energy E_{D^0} for donors at the centre of GaAs wells of large width, surrounded by $\text{Ga}_{0.9}\text{Al}_{0.1}\text{As}$ barriers

resembles the one-particle wave function ψ for an electron without a donor present, for donors in the barrier, i.e. $r_d \lesssim 160 \text{ \AA}$. As the donor approaches the electron wave function, i.e. nears the barrier-well interface at $z = 200 \text{ \AA}$, then the electron is drawn distinctly to the left towards the donor. The influence of the hydrogenic factor $e^{-r'/\lambda}$ can be seen for the donor positions $r_d = 220$ and 230 \AA within the well.

In the 2D case, the total wave function was given by:

$$\Psi(z) = \chi(z)e^{-\frac{\sqrt{x^2+y^2}}{\lambda}} \quad (5.93)$$

and hence the z -dependence is merely $\Psi(z) = \chi(z)$. Furthermore, it was found that the numerically determined envelope $\chi(z)$ was a very close approximation to the electron wave function $\psi(z)$ without the donor present. In this case, as is clear from Fig. 5.11, $\Psi(z) \neq \chi(z)$; this is illustrated more clearly in Fig. 5.12.

Moving on to the $\text{GaAs}/\text{Ga}_{1-x}\text{Al}_x\text{As}$ material system and employing the bulk values of $m^* = 0.067m_0$ and $\epsilon = 13.18$, Fig. 5.13 shows the effect of well width on the neutral donor binding energy, for donors at the centre of the well, for a variety of barrier compositions. As would be expected from earlier results, E_{D^0} peaks at a narrow well width and then tails off towards the bulk value. This important limit is explored further in Fig. 5.14; it is clear from this figure that the convergence is very close, and this helps give justification to the methods developed. The variation in the Bohr radius is displayed in Fig. 5.15, and it too converges well to a value of 104 \AA at very large well widths. This compares admirably with the value deduced from the simple bulk hydrogenic model of 103 \AA , calculated at the beginning of the chapter. The very small difference could arise from the finite λ increment employed of 1 \AA .

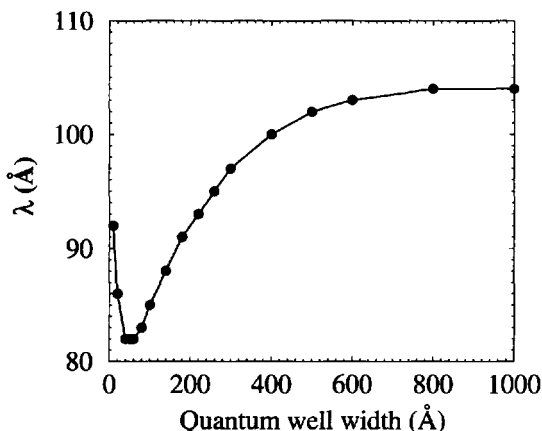


Figure 5.15 The Bohr radius λ as a function of well width for donors at the centres of GaAs wells surrounded by $\text{Ga}_{0.9}\text{Al}_{0.1}\text{As}$ barriers

5.5 VARIABLE-SYMMETRY TRIAL WAVE FUNCTION

In the two previous sections, simple high-symmetry trial wave functions have been chosen to illustrate the method. Indeed, often they provide approximate, but quick, numerical results that may generally suffice. However, a little thought highlights their deficiencies. The aim of this work, like the solution of many physical systems with the variational principle, is to choose a trial wave function which is applicable no matter what the system parameters may be. Which in this case would mean, no matter what the well width, barrier height and donor position are.

The spherical 3D trial wave function was chosen as this is the form found in bulk materials, i.e. systems which do not have a confining potential. Hence it might be expected that this trial wave function is most appropriate for systems which most closely resemble bulk systems i.e. those with only small confinement, such as wide wells and/or low barrier heights. For example, for a donor in the centre of a wide well, the envelope $\chi(z)$ would look like $\cos kz$, which would be slowly varying, and hence the total wave function Ψ would resemble the bulk. Conversely, a large confinement potential might be expected to give a more 2D-like wave function.

It is clear, therefore that while the two trial wave functions considered so far are useful, a more general form can be taken for the hydrogenic factor, a variable symmetry term, which can move between the 2D and 3D forms should it prove energetically favourable to do so. Equation (5.94) summarizes the most general choice for the hydrogenic factor of the trial wave function:

$$r'' = \sqrt{x^2 + y^2 + \zeta^2(z - r_d)^2} \quad (5.94)$$

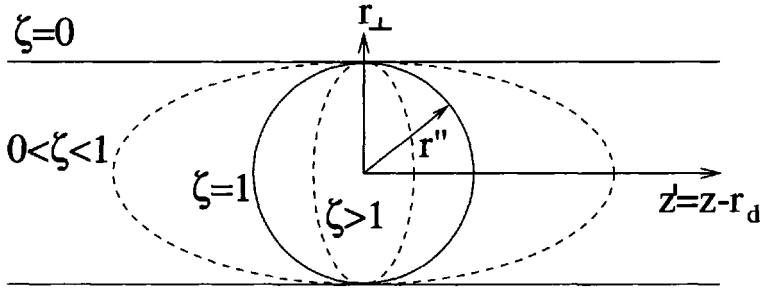


Figure 5.16 Schematic illustration of the variable symmetry relative motion term in the donor trial wave function

where ζ is a second variational parameter, which like λ will be adjusted systematically in order to minimize the total energy E of the system. Hence:

$$\xi = \exp \left(-\frac{\sqrt{x^2 + y^2 + \zeta^2 z'^2}}{\lambda} \right) \quad (5.95)$$

The effect of the parameter ζ on the symmetry of ξ is illustrated in Fig. 5.16.

Again, the investment made earlier, in structuring the problem in simple units, means that all that needs to be done in order to specify all of the quantities in the differential equation (5.23) exactly, is to evaluate the integrals I_j with the new form for r'' . Consider I_1 in the usual plane polar coordinates, i.e.

$$I_1 = 2\pi \int_0^\infty e^{-\frac{2r''}{\lambda}} r_\perp dr_\perp \quad (5.96)$$

Now $r''^2 = r_\perp^2 + \zeta^2 z'^2$, and therefore $r'' dr'' = r_\perp dr_\perp$, considering the limits, i.e.

$$\text{when } r_\perp = 0, \quad r'' = \zeta |z'|$$

$$\text{and when } r_\perp = \infty, \quad r'' = \infty$$

Therefore:

$$I_1 = 2\pi \int_{\zeta |z'|}^\infty e^{-\frac{2r''}{\lambda}} r'' dr' \quad (5.97)$$

As previously, this can be integrated by parts. Choosing:

$$u = r'', \quad \text{and} \quad \frac{dv}{dr''} = e^{-\frac{2r''}{\lambda}}$$

then:

$$\frac{du}{dr''} = 1, \quad \text{and} \quad v = -\frac{\lambda}{2} e^{-\frac{2r''}{\lambda}}$$

Therefore:

$$I_1 = 2\pi \left\{ \left[-\frac{r''\lambda}{2} e^{-\frac{2r''}{\lambda}} \right]_{\zeta|z'|}^{\infty} - \int_{\zeta|z'|}^{\infty} \left(-\frac{\lambda}{2} \right) e^{-\frac{2r''}{\lambda}} dr'' \right\} \quad (5.98)$$

and so:

$$I_1 = 2\pi \left[\left(-\frac{r''\lambda}{2} - \frac{\lambda^2}{4} \right) e^{-\frac{2r''}{\lambda}} \right]_{\zeta|z'|}^{\infty} \quad (5.99)$$

leading finally to:

$$I_1 = 2\pi \left(\frac{\zeta|z'|\lambda}{2} + \frac{\lambda^2}{4} \right) e^{-\frac{2\zeta|z'|}{\lambda}} \quad (5.100)$$

Secondly:

$$I_2 = 2\pi \int_0^{\infty} e^{-\frac{r''}{\lambda}} \nabla_z e^{-\frac{r''}{\lambda}} r_{\perp} dr_{\perp} \quad (5.101)$$

In order to proceed, the differential must be evaluated, i.e.

$$\nabla_z e^{-\frac{r''}{\lambda}} = \frac{\partial}{\partial z} e^{-\frac{r''}{\lambda}} = \frac{\partial}{\partial z''} e^{-\frac{r''}{\lambda}} \frac{\partial z''}{\partial z}, \quad \text{where } z'' = \zeta z' \quad (5.102)$$

With this final substitution, i.e. the use of z'' , then this form resembles the 3D version from earlier, and use can be made of some of the results. For example, using equation (5.80), then:

$$\nabla_z e^{-\frac{r''}{\lambda}} = -\frac{z''}{r''\lambda} e^{-\frac{r''}{\lambda}} \frac{\partial z''}{\partial z} \quad \left(\text{note that } \frac{\partial z''}{\partial z} = \zeta \right) \quad (5.103)$$

Therefore equation 5.101 becomes

$$I_2 = 2\pi \int_0^{\infty} e^{-\frac{r''}{\lambda}} \left(-\frac{\zeta z''}{r''\lambda} \right) e^{-\frac{r''}{\lambda}} r_{\perp} dr_{\perp} \quad (5.104)$$

Returning to z' and substituting r'' for r_{\perp} , then obtain:

$$I_2 = 2\pi \int_{\zeta|z'|}^{\infty} \left(-\frac{\zeta^2 z'}{r''\lambda} \right) e^{-\frac{2r''}{\lambda}} r'' dr'' \quad (5.105)$$

As the values r'' cancel, the integration then becomes trivial, and indeed:

$$I_2 = 2\pi \left(-\frac{\zeta^2 z'}{2} \right) e^{-\frac{2\zeta|z'|}{\lambda}} \quad (5.106)$$

In order to evaluate I_3 , it is again necessary to deduce $\nabla^2 e^{-\frac{r''}{\lambda}}$ first. In a similar manner to equation (5.102), then:

$$\nabla_z^2 e^{-\frac{r''}{\lambda}} = \nabla_{z''}^2 e^{-\frac{r''}{\lambda}} \left(\frac{\partial z''}{\partial z} \right)^2 = \zeta^2 \nabla_{z''}^2 e^{-\frac{r''}{\lambda}} \quad (5.107)$$

Considering z'' as the direct analogy of z' for the 3D case, then use can be made of the earlier result in equation (5.82), hence:

$$\nabla_z^2 e^{-\frac{r''}{\lambda}} = \zeta^2 \left(-\frac{1}{r''\lambda} + \frac{z''^2}{r''^3\lambda} + \frac{z''^2}{r''^2\lambda^2} \right) e^{-\frac{r''}{\lambda}} \quad (5.108)$$

Using the equivalent forms for $\nabla_x^2 e^{-\frac{r''}{\lambda}}$ and $\nabla_y^2 e^{-\frac{r''}{\lambda}}$, also given in equation (5.82) and summing gives:

$$\nabla^2 e^{-\frac{r''}{\lambda}} = \left[\frac{-1 - \zeta^2}{r''\lambda} + \frac{1}{\lambda^2} + \frac{(\zeta^2 - 1)z''^2}{r''^3\lambda} + \frac{(\zeta^2 - 1)z''^2}{r''^2\lambda^2} \right] e^{-\frac{r''}{\lambda}} \quad (5.109)$$

Noting that $z'' = \zeta z'$, then:

$$\nabla^2 e^{-\frac{r''}{\lambda}} = \left[\frac{-1 - \zeta^2}{r''\lambda} + \frac{1}{\lambda^2} + \frac{(\zeta^4 - \zeta^2)z'^2}{r''^3\lambda} + \frac{(\zeta^4 - \zeta^2)z'^2}{r''^2\lambda^2} \right] e^{-\frac{r''}{\lambda}} \quad (5.110)$$

As a check on the analysis so far, putting $\zeta = 0$ and $r'' = r_\perp$ does indeed yield the equivalent equation for the 2D case (equation (5.48)), while putting $\zeta = 1$ and $r'' = r'$ gives the 3D case expressed in equation (5.84).

Therefore, everything is in place for the evaluation of the third integral, which now becomes:

$$I_3 = 2\pi \int_0^\infty \left[\frac{-1 - \zeta^2}{r''\lambda} + \frac{1}{\lambda^2} + \frac{(\zeta^4 - \zeta^2)z'^2}{r''^3\lambda} + \frac{(\zeta^4 - \zeta^2)z'^2}{r''^2\lambda^2} \right] e^{-\frac{2r''}{\lambda}} r_\perp dr_\perp \quad (5.111)$$

Unfortunately each of the four terms needs to be handled separately, so writing equation (5.111) as:

$$I_3 = 2\pi (I_{31} + I_{32} + I_{33} + I_{34}) \quad (5.112)$$

Then the first integral is trivial, while the second can be solved by parts, thus giving:

$$I_{31} = \frac{-1 - \zeta^2}{2} e^{-\frac{2\zeta|z'|}{\lambda}}, \quad \text{and} \quad I_{32} = \left(\frac{\zeta|z'|}{2\lambda} + \frac{1}{4} \right) e^{-\frac{2\zeta|z'|}{\lambda}} \quad (5.113)$$

The third in this short series cannot be solved analytically; however it can again be manipulated into an integral between 0 and 1, and hence quickly and accurately evaluated. It is a worthwhile exercise to follow this one through; by changing the variable of integration from r_\perp to r'' , then obtain:

$$I_{33} = \int_{\zeta|z'|}^\infty \left[\frac{(\zeta^4 - \zeta^2)z'^2}{r''^3\lambda} \right] e^{-\frac{2r''}{\lambda}} r'' dr'' \quad (5.114)$$

By putting $r'' = \zeta|z'| \cosh \theta$, then $dr'' = \zeta|z'| \sinh \theta d\theta$, and it follows that:

$$I_{33} = \int_0^\infty \frac{(\zeta^4 - \zeta^2)z'^2}{\lambda \zeta|z'| \cosh^2 \theta} e^{-\frac{2\zeta|z'| \cosh \theta}{\lambda}} \sinh \theta d\theta \quad (5.115)$$

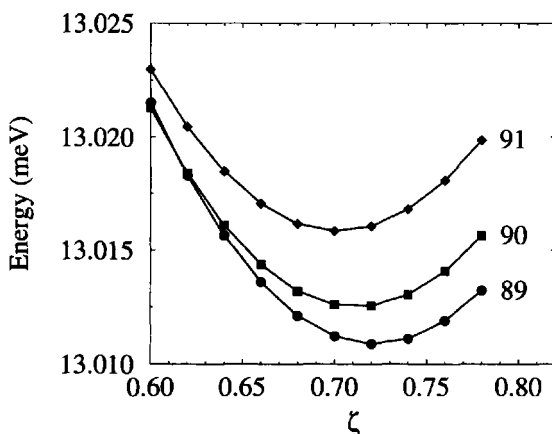


Figure 5.17 Total energy E of an electron bound to a donor as a function of the variable symmetry parameter ζ for $\lambda = 89, 90$ and 91 \AA , thus illustrating the convergence of the two parameter variational calculation

Again, making the substitution $w = e^{-\theta}$, and rewriting the sinh and cosh terms as in equation (5.57), yields the final form:

$$I_{33} = \int_0^1 \frac{2(\zeta^3 - \zeta)|z'|}{\lambda} \exp \left[-\frac{\zeta|z'|(\frac{1}{w} + w)}{\lambda} \right] \frac{1 - w^2}{(1 + w^2)^2} dw \quad (5.116)$$

Similarly

$$I_{34} = \int_0^1 \frac{2(\zeta^4 - \zeta^2)z'^2}{\lambda^2} \exp \left(-\frac{\zeta|z'|(\frac{1}{w} + w)}{\lambda} \right) \frac{1 - w^2}{w(1 + w^2)} dw \quad (5.117)$$

At last, finally I_4 , it follows similarly, with the variable of integration r_{\perp} needing to be switched in order to match the denominator r' , thus allowing manipulation to give:

$$I_4 = 2\pi \int_0^1 \exp \left[-\frac{2|z'|\sqrt{(\frac{1-w^2}{2w})^2 + \zeta^2}}{\lambda} \right] |z'| \frac{1 - w^2}{2w^2} dw \quad (5.118)$$

It should be noted that putting $\zeta=1$ in the above integrals, I_1 , I_2 , and I_3 , reassuringly gives the same result as those found for the 3D case. Unfortunately the manipulation of I_4 with $\zeta=1$ is much more difficult and lies beyond the scope of this present work. However numerical convergence tests have shown that putting $\zeta=1$ does yield the 3D energies, while putting $\zeta=0$ gives the 2D energies, although in both cases it is necessary to use a fine mesh, i.e. a small value of δz in equation (5.28).

Table 5.1 Total energies for an electron bound to a donor as a function of donor position in a 60 Å CdTe quantum well surrounded by Cd_{0.9}Mn_{0.1}Te barriers, for the three different trial wave functions (data first appeared in [108], data reproduced by permission of Academic Press)

r_d (Å)	$E(\zeta = 0)$ (meV)	$E(\zeta = 1)$ (meV)	$E(\zeta)$ (meV)
150	30.72	30.48	30.44
160	29.89	29.60	29.56
170	28.83	28.48	28.44
180	27.45	27.01	26.97
190	25.62	25.05	25.02
200	23.22	22.50	22.47
210	20.49	19.71	19.67
220	18.30	17.58	17.53
230	17.47	16.80	16.74

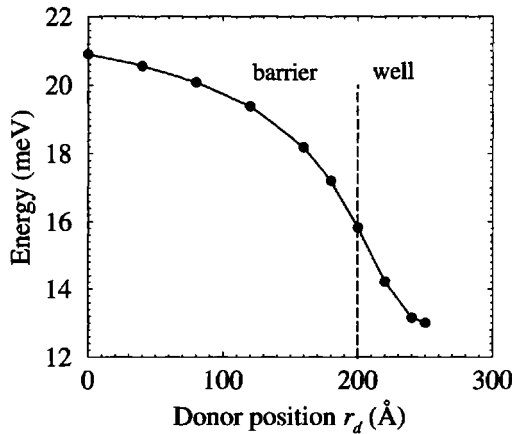


Figure 5.18 Energy E as a function of donor position across a 100 Å GaAs well surrounded by Ga_{0.9}Al_{0.1}As barriers

Figure 5.17 gives an example of the output from this two-parameter variational calculation. For the record, this calculation is for a donor at the centre of a 100 Å GaAs well surrounded by Ga_{0.9}Al_{0.1}As barriers. For the sake of clarity, only the Bohr radii corresponding to the minima (at $\lambda = 89$ Å) and above are shown. It can be seen from this figure that the change in energy due to the ζ variation is very

small, and certainly beyond the scope of experimental measurement. Nonetheless, the graph shows that the wave function will assume a value of $\zeta \approx 0.72$. This will have implications for systems where the extent and nature of the wave function are important, e.g. electron scattering from neutral donors, donor-bound excitons, and the electrical excitation of impurities.

Table 5.1 compares the energies obtained with all three trial wave functions for donor positions across a 60 Å CdTe well, surrounded by 200 Å $\text{Cd}_{0.9}\text{Mn}_{0.1}\text{Te}$ barriers (hence the centre of the well is at 230 Å). Note, as mentioned above, that the trial wave functions which contain integrals that require numerical evaluation, i.e. the 2D and the variable symmetry case which contain the integrals between 0 and 1 with respect to w , demand around eight wave function points per Å for complete convergence. The data displayed earlier in Fig. 5.10 as a comparison between the 2D and the 3D case, was obtained quickly with the standard mesh of 1 point per Å, i.e. $\delta z = 1$ Å. This was acceptable at that point in order to illustrate merely that the 3D case gave lower variational energies than the 2D case. However, for the purposes of demonstrating that the variable symmetry case gives lower energies still, more accurate calculations were necessary, and indeed for Table 5.1, δz in the iterative equation (equation (5.28)) was required to be 0.1 Å. The disadvantage to moving to such accuracy is, of course, the increased computational effort, which is roughly inversely proportional to δz ; hence an increase by a factor of 10 in the number of wave function points leads to a similar increase in computational time. Table 5.1 clearly illustrates that the variable symmetry trial function gives the lowest energies, for all donor positions, although in terms of energy the 3D wave function is a very good approximation also, with energies differing by less than 0.1 meV. In comparison with the more accurate evaluation of the energy for the 2D case, this differs by up to 0.7 meV.

Figures 5.18 and 5.19 illustrate the results of detailed calculations for the GaAs/ $\text{Ga}_{1-x}\text{Al}_x\text{As}$ system. The energies in Fig. 5.18 are similar in form to those obtained earlier, with the interest here lying with the symmetry of the wave function, as given in Fig. 5.19. Most obviously, it is clear that the values of ζ imply that the donor wave function will assume symmetries that are neither 2D or 3D, thus justifying the investment in the variable symmetry analysis. The value of ζ is non-monotonic as a function of donor position, with a local minimum for donors at the centre of the well and a peak for donors near the well-barrier interface; this is the exact same behaviour as reported by Roberts *et al.* [108] for the CdTe- $\text{Cd}_{1-x}\text{Mn}_x\text{Te}$ system and hence appears to be a generic result. Typically, it is found that $\zeta \sim 0.7$, and hence the variable-symmetry trial wave function looks like the following:

$$\Psi = \chi(z) \exp \left(\frac{\sqrt{x^2 + y^2 + z^2/2}}{\lambda} \right) \quad (5.119)$$

i.e. the coefficient of z^2 is half that of x and y . Therefore, the hydrogenic factor is midway between the 2D and 3D case, with a very definite prolate spheroid shape (refer again to Fig. 5.16).

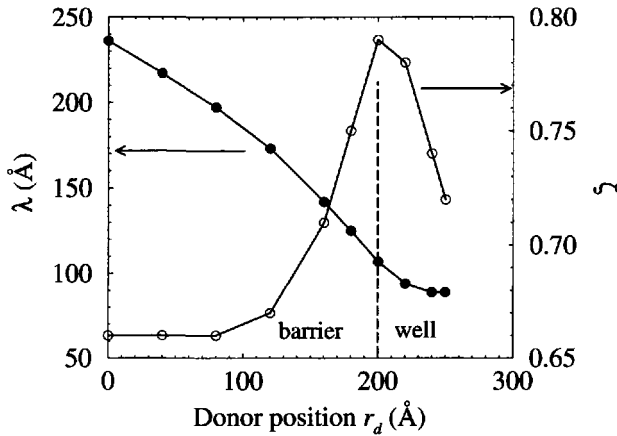


Figure 5.19 Bohr radius λ and symmetry parameter ζ as a function of donor position

In conclusion, regarding the various advantages and disadvantages of the three trial wave functions, clearly the last, namely the variable symmetry case, gives the lowest variational energies and hence is the most accurate physical representation. Without such an investment in complexity and computational time it would not have been apparent that donors assume this mixed 2D and 3D symmetry, i.e. wave functions resembling prolate spheroids. However, the computational demands are *substantial* and it has been shown that donor energies can be obtained within 0.1 meV of the lowest possible value, by using a simple 3D (spherical) trial wave function. This wave function has the advantage that all of the relevant integrations can be performed analytically, which leads to a quick evaluation of the total energy E .

5.6 INCLUSION OF A CENTRAL CELL CORRECTION

As mentioned earlier, the use of the bulk dielectric constant has to be employed with a note of caution. In the majority of cases its use is probably fine; however, it is worthwhile noting that there is a scheme for accounting for a change in the permittivity of the material as the electron approaches the donor. This occurs as the degree of electronic shielding is reduced and is known as the *central cell correction*.

This extension takes the form of an additional factor in the Coulombic potential term, which, for the purpose of illustration, is taken to be as follows:

$$-\frac{e^2}{4\pi\epsilon r'} \longrightarrow -\frac{e^2}{4\pi\epsilon r'} \left(1 + \Xi e^{-\frac{r'}{\lambda}}\right) \quad (5.120)$$

where Λ is a parameter describing the extent of the correction; the latter can be thought of as a screening length. As $r' \rightarrow 0$, it is expected that:

$$-\frac{e^2}{4\pi\epsilon r'} \left(1 + \Xi e^{-\frac{r'}{\Lambda}}\right) \rightarrow -\frac{e^2}{4\pi\epsilon_0 r'} \quad (5.121)$$

where ϵ_0 is the permittivity of free space. Taking this limit gives:

$$\frac{1}{\epsilon} (1 + \Xi) = \frac{1}{\epsilon_0} \quad (5.122)$$

which when rearranged gives:

$$\Xi = \frac{\epsilon}{\epsilon_0} - 1 \quad (5.123)$$

The effect of these changes on the analysis carried out above occurs solely in the integral I_4 , which for the most general case would become:

$$I'_4 = 2\pi \int_0^\infty \left(1 + \Xi e^{-\frac{r'}{\Lambda}}\right) \frac{e^{-\frac{2r'}{\Lambda}}}{r'} r_\perp dr_\perp \quad (5.124)$$

If the particular 3D case were chosen, then this gives:

$$I'_4 = I_4 + 2\pi\Xi \frac{\Lambda\lambda}{\lambda + 2\Lambda} e^{-\frac{|\mathbf{z}'|}{\Lambda}} e^{-\frac{|\mathbf{z}'|}{\lambda}} \quad (5.125)$$

Recent work has demonstrated the effect of such a central cell correction for this 3D case [109]. While the physical reasoning behind the addition of such a correction seems sound, here, as always, caution needs to be taken in choosing a value for Λ . Cynically, Λ could be looked upon as merely another parameter (although it is *not* a variational parameter), that can be varied in order to produce agreement between experiment and theory.

The evaluation of I'_4 for the most general, variable symmetry case will be left as an exercise for the reader!

5.7 SPECIAL CONSIDERATIONS FOR ACCEPTORS

Hitherto concentration has been focused on donor energies in heterostructures, although, as discussed earlier in this chapter, *in principle* all of the analysis above is also applicable to the calculation of acceptor levels. Agreement between the hydrogenic model of the acceptor in bulk and experiment needs to account for both the degeneracy of the valence band and the reduced screening of the negatively charged acceptor ion potential. In heterostructures however, the confinement potential lifts the degeneracy of light- and heavy-holes, so the only additional consideration that needs to be made relates to the permittivity.

The procedure would be to deduce empirically the central cell correction parameter Λ , by comparing the calculated acceptor binding energy E_{A0} predicted by the above

analysis and adjusting Λ to give agreement with experiment, for any well characterised quantum well system. For a particular material system and acceptor atom, the screening length Λ *should be* constant. Therefore it can be used in further predictive calculations for different well widths, barrier heights, dopant positions, etc.

Acceptor levels within quantum wells have been calculated successfully, as discussed by Bastard [18]. Masselink *et al.* [110] have achieved good agreement with the experimental measurements of Miller *et al.* [111] for the binding energy of carbon acceptors in GaAs-Ga_{1-x}Al_xAs quantum wells.

5.8 EFFECTIVE MASS AND DIELECTRIC MISMATCH

A great deal of attention has been paid in the literature to the role of effective mass mismatch at interfaces in semiconductor heterostructures (see Bastard [18]) and to the effect that this may have on donor energies. In addition, Fraizzoli *et al.* [112] have studied in detail the role of a dielectric constant mismatch at interfaces between dissimilar materials and its effect on shallow donor impurity levels in GaAs-Ga_{1-x}Al_xAs quantum well structures. Such considerations can be readily incorporated in the present approach. At first sight it might be thought that this can be achieved by making allowance for the fact that both m^* and ϵ , which appear in the original Schrödinger equation, are a function of z . In as far as constructing the Schrödinger equation, this is true for the latter. However, as discussed in detail in Chapter 3, the change to a variable effective mass necessitates substantial alterations to the quantum mechanical kinetic energy operator. Again, as before the first term in equation (5.4) would need to be substituted with:

$$-\frac{\hbar^2}{2m^*} \frac{\partial^2}{\partial z^2} \longrightarrow -\frac{\hbar^2}{2} \frac{\partial}{\partial z} \frac{1}{m^*(z)} \frac{\partial}{\partial z} \quad (5.126)$$

In principle the analysis could be followed through again, with re-evaluation of the integrals affected by this substitution.

The generalisation to include the material dependency of the permittivity ϵ is simpler to deal with. It can be achieved by absorbing this function into the integral I_4 which arises from the Coulombic term, i.e. with the appropriate change in the coefficient, I_4 would become:

$$I'_4 = 2\pi \int_0^\infty \frac{e^{-\frac{2r''}{\lambda}}}{\epsilon(z)r'} r_\perp dr_\perp \quad (5.127)$$

as ϵ depends on the electron position and not simply the electron-donor separation as usual. It is clear, however, that evaluation of this integral is non-trivial. To the author's knowledge, neither of these generalisations to the theoretical approach in this chapter have been implemented.

While some authors deem it necessary to include such dependencies in calculations [103], it must be remembered that often the interest lies in systems of weak alloys

such as $\text{Ga}_{1-x}\text{Al}_x\text{As}$ and $\text{Cd}_{1-x}\text{Mn}_x\text{Te}$ where the effective mass change between the well and barrier is relatively small. Nonetheless, the mechanisms by which the theory presented here can be extended have been mapped out and here lies an opportunity for the interested reader to explore such systems further.

5.9 BAND NON-PARABOLICITY

Small well widths and large potential barriers could require the inclusion of the non-parabolicity of the conduction band [106]. Ekenberg [113] described the inclusion of non-parabolicity on the subband structure of quantum wells. This method can account accurately for a variety of physical phenomena, but is analytically complicated. Simpler procedures have been proposed by various authors for the more complex problem of a donor in a quantum well. For example, Chaudhuri and Bajaj [106] have used the following simple replacement:

$$m^* \longrightarrow m^*(E) = a_0 + a_1 E + a_2 E^2 + a_3 E^3 + \cdots + a_n E^n \quad (5.128)$$

where a_n represents a series of constants. Given the values of these constants, the effects of non-parabolicity can be incorporated into equation (5.17) simply by making the effective mass a function of the energy E . This extension is certainly much more straightforward than the two described in the previous section.

In relation to the calculations below, it should be noted that Chaudhuri and Bajaj [106] showed that, even with relatively large potential barriers, band non-parabolicity was only significant for wells narrower than half the Bohr radius of the neutral donor, which, for the case of $\text{Ga}_{1-x}\text{Al}_x\text{As}$, would be $<50 \text{ \AA}$, and for $\text{Cd}_{1-x}\text{Mn}_x\text{Te}$, would be $<35 \text{ \AA}$.

5.10 EXCITED STATES

Just as there are many solutions to the Schrödinger equation for an isolated hydrogen atom, there are also many more solutions representing excited energy states of a donor in a heterostructure. Recalling the hydrogen-atom solutions [4]:

$$\psi_{1s} = e^{-\frac{r}{\lambda}}; \quad \psi_{2s} = \left(2 - \frac{r}{\lambda}\right) e^{-\frac{r}{2\lambda}}; \quad \psi_{2p_x} = x e^{-\frac{r}{2\lambda}} \quad (5.129)$$

and similarly for $2p_y$ and $2p_z$. The corresponding eigenenergy involves only the ground state and the principle quantum number, i.e. $E = E_1/n^2$, so very simply $E_2 = E_1/4$, $E_3 = E_1/9$, etc. In bulk semiconductors, the neutral donor does also exhibit these states (see [2], p. 314).

The situation is more complex in semiconductor heterostructures as the one-dimensional potential due to the layer structure breaks the symmetry of the spherical potential, and hence the wave function is more complex than for the hydrogen atom. Much detail has been given as to solving the corresponding ground state under these

conditions. In an analogy with the hydrogen atom the first excited state of the donor might be written as follows:

$$\Psi_{2s} = \chi(z) \left(1 - \frac{\alpha r''}{\lambda_{2s}} \right) e^{-\frac{r''}{\lambda_{2s}}} \quad (5.130)$$

where λ_{2s} has been labelled specifically as it cannot be assumed, a priori, that $\lambda_{2s} = 2\lambda_{1s}$. The constant α has been introduced, and is determined by ensuring orthogonality between the ground state and this the first excited state, i.e.

$$\langle \Psi_{1s} | \Psi_{2s} \rangle = \int_0^\infty \Psi_{1s}^* \Psi_{2s} d\tau = 2\pi \int_{-\infty}^\infty \int_0^\infty \Psi_{1s}^* \Psi_{2s} dr_\perp dz = 0 \quad (5.131)$$

Studies do exist in the literature of the excited states of donors in heterostructures [114], but to the author's present knowledge the extension utilizing the general form for $\Psi = \chi(z)\xi(x, y, z - r_d)$ has not yet been attempted. A simple alternative approach which yields reasonable results is presented later in this chapter.

5.11 APPLICATION TO SPIN-FLIP RAMAN SPECTROSCOPY IN DILUTED MAGNETIC SEMICONDUCTORS

5.11.1 Diluted magnetic semiconductors

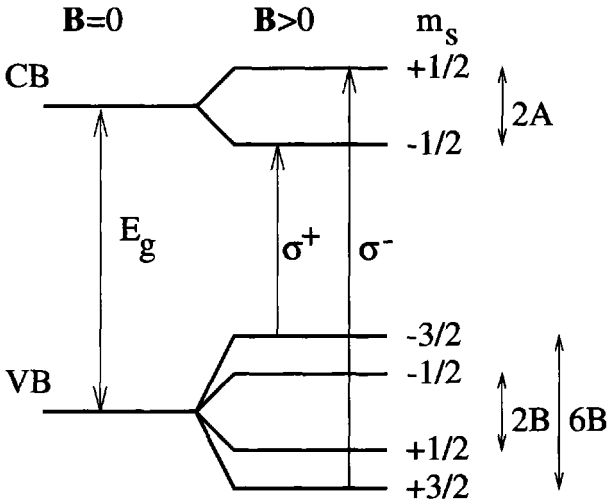


Figure 5.20 Zeeman splitting of conduction (CB) and valence (VB) bands of a diluted magnetic semiconductor within an external magnetic field B

Diluted magnetic semiconductors (DMSs) [115, 116] are important because of the strong exchange interaction between the hybridised sp^3 - d orbitals of the magnetic

ions and charge carriers. This manifests itself most clearly in the giant Zeeman splittings observed in both the conduction and valence bands when the material is placed in an external magnetic field. At low magnetic ion concentrations, the materials generally exhibit 'frustrated' paramagnetism, with the number of spin singlet states being reduced by nearest-neighbour anti-ferromagnetic spin-pairing [117]. At low magnetic fields (< 8 T) these spins remain locked and cannot contribute to the paramagnetism of the material; however experiments carried out under very high magnetic fields have been able to break these spin-doublets. The magnetic ion itself, usually Mn^{2+} , sits substitutionally on a cation site and can generally be incorporated to high concentrations. The most common DMS is $\text{Cd}_{1-x}\text{Mn}_x\text{Te}$, while others include $\text{Zn}_{1-x}\text{Mn}_x\text{S}$, and more recently $\text{Ga}_{1-x}\text{Mn}_x\text{As}$. Fig. 5.20 shows the Zeeman effect in a DMS material, with the vertical arrows linking the heavy-hole states ($\pm 3/2$) and the electron states, thus illustrating the allowed interband transitions under circularly polarized light, i.e. $|\text{hh}_{+3/2}\rangle \rightarrow |\text{e}_{+1/2}\rangle$ is the σ^- transition, and $|\text{hh}_{-3/2}\rangle \rightarrow |\text{e}_{-1/2}\rangle$ is the σ^+ transition.

The magnitudes of the conduction and valence band splittings are given in terms of the variables A and B as follows:

$$A = -\frac{1}{6}N_0\alpha x \langle S_z \rangle; \quad B = -\frac{1}{6}N_0\beta x \langle S_z \rangle \quad (5.132)$$

where $N_0\alpha$ and $N_0\beta$ are constants (220 and 880 meV respectively in $\text{Cd}_{1-x}\text{Mn}_x\text{Te}$). The expectation value of the magnetic ion spin along the z -axis $\langle S_z \rangle$ is given by:

$$\langle S_z \rangle = S_0(x)B_J(\mathbf{B}, T_{\text{eff}}) \quad (5.133)$$

where $S_0(x)$ is the effective spin of the magnetic ions and B_J is a Brillouin function describing the response of the spins in a magnetic fields \mathbf{B} . The effective spin S_0 accounts for the proportion of magnetic ions which are spin-paired with a nearest neighbour and can not respond to the alignment induced by the magnetic field. Alternatively the effective spin can be considered as the concentration of spin-singlet states, i.e.

$$xS_0(x) = \frac{5}{2}\bar{x} \quad (5.134)$$

where the spin of the Mn^{2+} ions are $5/2$. Recent theoretical studies [117] have calculated \bar{x} in agreement with experiment [118] and shown that for moderate fields (≈ 8 T), i.e. when the splittings have saturated, but before the nearest neighbour spin pairings are broken, the maximum value of $x \langle S_z(x) \rangle$ occurs at a manganese concentration $x \approx 0.15$ and is equal to 0.105. Hence, in $\text{Cd}_{1-x}\text{Mn}_x\text{Te}$ the maximum splitting in the conduction band is ≈ 23 meV, and for the heavy-holes in the valence band it is ≈ 92 meV. The paramagnetic behaviour falls off with increasing temperature.

When applying a magnetic field to a semiconductor heterostructure, the direction of the field becomes important. For fields parallel to the growth axis (z -) (the Faraday configuration), the splittings are still well represented by Fig. 5.20. However, application of the magnetic field along the plane of the wells, i.e. the Voigt configuration, leads to mixing of the light- and heavy-hole valence states, thus producing a

much more complex band structure. However, this lies beyond the scope of this short introduction.

Thus far, the $\text{CdTe}/\text{Cd}_{1-x}\text{Mn}_x\text{Te}$ system has been employed to illustrate donor binding energy calculations; now, however, specific use will be made of these magnetic properties in an application of the binding-energy calculations, namely spin-flip Raman spectroscopy.

5.11.2 Spin-flip Raman spectroscopy

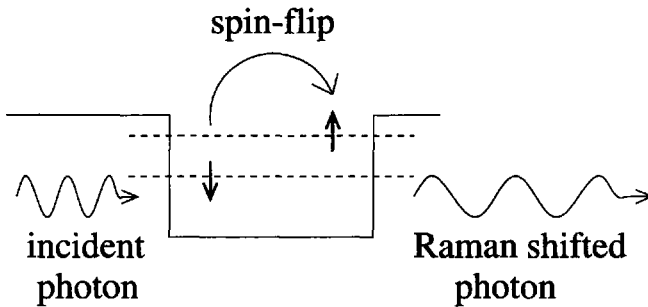


Figure 5.21 Schematic illustration of the spin-flip process

The application of a magnetic field to a heterostructure leads to a Zeeman splitting of the confined energy states, where in the case of an n -type material (i.e. material with donors present), two electron states are formed with the spin-up states having a higher energy than the spin-down states. At low temperatures, the carriers populate the lowest energy state, i.e. the spin-down, but can be excited by photons into the higher energy state. This excitation can be detected as a Raman shift [119], as illustrated schematically in Fig. 5.21.

This is a very powerful technique because the spectroscopy depends only upon one carrier type and theoretical modelling of the experimental data requires the parameter set of only one band. In this case, the parameters consist of the relative permittivity of the material, and the electron effective mass from bulk, together with the conduction band offset between the non-magnetic CdTe well and the magnetic $\text{Cd}_{1-x}\text{Mn}_x\text{Te}$ barriers [120].

The Raman shift is simply the energy difference between the two electron-donor spin states, $E_{sf} = E^{\uparrow} - E^{\downarrow}$. Fig. 5.22 displays the energy of these two spin states for a lightly n -doped $\text{CdTe}/\text{Cd}_{1-x}\text{Mn}_x\text{Te}$ system, calculated using the 3D trial wave function. The magnetic ion concentration in the barriers has been chosen as 15 % in order to maximize the energy difference between the states. Fig. 5.23 displays the spin-flip energy for this system, again as a function of donor position.

If the donors are assumed to be uniformly distributed across the entire heterostructure, it is possible to represent the data of Fig. 5.23 in the form of an intensity I versus spin-flip energy E Raman spectrum, by assigning a Gaussian distribution to

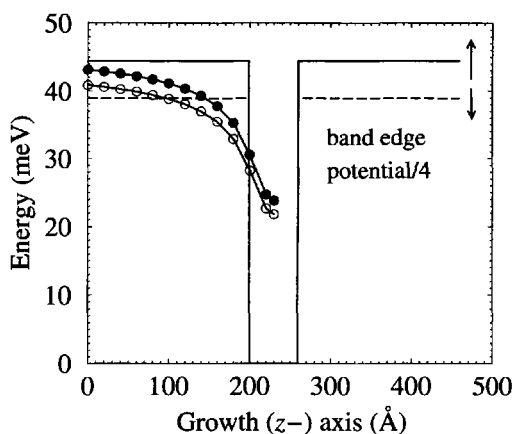


Figure 5.22 Energy of the spin-up (solid circles) and spin-down (open circles) states at a magnetic field of 8 T, for a range of donor positions across a 60 Å CdTe well surrounded by Cd_{0.85}Mn_{0.15}Te barriers

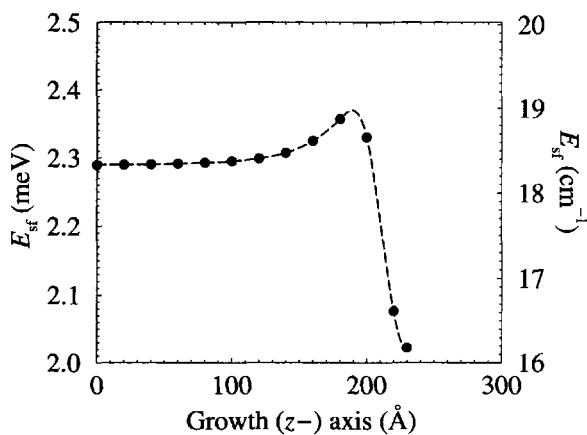


Figure 5.23 Spin-flip energy as a function of donor position

each point in energy space with a certain linewidth, i.e.

$$I = \sum_{r_d} \frac{1}{\sigma \sqrt{2\pi}} \exp \left[-\frac{(E - E_{sf}(r_d))^2}{2\sigma^2} \right] \quad (5.135)$$

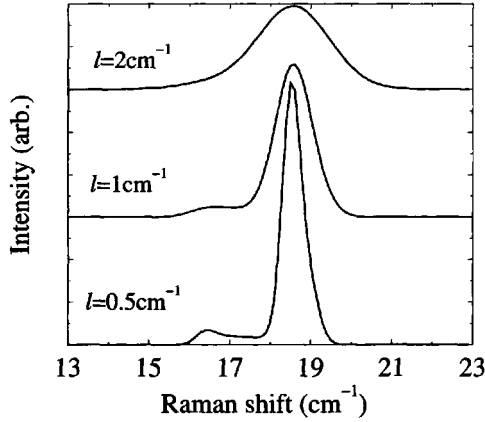


Figure 5.24 Intensity of Raman signal versus energy shift

where $E_{sf}(r_d)$ is the spin-flip energy of the donor at position r_d . The finite linewidth of any spin-flip signal will arise from microscopic fluctuations of the material parameters. For example, the well width, alloy fluctuations in the barrier, and the random nature of the donor distribution itself, will all give rise to a broadening of the signal from each of the donor positions r_d along the axis of the quantum well structure. The standard deviation σ is related to the linewidth l (full width at half-maximum) by:

$$\sigma = \frac{l}{2\sqrt{2 \ln 2}} \quad (5.136)$$

In order to save considerable computational effort, the donor calculations were performed only at the points marked by circles on Fig. 5.23. A spline of this data was produced, given by the dashed line, and then this more detailed curve was used to produce the intensity versus energy data of Fig. 5.24, which also shows the effect of different linewidths on the predicted Raman spectrum of this single quantum well.

In the lowest curve ($l = 0.5 \text{ cm}^{-1}$) it is possible in principle to resolve the donors in the well from those in the barrier, as observed experimentally [120]. Comparing Figs 5.23 and 5.24, it can be seen that the central peak at around 18.5 cm^{-1} corresponds to spin-flips of electrons bound to donors *in the barrier*. This peak in intensity is due entirely to the proportionately larger fraction of donors in the barrier compared to those in the well. In addition, spin-flips from electrons bound to donors in the well are clearly resolved as a small peak at about 16 cm^{-1} .

As the linewidth l of the signals is allowed to increase, the resolution and information in the simulated spectrum decreases. At a linewidth of 1 cm^{-1} , spin-flips due to donors in the well appear as just a shoulder on the larger 'barrier' peak, while by $l = 2 \text{ cm}^{-1}$ this information is lost altogether, and a broad central peak ensues.

In conclusion, the theoretical donor binding-energy calculations outlined above have been shown to be of direct relevance to simulating spin-flip Raman spectroscopy in diluted magnetic semiconductors. This allows the spatial distribution of donors to be investigated, although the importance of high quality samples has been shown to be significant.

It has been proposed that the magnetic behaviour of the first few monolayers of a dilute magnetic semiconductor adjacent to an interface with a non-magnetic semiconductor, could be significantly different from that of the bulk [121–123]. One contribution to this effect arises from a reduction in the number of antiferromagnetically coupled pairs due to a decrease in the number of nearest neighbour magnetic ions [121]. A single layer of donors, δ -doped into a quantum well structure in the region of the well–barrier interface could provide a useful probe of the magnetism—via the observations made from spin-flip Raman spectroscopy.

5.12 ALTERNATIVE APPROACH TO EXCITED IMPURITY STATES

The earlier method of calculating the energy levels of impurities within semiconductor heterostructures requires major mathematical analysis in order to extend to calculate excited states. In this section, a simpler, more general approach is developed that can handle the ground and excited states as well as the spatially degenerate states ($2s$, $2p_x$, $2p_z$) within a general heterostructure potential, that might also include an electric field (bias).

As in equation (5.4) the Hamiltonian of an impurity atom within a semiconductor heterostructure under the single-band effective mass and envelope function approximations is:

$$\mathcal{H} = -\frac{\hbar^2}{2} \frac{\partial}{\partial z} \frac{1}{m^*} \frac{\partial}{\partial z} + V(z) - \frac{e^2}{4\pi\epsilon r} \quad (5.137)$$

where again m^* is the effective mass of the charge carrier, $V(z)$ is the electrostatic potential which defines the heterostructure (and may include a bias $-eFz$) and r is the distance between the impurity and the charge carrier. Placing the x and y origins on the impurity atom, which is at a position r_i , then:

$$r^2 = x^2 + y^2 + (z - r_i)^2 \quad (5.138)$$

Taking a trial wave function of a charge carrier (electron or hole):

$$\Psi = \psi(z)\xi(r) \quad (5.139)$$

where $\psi(z)$ is the wave function of the electron (or hole) in the same heterostructure but *without* the impurity present and $\xi(r)$ is a hydrogenic like term describing the interaction between the electron (or hole) and the donor (or acceptor) ion.

For the $1s$ (ground) state of the impurity then the interaction term would be given by:

$$\xi(r) = \exp\left(-\frac{r}{\lambda}\right) \quad (5.140)$$

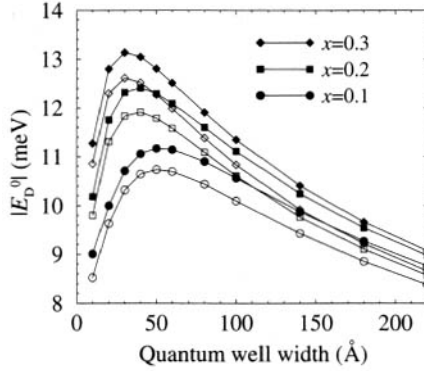


Figure 5.25 Comparison of the results of calculations using the approach developed in this section (open symbols) with the earlier method (closed symbols), presented in Fig. 5.13. The system is a donor at the centre of a GaAs quantum well surrounded by $\text{Ga}_{1-x}\text{Al}_x\text{As}$ barriers.

where λ is a variational parameter.

The variational calculation is implemented by adjusting λ in order to minimise the expectation value of the Hamiltonian operator (the total energy):

$$E = \frac{\langle \Psi | \mathcal{H} | \Psi \rangle}{\langle \Psi | \Psi \rangle} \quad (5.141)$$

The energy E is evaluated for different values of λ by direct numerical integration of the numerator and denominator in the above equation. For example,

$$\langle \Psi | \mathcal{H} | \Psi \rangle = \int_0^\infty \int_0^\infty \int_0^\infty \Psi \left(-\frac{\hbar^2}{2} \frac{\partial}{\partial x} \frac{1}{m^*} \frac{\partial}{\partial x} + V(z) - \frac{e^2}{4\pi\epsilon r} \right) \Psi \, dx \, dy \, dz \quad (5.142)$$

These integrals are calculated using a simple strip summation over a three-dimensional uniform mesh, with the differentials in the kinetic energy component evaluated using finite difference expansions.

5.13 DIRECT EVALUATION OF THE EXPECTATION VALUE OF THE HAMILTONIAN FOR THE GROUND STATE

Fig. 5.25 shows a comparison between the two numerical approaches of the ground state binding energy of a donor at the centre of a GaAs quantum well surrounded by $\text{Ga}_{1-x}\text{Al}_x\text{As}$ barriers. The closed symbols represent the data calculated using the three-dimensional trial wave function in Section 5.4, and presented in Fig. 5.13 while the open symbols show the results of calculations with the more direct numerical integration approach just developed.

It can be seen that for all three barrier heights (defined by three different Al concentrations, $x = 0.1, 0.2$ and 0.3), the function form of the binding energies with the quantum well width, is very similar. However, in all cases, the new data are around 0.5 meV below the earlier calculations.

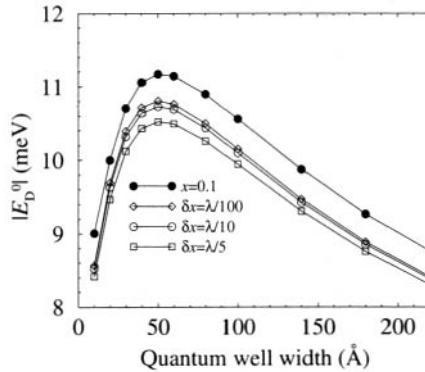


Figure 5.26 The effect of the numerical integration step length δx on the binding energy of a donor at the centre of a GaAs quantum well surrounded by $\text{Ga}_{0.9}\text{Al}_{0.1}\text{As}$ barriers.

Figs 5.26 and 5.28 explore the effect of two numerical parameters which could affect the accuracy of the calculations in this section.

In Fig. 5.26, the in-plane integration step length δx (equal to δy) was taken as $1/5$ th, $1/10$ th and $1/100$ th of the Bohr radius λ . The results of the calculations are compared to the $x = 0.1$ data from the original method. It can be seen that decreasing the size of the integration step length δx (which will increase the accuracy of any numerical integrations) does indeed move the new data closer to that of the earlier method. However, it also increases computational time, and given that this represents integration over a plane, increasing the number of steps by a factor of 10 (from $\lambda/10$ to $\lambda/100$, say), increases the computational time by a factor of 100. The step length $\delta x = \lambda/10$ is a good compromise and is adopted in all subsequent calculations.

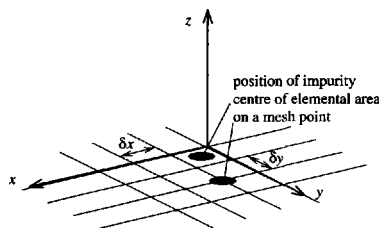


Figure 5.27 Illustration of the possible choices of the position of the impurity within the integration mesh.

The second computational parameter is the position of the impurity itself within the integration mesh. This would not normally be an issue, but in this case with impurities, the Coulomb potential $-e^2/(4\pi\epsilon r)$ has a singularity at $r = 0$. This can be avoided if the impurity is placed anywhere *within* the elemental area $\delta x \delta y$, and not on a mesh point itself, see Fig. 5.27.

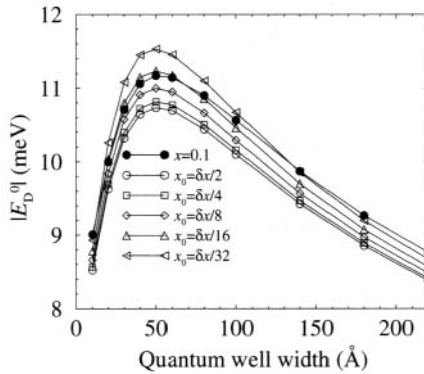


Figure 5.28 The effect of the position of the origin within the elemental volume on the binding energy of a donor at the centre of a GaAs quantum well surrounded by $\text{Ga}_{0.9}\text{Al}_{0.1}\text{As}$ barriers.

Fig. 5.28 shows the results of calculations of the effect of the origin (x_0, y_0) of the in-plane integration mesh with respect to the position of the impurity. It can be seen that the position of the origin can make up to nearly 1 meV difference in the binding energy, and as the impurity is brought closer and closer to a mesh point ($\delta x \rightarrow 0$) the binding energy increases. The latter is not surprising because the mesh points represent the points at which the deep Coulomb potential is sampled. The closer it is sampled to the impurity then the deeper the potential and this value makes a greater contribution to the integral. In all subsequent calculations the mesh origin was chosen so that the impurity lay in the centre of an elemental area, this also has the additional justification of symmetry.

5.14 VALIDATION OF THE MODEL FOR THE POSITION DEPENDENCE OF THE IMPURITY

It has thus been established that this approach to the calculation of the ground state energy level of an impurity at the centre of a quantum well is in good agreement with the earlier technique. However, it is now important to validate the new approach for various impurity positions—not restricting the impurity to the highly symmetric position at the centre of a quantum well. This is particularly important because of the choice of the impurity wave function as the product of the envelope function without

the impurity present with the hydrogenic term i.e. $\Psi = \psi(z)\xi(x, y, z, r_i)$, rather than the original method which recalculated the envelope function.

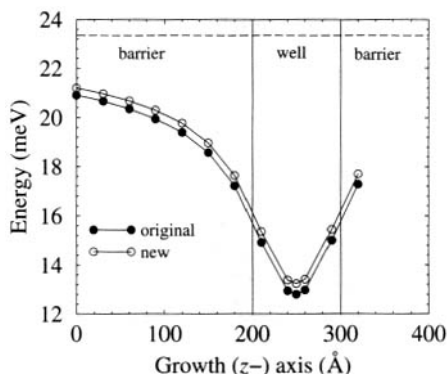


Figure 5.29 Comparison of the results of calculations using the approach developed in this section (open symbols) with the earlier method (closed symbols). The figure shows the total energy of an electron in a 100 Å GaAs quantum well surrounded by 200 Å $\text{Ga}_{0.9}\text{Al}_{0.1}\text{As}$ barriers, as a function of the position of the donor across the barrier–well–barrier system.

Fig. 5.29 shows the results of a series of calculations of the ground state donor energy for a range of donor positions across the barrier, well and barrier regions of a single 100 Å GaAs quantum well surrounded by 200 Å barriers of $\text{Ga}_{0.9}\text{Al}_{0.1}\text{As}$. The figure also shows the results from the earlier method and it can be seen that the difference in the energy of the donor is around 0.5 meV, which is quite acceptable in device design.

The calculations span the quantum well to show that the expected symmetry of the energy with position is reproduced. The horizontal dashed line near the top of the figure shows the energy of an electron in the same heterostructure but without a donor present.

5.15 EXCITED STATES

Thus, it now remains to move onto the initial purpose of this work which is to develop a model of the energy levels of *excited* impurity states. Taking the hydrogenic factor ξ of the total impurity wave function as:

$$\xi = \left(1 - \frac{r}{\lambda}\right) \exp\left(-\frac{r}{\lambda}\right) \quad (5.143)$$

where $r = \sqrt{x^2 + y^2 + (z - r_i)^2}$, the energy level of the excited 2s state of a donor in the centre of a GaAs quantum well can be calculated. Strictly speaking the total impurity wave function $\Psi = \psi(z)\xi(x, y, z, r_i)$ for this excited 2s state is not orthogonal

to the corresponding 1s wave function and a more complicated approach incorporating a factor ' α ' in ξ should be used. However, the results that follow indicate that the simple form taken in equation (5.143) is a good approximation for obtaining the energy of excited impurity states in heterostructures.

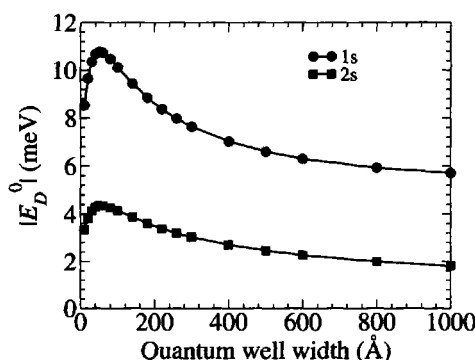


Figure 5.30 The well width dependence of the binding energies of the 1s and 2s states of donors at the centre of GaAs quantum wells surrounded by $\text{Ga}_{1-x}\text{Al}_x\text{As}$ barriers.

Fig. 5.30 shows the binding energy of the 1s and the 2s states as a function of the width of the GaAs quantum well. It can be seen that the binding energy of the 1s state is much larger than that of the 2s (as expected) and that the functional form of both curves is the same. Fig. 5.31 shows the ratio of these two binding energies. Bohr theory [4] gives the (binding) energy of the hydrogen atom as $E_n = E_1/n^2$, where E_1 is the (binding) energy of the lowest ($n = 1$) state. Thus for a hydrogen atom the ratio of the energy of the 1s state to the 2s state is exactly 4. In the figure, this number is much smaller, however as the width of the well is increased, and the system tends towards bulk, this ratio does begin to move towards 4. This is supporting evidence in favour of the application of this method to the calculation of the binding energy of excited impurity states in heterostructures.

Fig. 5.32 shows the binding energies of the 2s and $2p_x$ impurity levels for a donor at the centre of a GaAs quantum well, as a function of the quantum well width, again the barrier material is $\text{Ga}_{0.9}\text{Al}_{0.1}\text{As}$. The hydrogenic factor for the $2p_x$ state was taken as:

$$\xi = x \exp\left(-\frac{r}{\lambda}\right) \quad (5.144)$$

It can be seen that the binding energy of the $2p_x$ state is considerably less than that of the 2s state—only around half around the peak at 100 Å. However, as the width of the quantum well is increased the binding energy of the $2p_x$ level converges towards

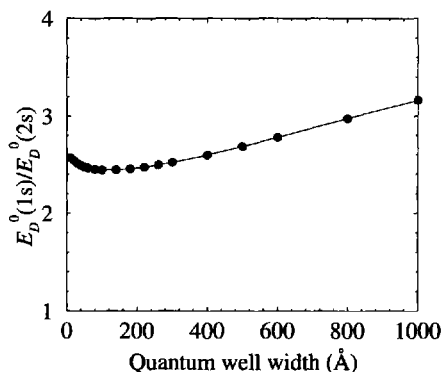


Figure 5.31 The well width dependence of the ratio of the binding energies of the 1s and 2s states of donors at the centre of GaAs quantum wells surrounded by $\text{Ga}_{1-x}\text{Al}_x\text{As}$ barriers.

that of the 2s level, as would be expected, because in the limit of an infinitely wide well (i.e. bulk material), these states should be degenerate.

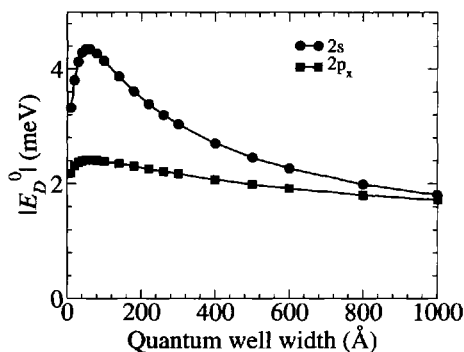


Figure 5.32 The well width dependence of the binding energies of the 2s and $2p_x$ states of donors at the centre of GaAs quantum wells surrounded by $\text{Ga}_{1-x}\text{Al}_x\text{As}$ barriers.

Figs 5.33 shows the Bohr radii (the value of the variational parameter) λ for the 1s and $2p_x$ states. Fig. 5.34 shows the ratio of these two radii. It can be seen that it is very close to 2—as expected from Bohr theory. In fact, the $2p_x$ state is a very good one to calculate because near the impurity, where the singularity exists, the functional form

of the hydrogenic term, i.e. $x \exp(-r/\lambda)$, naturally tends to zero, hence reducing any inaccuracies that might arise because of integrating over the singularity.

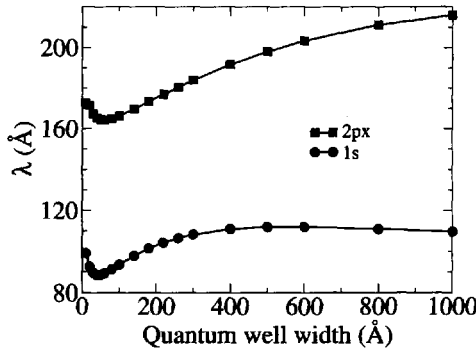


Figure 5.33 The well width dependence of the Bohr radius λ of the 1s and $2p_x$ states of donors at the centre of GaAs quantum wells surrounded by $\text{Ga}_{1-x}\text{Al}_x\text{As}$ barriers.

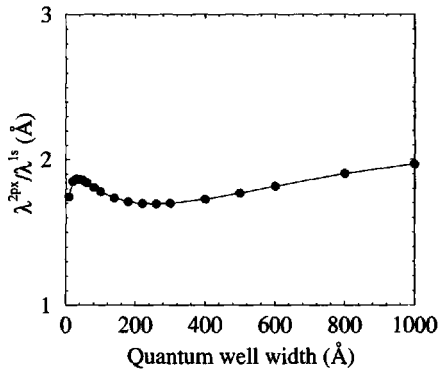


Figure 5.34 The well width dependence of the *ratio* of the Bohr radii of the 1s and $2p_x$ states of donors at the centre of GaAs quantum wells surrounded by $\text{Ga}_{1-x}\text{Al}_x\text{As}$ barriers.

Fig. 5.35 shows the binding energy of the $2p_z$ impurity state, calculated using a hydrogenic term analogous as that for the $2p_x$ state, as in equation (5.144), but with the 'lobes' aligned along the growth axis. The lobes now overlap with the repulsive potentials of the heterostructure barriers and hence the energy of the impurity state is increased. This *decreases* the binding energy, and in fact, as the figure shows, the

binding energy becomes negative, i.e. the state is unbound. This effect is reduced as the well becomes wider and the z -lobes can 'fit' inside the well. The binding energy of the $2p_z$ state tends to that of the $2s$ and $2p_x$ states as the well approaches 1000 \AA .

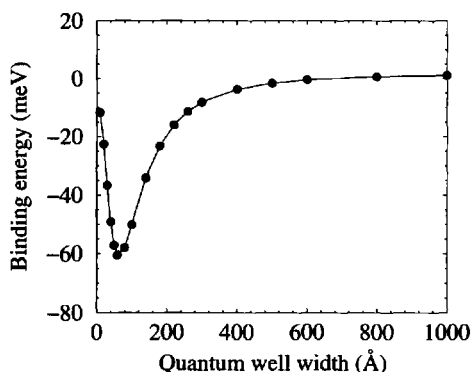


Figure 5.35 The well width dependence of the binding energy of the $2p_z$ state of donors at the centre of GaAs quantum wells surrounded by $\text{Ga}_{0.9}\text{Al}_{0.1}\text{As}$ barriers.

5.16 IMPURITY OCCUPANCY STATISTICS

Any book on solid state physics, for example Kasap [124] (page 314) or Sze [44] (page 16), gives the number of free electrons introduced into a crystal due to doping as the integral over the entire conduction band of the density of states multiplied by the probability of occupation:

$$n = \int_{E_c}^{\infty} \rho(E) f(E) dE \quad (5.145)$$

where the density of states $\rho(E)$ (derived earlier in Section 2.3) is written in relation to the conduction band edge E_c as:

$$\rho(E) = \frac{1}{2\pi^2} \left(\frac{2m^*}{\hbar^2} \right)^{\frac{3}{2}} (E - E_c)^{\frac{1}{2}} \quad (5.146)$$

and the Fermi–Dirac[†] distribution function $f(E)$ is approximated by a Boltzmann distribution function:

$$f(E) = \exp \left(-\frac{(E - E_F)}{kT} \right) \quad (5.147)$$

[†]This is the true Fermi–Dirac distribution function for a doped semiconductor and not the quasi-function introduced earlier to describe the distribution of electrons within a subband.

The number of free electrons (which is the same as the number of ionised donors) then follows as:

$$n = \frac{1}{2\pi^2} \left(\frac{2m^*}{\hbar^2} \right)^{\frac{3}{2}} \int_{E_c}^{\infty} (E - E_c)^{\frac{1}{2}} \exp \left(-\frac{(E - E_F)}{kT} \right) dE \quad (5.148)$$

which leads to (see Kasap [124] (page 314)):

$$n = N_c \exp \left(-\frac{(E_c - E_F)}{kT} \right) \quad (5.149)$$

where N_c is known as the 'effective density of states' and is given by:

$$N_c = 2 \left(\frac{2\pi m^* kT}{h^2} \right)^{\frac{3}{2}} \quad (5.150)$$

and the temperature dependent Fermi energy for a donor volume density N_d and energy E_d is given by:

$$E_F = \frac{E_c + E_d}{2} + \frac{1}{2} kT \ln \left(\frac{N_d}{2N_c} \right) \quad (5.151)$$

see for example Kasap [124] (page 335). It can be seen from equation (5.151) that at low temperatures E_F would be halfway between the donor energy levels E_d and the conduction band edge E_c as expected.

Figure 5.36 shows the results of calculations of the number of ionised donors N_d^+ ($=n$) as a function of the temperature T for several different doping densities N_d . It can be seen from the figure that as the doping density increases the proportion of donors ionised decreases at any given temperature. This is just a result of statistical mechanics—the effective density of states N_c remains constant, so as more donors are added to the system a smaller *proportion* of them can find empty states for their electrons to move to. It is interesting to note that at the typical doping density of 10^{18} cm^{-3} only around one third of donors are ionised at room temperature.

The focus of this book is on low-dimensional systems so generalising to the two dimensions of a quantum well heterostructure, then the density of states would become the two-dimensional density of states again derived in Section 2.3:

$$\rho^{2D}(E) = \frac{m^*}{\pi \hbar^2} \quad (5.152)$$

hence:

$$n = \frac{m^*}{\pi \hbar^2} \int_{E_c}^{\infty} \exp \left(-\frac{(E - E_F)}{kT} \right) dE = \frac{m^*}{\pi \hbar^2} \left[-kT \exp \left(-\frac{(E - E_F)}{kT} \right) \right]_{E_c}^{\infty} \quad (5.153)$$

$$\therefore n = \frac{m^* kT}{\pi \hbar^2} \exp \left(-\frac{(E_c - E_F)}{kT} \right) \quad (5.154)$$

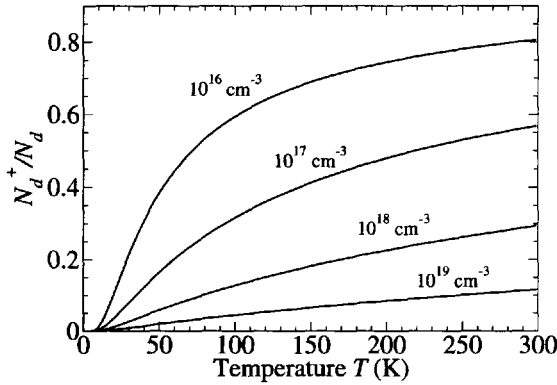


Figure 5.36 The proportion of ionised donors N_d^+ from a doping concentration N_d as a function of temperature for a bulk semiconductor, for several different dopant densities. The material system is GaAs with a donor energy 5.8 meV below the conduction band edge and an effective mass of $0.067 m_0$.

which in analogy to three dimensions, could also be written:

$$n = N_c^{2D} \exp \left(-\frac{(E_c - E_F)}{kT} \right) \quad (5.155)$$

where the two-dimensional effective density of states:

$$N_c^{2D} = \frac{m^* kT}{\pi \hbar^2} \quad (5.156)$$

and the Fermi energy would be equivalent to that in equation (5.151) but with N_c^{2D} instead of N_c and the donor density N_d would be taken as a *sheet density*.

Figure 5.37 shows the corresponding curve to Fig. 5.36 for a two-dimensional system. The generic behaviour is the same as in bulk, i.e. that the higher the temperature the greater proportion of donors are ionised and the greater the doping density the lower the proportion of ionised donors. Again the number of free carriers generated by doping might seem surprisingly small.

Although this view has been recently reinforced [125] it is actually too simplistic and the proportion of carriers ionised is too small. The reason is that the model assumes that the impurity ionisation energy (which in this section would be $E_c - E_d$) is left as a constant, whereas it was found that screening should see it reduce as the impurity concentration increases [126]. Pearson and Bardeen [126] argued that the ionisation energy of an impurity would be reduced by an energy inversely proportional to the distance between the impurities, hence the ionisation energy should be equal

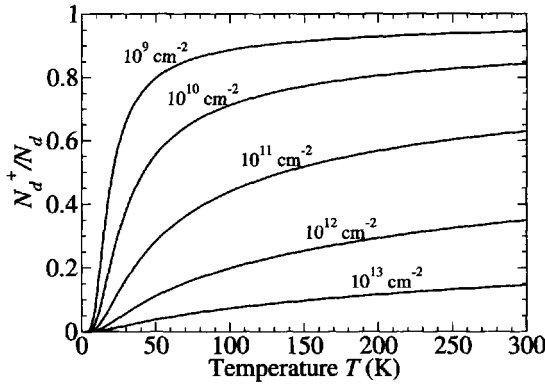


Figure 5.37 The proportion of ionised donors N_d^+ from a doping concentration N_d as a function of temperature for a two-dimensional (quantum well) system, for several different dopant densities. The material system is GaAs with a donor energy 5.8 meV below the conduction band edge and an effective mass of $0.067 m_0$.

to:

$$|E_D^0| - aN_d^{\frac{1}{3}}$$

where E_D^0 is the ionisation (binding) energy of the isolated impurity and N_d is the impurity (donor) concentration. The constant a depends on the impurity species and the host and was deduced experimentally by Pearson and Bardeen for boron and phosphorus in silicon.

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CHAPTER 6

EXCITONS

6.1 EXCITONS IN BULK

If photons of energy comparable to the band gap are incident on a semiconductor, then they can be absorbed by the electrons forming atomic bonds between neighbouring atoms, and so provide them with enough energy to break free and move around in the body of the crystal. Within the band theory of solids, this would be described as 'exciting an electron from the valence band across the band gap into the conduction band'. If the energy of the photon is larger than the band gap, then a free electron is created *and* an empty state is left within the valence band (see 'high energy excitation' in Fig. 6.1). The empty state within the valence band behaves very much like an air bubble in a liquid and rises to the top—the lowest energy state. This 'hole' behaves as though it were positively charged and hence often forms a bond with a conduction-band electron (see 'exciton formation' in Fig. 6.1). The attractive potential leads to a reduction (by an amount E_{X^0}) in the total energy of the electron and hole. This bound electron–hole pair is known as an 'exciton'. Photons of energy just below the band gap can be absorbed, thus creating excitons directly (see 'resonant excitation').

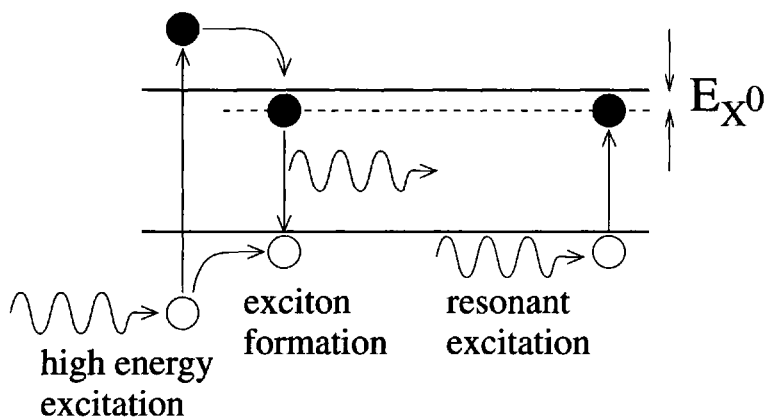


Figure 6.1 Schematic representation of the non-resonant and resonant generation of excitons

As the hole mass is generally much greater than the electron mass, then the two-body system resembles a hydrogen atom, with the negatively charged electron orbiting the positive hole. The exciton is quite stable and can have a relatively long lifetime, of the order of hundreds of ps to ns. Exciton recombination is an important feature of low temperature photoluminescence, although as the binding energies are relatively low, i.e. a few meV to a few tens of meV, they tend to dissociate at higher temperatures.

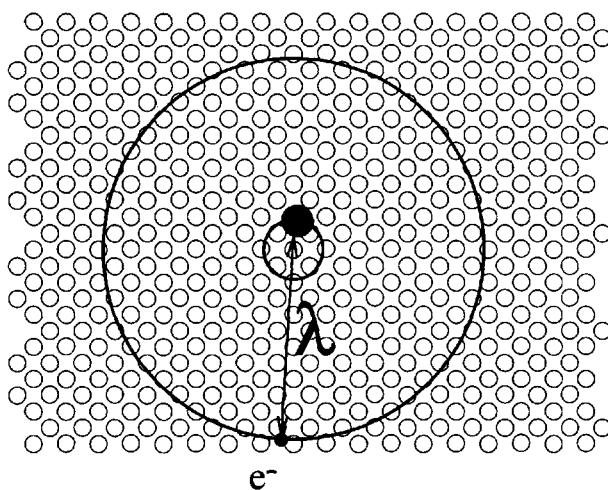


Figure 6.2 Schematic diagram of an exciton in bulk material, with the hole (filled circle near centre) and electron 'separated' by a Bohr radius λ , orbiting around the centre of mass

Therefore, in a similar manner to the hydrogenic impurities discussed in Chapter 5, the binding energy and orbital radius can be represented well by Bohr theory, *with*

the correction for the finite mass of the central charge [3,4]. This is implemented by exchanging the (in the case of a hydrogen atom, or donor) orbiting electron mass, with the reduced mass of the two-body system, in this case, the electron-hole pair. The reduced mass is given by:

$$\frac{1}{\mu} = \frac{1}{m_e^*} + \frac{1}{m_h^*} \quad (6.1)$$

Hence, the binding energy becomes:

$$E_{X^0} = -\frac{\mu e^4}{32\pi^2 \hbar^2 \epsilon_r^2 \epsilon_0^2} \quad (6.2)$$

and the Bohr radius follows as:

$$\lambda = \frac{4\pi\epsilon_r\epsilon_0\hbar^2}{\mu e^2} \quad (6.3)$$

Taking typical values for bulk GaAs, i.e. the Γ valley electron and heavy-hole effective masses, $m_e^* = 0.067m_0$ and $m_{hh}^* = 0.62m_0$, respectively, then $\mu = 0.060m_0$. Using the static dielectric constant $\epsilon_r = 13.18$ [14], then the exciton binding energy and Bohr radius follow respectively, as:

$$E_{X^0} = -4.7 \text{ meV} \quad \text{and} \quad \lambda = 115 \text{ \AA} \quad (6.4)$$

which are exactly the same as the low-temperature measured values (see [14], p.420). Correspondingly for CdTe, $m_e^* = 0.096m_0$ and $m_{hh} = 0.6m_0$, therefore $\mu = 0.083m_0$, with $\epsilon_r = 10.6$, then:

$$E_{X^0} = -10.1 \text{ meV} \quad \text{and} \quad \lambda = 67 \text{ \AA} \quad (6.5)$$

which again agree with experiment [127].

6.2 EXCITONS IN HETEROSTRUCTURES

In the same way as in bulk, excitons can be formed by the bonding of free electron-free hole pairs or through resonant excitation. Whereas in bulk, the total energy of the exciton is simply the energy of the free electron-free hole pair (i.e. the band gap) plus the exciton binding energy E_{X^0} , in a heterostructure there are additional components due to the electron and hole confinement energies, i.e.

$$E = E_g + E_{X^0} \text{ (bulk)} \quad E = E_g + E_e + E_h + E_{X^0} \text{ (heterostructure)} \quad (6.6)$$

The total exciton energy is clearly a function of structure because of the structural dependency of the confinement energies. In addition, it must be expected that the Coulombic potential energy, i.e. E_{X^0} , also depends upon the structure. This latter

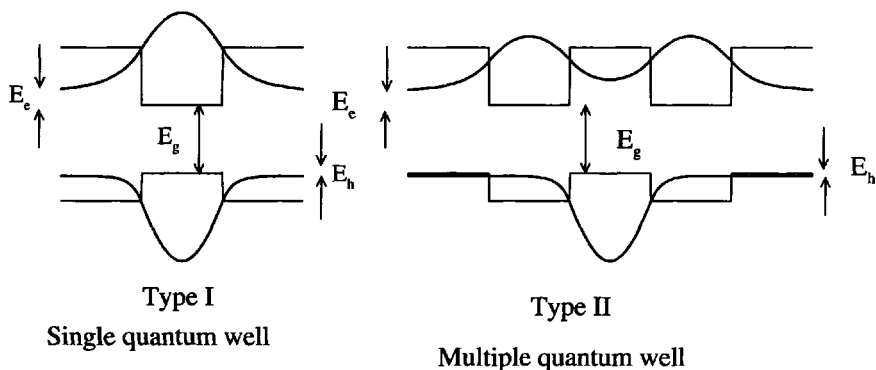


Figure 6.3 Electrons and holes in Type-I and Type-II systems

effect arises because the electron-hole separation can vary quite considerably between heterostructures. Fig. 6.3 illustrates this schematically; clearly the electron and hole separation is much smaller in the Type-I quantum well where both particles are localised in the same layer of semiconductor, than in the Type-II system, where they are localised in different layers. Hence, the exciton binding energy will be larger in the former.

6.3 EXCITON BINDING ENERGIES

The Hamiltonian representing the interacting two-body electron-hole complex can be considered to be the sum of three terms

$$\mathcal{H} = \mathcal{H}_e + \mathcal{H}_h + \mathcal{H}_{e-h} \quad (6.7)$$

where \mathcal{H}_e and \mathcal{H}_h are the one-particle Hamiltonians appropriate to the conduction and valence bands, respectively, of the particular microstructure of interest (as in Chapter 2). However, in the context of the exciton a constant effective mass along the growth (z -) axis must be assumed, i.e. the one-particle Hamiltonians are written:

$$\mathcal{H}_e = -\frac{\hbar^2}{2m_e^*} \frac{\partial^2}{\partial z^2} + V_{CB}(z) \quad ; \quad \mathcal{H}_h = -\frac{\hbar^2}{2m_h^*} \frac{\partial^2}{\partial z^2} + V_{VB}(z) \quad (6.8)$$

The third term, \mathcal{H}_{e-h} represents the electron-hole interaction, and is itself composed of two terms. One of these corresponds to the kinetic energy of the relative motion of the electron and hole in the x - y plane (perpendicular to the growth axis), while the

other represents the Coulombic potential energy, i.e.

$$\mathcal{H}_{e-h} = \frac{\mathcal{P}_{\perp}^2}{2\mu_{\perp}} - \frac{e^2}{4\pi\epsilon r} \quad (6.9)$$

where \mathcal{P}_{\perp} is the quantum mechanical momentum operator for the in-plane component of the relative motion. Now:

$$\mathcal{P} = -i\hbar\nabla = -i\hbar\left(\frac{\partial}{\partial x}\mathbf{i} + \frac{\partial}{\partial y}\mathbf{j} + \frac{\partial}{\partial z}\mathbf{k}\right) \quad (6.10)$$

$$\therefore \mathcal{P}_{\perp}^2 = -\hbar^2\left(\frac{\partial^2}{\partial x^2} + \frac{\partial^2}{\partial y^2}\right) \quad (6.11)$$

and r is simply the electron-hole separation, given by:

$$r^2 = (x_e - x_h)^2 + (y_e - y_h)^2 + (z_e - z_h)^2 \quad (6.12)$$

Note, that as the main motivation is concerned with the optical properties of excitons in quantum well systems, the Hamiltonian then contains no term for the motion of the centre of mass of the exciton in the x - y plane, i.e. the exciton is assumed to be at rest within the plane of the well.

The problem, as always, is to find the eigenfunctions Ψ and eigenvalues E of the Schrödinger equation:

$$\mathcal{H}\Psi = E\Psi \quad (6.13)$$

which describes the system. Following standard procedures, the two-body exciton wave function Ψ is chosen to be a product of three factors, as follows:

$$\Psi = \psi_e(z_e)\psi_h(z_h)\psi_r \quad (6.14)$$

where ψ_r represents the electron and hole relative motion. Following Hilton *et al.* [128, 129] and Harrison *et al.* [130, 131] ψ_r will be a variational wave function employed to minimize the total energy E of the system. The specific form of ψ_r will be discussed later. The other two factors, $\psi_e(z_e)$ and $\psi_h(z_h)$ are simply the eigenfunctions of the one-particle Hamiltonians of the heterostructure:

$$\mathcal{H}_e\psi_e(z_e) = E_e\psi_e(z_e) \quad ; \quad \mathcal{H}_h\psi_h(z_h) = E_h\psi_h(z_h) \quad (6.15)$$

One of the main advantages of this formalism is that it is independent of the form of the one-particle Hamiltonians \mathcal{H}_e and \mathcal{H}_h , and indeed calculations can be performed on any system in which the standard electron and hole wave functions can be calculated [77, 132, 133].

Multiplying equation (6.13) on the left by Ψ and integrating over all space, then the total exciton energy follows simply as the expectation value:

$$E = \frac{\langle \Psi | \mathcal{H} | \Psi \rangle}{\langle \Psi | \Psi \rangle} \quad (6.16)$$

Now, whatever the form of the relative motion term ψ_r of equation (6.14), it will be a function of the electron-hole separation, which is quantified in terms of the three variables, $x = |x_e - x_h|$, $y = |y_e - y_h|$ and $a = |z_e - z_h|$. The denominator \mathcal{D} in equation (6.16) can therefore be written as:

$$\mathcal{D} = \langle \Psi | \Psi \rangle = \int_{\text{all space}} |\psi_e(z_e)|^2 |\psi_h(z_h)|^2 |\psi_r|^2 \, dx \, dy \, dz_e \, dz_h \quad (6.17)$$

Since ψ_e and ψ_h are functions of z only, then the integrations over the x - y plane will involve only the relative motion term ψ_r . Correspondingly, the integrations over the electron and hole coordinates, z_e and z_h will yield a result which is a function of the electron and hole separation (a) only, i.e. it is possible to write:

$$\int |\psi_r|^2 \, dx \, dy = F(a) \quad (6.18)$$

Although the one-particle wave functions, ψ_e and ψ_h , in principle, extend to $\pm\infty$, in practice, from a computational viewpoint, it is necessary to define them between 'effective infinities'. As discussed in Chapter 3, these are defined so that:

$$\int_{\mu}^{\nu} \psi_e(z_e) \, dz_e \approx \int_{-\infty}^{+\infty} \psi_e(z_e) \, dz_e \quad (6.19)$$

and similarly for ψ_h . Using these results, the denominator \mathcal{D} in equation (6.17) can be re-written as:

$$\begin{aligned} \mathcal{D} = & \int_0^{\nu-\mu} \int_{\mu}^{\nu} \int_{\mu}^{\nu} |\psi_e(z_e)|^2 |\psi_h(z_h)|^2 F(a) \, dz_e \, dz_h \\ & \times [\delta(z_e - z_h - a) + \delta(z_h - z_e - a)] \, da \end{aligned} \quad (6.20)$$

where the Dirac δ -functions have been introduced in order to ensure that contributions are included from both $z_e - z_h = a$ and $z_h - z_e = a$. If the integration over z_h is performed first, then the first term only has a finite value when $z_h = z_e - a$ and the second term when $z_h = z_e + a$; hence:

$$\begin{aligned} \mathcal{D} = & \int_0^{\nu-\mu} \int_{\mu+a}^{\nu} |\psi_e(z_e)|^2 |\psi_h(z_e - a)|^2 F(a) \, dz_e \, da \\ & + \int_0^{\nu-\mu} \int_{\mu}^{\nu-a} |\psi_e(z_e)|^2 |\psi_h(z_e + a)|^2 F(a) \, dz_e \, da \end{aligned} \quad (6.21)$$

Making the substitution $z = z_e - a$ in the first term, and relabelling z as z_e in the second, finally gives the denominator in equation (6.16) as:

$$\mathcal{D} = \int_0^{\nu-\mu} p(a) F(a) \, da \quad (6.22)$$

where $p(a)$ represents the uncorrelated probability of finding the electron and hole separated by a distance a , i.e.

$$p(a) = \int_{\mu}^{\nu-a} |\psi_e(z+a)|^2 |\psi_h(z)|^2 + |\psi_e(z)|^2 |\psi_h(z+a)|^2 dz \quad (6.23)$$

Returning to the expression for the expectation value of the total exciton energy in equation (6.16) and using the three-term Hamiltonian of equation (6.7), the numerator \mathcal{N} can then be written as:

$$\mathcal{N} = \langle \Psi | \mathcal{H} | \Psi \rangle = \langle \Psi | \mathcal{H}_e | \Psi \rangle + \langle \Psi | \mathcal{H}_h | \Psi \rangle + \langle \Psi | \mathcal{H}_{e-h} | \Psi \rangle \quad (6.24)$$

Labelling these three terms as \mathcal{A} , \mathcal{B} and \mathcal{C} , respectively, i.e. $\mathcal{N} = \mathcal{A} + \mathcal{B} + \mathcal{C}$, then consider:

$$\mathcal{A} = \langle \Psi | \mathcal{T} | \Psi \rangle + \langle \Psi | \mathcal{V} | \Psi \rangle \quad (6.25)$$

where \mathcal{T} and \mathcal{V} are the kinetic and potential energy operators, respectively, of the electron one-particle Hamiltonian \mathcal{H}_e . Employing the standard constant-mass kinetic operator then (in an obvious notation) obtain:

$$\mathcal{A}_T = \int \psi_e^*(z_e) \psi_h^*(z_h) \psi_r^* \frac{-\hbar^2}{2m_e^*} \frac{\partial^2}{\partial z_e^2} \psi_e(z_e) \psi_h(z_h) \psi_r dz_e dz_h dx dy \quad (6.26)$$

On performing the differentiations:

$$\begin{aligned} \mathcal{A}_T = \int \psi_e^*(z_e) \psi_h^*(z_h) \psi_r^* \frac{-\hbar^2}{2m_e^*} [\psi_e''(z_e) \psi_h(z_h) \psi_r + 2\psi_e'(z_e) \psi_h(z_h) \psi_r' + \\ \psi_e(z_e) \psi_h(z_h) \psi_r''] dz_e dz_h dx dy \end{aligned} \quad (6.27)$$

which can be written as follows:

$$\begin{aligned} \mathcal{A}_T = \int \frac{-\hbar^2}{2m_e^*} [\psi_e^*(z_e) \psi_e''(z_e) |\psi_h(z_h)|^2 |\psi_r|^2 + \\ 2\psi_e^*(z_e) \psi_e'(z_e) |\psi_h(z_h)|^2 \psi_r^* \psi_r' + |\psi_e(z_e)|^2 |\psi_h(z_h)|^2 \psi_r^* \psi_r''] dz_e dz_h dx dy \end{aligned} \quad (6.28)$$

Note that the first term, when normalised by the denominator \mathcal{D} , merely represents the one-particle kinetic energy of the electron, which coupled together with $\langle \Psi | \mathcal{V} | \Psi \rangle$ in equation (6.25) gives the one-particle electron energy E_e as defined in equation (6.15). Furthermore, noting that for the problems of interest, i.e. stationary states within semiconductor microstructures, the wave functions are real and that the chain rule gives:

$$\psi_e(z_e) \psi_e'(z_e) = \frac{1}{2} \frac{\partial \psi_e^2(z_e)}{\partial z_e} \quad (6.29)$$

Equation (6.25) therefore becomes:

$$\mathcal{A} = E_e \mathcal{D} + \int \frac{-\hbar^2}{2m_e^*} \left\{ |\psi_h(z_h)|^2 \left[\frac{\partial \psi_e^2(z_e)}{\partial z_e} \right] \psi_r \psi_r' + \right.$$

$$|\psi_e(z_e)|^2 |\psi_h(z_h)|^2 \psi_r \psi_r'' \Big\} dz_e dz_h dx dy \quad (6.30)$$

Integrate, by parts, over dz_e the first term in the integrand, i.e.:

$$\int |\psi_h(z_h)|^2 \left[\frac{\partial |\psi_e(z_e)|^2}{\partial z_e} \right] \psi_r \psi_r' dz_e =$$

$$|\psi_h(z_h)|^2 [|\psi_e(z_e)|^2 \psi_r \psi_r']_\mu^\nu - \int (\psi_r'^2 + \psi_r \psi_r'') |\psi_e(z_e)|^2 |\psi_h(z_h)|^2 dz_e \quad (6.31)$$

Hence, substituting back into equation (6.30) gives:

$$\mathcal{A} = E_e \mathcal{D} + \int \frac{-\hbar^2}{2m_e^*} |\psi_h(z_h)|^2 \left\{ [|\psi_e(z_e)|^2 \psi_r \psi_r']_\mu^\nu \right.$$

$$\left. - \int [|\psi_e(z_e)|^2 \psi_r'^2 + |\psi_e(z_e)|^2 \psi_r \psi_r'' - |\psi_e(z_e)|^2 \psi_r \psi_r''] dz_e \right\} dz_h dx dy \quad (6.32)$$

The last two terms in the integration over z_e obviously cancel out, and in addition, examination of the first term on the right-hand side, shows that this term will not contribute, since whatever the form of the eigenfunction ψ_e , it will undoubtedly vanish at the effective infinities μ and ν . Therefore:

$$\mathcal{A} = E_e \mathcal{D} + \int_{\text{all space}} \frac{\hbar^2}{2m_e^*} |\psi_h(z_h)|^2 |\psi_e(z_e)|^2 \psi_r'^2 dz_e dz_h dx dy \quad (6.33)$$

Adopting a similar argument as above, the integral over the plane can then be written as:

$$G(a) = \int \left| \frac{\partial \psi_r}{\partial z_e} \right|^2 dx dy \quad (6.34)$$

and hence equation (6.33) gives the final expression for \mathcal{A} as follows:

$$\mathcal{A} = E_e \mathcal{D} + \frac{\hbar^2}{2m_e^*} \int_0^{\nu-\mu} p(a) G(a) da \quad (6.35)$$

An analogous expression exists for \mathcal{B} with the subscripts e and h , labelling electron and hole, respectively, being interchanged.

Now consider $\mathcal{C} = \langle \Psi | \mathcal{H}_{e-h} | \Psi \rangle$ as defined in equation (6.24). As mentioned earlier, \mathcal{H}_{e-h} is composed of two terms which represent the in-plane kinetic energy of the relative motion and the Coulombic potential energy between the electron and hole, i.e.

$$\mathcal{C} = \mathcal{C}_T + \mathcal{C}_V \quad (6.36)$$

By using the kinetic energy operator defined above in equation (6.11) then the first of these terms can then be written as:

$$\mathcal{C}_T = \int \psi_e^*(z_e) \psi_h^*(z_h) \psi_r^* \frac{-\hbar^2}{2\mu_\perp} \left(\frac{\partial^2}{\partial x^2} + \frac{\partial^2}{\partial y^2} \right) \psi_e(z_e) \psi_h(z_h) \psi_r dz_e dz_h dx dy \quad (6.37)$$

where the in-plane reduced mass μ_{\perp} is given by:

$$\frac{1}{\mu_{\perp}} = \frac{1}{m_e^{\perp}} + \frac{1}{m_h^{\perp}} \quad (6.38)$$

Again, whatever the functional form of the relative motion term ψ_r , it is apparent that the kinetic energy operator in \mathcal{C}_T will act only upon ψ_r , as the one-particle electron and hole wave functions, ψ_e and ψ_h , respectively, are not functions of x or y . Furthermore, the integration over the plane will result in an entity which is a function of a only, i.e. in the spirit of equation (6.35):

$$\mathcal{C}_T = -\frac{\hbar^2}{2\mu_{\perp}} \int_0^{\nu-\mu} p(a) J(a) da \quad (6.39)$$

where

$$J(a) = \int \psi_r \left(\frac{\partial^2}{\partial x^2} + \frac{\partial^2}{\partial y^2} \right) \psi_r dx dy \quad (6.40)$$

since ψ_r is real. In a similar manner, the potential energy term \mathcal{C}_V can be written as:

$$\mathcal{C}_V = -\frac{e^2}{4\pi\epsilon} \int_0^{\nu-\mu} p(a) K(a) da \quad (6.41)$$

where

$$K(a) = \int \frac{1}{r} \psi_r^2 dx dy \quad (6.42)$$

with $r^2 = x^2 + y^2 + a^2$.

To summarise then, the total exciton energy can be written as:

$$E = \frac{\mathcal{A} + \mathcal{B} + \mathcal{C}}{\mathcal{D}} \quad (6.43)$$

with \mathcal{A} given in equation (6.35), \mathcal{B} following analogously, \mathcal{C} given by equation (6.36) and \mathcal{D} by equation (6.22). Evaluation of the entities, $F(a)$, $G(a)$, $J(a)$ and $K(a)$, necessitates the introduction of a specific form for the relative motion term ψ_r . The exciton binding energy E_{X^0} follows simply from:

$$E = E_e + E_h + E_{X^0} \quad (6.44)$$

6.4 1S EXCITON

In line with other work in the literature, the choice of wave function representing the electron-hole interaction that is to be employed is a hydrogenic type, given by:

$$\psi_r = \exp\left(-\frac{r'}{\lambda}\right) \quad (6.45)$$

where the Bohr radius λ will be used as a parameter and systematically varied in order to minimize the total energy E of the system (which is equivalent to maximizing the exciton binding energy E_{X^0}). However, where this work differs is in the precise form of the choice of the relative coordinate r' . Using the knowledge and experience derived from the considerations of impurities in Chapter 5, a variable symmetry-type relative motion term is chosen:

$$r'^2 = (x_e - x_h)^2 + (y_e - y_h)^2 + \zeta^2 (z_e - z_h)^2 \quad (6.46)$$

or by using $a = |z_e - z_h|$ as above, then:

$$r'^2 = r_{\perp}^2 + \zeta^2 a^2 \quad (6.47)$$

The second variational parameter, ζ , allows the exciton to assume any shape of wave function that is energetically favourable. Traditionally, the case with $\zeta = 0$ has become known as the two-dimensional exciton [134–138], and $\zeta = 1$, as the three-dimensional exciton [134, 139]. Cases where ζ is allowed to take values other than 0 and 1 are rarely found in the literature because of the increased complexity in the resulting mathematics of the problem [140–143].

In the present approach, ζ will be allowed to take all values from 0 upwards; however, due to complications in the mathematics, in particular deriving the limits of integration for the evaluation of $K(a)$ (see later), it is necessary to introduce the following transformations:

$$\text{for } 0 \leq \zeta \leq 1, \quad \zeta^2 = 1 - \beta^2, \quad \text{where } 0 \leq \beta \leq 1 \quad (6.48)$$

$$\text{for } 1 \leq \zeta \leq \infty, \quad \zeta^2 = 1 + \eta^2, \quad \text{where } 0 \leq \eta \leq \infty \quad (6.49)$$

In order to illustrate the technique, attention will be focused initially on the former of the two cases. First, consider evaluation of the entity $F(a)$ as defined in equation (6.18); using this particular choice of ψ_r and moving into plane polar coordinates gives:

$$F(a) = 2\pi \int_0^\infty \exp\left(-\frac{2r'}{\lambda}\right) r_{\perp} dr_{\perp} \quad (6.50)$$

Using equation (6.47) to substitute r_{\perp} with r' , noting that $r' dr' = r_{\perp} dr_{\perp}$, and accounting for the change in limits, i.e. when:

$$r_{\perp} = 0, \quad r' = \sqrt{1 - \beta^2} a \quad (6.51)$$

$$r_{\perp} = \infty, \quad r' = \infty \quad (6.52)$$

then obtain:

$$F(a) = 2\pi \int_{\sqrt{1 - \beta^2} a}^\infty \exp\left(-\frac{2r'}{\lambda}\right) r' dr' \quad (6.53)$$

and by integrating by parts:

$$F(a) = 2\pi \left(\frac{\lambda \sqrt{1 - \beta^2} a}{2} + \frac{\lambda^2}{4} \right) \exp\left(-\frac{2\sqrt{1 - \beta^2} a}{\lambda}\right) \quad (6.54)$$

Next consider $G(a)$ as defined in equation (6.34), then:

$$G(a) = 2\pi \int_0^\infty \left| \frac{\partial}{\partial z_e} \exp \left(-\frac{r'}{\lambda} \right) \right|^2 r_\perp dr_\perp \quad (6.55)$$

and so performing the differentiation and substituting r' for r_\perp as above, then:

$$G(a) = 2\pi \int_{\sqrt{1-\beta^2}a}^\infty \left| -\frac{1}{\lambda} \exp \left(-\frac{r'}{\lambda} \right) \frac{(1-\beta^2)a}{r'} \right|^2 r' dr' \quad (6.56)$$

Now let $r' = \sqrt{1-\beta^2}a \cosh \theta$, then:

$$G(a) = 2\pi \int_0^\infty \frac{1}{\lambda^2} (1-\beta^2)^2 a^2 \exp \left(-\frac{2\sqrt{1-\beta^2}a \cosh \theta}{\lambda} \right) \frac{\sqrt{1-\beta^2}a \sinh \theta}{\sqrt{1-\beta^2}a \cosh \theta} d\theta \quad (6.57)$$

Making the further substitution, $w = \exp(-\theta)$, then $d\theta = -dw/w$ and noting that:

$$\cosh \theta = \frac{1}{2} (e^\theta + e^{-\theta}) = \frac{1}{2} \left(\frac{1}{w} + w \right), \quad \text{and} \quad \sinh \theta = \frac{1}{2} \left(\frac{1}{w} - w \right) \quad (6.58)$$

then finally:

$$G(a) = 2\pi \int_0^1 \frac{(1-\beta^2)^2 a^2}{\lambda^2} \exp \left[-\frac{\sqrt{1-\beta^2}a}{\lambda} \left(\frac{1}{w} + w \right) \right] \left(\frac{1-w^2}{w(1+w^2)} \right) dw \quad (6.59)$$

An analogous argument shows that $G(a)$ appearing in \mathcal{B} is exactly equal to this form.

Next consider evaluation of the integral $J(a)$ as defined in equation (6.40). With this aim, note that:

$$\frac{\partial r'}{\partial x} = \frac{x}{r'}, \quad \text{and} \quad \frac{\partial^2 r'}{\partial x^2} = \frac{1}{r'} - \frac{x^2}{r'^3} \quad (6.60)$$

This then gives the following:

$$\frac{\partial^2 \psi_r}{\partial x^2} = \left(\frac{-1}{r'\lambda} + \frac{x^2}{r'^3\lambda} + \frac{x^2}{r'^2\lambda^2} \right) \psi_r \quad (6.61)$$

and hence:

$$\left(\frac{\partial^2}{\partial x^2} + \frac{\partial^2}{\partial y^2} \right) \psi_r = \left[\frac{1}{\lambda^2} - \frac{(1-\beta^2)a^2}{\lambda r'^3} - \frac{1}{r'\lambda} - \frac{(1-\beta^2)a^2}{r'^2\lambda^2} \right] \psi_r \quad (6.62)$$

Therefore, moving from Cartesian into plane polar coordinates:

$$J(a) = 2\pi \int_0^\infty \left[\frac{1}{\lambda^2} - \frac{(1-\beta^2)a^2}{\lambda r'^3} - \frac{1}{r'\lambda} - \frac{(1-\beta^2)a^2}{r'^2\lambda^2} \right] \exp \left(-\frac{2r'}{\lambda} \right) r_\perp dr_\perp \quad (6.63)$$

It is standard practice in the literature [144] to expand the expressions involving $(r')^{-n}$ as a power series in r_{\perp} and a , and then to perform the integration numerically. This involves summing over a series of terms, each of which must be integrated over a range from 0 to ∞ ; this procedure, however, can be avoided.

Writing equation (6.63) as $J(a) = J_1 + J_2 + J_3 + J_4$, where J_i represents the first, second, etc, terms, respectively, then:

$$J_1 + J_3 = 2\pi \int_0^{\infty} \left[\frac{1}{\lambda^2} - \frac{1}{\lambda r'} \right] \exp \left(-\frac{2r'}{\lambda} \right) r_{\perp} dr_{\perp} \quad (6.64)$$

Again, substituting r' for r_{\perp} :

$$J_1 + J_3 = 2\pi \int_{\sqrt{1-\beta^2}a}^{\infty} \left[\frac{1}{\lambda^2} - \frac{1}{\lambda r'} \right] \exp \left(-\frac{2r'}{\lambda} \right) r' dr' \quad (6.65)$$

and therefore:

$$J_1 + J_3 = 2\pi \left\{ \left[\frac{1}{2} \exp \left(-\frac{2r'}{\lambda} \right) - \frac{r'}{2\lambda} \exp \left(-\frac{2r'}{\lambda} \right) \right]_{\sqrt{1-\beta^2}a}^{\infty} + \frac{1}{2\lambda} \int_{\sqrt{1-\beta^2}a}^{\infty} \exp \left(-\frac{2r'}{\lambda} \right) dr' \right\} \quad (6.66)$$

which gives:

$$J_1 + J_3 = 2\pi \left(\frac{\sqrt{1-\beta^2}a}{2\lambda} - \frac{1}{4} \right) \exp \left(-\frac{2\sqrt{1-\beta^2}a}{\lambda} \right) \quad (6.67)$$

The two remaining terms of equation (6.63) give:

$$J_2 + J_4 = 2\pi \int_0^{\infty} \left[-\frac{(1-\beta^2)a^2}{\lambda r'^3} - \frac{(1-\beta^2)a^2}{\lambda^2 r'^2} \right] \exp \left(-\frac{2r'}{\lambda} \right) r_{\perp} dr_{\perp} \quad (6.68)$$

Again substituting r' for r_{\perp} , then:

$$J_2 + J_4 = 2\pi \int_{\sqrt{1-\beta^2}a}^{\infty} (1-\beta^2)a^2 \left(-\frac{1}{\lambda r'^2} - \frac{1}{\lambda^2 r'} \right) \exp \left(-\frac{2r'}{\lambda} \right) dr' \quad (6.69)$$

Making a further scale change of $r' = \sqrt{1-\beta^2}a \cosh \theta$, then:

$$J_2 + J_4 = 2\pi \int_0^{\infty} \left(-\frac{1}{\lambda \cosh^2 \theta} - \frac{\sqrt{1-\beta^2}a}{\lambda^2 \cosh \theta} \right) \exp \left(-\frac{2\sqrt{1-\beta^2}a \cosh \theta}{\lambda} \right) \times \sqrt{1-\beta^2}a \sinh \theta d\theta \quad (6.70)$$

With the final substitution of $w = \exp -\theta$, then $d\theta = -dw/w$, therefore:

$$J_2 + J_4 = 2\pi \int_0^1 \left[-\frac{1}{\frac{\lambda}{4} \left(\frac{1}{w} + w \right)^2} - \frac{\sqrt{1-\beta^2}a}{\frac{\lambda^2}{2} \left(\frac{1}{w} + w \right)} \right] \\ \times \exp \left[-\frac{\sqrt{1-\beta^2}a}{\lambda} \left(\frac{1}{w} + w \right) \right] \sqrt{1-\beta^2} \frac{a}{2} \left(\frac{1}{w^2} - 1 \right) dw \quad (6.71)$$

This last equation illustrates the advantage of the present formalism, namely that the computationally difficult integral of equation (6.63), which has hitherto been expanded into a infinite series and integrated to infinity, has been replaced with a simple integral over the range from 0 to 1. Even if the integrand had a finite number of singularities, this would still pose no problem in its evaluation.

Finally consider $K(a)$, as defined in equation (6.42), i.e.

$$K(a) = 2\pi \int_0^\infty \frac{1}{r} \exp \left(-\frac{2r'}{\lambda} \right) r_\perp dr_\perp \quad (6.72)$$

Recalling that $r^2 = r_\perp^2 + a^2$, then:

$$K(a) = 2\pi \int_a^\infty \exp \left(-\frac{2r'}{\lambda} \right) dr \quad (6.73)$$

The form of r' , i.e. $r'^2 = r^2 - \beta^2 a^2$, suggests the substitution $r = \beta a \cosh \theta$, which gives:

$$K(a) = 2\pi \int_{\cosh^{-1} \frac{1}{\beta}}^\infty \beta a \sinh \theta \exp \left(-\frac{2\beta a \sinh \theta}{\lambda} \right) d\theta \quad (6.74)$$

Again making use of the substitution $w = \exp -\theta$, this then necessitates evaluating w corresponding to $\theta = \cosh^{-1} \frac{1}{\beta}$, i.e. $\cosh \theta = \frac{1}{\beta} = \frac{1}{2} \left(w + \frac{1}{w} \right)$; this then yields the quadratic equation:

$$w^2 - \frac{2w}{\beta} + 1 = 0 \quad (6.75)$$

Since the product of the two roots of this equation is unity, one root must correspond to $\exp(-\theta)$ and the other to $\exp \theta$. It is readily ascertained that the $\exp(-\theta)$ root is as follows:

$$\exp(-\theta) = \frac{1}{\beta} - \frac{1}{\beta} \sqrt{1-\beta^2} \quad (6.76)$$

This follows since the limits on β are 0 and 1. Hence:

$$K(a) = 2\pi \frac{\beta a}{2} \int_0^{\frac{1}{\beta} - \frac{1}{\beta} \sqrt{1-\beta^2}} \exp \left[-\frac{\beta a}{\lambda} \left(\frac{1}{w} - w \right) \right] \left(\frac{1}{w^2} - 1 \right) dw \quad (6.77)$$

In a similar manner, the second form of ψ_r , with $r' = \sqrt{r_\perp^2 + (1+\eta^2)a^2}$, gives the same expressions for $F(a)$, $G(a)$ and $J(a)$ as above, but with the simple substitution,

$1 + \eta^2$, in place of $1 - \beta^2$. Only the expression for the last of the 'a' functions differs, in particular:

$$K(a) = 2\pi \frac{\eta a}{2} \int_0^{\frac{1}{\eta} \sqrt{1+\eta^2} - \frac{1}{\eta}} \exp \left[-\frac{\eta a}{\lambda} \left(\frac{1}{w} + w \right) \right] \left(\frac{1}{w^2} + 1 \right) dw \quad (6.78)$$

6.5 THE TWO-DIMENSIONAL AND THREE-DIMENSIONAL LIMITS

It is *always* worthwhile performing convergence tests, i.e. taking the theoretical-computational model to established, often analytical, limits. The idea is to increase confidence in the theory and, as ever, to demonstrate that the previous theories are limits of the new. For example, classical mechanics is recovered from relativistic mechanics, in the low-velocity limit.

Although not as grand an example, there exist two limits which the analysis above can be compared with. In the limit of very wide quantum wells, the exciton should look like a bulk exciton, in both its binding energy and Bohr radius. In addition in the limit of very narrow wells, the exciton should become two-dimensional in nature.

The bulk, or three-dimensional limiting case of hydrogenic two-body systems, such as impurities and excitons, has been discussed and used already. A transparent treatise of this Bohr model of the hydrogen atom is given by Weidner and Sells [4]. This approach can easily be adapted to the case of an electron orbiting a positively charged central infinite mass, with the orbit restricted to a single plane, i.e. it is two-dimensional (2D). The Schrödinger equation is then written as:

$$-\frac{\hbar^2}{2m} \left(\frac{\partial^2 \psi}{\partial x^2} + \frac{\partial^2 \psi}{\partial y^2} \right) - \frac{e^2}{4\pi\epsilon_r\epsilon_0 r_\perp} \psi = E\psi \quad (6.79)$$

where the electron-proton separation

$$r_\perp = \sqrt{x^2 + y^2} \quad (6.80)$$

For solution, the Cartesian coordinates need to be converted in to the plane polar co-ordinate r_\perp , with this aim, note:

$$\frac{\partial \psi}{\partial x} = \frac{\partial \psi}{\partial r_\perp} \frac{\partial r_\perp}{\partial x} = \frac{x}{r_\perp} \frac{\partial \psi}{\partial r_\perp} \quad (6.81)$$

and so:

$$\frac{\partial^2 \psi}{\partial x^2} = \frac{\partial}{\partial x} \left(\frac{x}{r_\perp} \frac{\partial \psi}{\partial r_\perp} \right) = \frac{1}{r_\perp} \frac{\partial \psi}{\partial r_\perp} - \frac{x}{r_\perp^2} \frac{\partial r_\perp}{\partial x} \frac{\partial \psi}{\partial r_\perp} + \frac{x}{r_\perp} \frac{\partial^2 \psi}{\partial r_\perp^2} \frac{\partial r_\perp}{\partial x} \quad (6.82)$$

$$\therefore \frac{\partial^2 \psi}{\partial x^2} = \frac{1}{r_\perp} \frac{\partial \psi}{\partial r_\perp} - \frac{x^2}{r_\perp^3} \frac{\partial \psi}{\partial r_\perp} + \frac{x^2}{r_\perp^2} \frac{\partial^2 \psi}{\partial r_\perp^2} \quad (6.83)$$

Thus:

$$\frac{\partial^2 \psi}{\partial x^2} + \frac{\partial^2 \psi}{\partial y^2} = \frac{2}{r_{\perp}} \frac{\partial \psi}{\partial r_{\perp}} - \frac{x^2 + y^2}{r_{\perp}^3} \frac{\partial \psi}{\partial r_{\perp}} + \frac{x^2 + y^2}{r_{\perp}^2} \frac{\partial^2 \psi}{\partial r_{\perp}^2} \quad (6.84)$$

$$\therefore \frac{\partial^2 \psi}{\partial x^2} + \frac{\partial^2 \psi}{\partial y^2} = \frac{1}{r_{\perp}} \frac{\partial \psi}{\partial r_{\perp}} + \frac{\partial^2 \psi}{\partial r_{\perp}^2} \quad (6.85)$$

Therefore the Schrödinger equation, (equation 6.79), in plane polar co-ordinates becomes:

$$\frac{1}{r_{\perp}} \frac{\partial \psi}{\partial r_{\perp}} + \frac{\partial^2 \psi}{\partial r_{\perp}^2} + \frac{2m}{\hbar^2} \left(E + \frac{e^2}{4\pi\epsilon_r\epsilon_0 r_{\perp}} \right) \psi = 0 \quad (6.86)$$

The standard technique for solving the hydrogen atom is to choose a spherically symmetric wave function of the form:

$$\psi = \exp \left(-\frac{r}{\lambda_{3D}} \right) \quad (6.87)$$

In analogy to this, the 2D equivalent is:

$$\psi = \exp \left(-\frac{r_{\perp}}{\lambda_{2D}} \right) \quad (6.88)$$

By calculating the derivatives:

$$\frac{\partial \psi}{\partial r_{\perp}} = -\frac{1}{\lambda_{2D}} \psi \quad \text{and} \quad \frac{\partial^2 \psi}{\partial r_{\perp}^2} = \frac{1}{\lambda_{2D}^2} \psi \quad (6.89)$$

and substituting into equation (6.86), then obtain:

$$\left(\frac{1}{\lambda_{2D}^2} + \frac{2mE}{\hbar^2} \right) \psi + \frac{1}{r_{\perp}} \left(\frac{2me^2}{\hbar^2 4\pi\epsilon_r\epsilon_0} - \frac{1}{\lambda} \right) \psi = 0 \quad (6.90)$$

The standard argument is that as this equation must be valid for all values of r_{\perp} , then divergence as $r_{\perp} \rightarrow 0$ can only be avoided if the second term is zero; this implies:

$$\lambda_{2D} = \frac{4\pi\epsilon_r\epsilon_0\hbar^2}{2me^2} = \frac{\lambda_{3D}}{2} \quad (6.91)$$

i.e. the Bohr radius of the 2D (planar) two-body system, is half that of the 3D (spherical) system. Using this form for the Bohr radius, the first term then yields the energy of the 2D system as follows:

$$E^{2D} = -4 \frac{me^4}{32\pi^2\hbar^2\epsilon_r^2\epsilon_0^2} = 4E^{3D} \quad (6.92)$$

As mentioned above, and as already utilized for neutral impurities in Chapter 5, the 3D (or bulk) limit can be approached by increasing the width of a *finite* quantum well. However, the finite depth of such a quantum well is not conducive to the 2D limit, for

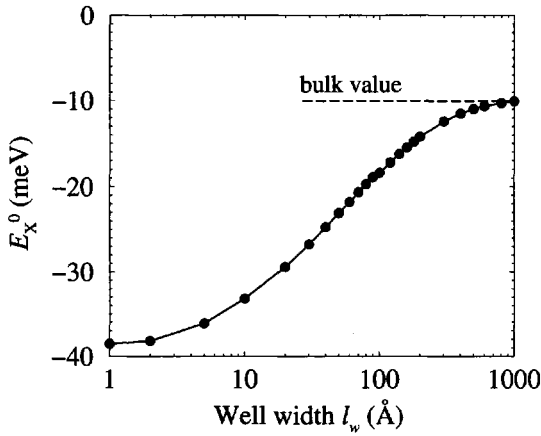


Figure 6.4 Exciton binding energy in an infinitely deep quantum well

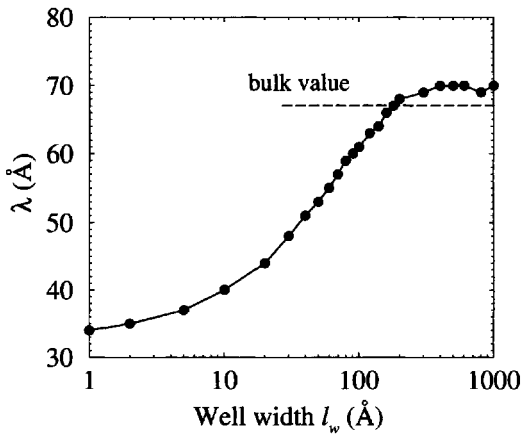


Figure 6.5 Bohr radius in an infinitely deep quantum well

as the well width decreases, the one-particle electron and hole wave functions begin to ‘spill out’ over the top of the well. The 2D limit can only be approached hypothetically in an infinitely deep quantum well. Fig. 6.4 illustrates the results of calculations of the exciton binding energy as a function of the width of an infinitely deep CdTe quantum well. The magnitude of the bulk exciton binding energy was calculated above in

Section 6.1 as 10.1 meV. The negative values on the graph illustrate that it is indeed a bound state.

Clearly, both limits are obeyed, i.e.

$$\lim_{l_w \rightarrow \infty} E_{X^0} = E_{X^0}^{3D} \quad \text{and} \quad \lim_{l_w \rightarrow 0} E_{X^0} = 4E_{X^0}^{3D} \quad (6.93)$$

Figure 6.5 displays the corresponding Bohr radii λ for the energies of Fig. 6.4. Remembering that the Bohr radius in bulk, $\lambda_{3D}=67 \text{ \AA}$, then the 2D limit, i.e.

$$\lim_{l_w \rightarrow 0} \lambda = \frac{\lambda_{3D}}{2} \quad (6.94)$$

is satisfied. The 3D limit is obeyed, although the data on the graph show a slight scatter around the bulk radii of 67 \AA . The source of this discrepancy is numerical accuracy. At the larger well widths, the wave function needs to be known at many points in order to calculate the binding energy to very high tolerances (thus leading to long computational times), as required here in these convergence tests. Despite this high accuracy, the Bohr radius still shows some deviation from the bulk, although greater numerical accuracy has reduced this towards the value of 67 \AA .

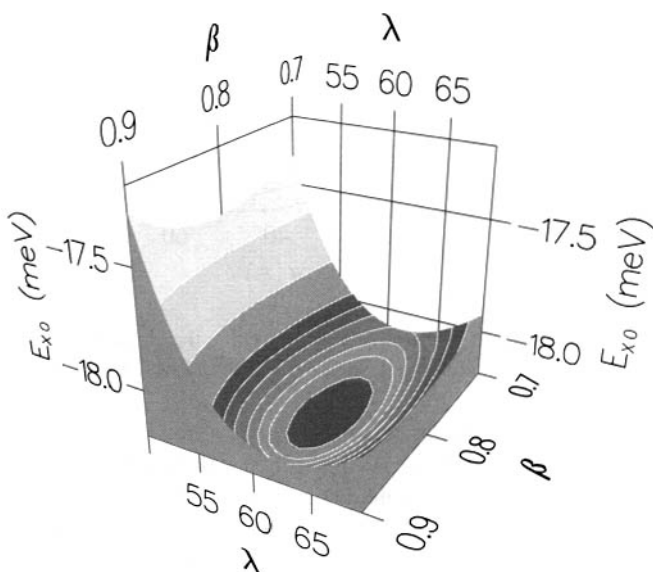


Figure 6.6 The exciton binding energy E_{X^0} in the two-dimensional λ - β parameter space

The calculations also showed that at the very narrow (5–40 \AA) well widths, the second parameter ζ , representing the symmetry of the relative motion term was close

to zero, i.e. a 2D wave function. As the width increased, ζ increased steadily towards unity, i.e. the 3D (spherical or bulk-like) exciton. This point that ζ generally lies in the limits $0 < \zeta < 1$ will be reinforced further by the use of several different example systems in the following discussions. For this reason, discussion of the symmetry of the wave function will use ζ and $\beta = \sqrt{1 - \zeta^2}$ interchangeably.

Figure 6.6 illustrates the variational technique in the case of the infinite well of width $l_w = 100 \text{ \AA}$. It can be seen that there is just one minimum in the exciton energy in the two-dimensional λ - β space. This is almost always the case. However, there is an exception to this rule [131]. In double quantum well systems where the one-particle electron wave function has a significant component in each of the wells, there exists two local energy minima within λ - β space. This implies that there are two possible exciton states which can be formed from the same one-particle wave functions. The two states can be thought of as originating from the localised hole binding separately with both of the 'lobes' of the electron wave function.

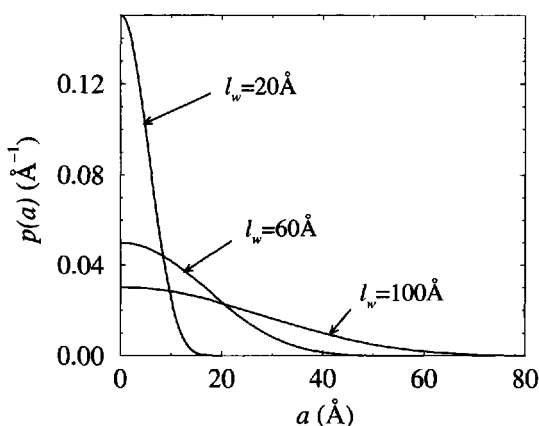


Figure 6.7 The uncorrelated probability $p(a)$ of finding the electron and hole separated by a distance a in the infinite quantum well, shown for different well widths

It is worthwhile just looking at the form of the function $p(a)$, which plays a significant rôle in the theoretical analysis. As mentioned at the point of its introduction, $p(a)$ represents the *uncorrelated* (i.e. calculated without taking account of the electron-hole Coulombic interaction) probability of finding the electron and hole separated by a distance a along the growth (z -) axis. While $p(a)$ is not a physical observable, often the form of it can help in understanding and provide an insight into the nature of the exciton. One particular example of that would be in the 'twin' exciton states of the double quantum well discussed above. Fig. 6.7 shows $p(a)$ for this simple case of the infinite quantum well, while later figures will compare it with other more complex heterostructures.

In conclusion, the validity of the technique presented above has been substantiated by the calculation of exciton binding energies and wave functions for the full width range of infinite quantum wells. In the sections that follow, examples of calculations will be presented which demonstrate the versatility of the method. In addition, some of the results are expected to be of interest in their own right.

6.6 EXCITONS IN SINGLE QUANTUM WELLS

Only in a very few circumstances are the band offsets in both the conduction and valence bands large enough that the system can be approximated with an infinite quantum well model. The vast majority of cases demand finite potentials. Reiterating, the approach developed above depends only upon the one-particle electron and hole wave functions, ψ_e and ψ_h , respectively, and no further knowledge of the system is required (except for the basic material parameters of electron and hole effective masses and dielectric constants).

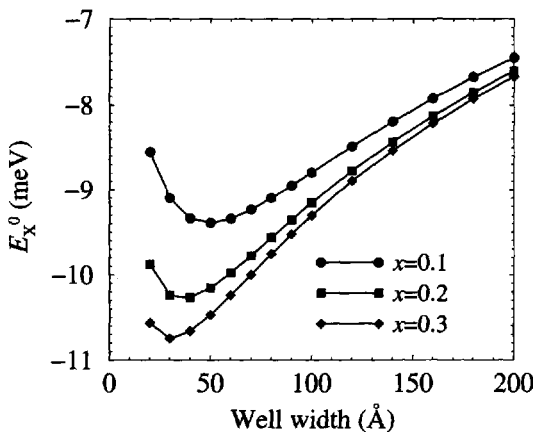


Figure 6.8 The exciton binding energy E_{X^0} in a GaAs quantum well surrounded by $\text{Ga}_{1-x}\text{Al}_x\text{As}$ barriers

To illustrate this, Figs 6.8–6.10 display the results of calculations of exciton binding energies in finite GaAs single quantum wells surrounded by $\text{Ga}_{1-x}\text{Al}_x\text{As}$ barriers. Unlike the infinite-well case, the exciton binding energy E_{X^0} is a non-monotonic function of well width. As in the impurity binding energy case of Chapter 5, this is due to the effect of the well width on the electron and hole confinement. At very narrow well widths, the one-particle states are ‘squeezed’ up the well to reside at energies just below the top of the barrier. The wave function tends to ‘spill’ over the top, thus leading to a reduced probability of the particle being within the region of

the quantum well. This non-monotonic behaviour has been observed in experiment [145]. In addition, increasing Al concentration in the barrier (both the conduction and valence band offsets are proportional to x) leads to an increase in the magnitude of the exciton binding energy for *all* well widths. However, there are two points to note regarding this. First, the effect of the barrier height is reduced at larger well widths. This substantiates the fact that a particular choice of barrier height (infinity) was chosen to illustrate the 3D limit in the previous section. It is clear from Fig. 6.8 that the exciton binding energy is tending towards its bulk value of 4.7 meV for *all* barrier heights. Secondly, the effect of an increasing barrier height is largest at smaller well widths, and for any given well width, further increases in the barrier height lead to smaller increases in the binding energy.

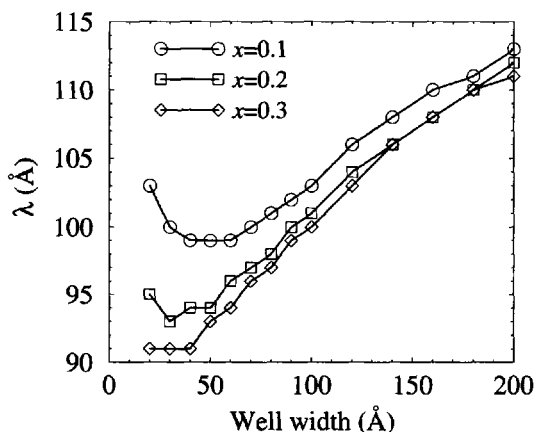


Figure 6.9 The exciton Bohr radius λ in a GaAs quantum well surrounded by $\text{Ga}_{1-x}\text{Al}_x\text{As}$ barriers

Figs 6.9 and 6.10 summarize the excitonic wave function corresponding to the minimized variational energies of Fig. 6.8. The binding energy is essentially calculated over a range of discrete values of both λ and (in this case) β , hence the accuracy of these parameters is known only to the resolution of the mesh. The most important result of the variational calculation is the energy, which around the minima is a weak function of the parameters, and thus even a coarse mesh would have little consequences for the binding energy; Fig. 6.6 illustrates this nicely. The minima can be determined to 0.001 meV with a λ resolution of 1 Å and a β resolution of 0.01. The symbols in Fig. 6.9 are drawn open and of a size in order to illustrate the accuracy in the determination of the Bohr radius λ . In many ways the functional dependencies of λ on well width and barrier height mirror that of the binding energy E_{X^0} —it is a non-monotonic function of well width, decreases with increasing barrier height and tends towards its bulk value of 115 Å at large well widths.

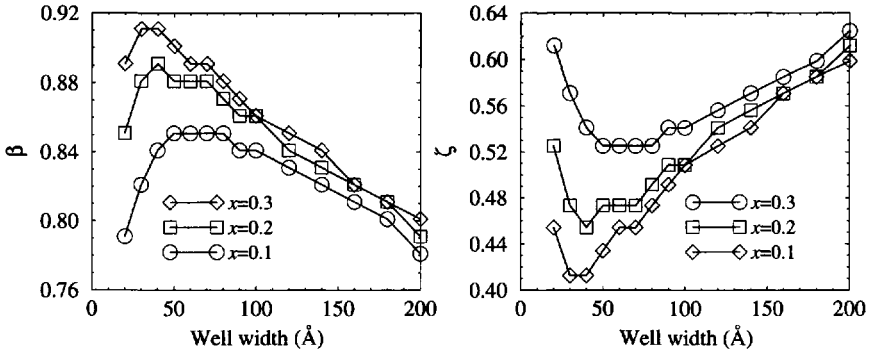


Figure 6.10 The symmetry parameters β and $\zeta (= \sqrt{1 - \beta^2})$ in a GaAs quantum well surrounded by $\text{Ga}_{1-x}\text{Al}_x\text{As}$ barriers

The symmetry parameter is displayed in Fig. 6.10 as both β (as calculated) and as its transformation into $\zeta (= \sqrt{1 - \beta^2})$. Recalling that the trial wave function of the relative motion is given by:

$$\psi_r = \exp \left[-\frac{\sqrt{r_{\perp}^2 + \zeta^2(z_e - z_h)^2}}{\lambda} \right] \quad (6.95)$$

then $\zeta \sim 0.5$ implies that:

$$\psi_r \sim \exp \left[-\frac{\sqrt{r_{\perp}^2 + \frac{(z_e - z_h)^2}{4}}}{\lambda} \right] \quad (6.96)$$

which is roughly midway between the two-dimensional case, i.e.

$$\psi_r \sim \exp \left(-\frac{r_{\perp}}{\lambda} \right) \quad (6.97)$$

and the three-dimensional case:

$$\psi_r \sim \exp \left[-\frac{\sqrt{r_{\perp}^2 + (z_e - z_h)^2}}{\lambda} \right] \quad (6.98)$$

While the implications for the binding energy, by using either one of the simpler wave functions above, might be relatively small, i.e. probably less than 1 meV, the fact remains that excitons in quantum wells do have mixed symmetry. In this example the ζ value implies that 'equi-surfaces' of ψ_r are prolate spheroids, i.e. they look like rugby footballs (American footballs), with the major axis along the growth (z -) direction of the heterostructure.

6.7 EXCITONS IN MULTIPLE QUANTUM WELLS

Semiconductor heterostructures are often more complex than single quantum wells. Perhaps the next stage in complexity of design is to incorporate many identical wells within the same grown layer. As discussed previously, if the barriers separating the wells are such that the one-particle wave functions in each well overlap, then the system is called a superlattice, while if they do not overlap, it is called a multiple quantum well system.

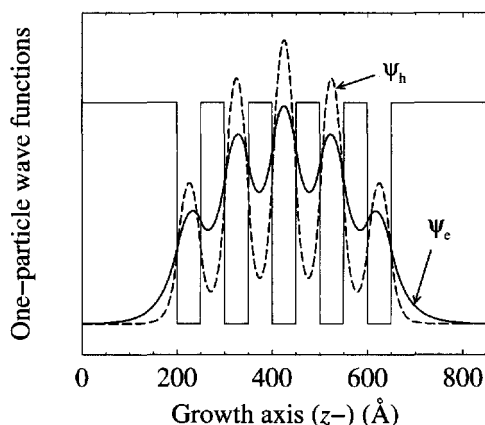


Figure 6.11 The electron and heavy-hole one-particle wave functions in a 5 period 50 Å GaAs/50 Å Ga_{0.9}Al_{0.1}As multiple quantum well

Figure 6.11 displays the electron and heavy-hole one-particle wave functions for a 5 period 50 Å GaAs/50 Å Ga_{0.9}Al_{0.1}As multiple quantum well. Clearly this system is a finite superlattice. The form of the wave functions in these finite superlattices has been discussed earlier in Chapter 3; however, a simple comparison of the two, indicates that the heavy-hole wave function is more localised than that of the electron on account of its larger mass.

Although it serves no *real* purpose, it is interesting to compare the functional form of $p(a)$ with that of the infinite well of earlier. The troughs in Fig. 6.12 indicate that there are electron-hole separations which are less favoured than others. These troughs at 50, 150 and 250 Å correspond to separations where one particle would be located in a well and the other in a barrier. As in the infinite well case, the electron-hole separation (a) with the highest probability is zero.

The exciton binding energy is displayed in Fig. 6.13 for a series of 5-period multiple quantum wells as a function of the equal well (l_w) and barrier (l_b) widths. In contrast to the finite well, the magnitude of E_{X^0} passes through a *minimum*, which in this material

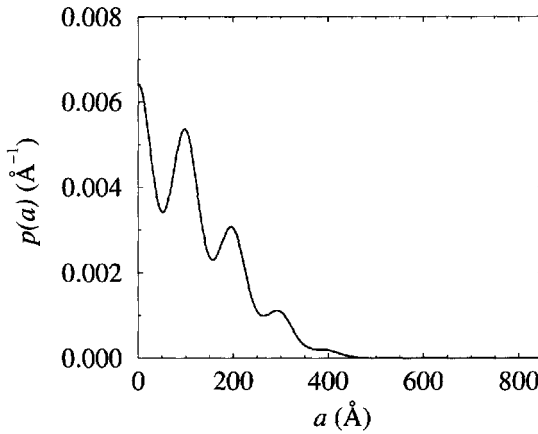


Figure 6.12 $p(a)$ for the 5 period GaAs-Ga_{0.9}Al_{0.1}As multiple quantum well

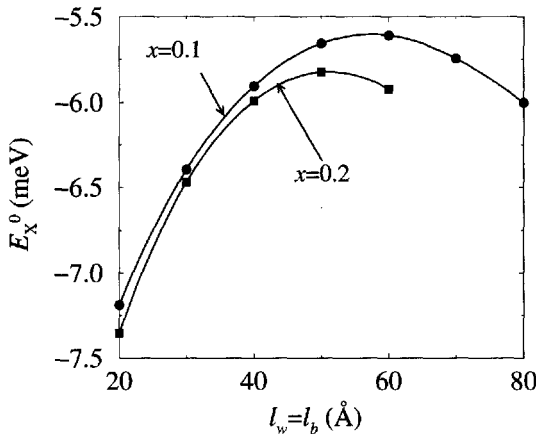


Figure 6.13 The exciton binding energy E_{X^0} as a function of well width in a 5-period GaAs-Ga_{1-x}Al_xAs multiple quantum well

system is at around 50–60 Å, before increasing again. In the GaAs-Ga_{1-x}Al_xAs material system here, larger well and barrier widths reduced the interaction between the states in adjacent wells to such an extent that the separation between the symmetric ground state (as displayed in Fig. 6.11) and the first excited state was less than 10^{-6} meV, and could not be resolved; hence the truncation of the data at 60 and 80 Å.

Such calculations have been performed at larger well and barrier widths in the CdTe- $\text{Cd}_{1-x}\text{Mn}_x\text{Te}$ system [131], where it was shown that beyond the minimum in the magnitude of E_{X^0} there is a maximum and then the binding energy of the multiple quantum well tends towards that of the finite well, as would be expected.

6.8 STARK LADDERS

As discussed previously (Chapter 3, in the context of unipolar (single-charge-carrier)), systems, when an electric field is applied along the growth (z -) axis of a (finite or infinite) superlattice, the eigenstates which previously extended over the whole system, begin to localise. As the field increases the miniband breaks up and a localised state forms in each quantum well. At this point, the energy separation between the states in adjacent wells is proportional to the electric field and is given simply by the difference in potential energy of the well centres, i.e.

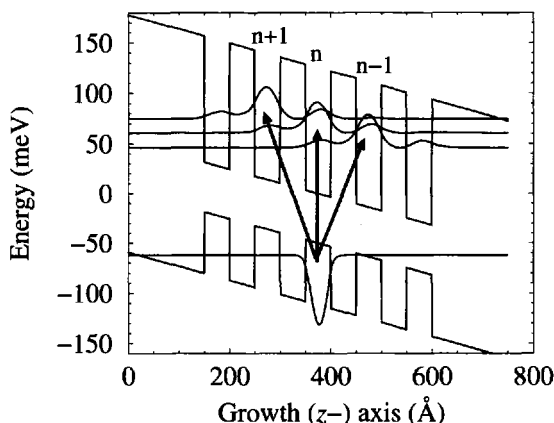


Figure 6.14 Illustration of the field (14 kVcm^{-1})-localised heavy-hole state recombining with the three closest electron states, which characterise the Stark ladder

$$E_{n+1} - E_n = eFL \quad (6.99)$$

where F is the electric field strength and L is the superlattice period. The conduction and valence band states thus form a series of steps similar to a 'ladder'.

In a bipolar system the hole wave functions tend to localise more rapidly than those of the electron's, hence a regime exists where a hole wave function centred in one particular well has a significant overlap with the electron wave functions from adjacent wells, in addition to its own. Therefore, the photoluminescence emission

and absorption spectra show a series of lines which represent exciton transitions from the hole in well n with electrons in well n , and in wells $n - 1$ and $n + 1$, as displayed in Fig. 6.14.

Given that the electron and hole energy separations between adjacent wells are proportional to the field, then the electron-hole separation (ignoring the exciton binding energy for now) will also be proportional to the field, as displayed in Fig. 6.15 (left). The addition of the exciton binding energy in this simplistic explanation complicates the situation slightly (see Fig. 6.15 (right)), but by the time that moderate electric fields are present, the spectral lines are equally spaced and radiating from the zero-field point. Mendez *et al.* [146] have observed very similar behaviour in a related system, namely a 30 Å GaAs/35 Å Ga_{0.65}Al_{0.35}As superlattice.

Figure 6.15 (right), however, does not tell the whole story. While it does indicate the position of the spectral lines, it does not give any information about their intensities. The oscillator strength (defined as follows) gives a measure of the expected intensities of the photoluminescence emission and photoluminescence excitation absorption lines:

$$O_s \propto \frac{\langle \psi_e | \psi_h \rangle^2}{\langle \Psi | \Psi \rangle} = \frac{O^2}{D} \quad (6.100)$$

Figure 6.16 displays the oscillator strength as a function of the applied electric field for the three excitonic transitions of interest. At zero field, all eigenstates have definite parity and hence the $|e_{n+1}\rangle \rightarrow |h_n\rangle$ and $|e_{n-1}\rangle \rightarrow |h_n\rangle$ transitions are forbidden; this result arises naturally from the analysis and is demonstrated numerically in the figure by a zero oscillator strength. As the field is increased, the system passes through the region of 'miniband breakup', with the effect of the symmetry breaking induced by the field being to allow the previously forbidden transitions. At larger fields still, the one-particle wave functions will tend towards complete localisation in a single

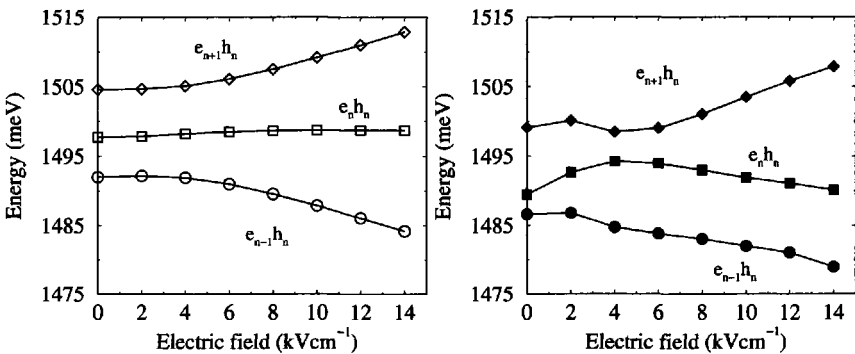


Figure 6.15 The excitonic transition energies characteristic of a Stark Ladder; (left) without exciton binding energies; (right) with exciton binding energies

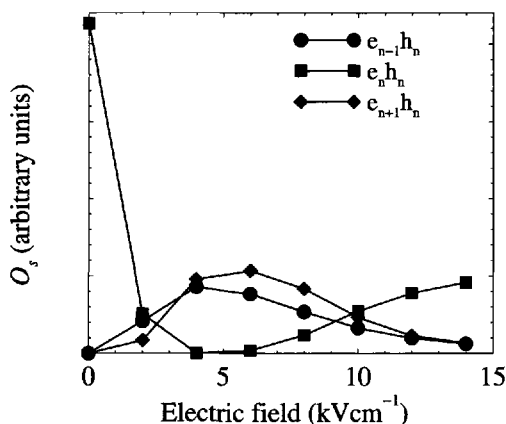


Figure 6.16 The oscillator strength O_s as function of electric field, for three of the Stark ladder transitions

well, and hence the oscillator strength of the inter-well transitions is expected (and confirmed by the calculations) to peak and then reduce. Meanwhile, the intra-well exciton transition $|e_n\rangle \rightarrow |h_n\rangle$ initially has the largest oscillator strength (obviously), which then temporarily decreases at low fields, before increasing steadily to dominate in the Stark ladder regime.

To date, no excitonic states have been calculated with the variable symmetry parameter $\zeta > 1$. However, for the work discussed in this present section some of the Stark ladder excitons have β values tending to zero, i.e. $\zeta = 1$.

6.9 SELF-CONSISTENT EFFECTS

In the analysis put forward in this present chapter and for the majority of the relevant literature, the exciton binding energy has been calculated by using a variational approach, based on the choice of trial wave function of the following form:

$$\Psi = \psi_e(z_e)\psi_h(z_h)\psi_r(x, y, z_e - z_h) \quad (6.101)$$

where most importantly, ψ_e and ψ_h were simply the one-particle eigenstates of the one-dimensional Schrödinger equation of the heterostructure; this means that ψ_e was calculated without the presence of the hole and ψ_h was calculated without the presence of the electron.

Attempts to improve upon this began with the simple idea of the Coulombic-hole model. This has been implemented [147] in the dilute magnetic semiconductor system CdTe-Cd_{1-x}Mn_xTe (see Section 5.11.1), which is thought to have a magnetic-field-

induced Type-I to Type-II transition. It is generally accepted that such corrections to the one-particle wave functions are more important in Type-II systems, as a well-confined electron may be able to ‘suck in’ the hole from the adjacent layer, thus having a large effect on the exciton binding energy.

A fully self-consistent exciton model has been proposed by Warnock *et al.* [148] and later extended to include the variable symmetry relative motion term of this analysis [149]. The self-consistent correction to the one-particle eigenfunctions, ψ_e and ψ_h , is incorporated by solving a new Schrödinger equation:

$$\begin{aligned} & \left(\int_{\text{all space}} \psi_h^* \psi_r^* \mathcal{H} \psi_h \psi_r \, dz_h \, dx \, dy \right) \psi_e \\ &= E \left(\int_{\text{all space}} \psi_h^* \psi_r^* \psi_h \psi_r \, dz_h \, dx \, dy \right) \psi_e \end{aligned} \quad (6.102)$$

where \mathcal{H} is the total Hamiltonian describing the system, as in equation (6.7). This yields the first improved iteration to the electron wave function, $\psi_e^{(1)}$ say. The latter is then employed to improve the hole wave function, as follows:

$$\begin{aligned} & \left(\int_{\text{all space}} \psi_e^{(1)*} \psi_r^* \mathcal{H} \psi_h \psi_r \, dz_h \, dx \, dy \right) \psi_h \\ &= E \left(\int_{\text{all space}} \psi_e^{(1)*} \psi_r^* \psi_h \psi_r \, dz_h \, dx \, dy \right) \psi_h \end{aligned} \quad (6.103)$$

Following, this both $\psi_e^{(1)}$ and $\psi_h^{(1)}$ are utilized in the standard exciton-binding-energy calculation to give the new relative motion term, characterised entirely in terms of the parameters λ and ζ . The procedure is repeated until the total energy of the exciton is minimized and λ and ζ have converged.

Piorek *et al.* [149] demonstrated that for Type-I systems, a variable-symmetry relative motion term is *more important* than self-consistency with simple 2D or 3D wave functions.

The self-consistent correction proved itself to be important in flat-band (i.e. a zero offset in the valence band) and Type-II systems. The simple one-particle solutions are usually localised in different semiconductor layers, thus leading to a quite small exciton binding energy. However, the addition of self-consistency allowed the electron and hole eigenfunctions to move closer together, thus increasing the binding energy and, most importantly, reducing the total energy of the exciton.

6.10 SPONTANEOUS SYMMETRY BREAKING

The aim of this book is to discuss the theoretical methods necessary to analyse semiconductor heterostructures. In addition, the computational implementation of these methods is paramount. This present section gives an example of purely computational

work giving rise to new science—something that could not have been predicted from theory.

In calculating the self-consistent exciton energy in series of multiple quantum wells, Piorek *et al.* [150] found that a 4-period multiple quantum well could have a larger exciton energy than the equivalent 3-period system. This seemed a contradiction as a 4-well system can be thought of as a 3-well system but with some of the potential barrier *removed*. Therefore it should have a lower exciton energy. This seemed to be a general result, i.e. that $2n$ -period multiple quantum wells had a larger energy than $2n - 1$ systems. Inspection of the self-consistently iterated electron and hole wave functions, $\psi_e^{(m)*}$ and $\psi_h^{(m)*}$, respectively, showed that in systems with odd numbers of wells, the electron and hole localised in the central well leading to a large binding energy. However, in systems with an even number of wells, the electron and hole localised in the *two* central wells, thus giving a smaller binding energy, and so leading to a higher total energy.

However, for certain calculations, exceptions to the above rule were identified. It was found that if there was the slightest asymmetry in the original electron and hole wave functions, ψ_e and ψ_h , then the self-consistent iteration repeatedly increased it. Such asymmetries in ψ_e and ψ_h needed only to be minute in the first instance, and could arise merely from a coarse finite difference mesh; in fact, a difference in the wave function maxima of 1 part in 10^{10} was enough to have an effect. Given these initial conditions, the iteration forcibly broke the symmetry of the electron and hole, thus forcing them to localise in the second well of a four-well system, for example.

A systematic study of the original multiple quantum wells found that if a very small asymmetry was introduced into the potential profile of the heterostructure (perhaps just one monolayer within the well, having an alloy concentration of 10^{-5} rather than zero), then it was always enough to spontaneously break the symmetry of the exciton [150], with the result being that the exciton energy *always* decreased as the number of wells in the system increased. A point to note here is the size of the asymmetry necessary in the potential to bring about localisation. This is very small and will always be present in even the highest quality semiconductor layers. However, in addition, there are many other mechanisms which could bring about the small asymmetry, e.g. the presence of a single impurity atom, a phonon, or another charge carrier, will all be sufficient to induce a tiny perturbation in either the electron or the hole wave function, which will then influence the final exciton state. Once the exciton begins to form, it 'pulls itself up by its own bootstraps' until it is fully localised within one well. Therefore, in conclusion, excitons within multiple quantum well systems will (in general) spontaneously localise into just one of the wells.

It must be noted for the record, however, that the spontaneous symmetry-breaking of excitons in multiple quantum wells has recently been disputed in the literature [151].

6.11 2S EXCITON

In a similar manner to both the cases of the hydrogen atom and impurities, excitons can also have excited states. While the electron and hole one-particle states remain unchanged, the corresponding choice for the relative motion factor of the 2s excited state is given by:

$$\psi_r^{2s} = \left(1 - \frac{\alpha r'}{\lambda_{2s}}\right) \exp\left(-\frac{r'}{\lambda_{2s}}\right) \quad (6.104)$$

where λ_{2s} is stated specifically to imply the Bohr radius of the 2s state and α is a (as yet undefined) parameter, chosen to ensure orthogonality of the 1s and 2s eigenstates, say, for example, as follows:

$$\mathcal{E} = \langle \Psi^{1s} | \Psi^{2s} \rangle = 0 \quad (6.105)$$

In the spirit of the above, this can be written as:

$$\mathcal{E} = \int_0^{\nu-\mu} p(a) L(a) da \quad (6.106)$$

where, in a similar manner to equation (6.18):

$$L(a) = \int \psi_r^{1s} \psi_r^{2s} dx dy \quad (6.107)$$

Using the specific forms for ψ_r^{1s} and ψ_r^{2s} as above, then:

$$L(a) = 2\pi \left\{ \sqrt{1 - \beta^2} a \lambda' + \lambda'^2 - \frac{\alpha}{\lambda_{2s}} \left[(1 - \beta^2) a^2 \lambda' + 2\sqrt{1 - \beta^2} a \lambda'^2 + 2\lambda'^3 \right] \right\} \\ \times \exp\left(-\frac{\sqrt{1 - \beta^2} a}{\lambda'}\right) \quad (6.108)$$

where $1/\lambda' = 1/\lambda_{1s} + 1/\lambda_{2s}$. Writing this as $L(a) = L_1 - \alpha L_2$, then obtain:

$$\alpha = \frac{\int_0^{\nu-\mu} p(a) L_1(a) da}{\int_0^{\nu-\mu} p(a) L_2(a) da} \quad (6.109)$$

The remaining functions, $F(a)$, $G(a)$, $J(a)$ and $K(a)$, can all be derived by using similar procedures to the 1s case, and after lengthy manipulation are given by:

$$F(a) = 2\pi \left\{ \frac{\sqrt{1 - \beta^2} a \lambda_{2s}}{2} + \frac{\lambda_{2s}^2}{4} \right. \\ \left. - \frac{2\alpha}{\lambda_{2s}} \left[\frac{(1 - \beta^2) a^2 \lambda_{2s}}{2} + \frac{\sqrt{1 - \beta^2} a \lambda_{2s}^2}{2} + \frac{\lambda_{2s}^3}{4} \right] \right\}$$

$$+ \frac{\alpha^2}{\lambda_{2s}^2} \left(\frac{(1-\beta^2)^{\frac{3}{2}} a^3 \lambda_{2s}}{2} + \frac{3(1-\beta^2) a^2 \lambda_{2s}^2}{4} + \frac{6\sqrt{1-\beta^2} a \lambda_{2s}^3}{8} + \frac{6\lambda_{2s}^4}{16} \right) \Bigg\} \\ \times \exp \left(-\frac{2\sqrt{1-\beta^2} a}{\lambda_{2s}} \right) \quad (6.110)$$

In addition, $G(a) = G_1 + G_2$ where:

$$G_1 = 2\pi \int_0^1 (\alpha+1)^2 \frac{(1-\beta^2)^2 a^2}{\lambda_{2s}^2} \exp \left[-\frac{\sqrt{1-\beta^2} a}{\lambda_{2s}} \left(\frac{1}{w} + w \right) \right] \\ \times \left[\frac{1-w^2}{w(1+w^2)} \right] dw \quad (6.111)$$

and:

$$G_2 = 2\pi \left[-\frac{\alpha(\alpha+1)}{\lambda_{2s}^2} + \frac{\alpha^2}{\lambda_{2s}^3} \left(\frac{\sqrt{1-\beta^2} a}{2} + \frac{\lambda_{2s}}{4} \right) \right] \\ \times \exp \left(-\frac{2\sqrt{1-\beta^2} a}{\lambda_{2s}} \right) (1-\beta^2)^2 a^2 \quad (6.112)$$

Continuing for the remaining functions:

$$J(a) = 2\pi \int_0^1 \left[1 - \frac{\alpha\sqrt{1-\beta^2} a \left(\frac{1}{w} + w \right)}{2\lambda_{2s}} \right] \\ \times \left\{ \left[\frac{2\alpha+1}{\lambda_{2s}^2} - \frac{\alpha\sqrt{1-\beta^2} a \left(\frac{1}{w} + w \right)}{2\lambda_{2s}^3} \right] \left(\frac{1-w^2}{1+w^2} \right)^4 \right. \\ \left. - \left[\frac{2(\alpha+1)w}{\lambda_{2s}\sqrt{1-\beta^2} a(1+w^2)} - \frac{\alpha}{\lambda_{2s}^2} \right] \left[1 + \frac{4w^2}{(1+w^2)^2} \right] \right\} \\ \times \exp \left[-\frac{\sqrt{1-\beta^2} a \left(\frac{1}{w} + w \right)}{\lambda_{2s}} \right] \left[\frac{(1-\beta^2)a^2}{4} \right] \left(\frac{1}{w} - w \right) \left(\frac{1}{w^2} + 1 \right) dw \quad (6.113)$$

and finally:

$$K(a) = 2\pi \frac{\beta a}{2} \int_0^{\left(\frac{1}{\beta} - \frac{1}{\beta} \sqrt{1-\beta^2} \right)} \left[1 - \frac{\alpha\beta a}{2\lambda_{2s}} \left(\frac{1}{w} - w \right) \right]^2 \\ \times \exp \left[-\frac{\beta a}{\lambda_{2s}} \left(\frac{1}{w} - w \right) \right] \left(\frac{1}{w^2} - 1 \right) dw \quad (6.114)$$

CHAPTER 7

STRAINED QUANTUM WELLS

contributed by V. D. Jovanović

7.1 STRESS AND STRAIN IN BULK CRYSTALS

A mechanical force acting on a crystal lattice changes the relative positions of the lattice points (sites) i.e. the positions of the atoms forming the crystal structure. This can be characterised by a vector \vec{u} , which defines the relative displacement of an atom into a new position $\vec{r}' = \vec{r} + \vec{u}$ in some arbitrary Cartesian coordinate system. Different crystal lattice points can have different relative displacements making the vector \vec{u} coordinate dependent i.e. $\vec{u} = \vec{u}(\vec{r})$. If the lattice points return to their original positions after the force is removed then the deformation is described as elastic.

Knowing the relative displacements of each lattice site, the state of the crystal deformation can be described by strain components defining a second-rank tensor

(for more about application of tensors in crystals see Nye [152]) as:

$$\epsilon_{ij} = \frac{1}{2} \left(\frac{\partial u_i}{\partial x_j} + \frac{\partial u_j}{\partial x_i} \right), \quad i, j = 1, 2, 3 \quad (7.1)$$

where u_1 , u_2 , and u_3 are the relative displacements of the crystal lattice points along the x_1 , x_2 and x_3 axes, respectively. The diagonal components represent extensions per unit length along the x_1 , x_2 and x_3 directions and are usually referred as 'stretches' (see Fig. 7.1), while the off-diagonal components ϵ_{ij} are related to 'rotations' e.g. the term $\partial u_i / \partial x_j$ represents a rotation about the x_3 axis toward x_1 of a line element parallel to x_2 . As such the angle between the two line elements parallel to x_1 and x_2 changes from $\pi/2$ before deformation to $\pi/2 - \epsilon_{ij}$ after *. The strain tensor is symmetrical ($\epsilon_{ij} = \epsilon_{ji}$) and can be written as:

$$\epsilon = \begin{pmatrix} \epsilon_{11} & \epsilon_{12} & \epsilon_{31} \\ \epsilon_{12} & \epsilon_{22} & \epsilon_{23} \\ \epsilon_{31} & \epsilon_{23} & \epsilon_{33} \end{pmatrix} \quad (7.2)$$

If the shear strain components are zero, the diagonal elements also determine the change in the crystal volume (also known as 'dilation') as:

$$\frac{\Delta V}{V} = \text{Tr}(\epsilon) = \epsilon_{11} + \epsilon_{22} + \epsilon_{33} \quad (7.3)$$

This can be explained in terms of the unity cube strained by the diagonal components. Assuming the strains are small, which is true in the limits of linear strain theory as well as in most applications, the distorted volume will then be $(1 + \epsilon_{11})(1 + \epsilon_{22})(1 + \epsilon_{33})$ giving a total change in the cube volume (to first-order) of $\epsilon_{11} + \epsilon_{22} + \epsilon_{33}$ (see Fig. 7.1).

It is also important to define the properties of the force causing the strain in the crystal lattice. If a crystal is acted on by an external force or if part of a crystal is applying a force on a neighbouring part then the crystal is said to be in the state of 'stress', where the stress is usually defined as the force per unit area of the crystal. If a unit cube is considered then normal stress components can be defined as σ_{ii} e.g. σ_{11} , σ_{22} , σ_{33} etc. and the shear stress components as σ_{ij} e.g. σ_{12} , σ_{21} , σ_{13} etc. as shown in Fig. 7.2. In the homogeneous case when the force is independent of the position on the crystal surface, the stress components form a symmetric second-rank tensor ($\sigma_{ij} = \sigma_{ji}$) as:

$$\sigma = \begin{pmatrix} \sigma_{11} & \sigma_{12} & \sigma_{31} \\ \sigma_{12} & \sigma_{22} & \sigma_{23} \\ \sigma_{31} & \sigma_{23} & \sigma_{33} \end{pmatrix} \quad (7.4)$$

The diagonal elements of the stress tensor are of special importance in applications. If only σ_{11} is a non-zero stress component then the stress is said to be uniaxial, while if both σ_{11} and σ_{22} are not equal to zero the stress is defined as 'biaxial'. The latter is the case common in quantum well structures as elaborated on in the following section.

*Note that in the literature an intuitive notation is occasionally seen replacing the indices 1, 2 and 3 with x , y and z , e.g. $\epsilon_{11} \rightarrow \epsilon_{xx}$.

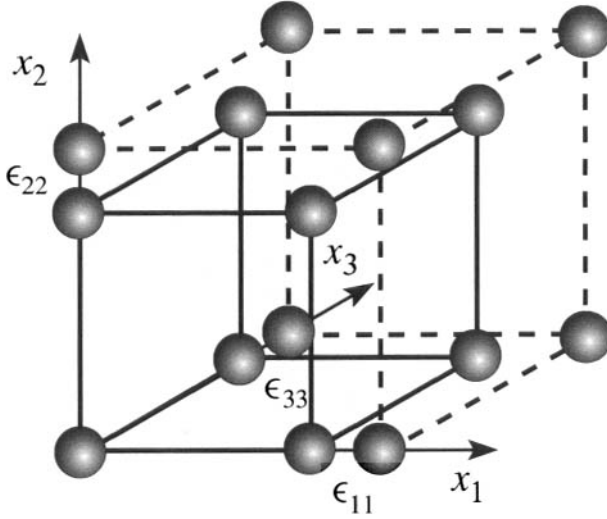


Figure 7.1 Schematic illustration of the influence of the diagonal strain components or stretches (ϵ_{11} , ϵ_{22} , ϵ_{33}) on the crystal lattice.

In the limit of linear elastic theory the stress and the strain are connected by Hooke's law which reads:

$$\epsilon_{ij} = \sum_{k=1}^3 \sum_{l=1}^3 S_{ijkl} \sigma_{kl} \quad (7.5)$$

where S_{ijkl} are the 'elastic compliance constants' which form a fourth-rank tensor. In the literature, stress is commonly presented as a function of strain and Hooke's law takes the similar form:

$$\sigma_{ij} = \sum_{k=1}^3 \sum_{l=1}^3 C_{ijkl} \epsilon_{kl} \quad (7.6)$$

where C_{ijkl} are the 'elastic stiffness constants' and also form a fourth-rank tensor.

A more concise, matrix (or Voigt's) notation is commonly used in the literature which reduces the number of tensor indices and transforms the strain, stress and elastic stiffness components as follows:

$$\begin{pmatrix} \epsilon_{11} & \epsilon_{12} & \epsilon_{31} \\ \epsilon_{12} & \epsilon_{22} & \epsilon_{23} \\ \epsilon_{31} & \epsilon_{23} & \epsilon_{33} \end{pmatrix} \rightarrow \begin{pmatrix} \epsilon_1 & \frac{1}{2}\epsilon_6 & \frac{1}{2}\epsilon_5 \\ \frac{1}{2}\epsilon_6 & \epsilon_2 & \frac{1}{2}\epsilon_4 \\ \frac{1}{2}\epsilon_5 & \frac{1}{2}\epsilon_4 & \epsilon_3 \end{pmatrix} \quad (7.7)$$

$$\begin{pmatrix} \sigma_{11} & \sigma_{12} & \sigma_{31} \\ \sigma_{12} & \sigma_{22} & \sigma_{23} \\ \sigma_{31} & \sigma_{23} & \sigma_{33} \end{pmatrix} \rightarrow \begin{pmatrix} \sigma_1 & \sigma_6 & \sigma_5 \\ \sigma_6 & \sigma_1 & \sigma_4 \\ \sigma_5 & \sigma_4 & \sigma_3 \end{pmatrix} \quad (7.8)$$

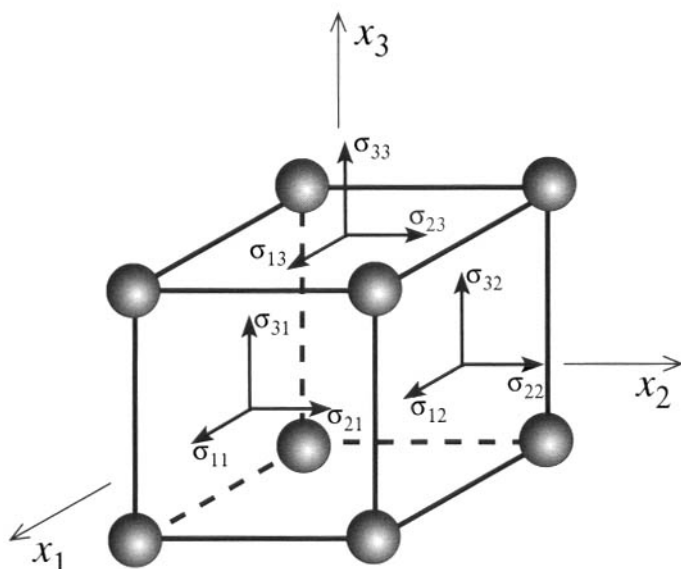


Figure 7.2 Directions of the stress components acting on a crystal lattice that are usually assumed.

$$C_{ijkl} \rightarrow C_{mn} \quad i, j, k, l = 1, 2, 3; m, n = 1, \dots, 6 \quad (7.9)$$

The $\frac{1}{2}$ terms have been introduced in order to give a clear relationship between the stress, strain and other parameters used. Hooke's law given by equation (7.6) in the matrix notation takes a simplified form and reads:

$$\sigma_i = \sum_{k=1}^6 C_{ik} \epsilon_k \quad (7.10)$$

or equivalently:

$$\begin{bmatrix} \sigma_1 \\ \sigma_2 \\ \sigma_3 \\ \sigma_4 \\ \sigma_5 \\ \sigma_6 \end{bmatrix} = \begin{bmatrix} C_{11} & C_{12} & C_{13} & C_{14} & C_{15} & C_{16} \\ C_{21} & C_{22} & C_{23} & C_{24} & C_{25} & C_{26} \\ C_{31} & C_{32} & C_{33} & C_{34} & C_{35} & C_{36} \\ C_{41} & C_{42} & C_{43} & C_{44} & C_{45} & C_{46} \\ C_{51} & C_{52} & C_{53} & C_{54} & C_{55} & C_{56} \\ C_{61} & C_{62} & C_{63} & C_{64} & C_{65} & C_{66} \end{bmatrix} \begin{bmatrix} \epsilon_1 \\ \epsilon_2 \\ \epsilon_3 \\ \epsilon_4 \\ \epsilon_5 \\ \epsilon_6 \end{bmatrix} \quad (7.11)$$

where C_{ik} is now the elastic stiffness matrix which is related to the elastic compliance matrix as $[C] = [S]^{-1}$ and defines the unique correlation between the elastic constants.

In order to define the elastic properties of the crystal 36 independent elastic stiffness constants are required. However, crystal lattices commonly exhibit certain symmetries which can be employed to reduce the number of constants necessary to describe

their elastic behaviour. The number of independent matrix components for the most used semiconductors are: 3 for the cubic (e.g. GaAs) and 5 for the hexagonal (e.g. wurtzite GaN) crystal geometries. Hence, the C matrices for these semiconductors are given as:

$$\begin{bmatrix} C_{11} & C_{12} & C_{12} & 0 & 0 & 0 \\ C_{12} & C_{11} & C_{12} & 0 & 0 & 0 \\ C_{12} & C_{12} & C_{11} & 0 & 0 & 0 \\ 0 & 0 & 0 & C_{44} & 0 & 0 \\ 0 & 0 & 0 & 0 & C_{44} & 0 \\ 0 & 0 & 0 & 0 & 0 & C_{44} \end{bmatrix} \quad (7.12)$$

and

$$\begin{bmatrix} C_{11} & C_{12} & C_{13} & 0 & 0 & 0 \\ C_{12} & C_{11} & C_{13} & 0 & 0 & 0 \\ C_{13} & C_{13} & C_{33} & 0 & 0 & 0 \\ 0 & 0 & 0 & C_{44} & 0 & 0 \\ 0 & 0 & 0 & 0 & C_{44} & 0 \\ 0 & 0 & 0 & 0 & 0 & \frac{1}{2}(C_{11} - C_{12}) \end{bmatrix} \quad (7.13)$$

for the cubic and the hexagonal crystals, respectively.

Finally, to deform the crystal lattice a certain amount of energy is needed. The sum of the work done by the stress components acting on the crystal defines the strain energy as another important parameter influencing the crystal strain state. In terms of the strain components with respect to the validity of Hooke's law, the strain energy per volume unit can be expressed in the matrix notation as:

$$W = \frac{1}{2} \sum_{i=1}^6 \sum_{j=1}^6 C_{ij} \epsilon_i \epsilon_j \quad (7.14)$$

The strain energy density always has to be greater than zero. Furthermore, the crystal system reaches the most stable strain state for the minimal value of the strain energy—a condition which is used in the derivation of strain balancing in Section 7.3.

For cubic semiconductors, equations (7.12) and (7.14) give the elastic energy as:

$$W_{\text{cubic}} = \frac{1}{2} C_{11} (\epsilon_1^2 + \epsilon_2^2 + \epsilon_3^2) + \frac{1}{2} C_{44} (\epsilon_4^2 + \epsilon_5^2 + \epsilon_6^2) + C_{12} (\epsilon_1 \epsilon_2 + \epsilon_2 \epsilon_3 + \epsilon_3 \epsilon_1) \quad (7.15)$$

whilst for hexagonal crystals it takes the form:

$$\begin{aligned} W_{\text{hex}} = & \frac{1}{2} C_{11} (\epsilon_1^2 + \epsilon_2^2) + \frac{1}{2} C_{33} \epsilon_3^2 + C_{12} \epsilon_1 \epsilon_2 + C_{13} \epsilon_3 (\epsilon_1 + \epsilon_2) \\ & + \frac{1}{2} C_{44} (\epsilon_4^2 + \epsilon_5^2) + \frac{1}{4} (C_{11} - C_{12}) \epsilon_6^2 \end{aligned} \quad (7.16)$$

7.2 STRAIN IN QUANTUM WELLS

The effects of strain are of particular interest in quantum well structures. If a thin epitaxial layer is deposited on a much thicker substrate (usually assumed to be infinitely thick in comparison to the epitaxial layer) then the lattice constant (a_l) in the growth plane (perpendicular to the growth direction) of the layer will be forced to change to try and equal the lattice constant of the substrate (a_0). As a consequence, the crystal lattice is under biaxial stress along the growth interface and while no force is applied along the growth direction the crystal is able to relax freely along that direction. Therefore, the stress has only two diagonal components σ_1 and σ_2 while σ_3 and the shear components are zero. Hence, in the matrix notation the stress in the epitaxial (quantum well) layer can be written as:

$$\sigma = \begin{pmatrix} \sigma_1 & 0 & 0 \\ 0 & \sigma_2 & 0 \\ 0 & 0 & 0 \end{pmatrix} \quad (7.17)$$

The existence of biaxial stress results in the appearance of an in-plane strain. Growth which allows the lattice constant of the epitaxial layer to fully equal (match) the substrate is usually referred as pseudomorphic growth. In such cases, the in-plane strain can be easily calculated as:

$$\epsilon_{\parallel} = \frac{a_0 - a_l}{a_l} \quad (7.18)$$

In quantum well systems the in-plane strain is usually of the order of 1% e.g. for an AlN layer grown on a GaN substrate $\epsilon_{\parallel} \approx 2\%$. However, some materials and their alloys have very similar lattice constants such as the GaAs/AlGaAs system for which $\epsilon_{\parallel} < 0.1\%$ allowing them to be considered as unstrained.

The lattice constant of the epitaxial layer can be either larger or smaller than the lattice constant of the substrate (see Fig. 7.3) defining a compressive ($\epsilon_{\parallel} < 0$) or a tensile ($\epsilon_{\parallel} > 0$) strain, respectively. A compressive strain will therefore force the lattice constant in the plane to shrink (see material B in Fig. 7.3) conversely tensile strain will force it to expand (see material A in Fig. 7.3). It is also reasonable to assume that the strain in the thin semiconductor layers is homogeneous and therefore constant throughout the layer.

If the epitaxial layer width is larger than some critical thickness the layer relaxes in the plane (perpendicular to the growth direction) restoring its original lattice constant. This is a violent process producing a large number of defects and imperfections (cracks) in the growth surface. The formation of defects and the existence of a critical thickness can be understood in term of the elastic energy. The strained layer system possesses a certain additional elastic energy which is a function of the layer thickness (approximately a product of the elastic energy density, the area and the layer width). If for a specific layer width the strain energy exceeds the energy required for the generation of defects, then the system will tend to relax to a new state with lower

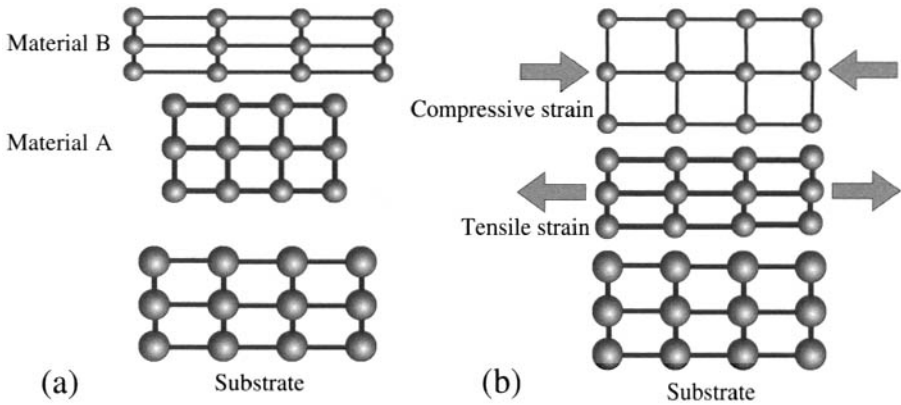


Figure 7.3 Schematic illustration of a substrate with two mismatched layers, (a) shows the free-standing unstrained layers with lattice constants smaller $a_l < a_0$ (material A) and larger $a_l > a_0$ (material B) than the lattice constant of the substrate and (b) shows the situation when either of the materials is grown on the substrate, i.e. material A is under tensile strain and material B is under compressive strain as their lattice constants are forced to be equal to the lattice constant of the substrate.

strain energy forming imperfections in the growth plane. The width of the layer for which this relaxation process occurs is commonly referred to as the 'critical thickness' (see Ref. [153]). Therefore, pseudomorphic growth is a necessary condition for the fabrication of good quality layers with a small number of intrinsic defects.

Though no stress exists ($\sigma_3 = 0$) in the growth direction the lattice constant is still forced to change due to the 'Poisson effect' (see Fig. 7.3). If the compressive strain forces the in-plane lattice constant to reduce then the lattice constant in the growth direction will increase and vice versa for tensile strain. Hence, strain exists in the growth direction as well. The ratio that determines the increase or decrease of the lattice constant due to the in-plane stress is called Poisson's ratio (ν) and it connects the in-plane and the perpendicular strains as:

$$\epsilon_3 = -\nu\epsilon_1 \quad (7.19)$$

For the commonly used cubic semiconductor materials grown along the [001] direction, the relationship between the stress and the strain in the epitaxial (quantum well) layers is given by Hooke's law (equations (7.6) and (7.12)) and for the biaxial

stress assumed reads:

$$\begin{bmatrix} \sigma_1 \\ \sigma_2 \\ \sigma_3 \\ 0 \\ 0 \\ 0 \end{bmatrix} = \begin{bmatrix} C_{11} & C_{12} & C_{12} & 0 & 0 & 0 \\ C_{12} & C_{11} & C_{12} & 0 & 0 & 0 \\ C_{12} & C_{12} & C_{11} & 0 & 0 & 0 \\ 0 & 0 & 0 & C_{44} & 0 & 0 \\ 0 & 0 & 0 & 0 & C_{44} & 0 \\ 0 & 0 & 0 & 0 & 0 & C_{44} \end{bmatrix} \begin{bmatrix} \epsilon_1 \\ \epsilon_2 \\ \epsilon_3 \\ \epsilon_4 \\ \epsilon_5 \\ \epsilon_6 \end{bmatrix} \quad (7.20)$$

The matrix equation defines a system of two linear independent equations. Knowing that the in-plane strain is actually equal to ϵ_1 , the strains in the epitaxial layers are given as:

$$\begin{aligned} \epsilon_1 &= \epsilon_2 = \epsilon_{\parallel} \\ \epsilon_3 &= -2 \frac{C_{12}}{C_{11}} \epsilon_1 \end{aligned} \quad (7.21)$$

or in matrix form:

$$\epsilon = \begin{pmatrix} \epsilon_1 & 0 & 0 \\ 0 & \epsilon_1 & 0 \\ 0 & 0 & -2 \frac{C_{12}}{C_{11}} \epsilon_1 \end{pmatrix} \quad (7.22)$$

where the factor $\nu = -2 \frac{C_{12}}{C_{11}}$ represents Poisson's ratio for cubic semiconductors in the [001] direction. Typical values of the strain components and the strain distribution in cubic InGaAs/AlGaAs quantum wells are presented in Fig. 7.4.

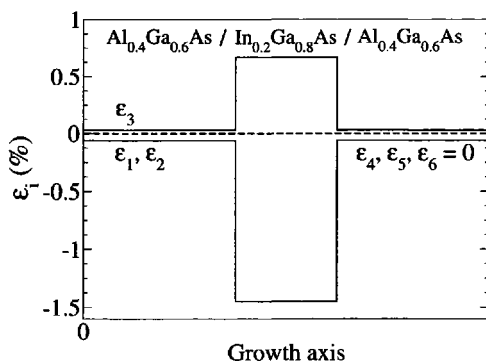


Figure 7.4 Values of the strain components in an $\text{In}_{0.2}\text{Ga}_{0.8}\text{As}$ quantum well surrounded by $\text{Al}_{0.4}\text{Ga}_{0.6}\text{As}$ barriers.

Knowing the strain components makes it possible to determine the relative change in the lattice volume (or dilation) using equation (7.3) which now reads:

$$\Theta = \frac{\Delta V}{V} = 2\epsilon_1 \left(1 - \frac{C_{12}}{C_{11}} \right) \quad (7.23)$$

For quantum wells with the layers based on the hexagonal crystal geometry grown in the [0001] direction a similar approach can be applied, modifying Hooke's law as:

$$\begin{bmatrix} \sigma_1 \\ \sigma_2 \\ \sigma_3 \\ 0 \\ 0 \\ 0 \end{bmatrix} = \begin{bmatrix} C_{11} & C_{12} & C_{13} & 0 & 0 & 0 \\ C_{12} & C_{11} & C_{13} & 0 & 0 & 0 \\ C_{13} & C_{13} & C_{33} & 0 & 0 & 0 \\ 0 & 0 & 0 & C_{44} & 0 & 0 \\ 0 & 0 & 0 & 0 & C_{44} & 0 \\ 0 & 0 & 0 & 0 & 0 & \frac{1}{2}(C_{11} - C_{12}) \end{bmatrix} \begin{bmatrix} \epsilon_1 \\ \epsilon_2 \\ \epsilon_3 \\ \epsilon_4 \\ \epsilon_5 \\ \epsilon_6 \end{bmatrix} \quad (7.24)$$

which implies the solutions for the strain components are:

$$\epsilon_1 = \epsilon_2 = \epsilon_{\parallel} \quad \text{and} \quad \epsilon_3 = -2 \frac{C_{12}}{C_{11}} \epsilon_1 \quad (7.25)$$

or in matrix form:

$$\epsilon = \begin{pmatrix} \epsilon_1 & 0 & 0 \\ 0 & \epsilon_1 & 0 \\ 0 & 0 & -2 \frac{C_{12}}{C_{11}} \epsilon_1 \end{pmatrix} \quad (7.26)$$

Similarly as for the cubic crystal geometry, for epitaxial layers based on hexagonal materials the relative change of the volume can be expressed as:

$$\Theta = \frac{\Delta V}{V} = 2\epsilon_1 \left(1 - \frac{C_{12}}{C_{33}} \right) \quad (7.27)$$

7.3 STRAIN BALANCING

In order to ensure pseudomorphic growth of multiple layers of quantum wells a technique called 'strain-balancing' is commonly employed. This implies a careful choice of the epitaxial layer compositions and widths as well as in the choice of the substrate (or buffer layer) parameters if possible, with the aim of engineering the strain state of the entire structure, and therefore preventing the appearance of misfit dislocations and other defects. However, the available margins for such parameter manipulation are quite small as the requirements of the quantum well structure enforce tight constraints.

The general method for evaluating a 'strain-balancing condition' is based on the fact that the stable strain state corresponds to the minimum of the elastic energy in each layer of the quantum well structure. However, when the multilayer system is considered such a state needs to be reached across the entire structure and not only in

a single layer so that a favourable strain distribution is reached overall and prevents the relaxation process. Such a condition can be met by minimising the average elastic energy with respect to the in-plane strain (known as the 'zero-stress' condition) [154] given as:

$$\bar{W} = \frac{\sum_{k=1}^n W_k l_k}{\sum_{k=1}^n l_k} \quad (7.28)$$

where l_k is the width and W_k the strain energy density of the k -th layer.

Recalling the elastic energy density definition stated in equation (7.14) the energy density of the k -th layer can be written as

$$W_k = \frac{1}{2} \sum_{i=1}^6 \sum_{j=1}^6 C_{ij}^{(k)} \epsilon_i^{(k)} \epsilon_j^{(k)} \quad (7.29)$$

where $C_{ij}^{(k)}$ is the elastic stiffness constant and $\epsilon_i^{(k)}$ is the strain component (in matrix notation) of the k -th layer.

For the commonly used cubic and hexagonal semiconductors the elastic energy densities are given by equations (7.15) and (7.16). If the nature of the strain in the epitaxial layers is assumed to be due to a biaxial stress, then the equations can be written in a simpler form by substituting equation (7.12) or (7.13) and equation (7.21) into equation (7.14) for the cubic and equivalently for the hexagonal crystals. The elastic energy density then reads:

$$W_k = A_k \cdot [\epsilon_1^{(k)}]^2 \quad (7.30)$$

where:

$$A_k = C_{11} + C_{12} - 2 \frac{C_{13}^2}{C_{33}} \quad (7.31)$$

for the cubic and:

$$A_k = C_{11} + C_{12} - 2 \frac{C_{13}^2}{C_{33}} \quad (7.32)$$

for the hexagonal crystal geometries.

With respect to equation (7.28), equation (7.29) can then written as:

$$\bar{W} = \frac{\sum_{k=1}^n A_k [\epsilon_1^{(k)}]^2 l_k}{\sum_{k=1}^n l_k} \quad (7.33)$$

The pseudomorphic condition implies that the modified lattice constants (a_k and a_{k-1}) of adjacent layers must be equal to that of the substrate. The in-plane strains in each of the layers can then be written as:

$$\epsilon_1^{(k)} = \frac{a_0 - a_k}{a_k} \quad (7.34)$$

and:

$$\epsilon_1^{(k-1)} = \frac{a_0 - a_{k-1}}{a_{k-1}} \quad (7.35)$$

which means that the in-plane strains in adjacent layers ($\epsilon_1^{(k)}$ and $\epsilon_1^{(k-1)}$) can be related as:

$$\epsilon_1^{(k)} = \frac{a_{k-1}}{a_k} \epsilon_1^{(k-1)} + \frac{a_{k-1} - a_k}{a_k} \quad (7.36)$$

and the average in-plane stress can follow as:

$$\bar{\sigma} = \frac{\partial \bar{W}}{\partial \epsilon_1^{(1)}} = \frac{2}{l_1 + l_2 + l_3 + \dots} \times \left\{ A_1 \epsilon_1^{(1)} l_1 + A_2 \epsilon_1^{(2)} l_2 \frac{\partial \epsilon_1^{(2)}}{\partial \epsilon_1^{(1)}} + A_3 \epsilon_1^{(3)} l_3 \frac{\partial \epsilon_1^{(3)}}{\partial \epsilon_1^{(1)}} + \dots \right\} \quad (7.37)$$

The zero-stress (or strain-balance) condition implies that the *in-plane* stress is zero ($\bar{\sigma} = 0$), and using $\partial \epsilon_1^{(k)} / \partial \epsilon_1^{(1)} = a_1 / a_k$ ($k = 2, 3, \dots$) then equation (7.37) gives:

$$A_1 \epsilon_1^{(1)} l_1 + A_2 \epsilon_1^{(2)} l_2 \frac{a_1}{a_2} + A_3 \epsilon_1^{(3)} l_3 \frac{a_1}{a_3} + \dots = 0 \quad (7.38)$$

which delivers the lattice constant of the substrate (or suitably grown buffer layer) necessary for strain-balancing the quantum well stack as:

$$a_0 = \frac{\sum_{k=1}^n A_k l_k / a_k}{\sum_{k=1}^n A_k l_k / a_k^2} \quad (7.39)$$

If this lattice constant is not equal to that of any readily available substrate then it can be achieved by growth of a suitable buffer layer. For example in the $\text{Si}_{1-x}\text{Ge}_x/\text{Si}$ material system strain-balancing is achieved by growing a buffer layer with the appropriate Ge composition, i.e. $\text{Si}_{1-y}\text{Ge}_y$. This is usually linearly graded from the pure silicon composition of the substrate to the required composition y .

If a multilayer structure has periodicity, the above expression should be understood in terms of the single period i.e. it should be applied to the n layers constituting a single period, and the whole structure will then clearly be strain balanced.

The previous derivation takes into account the difference in elastic properties of the layers (i.e. elastic stiffness constants). If the elastic constants are similar for all layers of the structure then the strain-balancing condition can be simplified by taking into account only the difference of the lattice constants (also known as the 'average lattice method') as:

$$a_0 = \frac{\sum_{k=1}^n l_k a_k}{\sum_{k=1}^n l_k} \quad (7.40)$$

If the substrate is fixed and cannot be engineered, the quantum well structure itself has to be 'tailored' to match the substrate, the complexity of which depends on the number of different layers per period and the boundaries imposed by the desired application. In something like a quantum well infrared photodetector (QWIP) this could be achieved by altering the width of the thick barrier layer separating the quantum well absorbing regions, as provided the thickness of this layer is above some minimum (to limit the dark current) it's precise value is not too important. In a quantum cascade laser the injector region offers some flexibility to allow a design to be created which also satisfies the strain-balancing condition.

7.4 EFFECT ON THE BAND PROFILE OF QUANTUM WELLS

The strain in epitaxial layers acts to change the crystal lattice geometry, i.e. to perturb its size and symmetry. This results in a change of the electronic structure in particular a modification of the conduction and valence band edges and therefore a corresponding shift in the energy levels. In order to give a quantitative measure of this further effect of strain, a general case of the strained crystal is considered under the framework of deformation potential theory. This section will consider the simple case of the conduction band Γ minimum, while more detailed analysis exploring the valence band will be presented in Chapter 10.

If the strain is assumed to be small, which is the case in semiconductor quantum wells ($\sim 1\%$), first order perturbation theory can be used to calculate the band shift. The Hamiltonian under strain can be expressed as the sum of the unperturbed Hamiltonian (\mathcal{H}_0) and the strain induced contribution (\mathcal{H}_ϵ), i.e.

$$\mathcal{H} = \mathcal{H}_0 + \mathcal{H}_\epsilon \quad (7.41)$$

where the dependence of \mathcal{H}_ϵ on the strain is:

$$\mathcal{H}_\epsilon = \sum_{i,j} \frac{\partial V}{\partial \epsilon_{ij}} \epsilon_{ij} \quad (7.42)$$

and V is the original (unperturbed) crystal potential. From perturbation theory it follows that the energy shift due to strain can be expressed in terms of the deformation potential components representing the matrix elements of the relative band shift due to strain (the $\frac{\partial V}{\partial \epsilon_{ij}}$ term in equation (7.42)) as:

$$\delta E_c = \sum_{i,j} D_{ij} \epsilon_{ij} \quad (7.43)$$

Note that the D_{ij} forms a second-rank tensor though the matrix representation is commonly used. The number of non-zero D_{ij} components is dependent on the crystal symmetry (similar to strain theory presented earlier) and the type of band minimum.

For the conduction band Γ point in cubic semiconductors all the off-diagonal deformation potential components vanish ($D_{ij}=0$) leaving only three equal diagonal

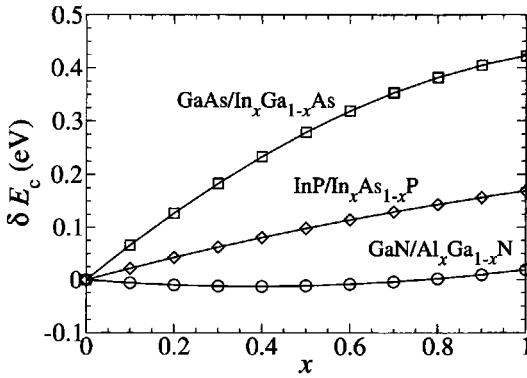


Figure 7.5 Conduction band energy shifts for strained layers based on different materials: AlGa_xN grown on GaN substrate (circles), InAs_xP grown on InP (diamonds) and InGa_xAs grown on GaAs (squares), as a function of the layer composition.

constants $D_{11} = D_{22} = D_{33} = a_c$ (the isotropic case) which are usually determined by fitting to experimental data. Equation (7.43), giving the shift of the conduction band energy due to strain, now reads:

$$\delta E_c^{\text{cubic}} = a_c(\epsilon_1 + \epsilon_2 + \epsilon_3) \quad (7.44)$$

Its clear from the equation that the energy band shift depends only on the strain state of the crystal given via the diagonal strain components.

Hexagonal crystals exhibit much less symmetry than cubic ones. Consequently, a certain anisotropic behaviour can be expected and this is reflected in the different deformation potential constants. For example, GaN in the growth plane has identical deformation potential constants $D_{11} = D_{22} = a_{c\perp}$ whilst in the growth direction the constant $D_{33} = a_{c\parallel}$ is almost three times larger. The total conduction band shift in GaN then follows as:

$$\delta E_c^{\text{hex}} = a_{c\perp}(\epsilon_1 + \epsilon_2) + a_{c\parallel}\epsilon_3 \quad (7.45)$$

The previous analysis can be applied to determine the energy shift of the conduction band edge in the vicinity of the Γ minimum in the epitaxial layer of the quantum well structure. For the layers based on cubic semiconductor material grown in the [001] direction substituting the strain components given by equation (7.21) the band edge energy shift becomes:

$$\delta E_c^{\text{cubic}} = 2a_c \left(1 - \frac{C_{12}}{C_{11}} \right) \epsilon_1 \quad (7.46)$$

where ϵ_1 is the in-plane strain due to lattice mismatch with the substrate and C_{ij} are the elastic stiffness constants as previously defined. Similar, for the hexagonal crystal geometry in the $[0001]$ direction the energy shift is given as:

$$\delta E_c^{\text{hex}} = 2 \left(a_{c\perp} - a_{c\parallel} \frac{C_{13}}{C_{33}} \right) \epsilon_1 \quad (7.47)$$

The conduction band shift depends significantly on the material deformation potential constants and the lattice mismatch, and can be up to a few hundreds of meV as shown in Fig. 7.5. It is interesting to see that although the wurtzite AlGaIn layer grown on GaN is highly strained in the growth plane, because of the anisotropy of the hexagonal crystals the overall shift is smaller.

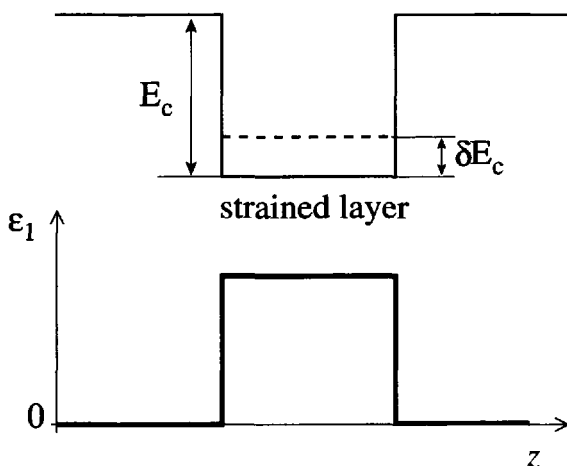


Figure 7.6 Schematic diagram of the conduction band edge of a quantum well with a strain induced energy band shift (δE_c). The original conduction band discontinuity is E_c . The lower diagram shows the corresponding in-plane strain distribution across the quantum well.

Consider now the influence of the conduction band edge shift on the electronic levels in a single quantum well. For simplicity it is assumed that the well material is strained while the barriers are matched to the substrate (Fig. 7.6). The conduction band edge shift implies the equivalent change in the barrier height, defining the new quantum well potential as:

$$U(z)_{\text{str}} = U(z) + \delta E_c(z) \quad (7.48)$$

which results in a shift of the discrete energy levels localised within the quantum well.

In order to illustrate the importance of strain in quantum wells Fig. 7.7 gives the energy of the ground state (with respect to the quantum well band edge) with and without the strain perturbation, as a function of the alloy composition (x) in the

InP/In_xAs_{1-x}P quantum well strained on an InP substrate. The calculation shows an average energy shift of around 10%, which can significantly influence the designs and applications. In addition strain will influence the energy difference between subbands so introducing a shift in intersubband transition energies. This implies that strain 'engineering' could be used to reach desired emission and detection wavelengths otherwise unattainable (see for example Faist *et al.* [155]).

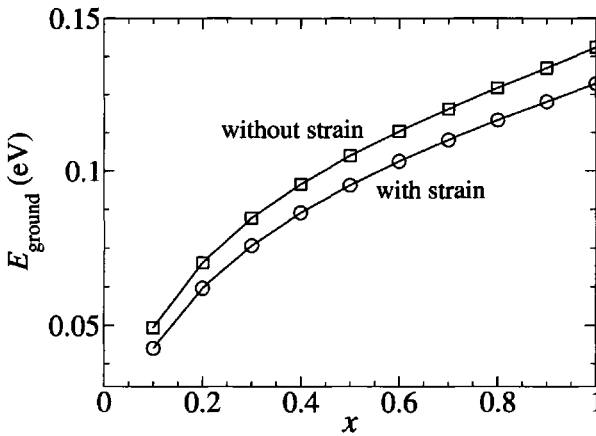


Figure 7.7 Energy of the ground state (with respect to the conduction band edge in the well) in the InP/In_xAs_{1-x}P system with and without strain, as a function of the alloy composition (x). The well width is set to 50 Å and the barrier width to 60 Å.

7.5 THE PIEZOELECTRIC EFFECT

Certain types of crystal materials exhibit a behaviour such that under stress, an extra electric charge gathers on their surfaces. The effect is called 'piezoelectricity' and is a consequence of a non-compensated electric polarisation generated in the volume of the crystal. Piezoelectric behaviour is exhibited in quartz, Rochelle Salt and Tourmaline, as well as in cubic and hexagonal semiconductors such as InGaAs and GaN. The changes in the crystal lattice geometry due to stress act to separate the charges and break the local electrostatic neutrality in the crystal, a schematic diagram illustrating this effect for wurtzite GaN is given in Fig. 7.8. The uncompensated positive and negative charges induce a global polarisation vector, the properties of which are defined by the material and the nature of the applied stress.

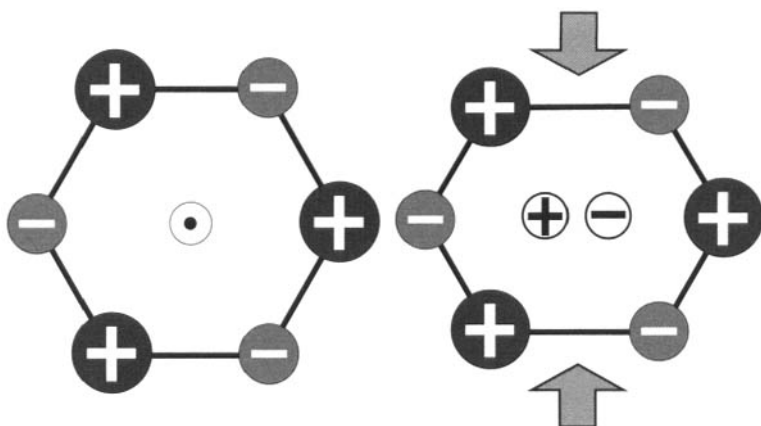


Figure 7.8 Schematic illustration of the charge separation in GaN grown along the [0001] direction which causes the appearance of an uncompensated polarisation. The left hand diagram shows the unstrained crystal lattice, and on the right the atomic positions when under strain.

If an arbitrary stress is acting on the crystal then the total induced polarisation (in matrix notation similar to that in Sec. 7.1) can be written as:

$$P_i = \sum_{k=1}^6 d_{ik} \sigma_k \quad (7.49)$$

where d_{ik} is the piezoelectric modulus. The piezoelectric polarisation is a vector which is described by three components, i.e. P_1 and P_2 in the growth plane and P_3 along the growth direction, which in turn define a 3×6 matrix of the piezoelectric moduli (d). As for the case of the elastic constants, the symmetry in the crystals reduces the number of independent elements of the d matrix. For cubic and hexagonal crystal geometries the number of independent moduli relies on the crystal class e.g. one modulus for InGaAs and three moduli for wurtzite GaN.

So it is stress induced symmetry breaking in a crystal which leads to a piezoelectric polarisation, hence the underlying symmetry of the crystal is important. A good example of this is InGaAs. When grown on a GaAs substrate in the [111] direction it exhibits a piezoelectric behaviour. However, layers grown along the [001] direction are polarisation free, while those grown along the [110] direction exhibit a piezoelectric polarisation in the plane of the layers.

In the following, the piezoelectric properties of epitaxial layers will be considered in more details for the example of the increasingly more widespread wurtzite GaN. In order to determine the piezoelectric polarisation for GaN grown in the [0001]

direction consider equation (7.49) which assuming a biaxial stress can be written as:

$$\begin{bmatrix} P_1 \\ P_2 \\ P_3 \end{bmatrix} = \begin{bmatrix} 0 & 0 & 0 & 0 & \frac{1}{2}d_{15} & 0 \\ 0 & 0 & 0 & \frac{1}{2}d_{15} & 0 & 0 \\ d_{31} & d_{31} & d_{33} & 0 & 0 & 0 \end{bmatrix} \begin{bmatrix} \sigma_1 \\ \sigma_2 \\ 0 \\ 0 \\ 0 \\ 0 \end{bmatrix} \quad (7.50)$$

Note that the factor $\frac{1}{2}$ is a consequence of adopting the matrix notation as discussed in Sec 7.1. The matrix equation can be decoupled into three linear equations defining the polarisation vector components, the solution of which gives a single non-zero component of the piezoelectric polarisation in the growth direction as:

$$P_3 = d_{31}(\sigma_1 + \sigma_2) = 2d_{31}\sigma_1 \quad (7.51)$$

Using equations (7.20) and (7.21), the stress component σ_1 can be expressed in terms of the strain as $\sigma_1 = \epsilon_1 \left(C_{11} + C_{12} - 2\frac{C_{13}^2}{C_{33}} \right)$. Hence, the final form of the in-plane piezoelectric polarisation of GaN reads:

$$P_3 = 2d_{31}\epsilon_1 \left(C_{11} + C_{12} - 2\frac{C_{13}^2}{C_{33}} \right) \quad (7.52)$$

where C_{ij} are the elastic stiffness constants. As the term $C_{11} + C_{12} - 2C_{13}^2/C_{33} > 0$ is always greater than zero, the sign of the polarisation is determined by the sign of the strain. Hence, P_3 is always positive for layers under biaxial compressive strain and negative for layers under tensile strain.

A more often used approach for describing piezoelectric properties of crystals than the moduli is that of the piezoelectric constants defined as:

$$e_{kl} = \sum_{j=1}^6 d_{kj} C_{jl} \quad (7.53)$$

where C_{jl} are the elastic stiffness constants. Then equation (7.49) for the piezoelectric polarisation can be rewritten as:

$$P_i = \sum_{k=1}^6 e_{ik} \epsilon_k \quad (7.54)$$

In terms of these piezoelectric constants, the polarisation in the wurtzite GaN layer under biaxial strain now reads:

$$P_3 = \epsilon_1 e_{31} + \epsilon_2 e_{32} + \epsilon_3 e_{33} = 2\epsilon_1 e_{31} + \epsilon_3 e_{33} = 2\epsilon_1 \left(e_{31} - e_{33} \frac{C_{13}}{C_{33}} \right) \quad (7.55)$$

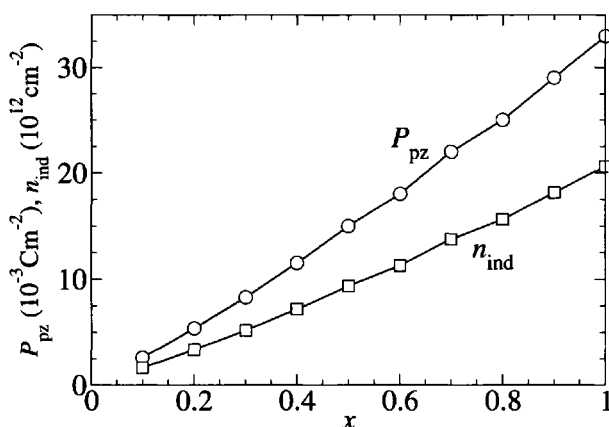


Figure 7.9 The piezoelectric polarisation and induced charge at the interface in a single $\text{Al}_x\text{Ga}_{1-x}\text{N}$ layer grown on a GaN substrate as a function of the layer composition x .

Fig. 7.9 shows the calculated piezoelectric polarisation and the induced charge ($n_{ind} = P_{pz}/q$) at the growth interface for a single AlGaIn layer grown strained on a GaN substrate. For example, for 50% of Al in the epitaxial layer the sheet electron density is around $1 \times 10^{13} \text{ cm}^{-2}$. This is a common case in GaN-based HEMTs where the induced charge can significantly influence the electronic properties of devices. Furthermore, in quantum well structures the piezoelectric polarisation can have a significant effect on the band profile through the electric fields induced.

7.6 INDUCED PIEZOELECTRIC FIELDS IN QUANTUM WELLS

Consider a single quantum well structure based on piezoelectric active material. The different piezoelectric and strain properties of the well and the barrier materials will result in different polarisations. Hence, at the interfaces a gradient of the piezoelectric polarisation appears which induces a fixed charge which gathers in the vicinity of the interface. The charge density is given by the equation:

$$\rho_p = -\nabla \vec{P} \quad (7.56)$$

Constant strain across the epitaxial layer implies a constant polarisation as well (see equation (7.55)). Hence, the change of the piezoelectric polarisation is abrupt and the induced charge can be found as the difference between the polarisations in the adjacent layers. Furthermore, this accumulated charge at the interfaces induces an electric field in order to satisfy Gauss's law.

Consider now the general case of the multilayer structure in Fig. 7.10 in order to estimate the intrinsic electric fields in the layers. The electrostatic boundary conditions

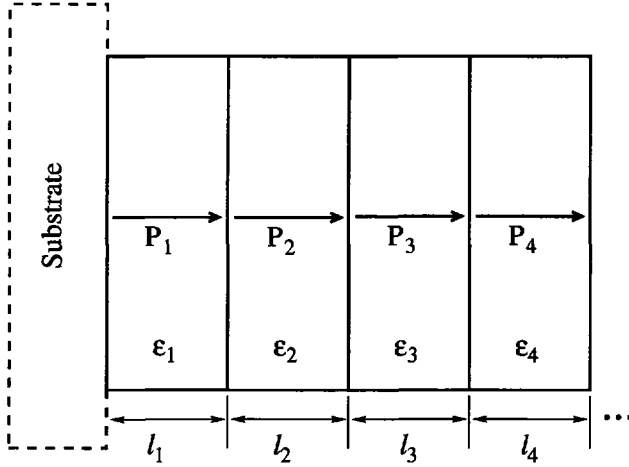


Figure 7.10 Schematic diagram of a multilayer structure based on piezoelectric material giving the notation used in text.

of the displacement vector (\vec{D}) at the adjacent interfaces are applied as:

$$\vec{D}_i = \vec{D}_{i+1} \quad (7.57)$$

where linear material properties[†] are assumed, i.e. $\vec{D}_i = \epsilon_i \vec{E}_i + \vec{P}_i$. Though the piezoelectric polarisation vector can have an arbitrary direction only the component along the growth direction will influence the electronic properties of a quantum well structure and therefore only this component will be considered here. For an arbitrary n -layer structure equation (7.57) gives a system of $n - 1$ linear equations:

$$\begin{aligned} \epsilon_1 F_1 + P_1 &= \epsilon_2 F_2 + P_2 \\ \epsilon_2 F_2 + P_2 &= \epsilon_3 F_3 + P_3 \\ &\vdots \\ \epsilon_{i-1} F_{i-1} + P_{i-1} &= \epsilon_i F_i + P_i \\ &\vdots \\ \epsilon_{n-1} F_{n-1} + P_{n-1} &= \epsilon_n F_n + P_n \end{aligned} \quad (7.58)$$

which can be solved provided appropriate boundary conditions across the structure are defined. Though there are no constraints on the choice of the boundary condition, the natural choice are the 'hard wall' boundary conditions, which enforce the total

[†] ϵ is used to represent the permittivity in this chapter to distinguish it from ϵ which is the commonly accepted symbol for the components of strain.

potential drop across the structure to be zero. All other possibilities would imply that the quantum well structure acts as a voltage source in the closed circuit. For a periodic structure (e.g. a superlattice) the hard wall and the periodic boundary conditions are equivalent.

Knowing the boundary condition the necessary n th equation of the system is defined and reads:

$$\sum_{i=1}^n F_i l_i = 0 \quad (7.59)$$

where L_i is the width of the i -th layer.

The solution of the system gives a rather simple equation:

$$F_j = \frac{\sum_k (P_k - P_j) \frac{l_k}{\epsilon_k}}{\epsilon_j \sum_k \frac{l_k}{\epsilon_k}} \quad (7.60)$$

from which the intrinsic electric field can be easily calculated. It is interesting to see that in the quantum wells the electric field exists even in the layers with zero piezoelectric polarisation, this is a consequence of the charge induced at the interfaces with the adjacent piezoelectric layers.

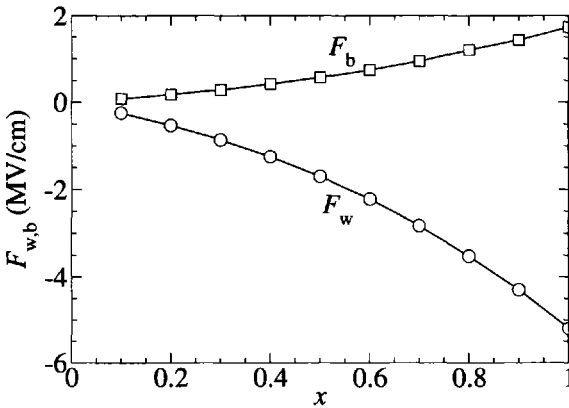


Figure 7.11 The induced piezoelectric field in a GaN well and $\text{Al}_x\text{Ga}_{1-x}\text{N}$ barrier layers grown on a GaN substrate as a function of the barrier layer composition x .

In the case of wurtzite GaN-based quantum wells, the piezoelectric field due to the large strain and piezoelectric constants, can be stronger than 1 MV/cm [156]. Fig. 7.11

shows such an example, where the piezoelectric field in the well and barrier layers of a single GaN/AlGaIn quantum well structure grown on a GaN substrate is calculated as a function of the barrier layer composition. The electric field in an unstrained 20 Å wide layer is as large as 5 MV/cm while in a thicker 60 Å strained AlGaIn layer it is proportionally smaller but still around 1 MV/cm. The piezoelectric field in GaN alloys is an order of magnitude larger than in another widely used material—cubic InGaAs which exhibits piezoelectric fields of around ~ 100 kV/cm [157].

7.7 EFFECT OF PIEZOELECTRIC FIELDS ON QUANTUM WELLS

As shown in the previous section, the piezoelectric effect manifests itself as an electric field in the layers of the quantum well structure. To explore the influence of such fields on the band profile and the electronic structure, the simplest case of a single quantum well will be considered. Also, without loss of generality a uniform dielectric constant can be assumed across the structure. Hence, the magnitude of the piezoelectric fields in the quantum well, assuming identical barriers, follows from equation (7.60) as:

$$F_{w,b} = \frac{(P_{b,w} - P_{w,b}) l_{b,w}}{\epsilon(l_w + l_b)} \quad (7.61)$$

where $P_{w,b}$ are the polarisation magnitudes in the well and the barrier respectively and ϵ is the constant permittivity. The electric fields in the well and the barrier layer must be of opposite sign as the overall potential drop across the quantum well has to be zero. This can be expressed analytically using equation (7.59) as:

$$F_w l_w + F_b l_b = 0 \quad (7.62)$$

This implies that the distribution of the electric field across the structure is proportional to the ratio of the layer widths. Consequently, the thinner epitaxial layer will have a higher electric field than that of the thicker.

The piezoelectric field alters the quantum well band profile which can be represented as a step-linear potential (with respect to the coordinate system as in Fig. 7.12):

$$V(z) = \begin{cases} eF_b z + V_b, & \text{for } z < 0 \\ eF_w z, & \text{for } 0 < z < l_w \\ eF_b z + F_w l_w + V_b & \text{for } z > l_w \end{cases} \quad (7.63)$$

where l_w is the well width and V_b is the barrier height. The previous potential definition is equivalent to a biased quantum well structure and the solutions can be found as a linear combination of Airy's functions as discussed in Chapter 2.

Another important effect is that the piezoelectric field can be screened by free electrons. The linear potential causes electrons to gather close to the interfaces, therefore inducing space-charges. If the carrier density in the quantum well is comparable to the density of the piezoelectric induced charges the electrostatic field due to the free electrons can suppress the piezoelectric field. However, as the induced charge at the

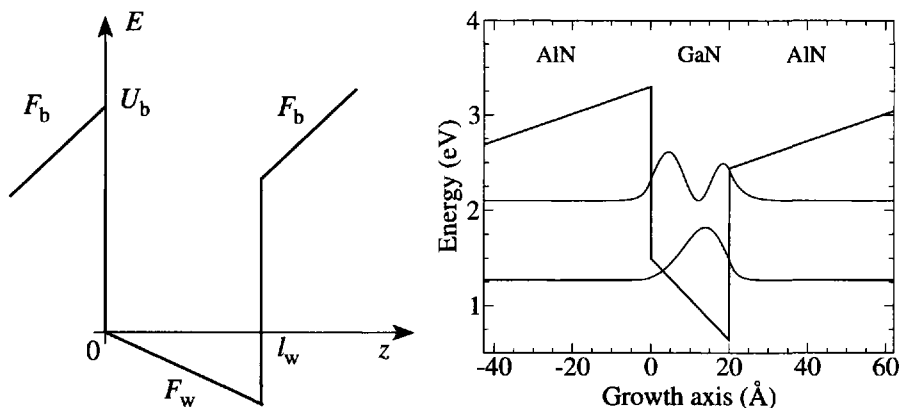


Figure 7.12 On the left hand side is a schematic diagram of typical induced electric fields in a quantum well based on piezoelectric materials and on the right hand side is the band profile of a GaN/AlN quantum well structure with wavefunction moduli of the first two bound states.

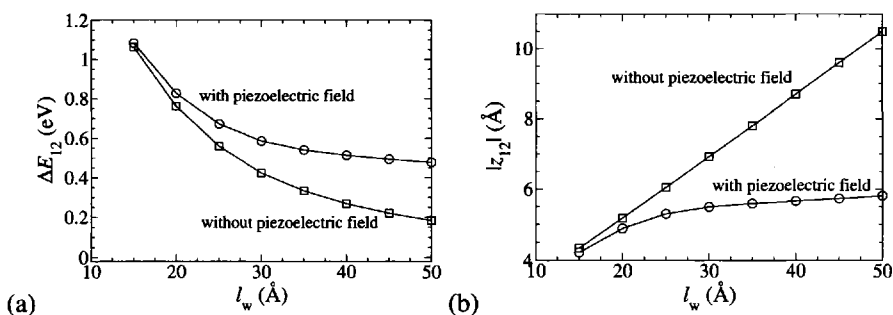


Figure 7.13 Illustration of the influence of the piezoelectric field on the electronic and optical properties of GaN/AlN quantum wells, (a) the energy difference and (b) the dipole matrix element, between the first two bound states with and without the piezoelectric field.

interfaces is usually much larger than the quantum well doping limit this screening effect is generally not pronounced in applications.

The change of the band profile due to a piezoelectric field can have a strong influence on the energies of intersubband transitions. This is illustrated in Fig. 7.13(a) which shows a comparison of the transition energies between the lowest two subbands in GaN/AlGaIn quantum wells with and without the internal electric field included. A substantial increase (blue shift) of the intersubband transition energies is clearly observed and has been observed experimentally [158].

The piezoelectric field can also considerably modify the localisation properties of the wavefunctions, increasing or decreasing the overlap between the ground and other excited states. This can reflect on the dipole matrix element (defined as $z_{ij} = \langle \Psi_i | z | \Psi_j \rangle$) and consequently on the intersubband optical properties of quantum well structures such as the intersubband absorption ($A_{ij} \sim |z_{ij}|^2$). For GaN/AlGaIn quantum wells the relative change in the dipole matrix element z_{12} is shown in Fig. 7.13(b). Clearly, the estimated decrease of up to 50% in z_{ij} is enough to significantly deteriorate the performance of a device in a potential application. A similar problem is present in quantum well lasers based on interband transitions between the conduction and the valence bands (see [159]).

The previously mentioned effects on optical transition energies and absorption strengths highlight the importance of a thorough understanding of the piezoelectric field properties for the modelling and design of optoelectronic devices.

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CHAPTER 8

QUANTUM WIRES AND DOTS

8.1 FURTHER CONFINEMENT

It has already been shown that the reduction in dimensionality produced by confining electrons (or holes) to a thin semiconductor layer leads to a dramatic change in their behaviour. This principle can be developed by further reducing the dimensionality of the electron's environment from a two-dimensional quantum well to a one-dimensional quantum wire and eventually to a zero-dimensional quantum dot. In this context, of course, the dimensionality refers to the number of degrees of freedom in the electron momentum; in fact, within a quantum wire, the electron is confined across two directions, rather than just the one in a quantum well, and, so, therefore, reducing the degrees of freedom to one. In a quantum dot, the electron is confined in all three-dimensions, thus reducing the degrees of freedom to zero. If the number of degrees of freedom are labelled as \mathcal{D}_f and the number of directions of confinement are labelled as \mathcal{D}_c , then clearly:

$$\mathcal{D}_f + \mathcal{D}_c = 3 \quad (8.1)$$

for all solid state systems. These values are highlighted for the four possibilities shown in Table 8.1. Tradition has determined that the reduced-dimensionality systems are labelled by the remaining degrees of freedom in the electron motion, i.e. \mathcal{D}_f , rather than the number of directions with confinement \mathcal{D}_c .

Table 8.1 The number of degrees of freedom \mathcal{D}_f in the electron motion, together with the extent of the confinement \mathcal{D}_c , for the four basic dimensionality systems

System	\mathcal{D}_c	\mathcal{D}_f
Bulk	0	3
Quantum well	1	2
Quantum wire	2	1
Quantum dot	3	0

Fig. 8.1 gives a simple outline of how quantum wires *might* be fabricated, although note that there is more than one method and the interested reader should refer to a specialist growth treatise for further details. A standard quantum well layer can be patterned with photolithography or perhaps electron-beam lithography, and etched to leave a free standing strip of quantum well material; the latter may or may not be filled in with an overgrowth of the barrier material (in this case, $\text{Ga}_{1-x}\text{Al}_x\text{As}$). Any charge carriers are still confined along the heterostructure growth (z -) axis, as they were in the quantum well, but in addition (provided the strip is narrow enough) they are now confined along an additional direction, either the x - or the y -axis, depending on the lithography.

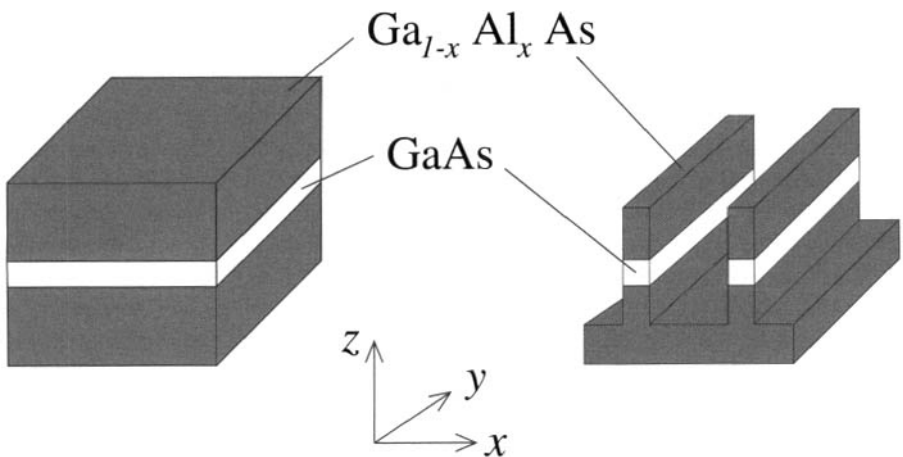


Figure 8.1 Fabrication of quantum wires

Fig. 8.2 shows an expanded view of a single quantum wire, where clearly the electron (or hole) is free to move in only one direction, in this case along the y -axis. Within the effective-mass approximation the motion along the axis of the wire can still be described by a parabolic dispersion, i.e.

$$E = \frac{\hbar^2 k^2}{2m^*} \quad (8.2)$$

just as in bulk and for the in-plane motion within a quantum well.

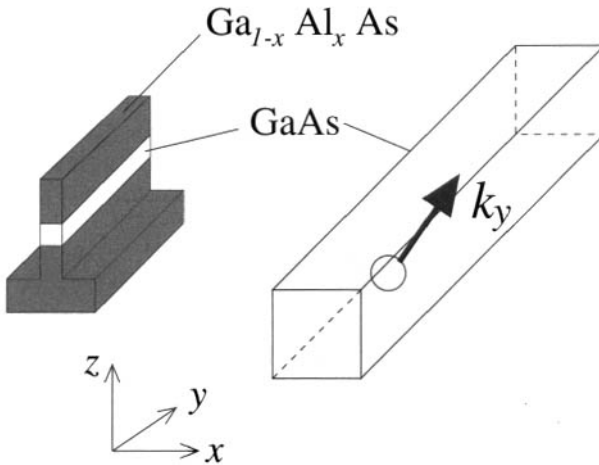


Figure 8.2 A single wire and an expanded view showing schematically the single degree of freedom in the electron momentum

Another class of quantum wire can be formed by patterning the substrate *before* growth. This leads to the formation of so-called *V-grooved quantum wires*, see e.g. [7], p. 35; the solution of these has been dealt with by Gangopadhyay and Nag [160], and will also be touched upon later in Section 13.4.

Quantum dots (see [8] for an introduction to their applications) *can* again be formed by further lithography and etching, e.g. if a quantum well sample is etched to leave pillars rather than wires, then a charge carrier can become confined in all three dimensions, as illustrated in Fig. 8.3.

Under certain growth conditions, when a thin layer of a semiconductor is grown on top of a substrate which has a quite different lattice constant, then in an attempt to minimize the total strain energy between the bonds, the thin layer spontaneously orders, or *self-assembles* into quantum dots. Microscopy has shown the dots to take the shape of pyramids, or square based ‘tetrahedron’ [161, 162].

In the following sections example solutions will be derived for some of the simpler geometries; some of the more complex ones will have to wait until Chapter 13.

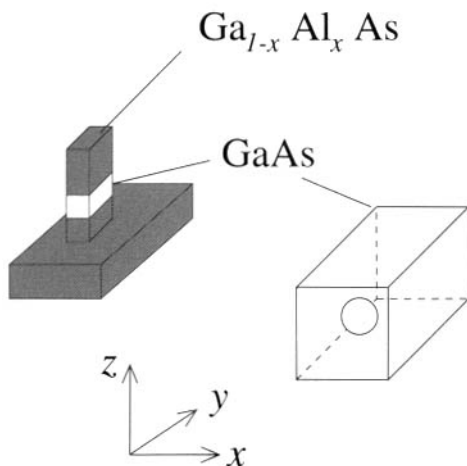


Figure 8.3 A single free standing pillar containing a quantum dot and an expanded view which shows schematically the removal of all degrees of freedom for the electron momentum

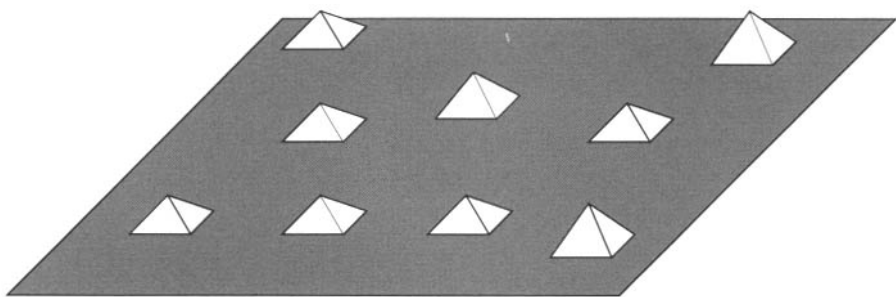


Figure 8.4 Schematic representation of the pyramidal shape of self-assembled quantum dots in highly lattice mismatched systems

8.2 SCHRÖDINGER'S EQUATION IN QUANTUM WIRES

The general three-dimensional Schrödinger equation for constant effective mass is:

$$-\frac{\hbar^2}{2m^*} \nabla^2 \psi(x, y, z) + V(x, y, z) \psi(x, y, z) = E \psi(x, y, z) \quad (8.3)$$

In analogy to the in-plane dispersion discussed in Section 2.2, in a quantum wire it is possible to decouple the motion along the length of the wire. Taking the axis of the wire along x , then the total potential $V(x, y, z)$ can always be written as the sum of a two-dimensional confinement potential plus the potential along the wire (which happens to be zero in this case), i.e.

$$V(x, y, z) = V(x) + V(y, z) \quad (8.4)$$

The eigenfunction can then be written as a product of two components:

$$\psi(x, y, z) = \psi(x)\psi(y, z) \quad (8.5)$$

Substituting both equation (8.4) and equation (8.5) into equation (8.3), then:

$$\begin{aligned} -\frac{\hbar^2}{2m^*} \left(\frac{\partial^2}{\partial x^2} + \frac{\partial^2}{\partial y^2} + \frac{\partial^2}{\partial z^2} \right) \psi(x)\psi(y, z) + (V(x) + V(y, z)) \psi(x)\psi(y, z) \\ = E\psi(x)\psi(y, z) \end{aligned} \quad (8.6)$$

Writing the energy as a sum of terms associated with the two components of the motion, then:

$$\begin{aligned} -\frac{\hbar^2}{2m^*} \left[\psi(y, z) \frac{\partial^2 \psi(x)}{\partial x^2} + \psi(x) \frac{\partial^2 \psi(y, z)}{\partial y^2} + \psi(x) \frac{\partial^2 \psi(y, z)}{\partial z^2} \right] \\ + \psi(y, z)V(x)\psi(x) + \psi(x)V(y, z)\psi(y, z) = (E_x + E_{y,z})\psi(x)\psi(y, z) \end{aligned} \quad (8.7)$$

It is now possible to associate distinct kinetic and potential energies on the left-hand side of equation (8.7), with the components E_x and $E_{y,z}$ on the right-hand side, thus giving two *decoupled* equations, as follows:

$$-\frac{\hbar^2}{2m^*} \psi(y, z) \frac{\partial^2 \psi(x)}{\partial x^2} + \psi(y, z)V(x)\psi(x) = \psi(y, z)E_x\psi(x) \quad (8.8)$$

$$\begin{aligned} -\frac{\hbar^2}{2m^*} \left[\psi(x) \frac{\partial^2 \psi(y, z)}{\partial y^2} + \psi(x) \frac{\partial^2 \psi(y, z)}{\partial z^2} \right] \\ + \psi(x)V(y, z)\psi(y, z) = \psi(x)E_{y,z}\psi(y, z) \end{aligned} \quad (8.9)$$

In the above $\psi(y, z)$ is not acted upon by any operator in the first equation, and similarly for $\psi(x)$ in the second equation, and thus they can be divided out. In addition, as mentioned above, the potential component along the axis of the wire $V(x) = 0$, thus giving the final decoupled equations of motion as follows:

$$-\frac{\hbar^2}{2m^*} \frac{\partial^2 \psi(x)}{\partial x^2} = E_x\psi(x) \quad (8.10)$$

$$-\frac{\hbar^2}{2m^*} \left[\frac{\partial^2 \psi(y, z)}{\partial y^2} + \frac{\partial^2 \psi(y, z)}{\partial z^2} \right] + V(y, z)\psi(y, z) = E_{y,z}\psi(y, z) \quad (8.11)$$

Clearly, the first of these equations is satisfied by a plane wave of the form $\exp(ik_x x)$, thus giving the standard dispersion relationship:

$$E_x = \frac{\hbar^2 k_x^2}{2m^*} \quad (8.12)$$

The second of these equations of motion, equation (8.11), is merely the Schrödinger equation for the two-dimensional confinement potential characterising a quantum wire. For a general cross-sectional wire, equation (8.11) should really be solved by using a full two-dimensional solution, which lies beyond the scope of this present work. For the purpose here, special cases of the solution of equation (8.11), for the relevant commonly found geometries, will be illustrated. Such particular solutions rely upon the ability to further decouple the motion into independent components.

8.3 INFINITELY DEEP RECTANGULAR WIRES

Perhaps the simplest quantum wire geometry would be a rectangular cross-section surrounded by infinite barriers. This is illustrated schematically in Fig. 8.5 and can be considered to be the two-dimensional analogy to the one-dimensional confinement potential of the standard infinitely deep quantum well.

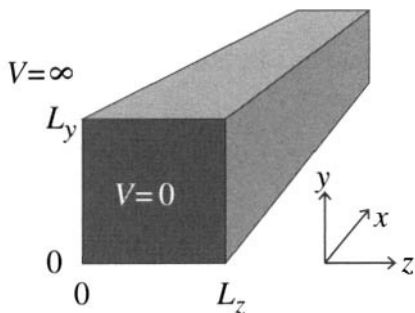


Figure 8.5 The infinitely deep rectangular cross-section quantum wire

Within the quantum wire, the potential is zero, while outside the wire it is infinite; thus in the latter case the wave function is zero. Hence, the Schrödinger equation is only defined within the wire for the motion in the two confined y - and z -directions, i.e. equation (8.11) becomes:

$$-\frac{\hbar^2}{2m^*} \left[\frac{\partial^2 \psi(y, z)}{\partial y^2} + \frac{\partial^2 \psi(y, z)}{\partial z^2} \right] = E_{y,z} \psi(y, z) \quad (8.13)$$

The form of the potential in this Schrödinger equation (i.e. zero!) allows the motions to be decoupled further, writing:

$$\psi(y, z) = \psi(y)\psi(z) \quad (8.14)$$

and then:

$$-\frac{\hbar^2}{2m^*} \psi(z) \frac{\partial^2 \psi(y)}{\partial y^2} - \frac{\hbar^2}{2m^*} \psi(y) \frac{\partial^2 \psi(z)}{\partial z^2} = E_{y,z} \psi(y)\psi(z) \quad (8.15)$$

Again, it is possible to associate the individual kinetic energy terms on the left-hand side of equation (8.15) with separate energy components, i.e. by writing $E_{y,z} = E_y + E_z$, then:

$$-\frac{\hbar^2}{2m^*} \psi(z) \frac{\partial^2 \psi(y)}{\partial y^2} - \frac{\hbar^2}{2m^*} \psi(y) \frac{\partial^2 \psi(z)}{\partial z^2} = \psi(z) E_y \psi(y) + \psi(y) E_z \psi(z) \quad (8.16)$$

The decoupling is completed with the following:

$$-\frac{\hbar^2}{2m^*} \psi(z) \frac{\partial^2 \psi(y)}{\partial y^2} = \psi(z) E_y \psi(y) \quad (8.17)$$

$$-\frac{\hbar^2}{2m^*}\psi(y)\frac{\partial^2\psi(z)}{\partial z^2} = \psi(y)E_z\psi(z) \quad (8.18)$$

Dividing the first of this pair of equations by $\psi(z)$ and the second by $\psi(y)$ gives:

$$-\frac{\hbar^2}{2m^*}\frac{\partial^2\psi(y)}{\partial y^2} = E_y\psi(y) \quad (8.19)$$

$$-\frac{\hbar^2}{2m^*}\frac{\partial^2\psi(z)}{\partial z^2} = E_z\psi(z) \quad (8.20)$$

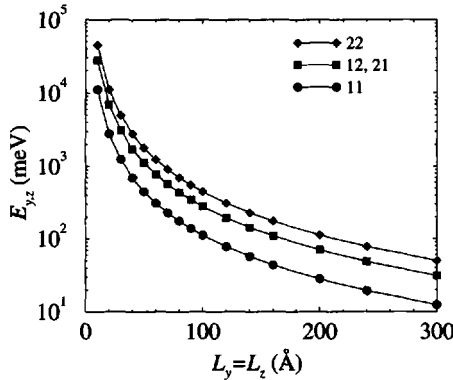


Figure 8.6 The confinement energy in an infinitely deep quantum wire with a square cross-section

Given that the potential outside the wire is infinite, then the standard boundary condition of continuity in the wave function implies that *both* $\psi(y)$ and $\psi(z)$ are zero at the edges of the wire. Thus, equations (8.19) and (8.20) are identical to those of the one-dimensional infinitely deep quantum well—the two-dimensional Schrödinger equation has been decoupled into two one-dimensional equations. Given the origin in a ‘corner’, and the wire dimensions L_y and L_z , as in Fig. 8.5, then the solutions follow as:

$$\psi(y) = \sqrt{\frac{2}{L_y}} \sin\left(\frac{\pi n_y y}{L_y}\right) \quad (8.21)$$

and

$$\psi(z) = \sqrt{\frac{2}{L_z}} \sin\left(\frac{\pi n_z z}{L_z}\right) \quad (8.22)$$

which give the components of energy as:

$$E_y = \frac{\hbar^2 \pi^2 n_y^2}{2m^* L_y^2} \quad (8.23)$$

and

$$E_z = \frac{\hbar^2 \pi^2 n_z^2}{2m^* L_z^2} \quad (8.24)$$

Thus, the total energy due to confinement, $E_{y,z} = E_y + E_z$, is:

$$E_{y,z} = \frac{\hbar^2 \pi^2}{2m^*} \left(\frac{n_y^2}{L_y^2} + \frac{n_z^2}{L_z^2} \right) \quad (8.25)$$

The confined states of a quantum wire are therefore described by the two principal quantum numbers n_y and n_z , in contrast to the sole number required for the one-dimensional confinement potential in quantum wells.

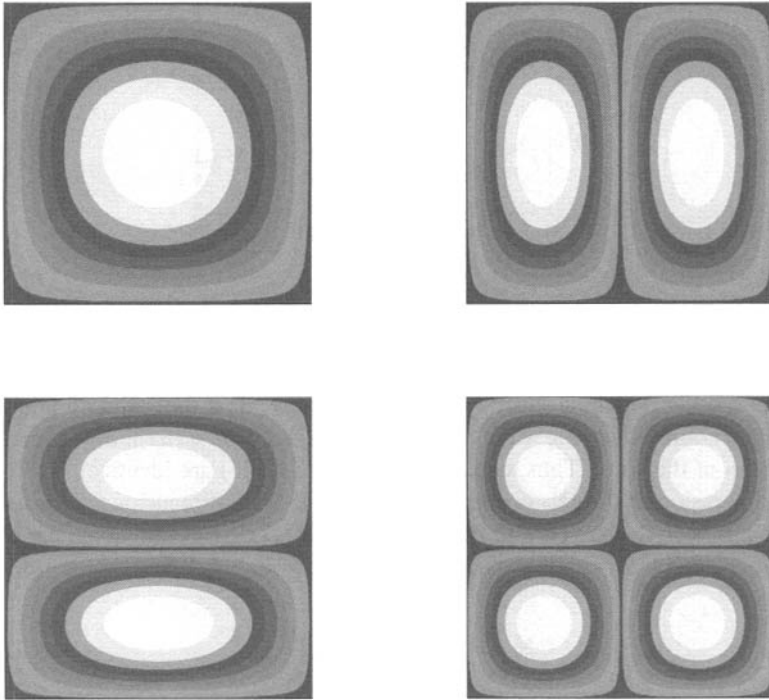


Figure 8.7 The charge densities of the four lowest energy confined states within a infinitely deep quantum wire; (top left) $n_y=1, n_z=1$; (top right) $n_y=1, n_z=2$; (bottom left) $n_y=2, n_z=1$; (bottom right) $n_y=2, n_z=2$

Fig. 8.6 displays the confinement energies $E_{y,z}$ for (n_y, n_z) equal to (1,1), (1,2), (2,1), and (2,2) as a function of the side length $L_y = L_z$ for a *square* cross-section infinitely deep quantum wire. In this case of a square cross-sectional wire, the confinement energies of the (1,2) and (2,1) states are equal; however clearly this will not be the case for a rectangular cross-section wire, which has $L_y \neq L_z$. Just as in the

quantum wells met previously, the confinement energy decreases as the size of the system increases.

The wave function for a stationary (wave function along the length of the wire, independent of position) confined particle within the wire is real, and hence the charge density is simply $[\psi(y)\psi(z)]^2$. This is plotted in Fig. 8.7 for the four lowest confined states over a cross-sectional plane. The spatial distribution of the charge density is dependent upon the principal quantum numbers n_y and n_z (as would be expected), and the number of anti-nodes (local maxima) is equal to $n_y n_z$.

8.4 SIMPLE APPROXIMATION TO A FINITE RECTANGULAR WIRE

More relevant to real devices would be a rectangular cross-sectional quantum wire, but with finite height barriers, as would be fabricated with a post-etch overgrowth. Fig. 8.8 (left) illustrates the two-dimensional confinement potential $V(y, z)$ for this system. With this configuration it is not possible to write the potential $V(y, z)$ as a sum of two independent potentials $V(y)$ and $V(z)$, and thus it is not possible to separate the y - and z -motions.

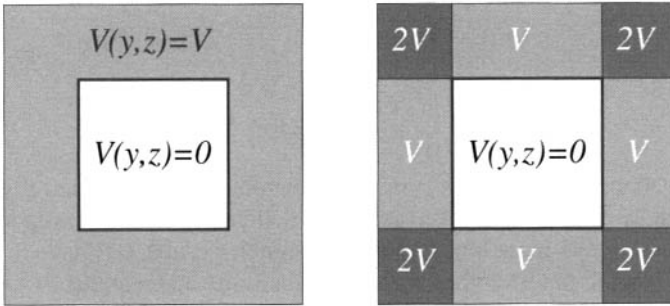


Figure 8.8 The rectangular cross-section quantum wire with finite barriers (left) and an approximate form for the potential (right), suitable for decoupling the motion

However, a very loose approximation may be to write the potential as in Fig. 8.8 (right). With this form, $V(y, z)$ does equal $V(y) + V(z)$, where $V(y)$ and $V(z)$ are independent finite well potentials, as in Sections 2.5 and 2.6. The approximation occurs in the 'corner regions' outside of the wire where the two quantum well potential barrier heights V sum to give $2V$. This is in areas which are not expected to be sampled too much by the eigenfunctions, particularly those in wide wires and the lower energy states.

So proceeding, the Schrödinger equation for the y - z cross-sectional confined motion in a general quantum wire is given by equation (8.11), i.e.

$$-\frac{\hbar^2}{2m^*} \left[\frac{\partial^2 \psi(y, z)}{\partial y^2} + \frac{\partial^2 \psi(y, z)}{\partial z^2} \right] + V(y, z) \psi(y, z) = E_{y,z} \psi(y, z) \quad (8.26)$$

Therefore, writing the potential $V(y, z) = V(y) + V(z)$ and the wave function $\psi(y, z)$ as $\psi(y)\psi(z)$, gives:

$$-\frac{\hbar^2}{2m^*}\psi(z)\frac{\partial^2\psi(y)}{\partial y^2} - \frac{\hbar^2}{2m^*}\psi(y)\frac{\partial^2\psi(z)}{\partial z^2} + [V(y) + V(z)]\psi(y)\psi(z) = E_{y,z}\psi(y)\psi(z) \quad (8.27)$$

Again, by writing the energy $E_{y,z}$ as a sum of two components associated with the y - and z -motions, equation (8.27) can then be split into two, giving:

$$-\frac{\hbar^2}{2m^*}\psi(z)\frac{\partial^2\psi(y)}{\partial y^2} + \psi(z)V(y)\psi(y) = \psi(z)E_y\psi(y) \quad (8.28)$$

and

$$-\frac{\hbar^2}{2m^*}\psi(y)\frac{\partial^2\psi(z)}{\partial z^2} + \psi(y)V(z)\psi(z) = \psi(y)E_z\psi(z) \quad (8.29)$$

Dividing the first of these equations by $\psi(z)$, and the second by $\psi(y)$, gives familiar equations for straightforward one-dimensional potentials, i.e.

$$-\frac{\hbar^2}{2m^*}\frac{\partial^2\psi(y)}{\partial y^2} + V(y)\psi(y) = E_y\psi(y) \quad (8.30)$$

and

$$-\frac{\hbar^2}{2m^*}\frac{\partial^2\psi(z)}{\partial z^2} + V(z)\psi(z) = E_z\psi(z) \quad (8.31)$$

In this derivation, the kinetic energy operator appropriate for a constant effective mass has been employed, but equally well as there is nothing which depends upon this form, the variable effective mass kinetic energy operator, i.e. $(\partial/\partial z)(1/m^*(z))(\partial/\partial z)$, could have been used. The final outcome is the same, namely that as the independent potentials $V(y)$ and $V(z)$ are simply those of a finite quantum well, then the solutions for both the wave functions and the confinement energies follow as before (Sections 2.5 and 2.6).

Fig. 8.9 shows the equivalent of Fig. 8.6 but with finite barriers, corresponding to Al concentrations of 20 and 40 % in $\text{Ga}_{1-x}\text{Al}_x\text{As}$. The solutions were derived by combining the eigenvalues from two independent one-dimensional quantum well calculations, including the effective mass mismatch at the well (in this case wire) interface. The eigenvalues labelled '1,1' correspond to the ground state of the y -motion combined with the ground state of the z -motion. The others represent the combinations formed between the lowest two energy states in each direction. The behaviour of the energy with the length of the wire side is qualitatively similar to the infinite barrier case, i.e. there is degeneracy between the 1,2 and the 2,1 solutions, the ordering of the energy levels is the same and the eigenvalues decrease with increasing wire side.

The corresponding charge densities $[\psi(y, z)\psi(y, z)]$ are plotted in Fig. 8.10 for a cross-section of the wire for these same four eigenstates. Again the behaviour is

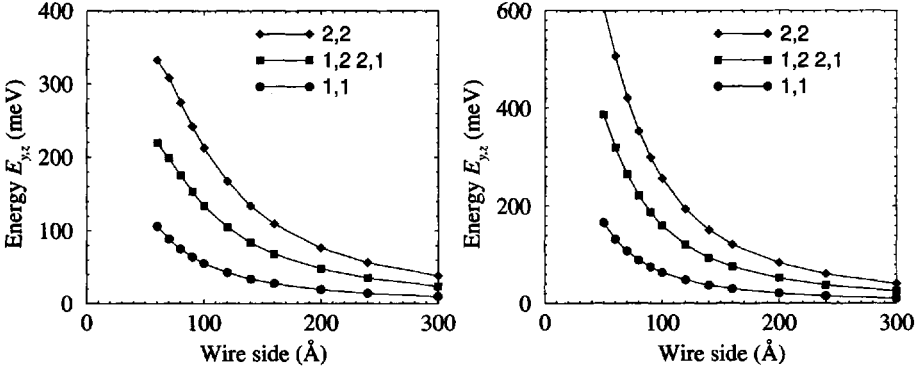


Figure 8.9 The confinement energy, $E_{y,z} = E_y + E_z$, in a square cross-section GaAs quantum wire surrounded by finite height barriers of $\text{Ga}_{0.8}\text{Al}_{0.2}\text{As}$ (left) and $\text{Ga}_{0.6}\text{Al}_{0.4}\text{As}$ (right)

similar to the infinitely deep wire case, with the same distribution of maxima and minima. However, the main difference is, of course, that the finite barrier height allows for a significant ‘leakage’ of the wave function into the surrounding material. It is this interaction with the ‘potential pillars’ located outside of the wire at each corner, which originate from the approximated potential in Fig. 8.8, which limits the applicability or accuracy of this simple model.

For this relatively narrow ($100 \text{ Å} \times 100 \text{ Å}$) wire, the scale next to the ground-state charge density in Fig. 8.10 (top left), implies that, roughly speaking, probably around 80–90 % of the charge is confined within the wire. Thus the effect of the approximate potential outside of the wire will not be too substantial. However for the higher-energy states more of the ‘lighter shade’ is outside the wire, and thus it would be expected that this approximation would be worse. The same effect would be true when reducing the length of the wire side.

This simple approach to the finite quantum wire is just a way of understanding the basic physics and being able to predict *qualitatively* how the electronic properties change with the system parameters. For a more complete description, a full two-dimensional Schrödinger solution may be required (see [160, 163]), or alternatively the empirical pseudopotential approach (see Chapter 13) may be employed.

The energy eigenvalues $E_{y,z}$ could be improved by considering a perturbation on the two-dimensional system which removed the ‘ $2V$ ’ potential pillars. Using first-order perturbation theory, the change in energy of a level would be given by:

$$\Delta E = \langle \psi(y, z) | V'(y, z) | \psi(y, z) \rangle \quad (8.32)$$

i.e.

$$\Delta E = \int_{-\infty}^{+\infty} \int_{-\infty}^{+\infty} \psi^*(y) \psi^*(z) V'(y, z) \psi(y) \psi(z) \, dy \, dz \quad (8.33)$$

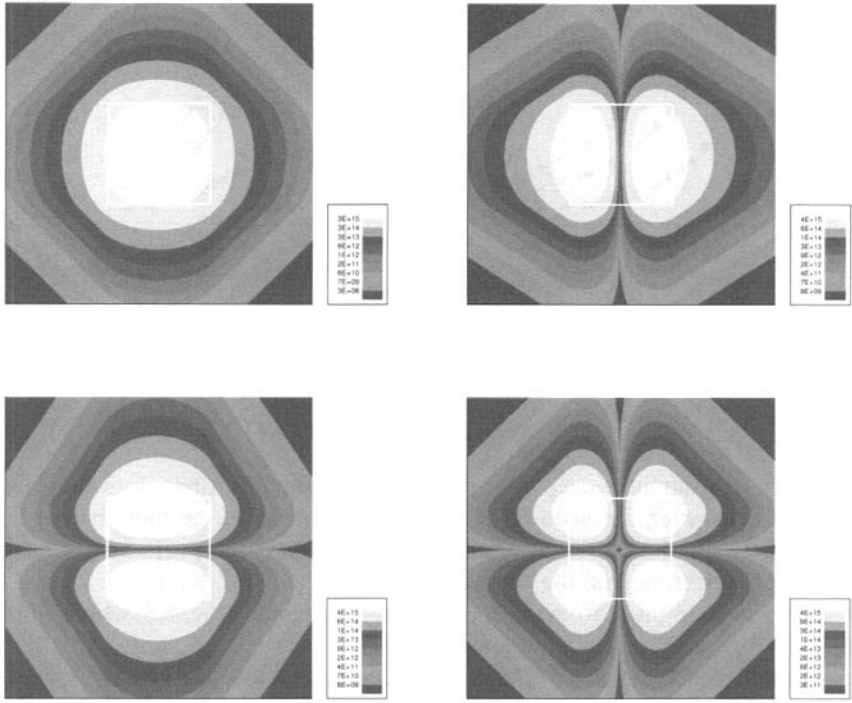


Figure 8.10 The charge densities of the four lowest energy confined states of a GaAs quantum wire of side 100 \AA with finite $\text{Ga}_{0.8}\text{Al}_{0.2}\text{As}$ barriers, (top left) $n_y=1, n_z=1$; (top right) $n_y=1, n_z=2$; (bottom left) $n_y=2, n_z=1$; (bottom right) $n_y=2, n_z=2$; the edges of the wire are indicated by the boxes

The perturbation to the potential $V'(y, z)$ would be negative and of magnitude V , and employing the four fold symmetry of this *square* cross-sectional wire, would give:

$$\Delta E = -4V \int_{L_y}^{+\infty} \psi^*(y)\psi(y) \, dy \int_{L_z}^{+\infty} \psi^*(z)\psi(z) \, dz \quad (8.34)$$

which is relatively straightforward to evaluate. Califano and Harrison [164] have demonstrated that this can be a quite useful approach to the solution of finite barrier quantum wires and dots.

8.5 CIRCULAR CROSS-SECTION WIRE

Consider again the Schrödinger equation for the motion in the confined cross-sectional plane of a quantum wire, as given earlier in equation (8.11), i.e.

$$-\frac{\hbar^2}{2m^*} \left[\frac{\partial^2 \psi(y, z)}{\partial y^2} + \frac{\partial^2 \psi(y, z)}{\partial z^2} \right] + V(y, z)\psi(y, z) = E_{y,z}\psi(y, z) \quad (8.35)$$

Given the cylindrical symmetry of the quantum wire, as shown in Fig. 8.11, it would

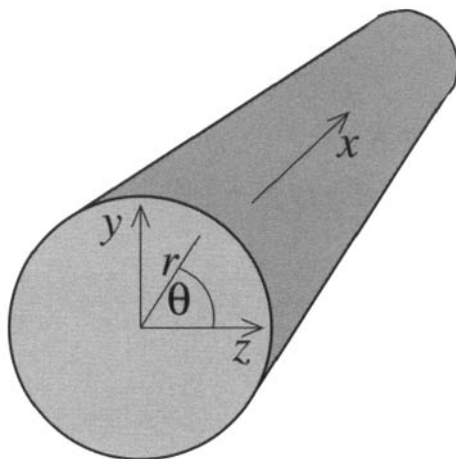


Figure 8.11 Schematic illustration of a circular cross-section quantum wire

seem advantageous to move into polar coordinates for the cross-sectional motion. With the definitions of the modulus r and angle θ as in the figure, the Cartesian coordinates then follow as:

$$y = r \sin \theta \quad \text{and} \quad z = r \cos \theta \quad (8.36)$$

$$\therefore r = \sqrt{y^2 + z^2} \quad (8.37)$$

The wave function $\psi(y, z)$ can clearly be written in terms of the new variables r and θ ; however, given the circular symmetry the wave functions should not have a dependence on the angle θ . Thus, the wave function can actually be written as $\psi(r)$, and the Schrödinger equation therefore becomes:

$$-\frac{\hbar^2}{2m^*} \left(\frac{\partial^2}{\partial y^2} + \frac{\partial^2}{\partial z^2} \right) \psi(r) + V(r)\psi(r) = E_r\psi(r) \quad (8.38)$$

where the index on E_r just indicates that this eigenvalue is associated with the confined cross-sectional motion, as opposed to the unconfined motion along the axis of the

wire. In addition, the circular symmetry of the potential which defines the wire can be written explicitly as $V(r)$. Now:

$$\frac{\partial}{\partial y} \psi(r) = \frac{\partial}{\partial r} \psi(r) \times \frac{\partial r}{\partial y} \quad (8.39)$$

Differentiating both sides of equation (8.37) with respect to y , gives:

$$\frac{\partial r}{\partial y} = \frac{1}{2} (y^2 + z^2)^{-\frac{1}{2}} \times 2y = \frac{y}{r} \quad (8.40)$$

Hence:

$$\frac{\partial}{\partial y} \psi(r) = \frac{\partial}{\partial r} \psi(r) \times \frac{y}{r} \quad (8.41)$$

The second derivative is then:

$$\frac{\partial}{\partial y} \frac{\partial}{\partial y} \psi(r) = \frac{\partial}{\partial y} \left[\frac{\partial}{\partial r} \psi(r) \times \frac{y}{r} \right] \quad (8.42)$$

$$\therefore \frac{\partial^2}{\partial y^2} \psi(r) = \frac{\partial^2}{\partial r^2} \psi(r) \times \frac{\partial r}{\partial y} \frac{y}{r} + \frac{\partial}{\partial r} \psi(r) \frac{\partial}{\partial y} \left(\frac{y}{r} \right) \quad (8.43)$$

and thus:

$$\frac{\partial^2}{\partial y^2} \psi(r) = \frac{y^2}{r^2} \frac{\partial^2}{\partial r^2} \psi(r) + \frac{\partial}{\partial r} \psi(r) \left(\frac{1}{r} - \frac{y}{r^2} \frac{\partial r}{\partial y} \right) \quad (8.44)$$

Finally:

$$\frac{\partial^2}{\partial y^2} \psi(r) = \frac{1}{r} \frac{\partial}{\partial r} \psi(r) - \frac{y^2}{r^3} \frac{\partial}{\partial r} \psi(r) + \frac{y^2}{r^2} \frac{\partial^2}{\partial r^2} \psi(r) \quad (8.45)$$

and similarly for z , hence:

$$\left(\frac{\partial^2}{\partial y^2} + \frac{\partial^2}{\partial z^2} \right) \psi(r) = \frac{2}{r} \frac{\partial}{\partial r} \psi(r) - \frac{(y^2 + z^2)}{r^3} \frac{\partial}{\partial r} \psi(r) + \frac{(y^2 + z^2)}{r^2} \frac{\partial^2}{\partial r^2} \psi(r) \quad (8.46)$$

Recalling that $y^2 + z^2 = r^2$, then:

$$\left(\frac{\partial^2}{\partial y^2} + \frac{\partial^2}{\partial z^2} \right) \psi(r) = \frac{1}{r} \frac{\partial}{\partial r} \psi(r) + \frac{\partial^2}{\partial r^2} \psi(r) \quad (8.47)$$

Substituting into equation (8.38) gives the final form for the Schrödinger equation as follows:

$$-\frac{\hbar^2}{2m^*} \left(\frac{1}{r} \frac{\partial}{\partial r} + \frac{\partial^2}{\partial r^2} \right) \psi(r) + V(r) \psi(r) = E_r \psi(r) \quad (8.48)$$

In this case, reliance has been made on the specific form of the kinetic energy operator, unlike the earlier example of the rectangular cross-section quantum wire, and hence this Schrödinger equation is only valid for a constant effective mass.

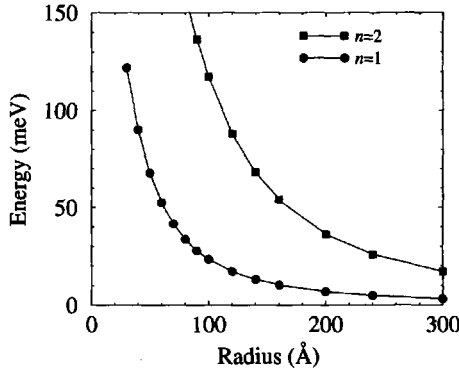


Figure 8.12 The confinement energy in a finite barrier circular cross-section quantum wire

One numerical approach for solving equation (8.48) would be to follow a similar procedure to that employed in Section 3.1, i.e. expand the derivatives in terms of the standard finite differences equivalents:

$$\frac{\partial}{\partial r} \psi(r) = \frac{\psi(r + \delta r) - \psi(r - \delta r)}{2\delta r} \quad (8.49)$$

$$\frac{\partial^2}{\partial r^2} \psi(r) = \frac{\psi(r + \delta r) - 2\psi(r) + \psi(r - \delta r)}{(\delta r)^2} \quad (8.50)$$

and then rearrange to obtain a shooting iterative equation. This is achieved by substituting the above into equation (8.48) to give:

$$\begin{aligned} \frac{1}{r} \left[\frac{\psi(r + \delta r) - \psi(r - \delta r)}{2\delta r} \right] + \frac{\psi(r + \delta r) - 2\psi(r) + \psi(r - \delta r)}{(\delta r)^2} \\ = \frac{2m^*}{\hbar^2} [V(r) - E_r] \psi(r) \end{aligned} \quad (8.51)$$

Multiplying both sides by $2r(\delta r)^2$ gives:

$$\begin{aligned} [\psi(r + \delta r) - \psi(r - \delta r)] \delta r + 2r [\psi(r + \delta r) - 2\psi(r) + \psi(r - \delta r)] \\ = 2r(\delta r)^2 \frac{2m^*}{\hbar^2} [V(r) - E_r] \psi(r) \end{aligned} \quad (8.52)$$

Gathering terms in $\psi(r + \delta r)$, $\psi(r)$, and $\psi(r - \delta r)$, then:

$$\begin{aligned} (2r + \delta r) \psi(r + \delta r) \\ = 2r \left\{ (\delta r)^2 \frac{2m^*}{\hbar^2} [V(r) - E_r] + 2 \right\} \psi(r) + (-2r + \delta r) \psi(r - \delta r) \end{aligned} \quad (8.53)$$

Then finally, obtain:

$$\psi(r + \delta r) = \frac{2r \{2m^* (\delta r / \hbar)^2 [V(r) - E_r] + 2\} \psi(r) + (-2r + \delta r) \psi(r - \delta r)}{2r + \delta r} \quad (8.54)$$

which is an iterative shooting equation, similar to those met earlier in Chapter 3. It can be solved according to the standard boundary condition, $\psi(r) \rightarrow 0$ as $r \rightarrow \infty$, as before. Now in regions of constant potential, the wave functions are in general continuous, and therefore for the particular case of a straight line perpendicular to and through the wire (x -) axis, the wave function must also be continuous. Thus, when crossing the wire axis, the radial component of the wave function $\psi(r)$ must have a derivative of zero, i.e. a local maxima or minima. This allows the iterative starting conditions to be chosen as:

$$\psi(0) = 1 \quad \text{and} \quad \psi(\delta r) = 1 \quad (8.55)$$

These can look a little simplistic, but in the limit of decreasing step length δr , the maxima or minima is clearly flat.

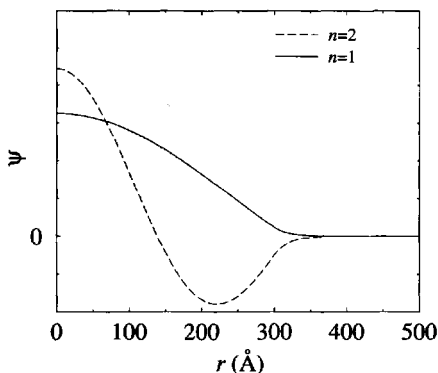


Figure 8.13 The radial component of the wave function $\psi(r)$ for the lowest two eigenstates in a finite-barrier quantum wire with radius 300 Å of circular cross-section

Fig. 8.12 displays the results of calculations of the electron confinement energy versus the wire radius, for a GaAs wire surrounded by $\text{Ga}_{0.8}\text{Al}_{0.2}\text{As}$, for constant effective mass. As expected, the confinement energy decreases with increasing radius and the odd-parity eigenstate is of higher energy than the even. This latter point is highlighted in Fig. 8.13, which plots the radial motion $\psi(r)$ for the 300 Å radius wire. The even- ($n = 1$) and odd- ($n = 2$) parity nature of the eigenstates can clearly be seen.

8.6 QUANTUM BOXES

Cuboid quantum dots, perhaps more specifically designated as *quantum boxes* can be thought of as simply a generalisation of the rectangular cross-section quantum wires, in which there is now additional confinement along the remaining x -axis. This additional confinement removes the remaining degree of freedom in the particle's momentum and localises it in all directions. Thus the energy levels can no longer be referred to as *subbands* and are now known as *sublevels*.

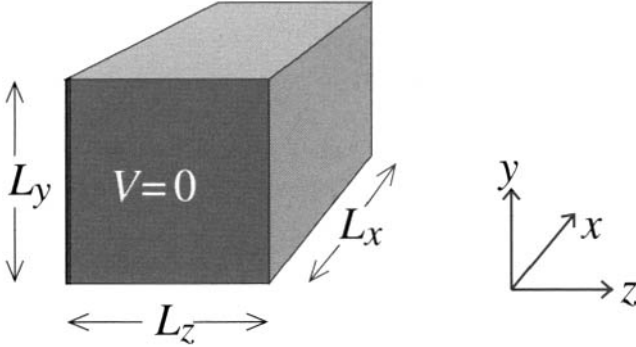


Figure 8.14 Schematic illustration of a quantum box with side L_x , L_y , and L_z

Considering the case of an infinite potential separating the inside of the box from the outside, then the three-dimensional Schrödinger equation within the box is simply:

$$-\frac{\hbar^2}{2m^*} \left(\frac{\partial^2}{\partial x^2} + \frac{\partial^2}{\partial y^2} + \frac{\partial^2}{\partial z^2} \right) \psi(x, y, z) = E_{x,y,z} \psi(x, y, z) \quad (8.56)$$

Again, writing the total energy $E_{x,y,z}$ as a sum of the three terms E_x , E_y , and E_z , then this single three-dimensional equation can be decoupled into three one-dimensional equations:

$$-\frac{\hbar^2}{2m^*} \frac{\partial^2}{\partial x^2} \psi(x) = E_x \psi(x) \quad (8.57)$$

$$-\frac{\hbar^2}{2m^*} \frac{\partial^2}{\partial y^2} \psi(y) = E_y \psi(y) \quad (8.58)$$

$$-\frac{\hbar^2}{2m^*} \frac{\partial^2}{\partial z^2} \psi(z) = E_z \psi(z) \quad (8.59)$$

which for the infinitely deep barrier case, are just the solutions to infinitely deep quantum wells, of widths L_x , L_y , and L_z respectively, i.e. in analogy to the infinitely deep quantum wire confinement energy of equation (8.25), the confinement energy within this quantum box follows as:

$$E_{x,y,z} = \frac{\hbar^2 \pi^2}{2m^*} \left(\frac{n_x^2}{L_x^2} + \frac{n_y^2}{L_y^2} + \frac{n_z^2}{L_z^2} \right) \quad (8.60)$$

The three-dimensional nature of the confinement thus requires three *quantum numbers*, i.e. n_x , n_y , and n_z to label each state.

Finite-barrier quantum boxes could be gleaned from three decoupled one-dimensional quantum well calculations in a similar manner to the finite barrier quantum wire discussed in Section 8.4. However, in this case any perturbative correction would have to account for eight corner-cubes of additional potential ' $2V$ ' and twelve edge-cuboids of additional potential ' V '. Alternatively, a full three-dimensional solution can be constructed by expanding the wave function as a linear combination of infinite well solutions (see Gangopadhyay and Nag [165]).

8.7 SPHERICAL QUANTUM DOTS

It is perhaps easier to deal with a finite barrier quantum dot with spherical rather than cuboid symmetry. The approach is rather similar to that derived earlier for the circular cross-section quantum wire.

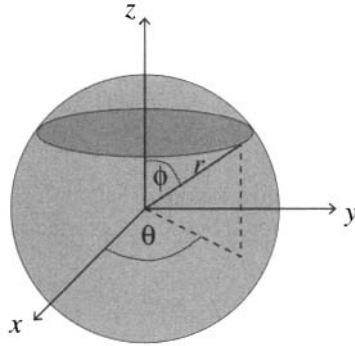


Figure 8.15 Schematic illustration of a spherical quantum dot

Given the spherical symmetry of the potential, then the wave function would also be expected to have spherical symmetry, hence the Schrödinger equation for a *constant* effective mass could be written:

$$-\frac{\hbar^2}{2m^*} \left(\frac{\partial^2}{\partial x^2} + \frac{\partial^2}{\partial y^2} + \frac{\partial^2}{\partial z^2} \right) \psi(r) + V(r)\psi(r) = E_r\psi(r) \quad (8.61)$$

where the index on E_r has been added just to indicate that this energy is associated with the confinement along the radius. In this case:

$$r = \sqrt{x^2 + y^2 + z^2} \quad (8.62)$$

The transition can be made from Cartesian (x, y, z) to spherical polar coordinates, in effect just r , in the same way as detailed in Section 8.5. Using equation (8.45), each

of the three Cartesian axes gives an equation of the following form:

$$\frac{\partial^2}{\partial x^2} \psi(r) = \frac{1}{r} \frac{\partial}{\partial r} \psi(r) - \frac{x^2}{r^3} \frac{\partial}{\partial r} \psi(r) + \frac{x^2}{r^2} \frac{\partial^2}{\partial r^2} \psi(r) \quad (8.63)$$

Therefore, the complete $\nabla^2 \psi(r)$ is given by:

$$\left(\frac{\partial^2}{\partial x^2} + \frac{\partial^2}{\partial y^2} + \frac{\partial^2}{\partial z^2} \right) \psi(r) = \frac{3}{r} \frac{\partial}{\partial r} \psi(r) - \frac{(x^2 + y^2 + z^2)}{r^3} \frac{\partial}{\partial r} \psi(r) + \frac{(x^2 + y^2 + z^2)}{r^2} \frac{\partial^2}{\partial r^2} \psi(r) \quad (8.64)$$

$$\therefore \left(\frac{\partial^2}{\partial x^2} + \frac{\partial^2}{\partial y^2} + \frac{\partial^2}{\partial z^2} \right) \psi(r) = \frac{2}{r} \frac{\partial}{\partial r} \psi(r) + \frac{\partial^2}{\partial r^2} \psi(r) \quad (8.65)$$

(as, for example, in [4], p. 188).

Substituting into the Schrödinger equation then:

$$-\frac{\hbar^2}{2m^*} \left(\frac{2}{r} \frac{\partial}{\partial r} + \frac{\partial^2}{\partial r^2} \right) \psi(r) + V(r) \psi(r) = E_r \psi(r) \quad (8.66)$$

Such spherically symmetric Schrödinger equations have been investigated before (see for example [29], p. 76). As an alternative to such a well established approach, and with the impetus in this work on simple numerical schemes, a shooting technique similar to that described for the circular cross-section quantum wire is sought. With this aim, expanding the first and second derivatives in terms of finite differences gives:

$$\begin{aligned} \frac{2}{r} \left[\frac{\psi(r + \delta r) - \psi(r - \delta r)}{2\delta r} \right] + \frac{\psi(r + \delta r) - 2\psi(r) + \psi(r - \delta r)}{(\delta r)^2} \\ = \frac{2m^*}{\hbar^2} [V(r) - E_r] \psi(r) \end{aligned} \quad (8.67)$$

Multiplying both sides by $r(\delta r)^2$ gives:

$$\begin{aligned} [\psi(r + \delta r) - \psi(r - \delta r)] \delta r + r [\psi(r + \delta r) - 2\psi(r) + \psi(r - \delta r)] \\ = r(\delta r)^2 \frac{2m^*}{\hbar^2} [V(r) - E_r] \psi(r) \end{aligned} \quad (8.68)$$

Gathering terms in $\psi(r + \delta r)$, $\psi(r)$, and $\psi(r - \delta r)$ then:

$$\begin{aligned} (r + \delta r) \psi(r + \delta r) \\ = r \left\{ (\delta r)^2 \frac{2m^*}{\hbar^2} [V(r) - E_r] + 2 \right\} \psi(r) + (-r + \delta r) \psi(r - \delta r) \end{aligned} \quad (8.69)$$

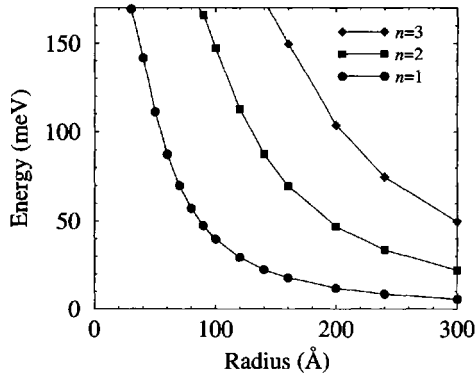


Figure 8.16 The confinement energy in a spherical GaAs quantum dot surrounded by a $\text{Ga}_{0.8}\text{Al}_{0.2}\text{As}$ barrier

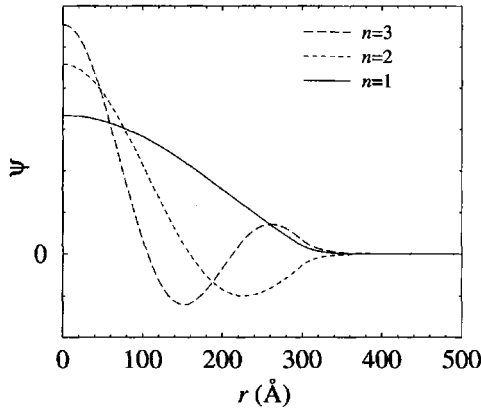


Figure 8.17 The wave functions of the three lowest energy states in the 300 Å spherical quantum dot

and finally:

$$\psi(r + \delta r) = \frac{r \{ 2m^* (\delta r / \hbar)^2 [V(r) - E_r] + 2 \} \psi(r) + (-r + \delta r) \psi(r - \delta r)}{r + \delta r} \quad (8.70)$$

which is again an iterative equation which can be solved with a numerical shooting technique according to the same boundary conditions as discussed in Section 8.5.

Fig. 8.16 shows the results of calculations of the three lowest energy levels of a spherical GaAs quantum dot surrounded by a finite barrier composed of $\text{Ga}_{0.8}\text{Al}_{0.2}\text{As}$, with a *sharp* boundary. In fact, the formalism above, as that of the circular cross-section quantum wire, is applicable for any radial potential profile $V(r)$, e.g. it is also valid for diffused interfaces. Again, the behaviour of the energies as a function

of the spatial dimension, as shown in Fig. 8.16, is as expected in confined systems, namely the confinement energy decreases as the size of the system increases.

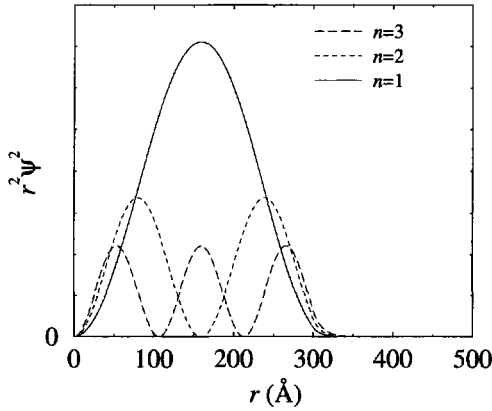


Figure 8.18 The probability density of the three lowest energy states in the 300 Å spherical quantum dot

Fig. 8.17 displays the corresponding radial components of the wave functions. It can be seen that they all have a maximum at the centre of the potential and that as the principal quantum number n increases, then the number of nodes increases. The nature of the states is perhaps better illustrated by considering the charge density. Given that successive increases in the radial coordinate r lead to spherical shells of volume proportional to r^2 , then the charge density between the radii r and $r + \delta r$ is proportional to $r^2 \psi^*(r) \psi(r)$; this is plotted in Fig. 8.18. The lowest energy state, i.e. $n = 1$, has a single anti-node which is close to half the radius of the potential, while the second has two maxima in the charge density, etc.

8.8 NON-ZERO ANGULAR MOMENTUM STATES

The states deduced in the previous section were of the form $\psi(r)$, and therefore were analogous to the 's' orbitals akin to hydrogen atoms. However, in atoms there exist other types of state, with the next simplest being that of the 'p' orbitals. These have wave functions of the following form:

$$\psi_{p_x} = x\psi(r) \quad (8.71)$$

$$\psi_{p_y} = y\psi(r) \quad (8.72)$$

$$\psi_{p_z} = z\psi(r) \quad (8.73)$$

(see any quantum theory book, e.g. [4], p. 190). These states are clearly not spherically symmetric, and have non-zero angular momenta.

Consider the first of these states in the (spherical polar coordinates) Schrödinger equation (equation (8.66)). As the orientation of the axes is purely arbitrary, then the radius r in any shooting equation *could* be aligned along the x -axis. With this simplification, the Schrödinger equation would become:

$$-\frac{\hbar^2}{2m^*} \left(\frac{2}{r} \frac{\partial}{\partial r} + \frac{\partial^2}{\partial r^2} \right) r\psi(r) + V(r)r\psi(r) = E_r r\psi(r) \quad (8.74)$$

Dropping the indices on ψ and V , then:

$$-\frac{\hbar^2}{2m^*} \left[\frac{2}{r} \left(\psi + r \frac{\partial \psi}{\partial r} \right) + \frac{\partial}{\partial r} \left(\psi + r \frac{\partial \psi}{\partial r} \right) \right] + Vr\psi = E_r r\psi \quad (8.75)$$

$$\therefore -\frac{\hbar^2}{2m^*} \left(\frac{2}{r} \psi + 2 \frac{\partial \psi}{\partial r} + \frac{\partial \psi}{\partial r} + \frac{\partial \psi}{\partial r} + r \frac{\partial^2 \psi}{\partial r^2} \right) + Vr\psi = E_r r\psi \quad (8.76)$$

and

$$-\frac{\hbar^2}{2m^*} \left(4 \frac{\partial \psi}{\partial r} + r \frac{\partial^2 \psi}{\partial r^2} \right) + \left(Vr - \frac{\hbar^2}{2m^*} \frac{2}{r} \right) \psi = E_r r\psi \quad (8.77)$$

Dividing through by r gives:

$$\frac{\partial^2 \psi}{\partial r^2} + \frac{4}{r} \frac{\partial \psi}{\partial r} + \left[\frac{2m^*}{\hbar^2} (E_r - V) + \frac{2}{r^2} \right] \psi = 0 \quad (8.78)$$

(which resembles equation (15.4) of Schiff [29]). The extra potential term ($1/r^2$) represents the contribution of the angular momentum to the energy, see [29], p. 81.

This current section has shown a possible way forward for deducing non-spherical eigenstates from a *general* spherically symmetric potential, as may be encountered in a quantum dot. Clearly, this work is incomplete and remains left open for the interested reader.

8.9 APPROACHES TO PYRAMIDAL DOTS

There has been a drive to simplify the fabrication of quantum dots, and in a particular kind of material system this has been achieved with the aid of a process known as *self-assembly*. This occurs when a thin (perhaps even sub-mono) layer of one material is deposited on top of a substrate which has a quite different lattice constant. In such a system, the strain energy in the bonds forms a substantial fraction of the total energy, and in an effort to minimize this the deposited atoms rearrange themselves, moving from a thin layer into ‘clumps’. This method has become known as the ‘Stranski-Krastanov’ growth mode [166]. Microscopy has shown that these small groups of atoms form pyramidal dots, as illustrated earlier in Fig. 8.4.

It appears that the pyramidal dot requires a fully three-dimensional solution of Schrödinger’s equation, as given by Cusack *et al.* [167]. However, a little thought

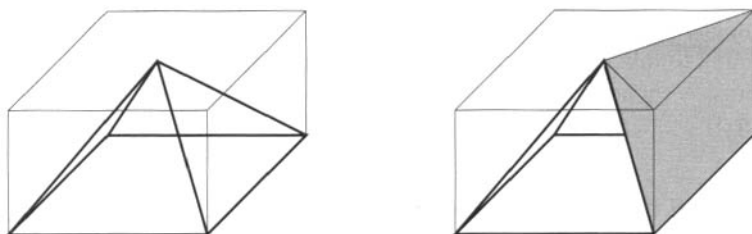


Figure 8.19 A possible perturbative approach for small pyramidal dots

allows the problem to be simplified into a combination of a series of decoupled Schrödinger equations and some perturbation theory.

For example, a pyramidal dot could be thought of as a finite-barrier cuboid dot, but with four additional perturbing potentials, as shown in Fig. 8.19 (right) (shaded area). These also take the shape of square based pyramids, with the base, in this example, vertical. However, these pyramids are not symmetric, having the apex centred on the upper edge rather than over the centre of the base.

If the finite dot was solved with decoupled one-dimensional wave functions, then for relatively small systems ($< 100 \text{ \AA}$) where the unperturbed wave function penetrates considerably into the barrier of the cuboid quantum dot, there is the possibility of a partial cancellation of the $-2V$ and $-V$ potentials at the corners and along the edges (see Section 8.6), with these $+V$ non-symmetric pyramidal potentials. Such a partial cancellation may be vital to ensure convergence of the perturbed energy levels. Recent work employing this approach has derived connection rules between the energy of a pyramidal dot with a cubic quantum dot of a different volume [164].

As mentioned above, lattice mismatch is what drives the self-assembly process, hence the atoms constituting self-assembled quantum dots are under the influence of significant amounts of strain. The effect of this strain is to alter the conduction and valence band profiles so they are no longer constant either inside or outside the quantum dot. The method for determining the strain field and the effect on the conduction and valence band edges has been summarised by Califano and Harrison [168] for both $\text{In}_{1-x}\text{Ga}_x\text{As}$ on GaAs and $\text{Si}_{1-x}\text{Ge}_x$ on silicon quantum dots.

8.10 MATRIX APPROACHES

Alternative solutions are being sought to this problem by using an extension of the method of Gangopadhyay and Nag cited earlier [165]. This involves expanding the wave function as a Fourier series of infinite quantum well solutions and has proved itself to be a workable and reliable technique [169, 170].

The basic approach is to expand the two- or three-dimensional wave function of the quantum wire or quantum dot as a linear combination of some basis functions. The basis functions are usually sine or plane waves of a larger box that encompasses the quantum wire or dot and has an infinite exterior potential. This leads to a formulation

where the energy solutions are the eigenvalues of a matrix equation. There is a thorough introduction to this method and a description on how to apply it to the solution of Schrödinger's equation in two and three dimensions in the book by Harrison [171] (Chapter 3).

8.11 FINITE DIFFERENCE EXPANSIONS

Although less important with the advent of desktop computers with more and more memory, the main problem with matrix methods for the solution of Schrödinger's equation in multiple dimensions is the sheer size of the expansion set. If in one-dimension it was found that 10 basis states were needed to reproduce the ground state energy and wave function of some perturbed potential, then in two dimensions for similar accuracy the basis set would have $10 \times 10 = 100$ components. In three dimensions this would become $10 \times 10 \times 10 = 1000$ states. Thus the matrix would be of order 100 or 1000 respectively and this is for the relatively small number of basis functions, in reality 20 or more in each direction may be necessary.

An alternative approach [172, 173] which is less demanding on memory is to return to the idea of a finite difference expansion of the derivatives in Schrödinger's equation, similar to that developed for the shooting method for one dimension in Chapter 3. Recalling the Schrödinger equation for the motion in the confined cross-sectional plane of a quantum wire, as given earlier in equation (8.11) and used again in equation (8.35), i.e.

$$-\frac{\hbar^2}{2m^*} \left[\frac{\partial^2 \psi(y, z)}{\partial y^2} + \frac{\partial^2 \psi(y, z)}{\partial z^2} \right] + V(y, z) \psi(y, z) = E_{y,z} \psi(y, z) \quad (8.79)$$

Expanding the two derivatives in terms of finite differences, as in equation (3.8), gives:

$$\begin{aligned} & -\frac{\hbar^2}{2m^*} \left[\frac{\psi(y + \delta y, z) - 2\psi(y, z) + \psi(y - \delta y, z)}{(\delta y)^2} \right. \\ & \left. + \frac{\psi(y, z + \delta z) - 2\psi(y, z) + \psi(y, z - \delta z)}{(\delta z)^2} \right] + V(y, z) \psi(y, z) = E_{y,z} \psi(y, z) \end{aligned} \quad (8.80)$$

Multiplying through by $(\delta y)^2 (\delta z)^2$:

$$\begin{aligned} & (\delta z)^2 [\psi(y + \delta y, z) - 2\psi(y, z) + \psi(y - \delta y, z)] \\ & + (\delta y)^2 [\psi(y, z + \delta z) - 2\psi(y, z) + \psi(y, z - \delta z)] \\ & = -\frac{2m^*}{\hbar^2} [E_{y,z} - V(y, z)] (\delta y \delta z)^2 \psi(y, z) \end{aligned} \quad (8.81)$$

Gathering terms together:

$$(\delta z)^2 [\psi(y + \delta y, z) + \psi(y - \delta y, z)] + (\delta y)^2 [\psi(y, z + \delta z) + \psi(y, z - \delta z)]$$

$$+ \left\{ \frac{2m^*}{\hbar^2} [E_{y,z} - V(y, z)] (\delta y \delta z)^2 - 2 [(\delta y)^2 + (\delta z)^2] \right\} \psi(y, z) = 0 \quad (8.82)$$

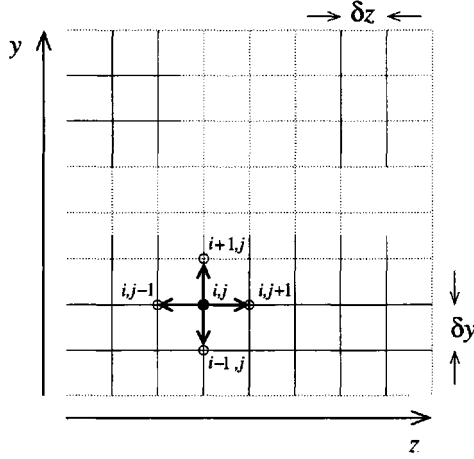


Figure 8.20 The two-dimensional mesh for the finite difference approach to the quantum wire

Thus the wave function at the general point $\psi(y, z)$ depends upon the values at 4 neighbouring points, see Fig. 8.20. Equation (8.82) can be written more succinctly in terms of the indices given in Fig. 8.20:

$$(\delta y)^2 (\psi_{i,j+1} + \psi_{i,j-1}) + (\delta z)^2 (\psi_{i+1,j} + \psi_{i-1,j}) + k\psi_{i,j} = 0 \quad (8.83)$$

where:

$$k = \frac{2m^*}{\hbar^2} [E_{y,z} - V(y, z)] (\delta y \delta z)^2 - 2 [(\delta y)^2 + (\delta z)^2] \quad (8.84)$$

Such an equation exists for each grid point on the mesh and to solve simultaneously they have to be written in the form of a matrix equation and solved according to the standard boundary conditions of the wave function and its first derivative tending to zero as the spatial coordinates tend to infinity. The memory saving occurs because the matrix is sparse. El-Moghraby *et al.* [172] map out this method in detail and apply it to rectangular and triangular cross-section quantum wires and pyramidal self-assembled quantum dots. In a later work El-Moghraby *et al.* [173] apply the method to vertically aligned coupled quantum dots.

8.12 DENSITY OF STATES

Just as there is a change in the density of states moving from the bulk (3D) crystal to a quantum well (2D), there is a further change in the density of states on moving to

quantum wires (1D) and quantum dots (0D). Recall from Section 2.3 that the density of states is defined as the number of states per unit energy per unit volume of real space, which was expressed mathematically in equation (2.33) as:

$$\rho(E) = \frac{dN}{dE} \quad (8.85)$$

In the bulk crystal, the three degrees of freedom for the electron momentum mapped out a sphere in \mathbf{k} -space, while in a quantum well the electron momenta fill successively larger circles. Continuing this argument for a quantum wire with just one degree of freedom, the electron momenta then fill states along a line.

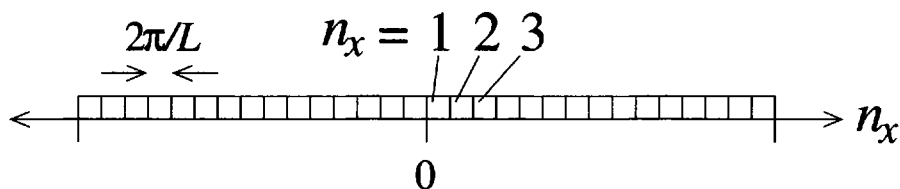


Figure 8.21 The occupation of states in \mathbf{k} -space along a one-dimensional quantum wire

Therefore, proceeding with the same argument as before, the total number of states N is then equal to the length of the line in \mathbf{k} -space ($2k$), divided by the length occupied by one state (again $2\pi/L$), and divided by the *length* in real space (see Fig. 8.21), i.e.

$$N^{1D} = 2 \times 2k \frac{1}{2\pi/L} \frac{1}{L} \quad (8.86)$$

where again the ‘2’ prefactor accounts for spin degeneracy. Therefore:

$$N^{1D} = \frac{4k}{2\pi} \quad (8.87)$$

and for later:

$$\frac{dN^{1D}}{dk} = \frac{2}{\pi} \quad (8.88)$$

In analogy to both the bulk and quantum-well cases, define the density of states for a one-dimensional wire as:

$$\rho^{1D}(E) = \frac{dN^{1D}}{dE} = \frac{dN^{1D}}{dk} \frac{dk}{dE} \quad (8.89)$$

As the along-axis dispersion curve can still be described as a parabola, (as given by equation (8.12)), then reuse can be made of equation (2.39), i.e.

$$\frac{dk}{dE} = \left(\frac{2m^*}{\hbar^2} \right)^{\frac{1}{2}} \frac{E^{-\frac{1}{2}}}{2} \quad (8.90)$$

This, taken together with the result in equation (8.88), then gives:

$$\rho^{1D}(E) = \frac{2}{\pi} \left(\frac{2m^*}{\hbar^2} \right)^{\frac{1}{2}} \frac{E^{-\frac{1}{2}}}{2} \quad (8.91)$$

$$\therefore \rho^{1D}(E) = \left(\frac{2m^*}{\hbar^2} \right)^{\frac{1}{2}} \frac{1}{\pi E^{\frac{1}{2}}} \quad (8.92)$$

where the energy E is measured upwards from a subband minimum. Therefore, comparing the density of states for bulk (3D), quantum wells (2D) and quantum wires (1D), as summarized in Table 8.2, it can be seen that successive reductions in degrees of freedom for the electron motion, lead to reductions in the functional form of $\rho(E)$ by factors of $E^{\frac{1}{2}}$.

Table 8.2 The density of states for reduced dimensionality systems, rewritten in a standard form

Dimensionality	$\rho(E)$
3D	$\frac{1}{2\pi^2} \left(\frac{2m^*}{\hbar^2} \right)^{\frac{3}{2}} E^{\frac{1}{2}}$
2D	$\frac{1}{2\pi} \left(\frac{2m^*}{\hbar^2} \right)^1 E^0$
1D	$\frac{1}{\pi} \left(\frac{2m^*}{\hbar^2} \right)^{\frac{1}{2}} E^{-\frac{1}{2}}$

If there are many (n) confined states within the quantum wire with subband minima E_i , then the density of states at any particular energy is the sum over all the subbands below that point, which can be written as:

$$\rho^{1D}(E) = \sum_{i=1}^n \left(\frac{2m^*}{\hbar^2} \right)^{\frac{1}{2}} \frac{1}{\pi(E - E_i)^{\frac{1}{2}}} \Theta(E - E_i) \quad (8.93)$$

Fig. 8.22 gives an example of the 1D density of states, for a 60×70 Å rectangular cross-section GaAs quantum wire surrounded by infinite barriers. In contrast to the bulk and 2D cases displayed earlier in Fig. 2.7, quantum wires show maxima in the density of states at around the subband minima, i.e. at around the point at which charge would be expected to accumulate. Therefore, interband (electron-hole) recombination will have a narrower linewidth than that of the 2D or 3D cases.

The situation for quantum dots is quite different. As the particles are confined in all directions, then there are no dispersion curves, and thus the density of states is just dependent upon the number of confined levels. One single isolated dot would therefore offer just two (spin-degenerate) states at the energy of each confined level, and a plot of the density of states versus energy would be a series of δ -functions.

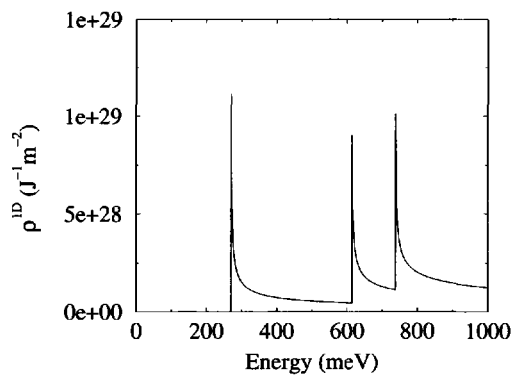


Figure 8.22 The density of states in a one-dimensional quantum wire

CHAPTER 9

CARRIER SCATTERING

9.1 FERMI'S GOLDEN RULE

If a charge carrier, i.e. an electron or a hole, is moving within the body of a perfect crystal lattice which is free from all defects and with all atoms stationary, then it will continue in that state *ad infinitum*. Of course, such a situation is never reached, which implies that the charge carrier will change its state—a process which is known as *scattering*.

Quantum mechanical scattering is usually summarized in terms of *Fermi's Golden Rule* [174] which states the following: if an electron (or hole) in a state $|i\rangle$ of energy E_i experiences a time-dependent perturbation $\tilde{\mathcal{H}}$ which could scatter (transfer) it into any one of the final states $|f\rangle$ of energy E_f , then the lifetime of the carrier in state $|i\rangle$ is given by:

$$\frac{1}{\tau_i} = \frac{2\pi}{\hbar} \sum_f \left| \langle f | \tilde{\mathcal{H}} | i \rangle \right|^2 \delta(E_f - E_i) \quad (9.1)$$

9.2 PHONONS

The massive atoms that constitute semiconductor crystals are all connected together by chemical bonds which are nominally covalent, although in compounds can have a degree of ionicity. These atoms are always in a state of continual motion, which because of the definite crystal lattice structure, is vibrational around an equilibrium position. The atoms vibrate even at the hypothetical zero of absolute temperature—the so-called *zero point energy* (see Section 3.5). In some ways, the vibrations of these interconnected quantum particles (atoms) resembles a classical (macroscopic) system of a series of masses connected by springs. There are basically four different modes of vibration, as illustrated in Figs 9.1 and 9.2, each one of which is referred to as a *phonon*.

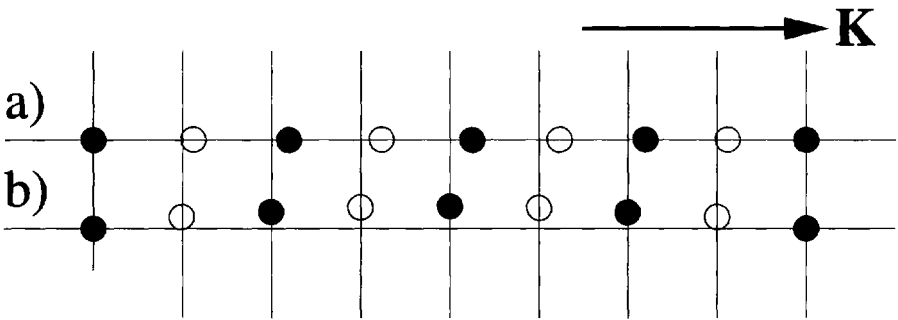


Figure 9.1 Schematic illustrations of the atomic displacements in (a) longitudinal acoustic (LA) and (b) transverse acoustic (TA) phonon modes

The acoustic modes shown in Fig. 9.1 are characterised by the neighbouring atoms being in phase. In the longitudinal mode the atomic displacements are in the same direction as the direction of energy transfer, while in the transverse mode the atomic displacements are perpendicular to this direction.

The longitudinal and transverse definitions also apply to the two types of optic phonon modes as illustrated in Fig. 9.2. However, in this type of lattice vibration the displacements of neighbouring atoms are in opposite phase.

The wave-like nature of the lattice vibrations allows them to be described, say, by an angular frequency ω and a wave vector \mathbf{K} . Thus the energy of a phonon is $\hbar\omega$ —the same as a photon of light. In addition, and in analogy to propagating electrons, the momentum of a phonon is said to be quantised and of value $\hbar\mathbf{K}$. Furthermore, phonons are diffracted by the crystal lattice just like electrons and holes, and thus a Brillouin zone type summary of the energy–momentum curves can be employed.

Fig. 9.3 shows schematically just such a set of phonon dispersion curves for a typical semiconductor. The form of the curves are reasonably similar, although with differing energy scales for the common semiconductors that are of interest in this work (see for example [7], p. 14), note, however, that the coupling (interaction)

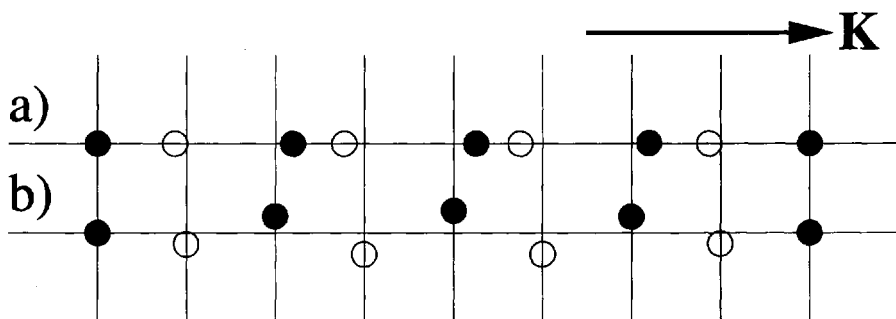


Figure 9.2 Schematic illustrations of the atomic displacements in (a) longitudinal optic (LO) and (b) transverse optic (TO) phonon modes

between the different phonon modes with charge carriers does differ between the various materials.

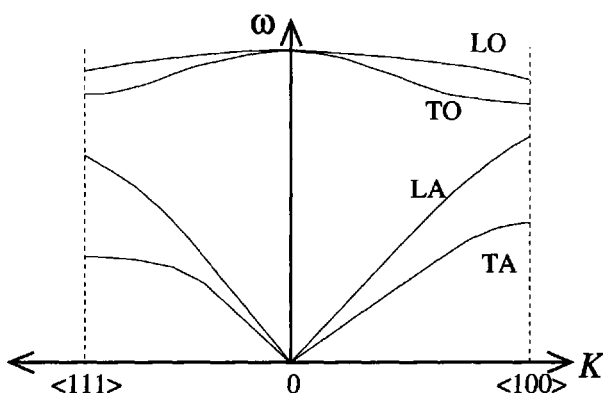


Figure 9.3 Phonon energy ($\hbar\omega$) versus momentum ($\hbar K$) curves for a typical semiconductor

Phonons are bosons, and hence their number per unit volume is given by the Bose-Einstein factor (see [175], p. 391 or [1], p. 454):

$$N_0 = \frac{1}{\exp(\hbar\omega/kT) - 1} \quad (9.2)$$

As the phonons themselves represent the motion of atoms which are centres of electric charge, then they also represent time-dependent perturbations \hat{H} of the crystal potential and can therefore scatter charge carriers. Many authors have previously considered scattering via phonons (see for example [176–178]); however, the spirit of this present work is to provide a *fully* documented derivation.

With this aim, consider a simple wave function of a phonon in bulk material*:

$$\phi = C(\mathbf{K})e^{-i\mathbf{K}\cdot\mathbf{r}} \quad (9.3)$$

Therefore, the electric field, which is the derivative of this wave function, can be described by the following relationship:

$$\mathbf{E} = \nabla\phi = -i\mathbf{K}\phi \quad (9.4)$$

and hence:

$$\mathbf{E}^*\mathbf{E} = |C(\mathbf{K})|^2 \mathbf{K}\cdot\mathbf{K} \quad (9.5)$$

The normalisation condition is therefore:

$$\frac{1}{2}\hbar\omega(\mathbf{K}) = \frac{1}{2}\omega(\mathbf{K}) \left. \frac{\partial\epsilon}{\partial\omega} \right|_{\mathbf{K}} \int \mathbf{E}^*\mathbf{E} \, d\tau \quad (9.6)$$

If V is the volume of the crystal, then as the integrand is independent of position (see equation (9.5)):

$$\frac{1}{2}\hbar\omega(\mathbf{K}) = \frac{1}{2}\omega(\mathbf{K}) \left. \frac{\partial\epsilon}{\partial\omega} \right|_{\mathbf{K}} V |C(\mathbf{K})|^2 \mathbf{K}\cdot\mathbf{K} \quad (9.7)$$

The majority of the interest here lies with heterostructures made from compound semiconductors. These materials are polar as the different electronegativities of the constituent atoms lead to a degree of ionicity in the chemical bonds (see Chapter 11). In such materials the dominant electron-phonon interaction (scattering) is with the *longitudinal optic phonon*, often referred to as the LO phonon. The LO phonon dispersion curve (see Fig. 9.3), is relatively flat, and hence it is possible to approximate it as being *dispersionless*, which gives $\omega\partial\epsilon/\partial\omega = 2/P$ [179], where:

$$P = \frac{1}{\epsilon_\infty} - \frac{1}{\epsilon_s} \quad (9.8)$$

with ϵ_∞ and ϵ_s being, respectively, the high- and low-frequency permittivities of the material. Therefore:

$$\frac{1}{2}\hbar\omega(\mathbf{K}) = \frac{1}{2} \frac{2}{P} V |C(\mathbf{K})|^2 \mathbf{K}\cdot\mathbf{K} \quad (9.9)$$

The normalisation coefficients $C(\mathbf{K})$ of the phonon wave functions are therefore given by:

$$|C(\mathbf{K})|^2 = \frac{\hbar\omega P}{2V|\mathbf{K}|^2} \quad (9.10)$$

i.e. the normalised wave function of a single dispersionless phonon is:

$$\phi = \left(\frac{\hbar\omega P}{2|\mathbf{K}|^2} \right)^{\frac{1}{2}} \frac{e^{-i\mathbf{K}\cdot\mathbf{r}}}{V^{\frac{1}{2}}} \quad (9.11)$$

*The author would like to thank Paul Kinsler for this contribution.

The total *phonon interaction term* is thus obtained by summing over all phonon wave vectors, i.e.

$$\tilde{\mathcal{H}} = e \sum_{\mathbf{K}} \phi \quad (9.12)$$

and therefore, by using equation (9.12), obtain:

$$\tilde{\mathcal{H}} = e \sum_{\mathbf{K}} \left(\frac{\hbar \omega P}{2|\mathbf{K}|^2} \right)^{\frac{1}{2}} \frac{e^{-i\mathbf{K} \cdot \mathbf{r}}}{V^{\frac{1}{2}}} \quad (9.13)$$

9.3 LONGITUDINAL OPTIC PHONON SCATTERING OF BULK CARRIERS

Although not the main emphasis of this present work, it is worthwhile deducing here the *scattering rates* ($1/\tau_i$) of electrons in bulk bands with these longitudinal optic (LO) phonons, as many of the mathematical techniques used will be required for the quantum well system that follows.

In a bulk crystal, the electron wave functions are simply given by:

$$|i\rangle = \psi_i = \frac{e^{-i\mathbf{k}_i \cdot \mathbf{r}}}{V^{\frac{1}{2}}} \quad (9.14)$$

Therefore, substituting for the electron wave function and the phonon interaction of equation (9.13) into Fermi's Golden Rule (equation (9.1)) gives:

$$\frac{1}{\tau_i} = \frac{2\pi}{\hbar} \sum_{\mathbf{k}_f} \left| \int \frac{e^{i\mathbf{k}_f \cdot \mathbf{r}}}{V^{\frac{1}{2}}} e \sum_{\mathbf{K}} \left(\frac{\hbar \omega P}{2|\mathbf{K}|^2} \right)^{\frac{1}{2}} \frac{e^{-i\mathbf{K} \cdot \mathbf{r}}}{V^{\frac{1}{2}}} \frac{e^{-i\mathbf{k}_i \cdot \mathbf{r}}}{V^{\frac{1}{2}}} d\mathbf{r} \right|^2 \delta(E_f^i - E_i^i) \quad (9.15)$$

The additional index on the initial and final state energies has been introduced in order to specify total energy, which could be the sum of a band minimum and the electron kinetic energy.

The above basically represents the lifetime for scattering for a total phonon population of 1, and although the sum over all of the phonon wave vectors should introduce a population term, it is simpler just to add it here manually, i.e.

$$\frac{1}{\tau_i} = \frac{2\pi e^2 \hbar \omega P}{2\hbar V} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \sum_{\mathbf{k}_f} \left| \int \sum_{\mathbf{K}} \frac{e^{-i(\mathbf{k}_i - \mathbf{k}_f + \mathbf{K}) \cdot \mathbf{r}}}{V|\mathbf{K}|} d\mathbf{r} \right|^2 \delta(E_f^i - E_i^i) \quad (9.16)$$

The factor $(N_0 + \frac{1}{2} \mp \frac{1}{2})$ represents the phonon density within the crystal. The upper sign of the \mp represents absorption, which reduces the phonon population from $(N_0 + 1)$ to N_0 , while the lower sign represents emission of a phonon which increases the number of phonons from N_0 to $(N_0 + 1)$. Just for convenience, absorb this factor into P and rewrite as P' , i.e.

$$P' = \left(\frac{1}{\epsilon_\infty} - \frac{1}{\epsilon_s} \right) \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \quad (9.17)$$

Then the integral over all space, specified by the $d\mathbf{r}$, of the exponential function $\exp[-i(\mathbf{k}_i - \mathbf{k}_f + \mathbf{K}) \cdot \mathbf{r}]$, can be converted into a δ -function by taking the surfaces of the volume V to be effectively at infinity, thus giving a factor of 2π per dimension, i.e. the lifetime of the carrier in state i is now given by:

$$\frac{1}{\tau_i} = \frac{\pi e^2 \omega P'}{V} \sum_{\mathbf{k}_f} \left| \sum_{\mathbf{K}} \frac{(2\pi)^3}{V} \delta(\mathbf{k}_i - \mathbf{k}_f + \mathbf{K}) \frac{1}{|\mathbf{K}|} \right|^2 \delta(E_f^t - E_i^t) \quad (9.18)$$

The sum over \mathbf{K} can be converted into an integral which introduces a factor of $L/2\pi$ per dimension (see Section 2.3); over all three-dimensions, this introduces a factor of $V/(2\pi)^3$, which cancels with the existing $(2\pi)^3/V$, and hence:

$$\frac{1}{\tau_i} = \frac{\pi e^2 \omega P'}{V} \sum_{\mathbf{k}_f} \left| \int \frac{\delta(\mathbf{k}_i - \mathbf{k}_f + \mathbf{K})}{|\mathbf{K}|} d\mathbf{K} \right|^2 \delta(E_f^t - E_i^t) \quad (9.19)$$

In a similar manner, the sum over \mathbf{k}_f can also be changed into an integral, again introducing a factor of $V/(2\pi)^3$:

$$\therefore \frac{1}{\tau_i} = \frac{\pi e^2 \omega P'}{V} \int \frac{V}{(2\pi)^3} \left| \int \frac{\delta(\mathbf{k}_i - \mathbf{k}_f + \mathbf{K})}{|\mathbf{K}|} d\mathbf{K} \right|^2 \delta(E_f^t - E_i^t) d\mathbf{k}_f \quad (9.20)$$

Expanding the modulus squared gives:

$$\begin{aligned} \frac{1}{\tau_i} &= \frac{\pi e^2 \omega P'}{(2\pi)^3} \int \left[\int \frac{\delta(\mathbf{k}_i - \mathbf{k}_f + \mathbf{K})}{|\mathbf{K}|} d\mathbf{K} \right] \\ &\quad \times \left[\int \frac{\delta(\mathbf{k}_i - \mathbf{k}_f + \mathbf{K}')}{|\mathbf{K}'|} d\mathbf{K}' \right] \delta(E_f^t - E_i^t) d\mathbf{k}_f \end{aligned} \quad (9.21)$$

Changing the order of integration, then:

$$\begin{aligned} \therefore \frac{1}{\tau_i} &= \frac{\pi e^2 \omega P'}{(2\pi)^3} \int \int \int \frac{1}{|\mathbf{K}||\mathbf{K}'|} \delta(\mathbf{k}_i - \mathbf{k}_f + \mathbf{K}) \delta(\mathbf{k}_i - \mathbf{k}_f + \mathbf{K}') \\ &\quad \times \delta(E_f^t - E_i^t) d\mathbf{K}' d\mathbf{k}_f d\mathbf{K} \end{aligned} \quad (9.22)$$

Now the integral over \mathbf{K}' only gives anything when $\mathbf{k}_i - \mathbf{k}_f + \mathbf{K}' = 0$, i.e. when $\mathbf{K}' = -\mathbf{k}_i + \mathbf{k}_f$, and therefore:

$$\frac{1}{\tau_i} = \frac{\pi e^2 \omega P'}{(2\pi)^3} \int \int \frac{1}{|\mathbf{K}||-\mathbf{k}_i + \mathbf{k}_f|} \delta(\mathbf{k}_i - \mathbf{k}_f + \mathbf{K}) \delta(E_f^t - E_i^t) d\mathbf{k}_f d\mathbf{K} \quad (9.23)$$

Similarly, the integral over \mathbf{k}_f only makes a contribution when $\mathbf{k}_i - \mathbf{k}_f + \mathbf{K} = 0$, i.e. when $\mathbf{k}_f = \mathbf{k}_i + \mathbf{K}$, and therefore:

$$\frac{1}{\tau_i} = \frac{\pi e^2 \omega P'}{(2\pi)^3} \int \frac{1}{|\mathbf{K}||-\mathbf{k}_i + \mathbf{k}_i + \mathbf{K}|} \delta(E_f^t - E_i^t) d\mathbf{K} \quad (9.24)$$

which gives:

$$\frac{1}{\tau_i} = \frac{\pi e^2 \omega P'}{(2\pi)^3} \int \frac{1}{|\mathbf{K}|^2} \delta(E_f^i - E_i^i) d\mathbf{K} \quad (9.25)$$

Now the total energy E_i^i of the system before the scattering event is equal to the energy of the charge carrier plus (or minus) the energy ($\hbar\omega$) of the phonon for absorption (or emission). In addition, the carrier energy can be split into its potential and kinetic components, i.e. for parabolic bands, the total energy of a carrier is equal to the energy of the band minimum plus a component proportional to the momentum squared, i.e.

$$E_i^i = E_i + \frac{\hbar^2 \mathbf{k}_i^2}{2m^*} \pm \hbar\omega \quad (9.26)$$

$$E_f^i = E_f + \frac{\hbar^2 \mathbf{k}_f^2}{2m^*} \quad (9.27)$$

where E_i and E_f represent the energy band minima of the initial (i) and final (f) states, respectively. A note of caution must be added here; the assumption of parabolic bands is fundamental, the derivation beyond this point is dependent upon it. While this is generally true for typical situations met in n -type material, care must be taken for the valence band, where non-parabolicity is paramount.

The optical branches of the phonon dispersion curves in typical III-V materials are quite flat, which means that the energy $\hbar\omega$ of the LO phonon is only a weak function of the phonon wave vector \mathbf{K} (see for example [1], p. 435 and [14] p. 70). Hence, as mentioned before, this derivation will follow the standard assumption that the phonon energy $\hbar\omega$ can be approximated well with a constant value (taken as 36 meV in GaAs ([14], p. 92)).

Substituting into equation (9.25) gives:

$$\frac{1}{\tau_i} = \frac{\pi e^2 \omega P'}{(2\pi)^3} \int \frac{1}{|\mathbf{K}|^2} \delta \left(E_f + \frac{\hbar^2 \mathbf{k}_f^2}{2m^*} - (E_i + \frac{\hbar^2 \mathbf{k}_i^2}{2m^*} \pm \hbar\omega) \right) d\mathbf{K} \quad (9.28)$$

$$\therefore \frac{1}{\tau_i} = \frac{\pi e^2 \omega P'}{(2\pi)^3} \int \frac{1}{|\mathbf{K}|^2} \delta \left(E_f - E_i + \frac{\hbar^2 \mathbf{k}_f^2}{2m^*} - \frac{\hbar^2 \mathbf{k}_i^2}{2m^*} \mp \hbar\omega \right) d\mathbf{K} \quad (9.29)$$

As this is scattering of electrons (carriers) in bulk material, then consider the simplest case where the carrier remains within the same *band*, i.e. $E_i = E_f$, then:

$$\frac{1}{\tau_i} = \frac{\pi e^2 \omega P'}{(2\pi)^3} \int \frac{1}{|\mathbf{K}|^2} \delta \left(\frac{\hbar^2 \mathbf{k}_f^2}{2m^*} - \frac{\hbar^2 \mathbf{k}_i^2}{2m^*} \mp \hbar\omega \right) d\mathbf{K} \quad (9.30)$$

where the argument of the δ -function can be factorised, thus giving:

$$\frac{1}{\tau_i} = \frac{\pi e^2 \omega P'}{(2\pi)^3} \int \frac{1}{|\mathbf{K}|^2} \delta \left(\frac{\hbar^2}{2m^*} \left(\mathbf{k}_f^2 - \mathbf{k}_i^2 \mp \frac{2m^* \omega}{\hbar} \right) \right) d\mathbf{K} \quad (9.31)$$

Then, employing the rule $\delta(ax) = \delta(x)/a$, with the constant $a = \hbar^2/2m^*$, obtain:

$$\frac{1}{\tau_i} = \frac{2\pi m^* e^2 \omega P'}{\hbar^2 (2\pi)^3} \int \frac{1}{|\mathbf{K}|^2} \delta \left(\mathbf{k}_f^2 - \mathbf{k}_i^2 \mp \frac{2m^* \omega}{\hbar} \right) d\mathbf{K} \quad (9.32)$$

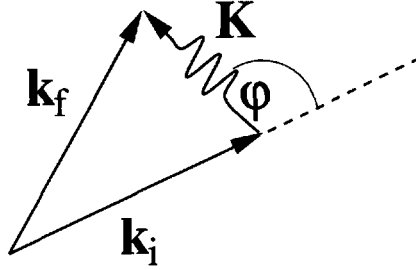


Figure 9.4 Momentum conservation in phonon scattering

Consider now the conservation of momentum between the two states of the charge carrier, labelled with their initial \mathbf{k}_i and final \mathbf{k}_f momenta, respectively, together with the momentum of the phonon \mathbf{K} , as illustrated in Fig. 9.4. The aim is to remove the dependency of the integrand on the final momentum state \mathbf{k}_f . This can be achieved by defining the angle between the initial momentum \mathbf{k}_i and the phonon momentum \mathbf{K} as ϕ , and then by using the cosine rule and the shorthand notation $|\mathbf{K}| = K$:

$$\mathbf{k}_f^2 = \mathbf{k}_i^2 + \mathbf{K}^2 - 2k_i K \cos(\pi - \phi) \quad (9.33)$$

which gives:

$$\mathbf{k}_f^2 = \mathbf{k}_i^2 + \mathbf{K}^2 + 2k_i K \cos \phi \quad (9.34)$$

Substituting for \mathbf{k}_f^2 in equation (9.32) then gives:

$$\frac{1}{\tau_i} = \frac{m^* e^2 \omega P'}{\hbar^2 (2\pi)^2} \int \frac{1}{|\mathbf{K}|^2} \delta \left(\mathbf{K}^2 + 2k_i K \cos \phi \mp \frac{2m^* \omega}{\hbar} \right) d\mathbf{K} \quad (9.35)$$

Converting the integral over all of the phonon momentum states \mathbf{K} into spherical polar coordinates, with the Cartesian z -axis along the initial momentum \mathbf{k}_i in order to conserve the angle ϕ , then the elemental volume becomes $K^2 dK \sin \phi d\phi d\theta$ therefore

$$\begin{aligned} \frac{1}{\tau_i} = \frac{m^* e^2 \omega P'}{\hbar^2 (2\pi)^2} \int_0^{2\pi} \int_0^\pi \int_0^\infty \frac{1}{|\mathbf{K}|^2} \delta \left(\mathbf{K}^2 + 2k_i K \cos \phi \mp \frac{2m^* \omega}{\hbar} \right) \\ \times K^2 dK \sin \phi d\phi d\theta \end{aligned} \quad (9.36)$$

Now clearly, $|\mathbf{K}|^2 = K^2$, as the latter is just a shorthand version, and in addition, the integral over the angle θ only introduces a factor of 2π as the integrand is independent of θ ; therefore:

$$\frac{1}{\tau_i} = \frac{m^* e^2 \omega P'}{\hbar^2 2\pi} \int_0^\pi \int_0^\infty \delta \left(K^2 + 2k_i K \cos \phi \mp \frac{2m^* \omega}{\hbar} \right) dK \sin \phi d\phi \quad (9.37)$$

Write the constant prefactor as:

$$\Upsilon' = \frac{m^* e^2 \omega P'}{2\pi \hbar^2} = \frac{m^* e^2 \omega P}{2\pi \hbar^2} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \quad (9.38)$$

and by considering the absorption process first (i.e. the upper sign in the \mp) then:

$$\frac{1}{\tau_i} = \Upsilon' \int_0^\pi \int_0^\infty \delta \left(K^2 + 2k_i K \cos \phi - \frac{2m^* \omega}{\hbar} \right) dK \sin \phi d\phi \quad (9.39)$$

Following the approach of Hagston, Piorek, and Harrison (for more details of this and related work see the PhD thesis of Piorek [180]), the argument of the δ -function can be factorised uniquely into:

$$(K - \alpha_1)(K + \alpha_2) = K^2 + (\alpha_2 - \alpha_1)K - \alpha_1 \alpha_2 \quad (9.40)$$

where the constants α_1 and α_2 are real and positive—this follows since the product $\alpha_1 \alpha_2$ must be negative. Hence, since K , which is the modulus of the phonon wave vector, must be greater than zero, then there is only one contribution to the δ -function, i.e.

$$\frac{1}{\tau_i} = \Upsilon' \int_0^\pi \int_0^\infty \delta((K - \alpha_1)(K + \alpha_2)) dK \sin \phi d\phi \quad (9.41)$$

Around the solution $K = \alpha_1$ the other factor $(K + \alpha_2)$ is clearly finite (and nearly constant!) and hence can be brought outside the δ -function, again by using the relation $\delta(ax) = \delta(x)/a$, thus giving:

$$\frac{1}{\tau_i} = \Upsilon' \int_0^\pi \int_0^\infty \frac{1}{K + \alpha_2} \delta(K - \alpha_1) dK \sin \phi d\phi \quad (9.42)$$

On performing the integration over K , the only contribution occurs when the argument of the δ -function is equal to zero, i.e. when $K = \alpha_1$, which results in:

$$\frac{1}{\tau_i} = \Upsilon' \int_0^\pi \frac{1}{\alpha_1 + \alpha_2} \sin \phi d\phi \quad (9.43)$$

Recalling the definition in equation (9.40) then it is worth noting for later that $\alpha_2 > \alpha_1$ for $0 < \phi < \pi/2$ and $\alpha_1 > \alpha_2$ for $\pi/2 < \phi < \pi$. The quadratic equation for K is then:

$$K^2 + 2k_i K \cos \phi - \frac{2m\omega}{\hbar} = 0 \quad (9.44)$$

Hence, application of the standard formula for the solution of quadratics gives:

$$K = -k_i \cos \phi \pm \sqrt{k_i^2 \cos^2 \phi + \frac{2m\omega}{\hbar}} \quad (9.45)$$

so therefore:

$$K = -k_i \cos \phi + \sqrt{k_i^2 \cos^2 \phi + \frac{2m\omega}{\hbar}} \quad (9.46)$$

and

$$K = -k_i \cos \phi - \sqrt{k_i^2 \cos^2 \phi + \frac{2m\omega}{\hbar}} \quad (9.47)$$

Now the constants α_1 and α_2 are equal to the root (K) and the negative root ($-K$) respectively, and therefore their sum is:

$$\alpha_1 + \alpha_2 = 2\sqrt{k_i^2 \cos^2 \phi + \frac{2m\omega}{\hbar}} \quad (9.48)$$

Substituting this result into equation (9.43) gives:

$$\frac{1}{\tau(p)} = \Upsilon' \int_0^\pi \frac{1}{2\sqrt{k_i^2 \cos^2 \phi + \frac{2m\omega}{\hbar}}} \sin \phi \, d\phi \quad (9.49)$$

Thus, the problem has been reduced to a single integral over the variable ϕ . One way to proceed from here is to introduce a new variable φ , defined as:

$$\varphi = -k_i \cos \phi + \sqrt{k_i^2 \cos^2 \phi + \frac{2m\omega}{\hbar}} \quad (9.50)$$

where the limits of φ follow as:

$$\varphi_{\min} = \varphi(\phi = 0) = -k_i + \sqrt{k_i^2 + \frac{2m\omega}{\hbar}} \quad (9.51)$$

and

$$\varphi_{\max} = \varphi(\phi = \pi) = +k_i + \sqrt{k_i^2 + \frac{2m\omega}{\hbar}} \quad (9.52)$$

Using equation (9.50) then:

$$\varphi + k_i \cos \phi = \sqrt{k_i^2 \cos^2 \phi + \frac{2m\omega}{\hbar}} \quad (9.53)$$

and squaring both sides:

$$\varphi^2 + 2\varphi k_i \cos \phi + k_i^2 \cos^2 \phi = k_i^2 \cos^2 \phi + \frac{2m\omega}{\hbar} \quad (9.54)$$

$$\therefore k_i \cos \phi = \frac{m\omega}{\hbar\varphi} - \frac{\varphi}{2} \quad (9.55)$$

Substituting $k_i \cos \phi$ back into equation (9.53) gives:

$$\varphi + \frac{m\omega}{\hbar\varphi} - \frac{\varphi}{2} = \sqrt{k_i^2 \cos^2 \phi + \frac{2m\omega}{\hbar}} \quad (9.56)$$

$$\therefore \frac{\varphi}{2} + \frac{m\omega}{\hbar\varphi} = \sqrt{k_i^2 \cos^2 \phi + \frac{2m\omega}{\hbar}} \quad (9.57)$$

Using equation (9.55) then:

$$\cos \phi = \frac{m\omega}{\hbar k_i \varphi} - \frac{\varphi}{2k_i} \quad (9.58)$$

and by differentiating both sides with respect to φ :

$$\sin \phi \, d\phi = \left(\frac{m\omega}{\hbar k_i \varphi^2} + \frac{1}{2k_i} \right) d\varphi \quad (9.59)$$

Using equations (9.57) and (9.59) in equation (9.49) gives:

$$\frac{1}{\tau_i} = \frac{\Upsilon'}{2} \int_{\varphi_{\min}}^{\varphi_{\max}} \frac{1}{\frac{\varphi}{2} + \frac{m\omega}{\hbar k_i \varphi}} \left(\frac{m\omega}{\hbar k_i \varphi^2} + \frac{1}{2k_i} \right) d\varphi \quad (9.60)$$

$$\therefore \frac{1}{\tau_i} = \frac{\Upsilon'}{2} \int_{\varphi_{\min}}^{\varphi_{\max}} \frac{1}{\frac{\varphi}{2} + \frac{m\omega}{\hbar k_i \varphi}} \times \frac{1}{k_i \varphi} \left(\frac{m\omega}{\hbar \varphi} + \frac{\varphi}{2} \right) d\varphi \quad (9.61)$$

and therefore:

$$\frac{1}{\tau_i} = \frac{\Upsilon'}{2} \int_{\varphi_{\min}}^{\varphi_{\max}} \frac{1}{k_i \varphi} d\varphi \quad (9.62)$$

which finally gives:

$$\frac{1}{\tau_i} = \frac{\Upsilon'}{2} \frac{1}{k_i} [\ln \varphi]_{\varphi_{\min}}^{\varphi_{\max}} \quad (9.63)$$

Now for emission, equation (9.37) becomes:

$$\frac{1}{\tau_i} = \Upsilon' \int_0^\pi \int_0^\infty \delta \left(K^2 + 2Kk_i \cos \phi + \frac{2m^* \omega}{\hbar} \right) dK \sin \phi \, d\phi \quad (9.64)$$

In analogy with absorption, the quadratic argument of the δ -function can be factorised. Considering the integration over the phonon momentum K first, then for $0 < \phi < \pi/2$, where $\cos \phi > 0$, the only possible factors are $(K + \alpha_1)(K + \alpha_2)$. These both imply roots for K which are less than zero and hence unphysical, and thus the argument of the δ -function in this ϕ domain is never zero and there are therefore no contributions to the integral.

However, when $\pi/2 < \phi < \pi$, $\cos \phi$ is negative and the quadratic in K can be factorised with two possible roots: i.e.

$$K = -k_i \cos \phi \pm \sqrt{k_i^2 \cos^2 \phi - \frac{2m\omega}{\hbar}} \quad (9.65)$$

For the phonon momentum K to remain real, the argument of the square root function must be greater than or equal to zero, i.e.

$$k_i^2 \cos^2 \phi - \frac{2m\omega}{\hbar} > 0 \quad (9.66)$$

$$\therefore k_i^2 \cos^2 \phi > \frac{2m\omega}{\hbar} \quad (9.67)$$

Thus, there is a minimum value for ϕ , given by:

$$\cos \phi_{\min} = -\sqrt{\frac{2m\omega}{\hbar k_i^2}} \quad (9.68)$$

where the negative square root has been specified as $\phi_{\min} > \pi/2$. Given that the angle ϕ belongs to the domain $\pi/2 < \phi < \pi$, the coefficient of K within the δ -function is negative, and hence the argument can be factorised into the form $(K - \alpha_1)(K - \alpha_2)$, where as before α_1 and α_2 are assumed to be real and positive. Therefore, equation (9.64) becomes:

$$\frac{1}{\tau_1} = \Upsilon' \int_{\phi_{\min}}^{\pi} \int_0^{\infty} \delta((K - \alpha_1)(K - \alpha_2)) dK \sin \phi d\phi \quad (9.69)$$

Equation (9.65) implies that the constants α_1 and α_2 are distinct, and thus there are two contributions to the integral. Choosing $\alpha_1 > \alpha_2$, then $\alpha_2 - \alpha_1$ will be negative, and hence around α_2 :

$$\delta((K - \alpha_1)(K - \alpha_2)) = \frac{\delta(K - \alpha_2)}{|K - \alpha_1|} \quad (9.70)$$

i.e.

$$\frac{1}{\tau_1} = \Upsilon' \int_{\phi_{\min}}^{\pi} \int_0^{\infty} \left[\frac{\delta(K - \alpha_1)}{K - \alpha_2} + \frac{\delta(K - \alpha_2)}{|K - \alpha_1|} \right] dK \sin \phi d\phi \quad (9.71)$$

Completing the integration over the phonon momentum K then gives:

$$\frac{1}{\tau_1} = \Upsilon' \int_{\phi_{\min}}^{\pi} \left(\frac{1}{\alpha_1 - \alpha_2} + \frac{1}{|\alpha_2 - \alpha_1|} \right) \sin \phi d\phi \quad (9.72)$$

Again in analogy to absorption, introduce a variable φ , which differs only in the sign of the $2m\omega/\hbar$ term, i.e.

$$\varphi = -k_i \cos \phi + \sqrt{k_i^2 \cos^2 \phi - \frac{2m\omega}{\hbar}} \quad (9.73)$$

As the integration over the angle ϕ is to be replaced with an integral over the new variable φ , then the new limits of integration have to be deduced. Now the minimum value of φ occurs when ϕ is at its minimum, as deduced above in equation (9.66); this occurs when the argument of the square root is zero, i.e.

$$\varphi_{\min} = \varphi(\phi_{\min}) = -k_i \cos \phi_{\min} \quad (9.74)$$

Using $\cos \phi_{\min}$ from equation (9.68), then:

$$\varphi_{\min} = -k_i \times -\sqrt{\frac{2m\omega}{\hbar k_i^2}} = \sqrt{\frac{2m\omega}{\hbar}} \quad (9.75)$$

Correspondingly, the maximum value of φ occurs when $\cos \phi_{\min}$ is at its maximum, i.e.

$$\varphi_{\max} = \varphi(\phi = \pi) = k_i + \sqrt{k_i^2 - \frac{2m\omega}{\hbar}} \quad (9.76)$$

Following a similar argument to that used for absorption eventually gives the scattering rate for emission as:

$$\frac{1}{\tau_i} = \Upsilon' \frac{1}{k_i} [\ln \varphi]_{\varphi_{\min}}^{\varphi_{\max}} \quad (9.77)$$

which is equivalent to that given by Lundstrom ([177], equation (2.76)) and can be manipulated further to give the more familiar form (for example [177], equation (2.79)).

9.4 LO PHONON SCATTERING OF TWO-DIMENSIONAL CARRIERS

Again following the method of Kinsler [179], the phonon interaction term is as for the case of bulk phonon modes, as defined in equation (9.13). However, in order to make use of the symmetry of a general heterostructure it can be split into components along the growth (z -) axis and in the (x - y) plane of the layers, i.e.

$$\tilde{\mathcal{H}} = e \sum_{\mathbf{K}} \left(\frac{\hbar \omega P}{2|\mathbf{K}|^2} \right)^{\frac{1}{2}} \frac{e^{-i\mathbf{K} \cdot \mathbf{r}}}{V^{\frac{1}{2}}} \quad (9.78)$$

which becomes:

$$\tilde{\mathcal{H}} = e \sum_{\mathbf{K}_{xy}} \sum_{K_z} \left(\frac{\hbar \omega P}{2(|\mathbf{K}_{xy}|^2 + |K_z|^2)} \right)^{\frac{1}{2}} \frac{e^{-i\mathbf{K}_{xy} \cdot \mathbf{r}_{xy}}}{A^{\frac{1}{2}}} \frac{e^{-iK_z z}}{L^{\frac{1}{2}}} \quad (9.79)$$

Now the electron (or hole) wave functions in a heterostructure are a product of an envelope along the growth axis and an in-plane travelling wave (see Section 2.2), i.e.

$$\psi = \psi(z) \frac{e^{-i\mathbf{k} \cdot \mathbf{r}_{xy}}}{A^{\frac{1}{2}}} \quad (9.80)$$

Therefore, substituting this new form for the phonon interaction and the electron wave function into Fermi's Golden Rule (equation 9.1), gives:

$$\begin{aligned} \frac{1}{\tau_i} = & \frac{2\pi}{\hbar} \sum_{\mathbf{k}_f} \left| \int \int \psi_f^*(z) \frac{e^{i\mathbf{k}_f \cdot \mathbf{r}_{xy}}}{A^{\frac{1}{2}}} e \sum_{\mathbf{K}_{xy}} \sum_{K_z} \left(\frac{\hbar \omega P}{2(|\mathbf{K}_{xy}|^2 + |K_z|^2)} \right)^{\frac{1}{2}} \right. \\ & \times \left. \frac{e^{-i\mathbf{K}_{xy} \cdot \mathbf{r}_{xy}}}{A^{\frac{1}{2}}} \frac{e^{-iK_z z}}{L^{\frac{1}{2}}} \psi_i(z) \frac{e^{-i\mathbf{k}_i \cdot \mathbf{r}_{xy}}}{A^{\frac{1}{2}}} dz d\mathbf{r}_{xy} \right|^2 \delta(E_f^i - E_i^i) \end{aligned} \quad (9.81)$$

where the electron wave vectors, \mathbf{k}_i and \mathbf{k}_f of the initial and final states, respectively, are taken explicitly to lie in the plane of the quantum wells only, i.e. $\mathbf{k} = \mathbf{k}(x, y)$ only.

This is a good place to introduce the phonon population factor ($N_0 + \frac{1}{2} \mp \frac{1}{2}$); again incorporating it within the factor P to give P' as in equation (9.17), and rearranging, then gives:

$$\frac{1}{\tau_i} = \frac{2\pi e^2 \hbar \omega P'}{2\hbar A} \sum_{\mathbf{k}_f} \left| \sum_{K_z} \frac{1}{L^{\frac{1}{2}}} \int \psi_f^*(z) e^{-iK_z z} \psi_i(z) dz \right. \\ \left. \times \sum_{\mathbf{K}_{xy}} \frac{1}{(|\mathbf{K}_{xy}|^2 + |K_z|^2)^{\frac{1}{2}}} \frac{1}{A} \int e^{-i(\mathbf{k}_i - \mathbf{k}_f + \mathbf{K}_{xy}) \cdot \mathbf{r}_{xy}} d\mathbf{r}_{xy} \right|^2 \delta(E_f^i - E_i^i) \quad (9.82)$$

As in the previous section, converting the integral over the x - y plane (denoted by $d\mathbf{r}_{xy}$) into a δ -function, then gives a factor of 2π per dimension, provided that the limits of integration are effectively at infinity. This results in the following:

$$\frac{1}{\tau_i} = \frac{\pi e^2 \omega P'}{AL} \sum_{\mathbf{k}_f} \left| \sum_{K_z} G_{if}(K_z) \right. \\ \left. \times \sum_{\mathbf{K}_{xy}} \frac{1}{(|\mathbf{K}_{xy}|^2 + |K_z|^2)^{\frac{1}{2}}} \frac{(2\pi)^2}{A} \delta(\mathbf{k}_i - \mathbf{k}_f + \mathbf{K}_{xy}) \right|^2 \delta(E_f^i - E_i^i) \quad (9.83)$$

where $G_{if}(K_z)$ is known as the *form factor* and is given by:

$$G_{if}(K_z) = \int \psi_f^*(z) e^{-iK_z z} \psi_i(z) dz \quad (9.84)$$

Often this form factor is normalised by dividing by some length, or square root of a length (see for example Lundstrom [177], equation (2.102)). However, it suits the purpose better here to leave it unnormalised as above.

Fig. 9.5 shows this form factor for the case of *intrasubband* scattering events, i.e. transitions where the carrier remains within the same quantum well subband. For this particular case, the example system is that of an electron in a 100 Å GaAs infinitely deep quantum well. As the form factor $G_{if}(K_z)$ always appears as a modulus squared, i.e. $|G_{if}(K_z)|^2$ (see later), and as the initial- and final-state wave functions, ψ_i and ψ_f , in these confined systems are real, then $|G_{if}(K_z)|^2$ is symmetric about $K_z = 0$ and hence it is plotted for positive K_z only.

The figure shows that $|G_{if}(K_z)|^2$ can have more than one maxima, with the overall maximum always for the zone-centre ($K_z = 0$) phonon, which implies that this scattering event is most likely to occur. The physical interpretation is that as the initial and final states of the carrier are the same, then no momentum change along the growth (z -) axis is required, and hence the dominance of the zero-momentum transition. There are additional local maxima, the number of which depend on the number of antinodes in the wave functions. The form factor squared for the 2→2 event, for example, has a total of two maxima, i.e. one at the zone centre and another

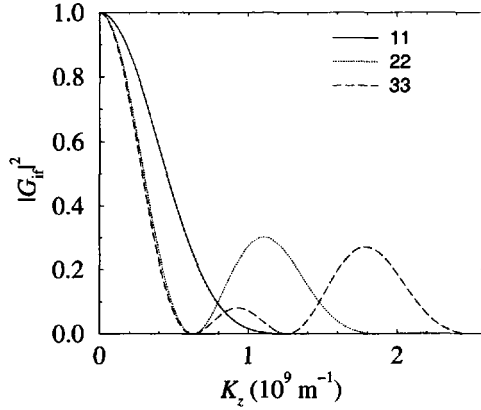


Figure 9.5 Intrasubband form factors

local maximum. This second local maximum *may* be thought of as corresponding to a scattering event from one of the wave function nodes to the other.

Fig. 9.6 shows the corresponding case for *intersubband* transitions, i.e. scattering events where the carrier changes from one confined level to another. These form factors are quite different in their behaviour to those of the intrasubband case in that there is no maximum at the zone centre—rather it is at a finite wave vector. The particular value corresponds to the difference in momenta along the z -axis of the carrier states. Thus, as ‘ i ’ increases, the wave vector of the antinode of the $i \rightarrow 1$ transition also increases.

Converting the summations over the phonon wave vectors \mathbf{K}_{xy} and K_z into integrals introduces factors of $A/(2\pi)^2$ and $L/(2\pi)$, respectively, from the density of states, and therefore:

$$\frac{1}{\tau_i} = \frac{\pi e^2 \omega P'}{AL} \sum_{\mathbf{k}_f} \left| \frac{L}{2\pi} \int \int G_{if}(K_z) \frac{\delta(\mathbf{k}_i - \mathbf{k}_f + \mathbf{K}_{xy})}{(|\mathbf{K}_{xy}|^2 + |K_z|^2)^{\frac{1}{2}}} dK_z d\mathbf{K}_{xy} \right|^2 \delta(E_f^t - E_i^t) \quad (9.85)$$

Finally, by changing the summation over the final in-plane electron wave vector \mathbf{k}_f into an integral, and also introducing a factor of $A/(2\pi)^2$ obtain:

$$\frac{1}{\tau_i} = \frac{\pi e^2 \omega P' L}{A(2\pi)^2} \frac{A}{(2\pi)^2} \int \left| \int \int G_{if}(K_z) \frac{\delta(\mathbf{k}_i - \mathbf{k}_f + \mathbf{K}_{xy})}{(|\mathbf{K}_{xy}|^2 + |K_z|^2)^{\frac{1}{2}}} dK_z d\mathbf{K}_{xy} \right|^2 d\mathbf{k}_f \times \delta(E_f^t - E_i^t) \quad (9.86)$$

The procedure followed is similar to the previous case for bulk electrons, expanding the modulus squared doubles the number of integrals over the phonon wave vectors,

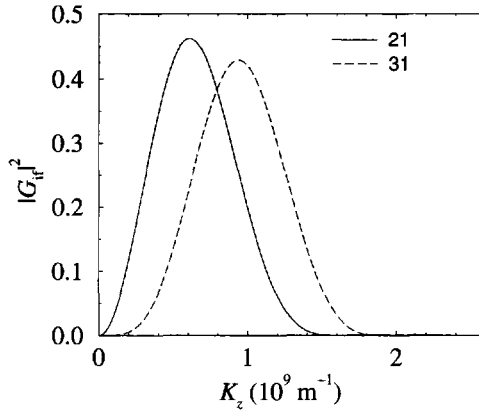


Figure 9.6 Intersubband form factors

i.e.

$$\frac{1}{\tau_i} = \frac{\pi e^2 \omega P' L}{(2\pi)^4} \int \left(\int \int G_{if}^*(K_z) \frac{\delta(\mathbf{k}_i - \mathbf{k}_f + \mathbf{K}_{xy})}{(|\mathbf{K}_{xy}|^2 + |K_z|^2)^{\frac{1}{2}}} dK_z d\mathbf{K}_{xy} \right) \times \left(\int \int G_{if}(K'_z) \frac{\delta(\mathbf{k}_i - \mathbf{k}_f + \mathbf{K}'_{xy})}{(|\mathbf{K}'_{xy}|^2 + |K'_z|^2)^{\frac{1}{2}}} dK'_z d\mathbf{K}'_{xy} \right) d\mathbf{k}_f \delta(E_f^t - E_i^t) \quad (9.87)$$

Again, as before change the order of the integrations in such a way as to perform the integration over \mathbf{K}'_{xy} first, i.e.

$$\frac{1}{\tau_i} = \frac{\pi e^2 \omega P' L}{(2\pi)^4} \int \int \int G_{if}^*(K_z) G_{if}(K'_z) \int \int \frac{\delta(\mathbf{k}_i - \mathbf{k}_f + \mathbf{K}_{xy})}{(|\mathbf{K}_{xy}|^2 + |K_z|^2)^{\frac{1}{2}}} \times \frac{\delta(\mathbf{k}_i - \mathbf{k}_f + \mathbf{K}'_{xy})}{(|\mathbf{K}'_{xy}|^2 + |K'_z|^2)^{\frac{1}{2}}} d\mathbf{K}'_{xy} d\mathbf{k}_f dK'_z dK_z d\mathbf{K}_{xy} \delta(E_f^t - E_i^t) \quad (9.88)$$

So by performing this integration over \mathbf{K}'_{xy} the second δ -function then limits the contribution to the point where $\mathbf{k}_i - \mathbf{k}_f + \mathbf{K}'_{xy} = 0$, i.e. when $\mathbf{K}'_{xy} = -\mathbf{k}_i + \mathbf{k}_f$, thus giving:

$$\frac{1}{\tau_i} = \frac{\pi e^2 \omega P' L}{(2\pi)^4} \int \int \int G_{if}^*(K_z) G_{if}(K'_z) \int \frac{\delta(\mathbf{k}_i - \mathbf{k}_f + \mathbf{K}_{xy})}{(|\mathbf{K}_{xy}|^2 + |K_z|^2)^{\frac{1}{2}}} \times \frac{1}{(|-\mathbf{k}_i + \mathbf{k}_f|^2 + |K'_z|^2)^{\frac{1}{2}}} d\mathbf{k}_f dK'_z dK_z d\mathbf{K}_{xy} \delta(E_f^t - E_i^t) \quad (9.89)$$

Now when performing the integration over the final electron in-plane wave vector \mathbf{k}_f , as denoted by the $d\mathbf{k}_f$, then again its contribution is limited by the remaining δ -function to the point where $\mathbf{k}_i - \mathbf{k}_f + \mathbf{K}_{xy} = 0$, i.e. when $\mathbf{k}_f = \mathbf{k}_i + \mathbf{K}_{xy}$, thus giving:

$$\begin{aligned} \frac{1}{\tau_i} &= \frac{\pi e^2 \omega P' L}{(2\pi)^4} \int \int \int G_{if}^*(K_z) G_{if}(K'_z) \frac{1}{(|\mathbf{K}_{xy}|^2 + |K_z|^2)^{\frac{1}{2}}} \\ &\times \frac{1}{(|-\mathbf{k}_i + \mathbf{k}_i + \mathbf{K}_{xy}|^2 + |K'_z|^2)^{\frac{1}{2}}} dK'_z dK_z d\mathbf{K}_{xy} \delta(E_f^t - E_i^t) \end{aligned} \quad (9.90)$$

and therefore:

$$\begin{aligned} \frac{1}{\tau_i} &= \frac{\pi e^2 \omega P' L}{(2\pi)^4} \int \int \int \frac{G_{if}^*(K_z) G_{if}(K'_z)}{(|\mathbf{K}_{xy}|^2 + |K_z|^2)^{\frac{1}{2}} (|\mathbf{K}_{xy}|^2 + |K'_z|^2)^{\frac{1}{2}}} dK'_z dK_z d\mathbf{K}_{xy} \\ &\times \delta(E_f^t - E_i^t) \end{aligned} \quad (9.91)$$

It is now necessary to perform a little mathematical trickery in order to force the problem through and obtain the *accepted* result (see Lundstrom [177], for example). Such steps are questionable and the possibility remains that the scattering rate should depend on the 'square of an integral' rather than the 'integral of a square'. Proceeding, however, rearrange equation (9.91) to give:

$$\begin{aligned} \frac{1}{\tau_i} &= \frac{\pi e^2 \omega P' L}{(2\pi)^4} \int \int \frac{G_{if}^*(K_z)}{(|\mathbf{K}_{xy}|^2 + |K_z|^2)^{\frac{1}{2}}} dK_z \int \frac{G_{if}(K'_z)}{(|\mathbf{K}_{xy}|^2 + |K'_z|^2)^{\frac{1}{2}}} dK'_z \\ &\times d\mathbf{K}_{xy} \delta(E_f^t - E_i^t) \end{aligned} \quad (9.92)$$

then:

$$\begin{aligned} \frac{1}{\tau_i} &= \frac{\pi e^2 \omega P' L}{(2\pi)^4} \int \int \frac{G_{if}^*(K_z)}{(|\mathbf{K}_{xy}|^2 + |K_z|^2)^{\frac{1}{2}}} \\ &\times \left[\int \frac{G_{if}(K_z)}{(|\mathbf{K}_{xy}|^2 + |K_z|^2)^{\frac{1}{2}}} \delta(K_z - K'_z) \right] dK_z dK'_z d\mathbf{K}_{xy} \delta(E_f^t - E_i^t) \end{aligned} \quad (9.93)$$

Note that the δ -function ensures that the term in the square brackets ($[\]$) only gives a contribution when $K_z = K'_z$, and when it does this term reverts to the integral over K'_z , as before. Changing the order of integration:

$$\begin{aligned} \frac{1}{\tau_i} &= \frac{\pi e^2 \omega P' L}{(2\pi)^4} \int \int \frac{G_{if}^*(K_z) G_{if}(K_z)}{(|\mathbf{K}_{xy}|^2 + |K_z|^2)^{\frac{1}{2}} (|\mathbf{K}_{xy}|^2 + |K_z|^2)^{\frac{1}{2}}} \\ &\times \int \delta(K_z - K'_z) dK'_z dK_z d\mathbf{K}_{xy} \delta(E_f^t - E_i^t) \end{aligned} \quad (9.94)$$

Performing the integration over K'_z first, then the δ -function has to yield a constant term with the units of a wave vector, e.g. $2\pi/L$, in order to satisfy dimensionality arguments; furthermore, introduction the notation $|\mathbf{K}| = K$ gives:

$$\frac{1}{\tau_i} = \frac{\pi e^2 \omega P' L}{(2\pi)^4} \int \int \frac{2\pi |G_{if}(K_z)|^2}{L K_{xy}^2 + K_z^2} dK_z d\mathbf{K}_{xy} \delta(E_f^t - E_i^t) \quad (9.95)$$

$$\therefore \frac{1}{\tau_i} = \frac{\pi e^2 \omega P'}{(2\pi)^3} \int \int \frac{|G_{if}(K_z)|^2}{K_{xy}^2 + K_z^2} dK_z d\mathbf{K}_{xy} \delta(E_f^t - E_i^t) \quad (9.96)$$

As in the bulk case, the total electron (or hole) energies are a sum of a band minimum and the kinetic energy within the band, the only difference being here that instead of the three-dimensional bulk bands, the carriers are in the two-dimensional subbands (with minima labelled E_f and E_i) of some general quantum well system. Thus:

$$E_i^t = E_i + \frac{\hbar^2 \mathbf{k}_i^2}{2m^*} \pm \hbar\omega \quad (9.97)$$

and

$$E_f^t = E_f + \frac{\hbar^2 \mathbf{k}_f^2}{2m^*} \quad (9.98)$$

where the upper sign in the $\pm \hbar\omega$ term accounts for scattering processes involving the absorption of a phonon and the lower sign represents emission. Again, it must be noted that this assumption of parabolic subbands could limit the range of applicability for the case of holes in the valence band. Exploiting the conservation of energy, as specified explicitly in the δ -function, by substituting E_f^t and E_i^t into equation (9.96) then gives:

$$\begin{aligned} \frac{1}{\tau_i} &= \frac{\pi e^2 \omega P'}{(2\pi)^3} \int \int \frac{|G_{if}(K_z)|^2}{K_{xy}^2 + K_z^2} \\ &\times \delta \left(E_f + \frac{\hbar^2 \mathbf{k}_f^2}{2m^*} - \left(E_i + \frac{\hbar^2 \mathbf{k}_i^2}{2m^*} \pm \hbar\omega \right) \right) dK_z d\mathbf{K}_{xy} \end{aligned} \quad (9.99)$$

By labelling $E_f - E_i \mp \hbar\omega$ as Δ (say), then:

$$\begin{aligned} \frac{1}{\tau_i} &= \frac{\pi e^2 \omega P'}{(2\pi)^3} \int \int \frac{|G_{if}(K_z)|^2}{K_{xy}^2 + K_z^2} \\ &\times \delta \left(\frac{\hbar^2 \mathbf{k}_f^2}{2m^*} - \frac{\hbar^2 \mathbf{k}_i^2}{2m^*} + \Delta \right) dK_z d\mathbf{K}_{xy} \end{aligned} \quad (9.100)$$

The physical requirement of conservation of energy has therefore advanced the mathematical derivation, and now it is the turn of the conservation of momentum. Consider again the application of the cosine rule to Fig. 9.4, with the additional feature that as the initial and final momenta of the carrier are in-plane, then the phonon momentum

\mathbf{K} must also be in the plane. This latter feature has been specified up until now with the additional index xy ; thus, in analogy to equation (9.34):

$$\mathbf{k}_f^2 = \mathbf{k}_i^2 + \mathbf{K}_{xy}^2 + 2k_i K_{xy} \cos \phi \quad (9.101)$$

Substituting for \mathbf{k}_f^2 in the δ -function therefore gives:

$$\begin{aligned} \frac{1}{\tau_1} &= \frac{\pi e^2 \omega P'}{(2\pi)^3} \iint \frac{|G_{if}(K_z)|^2}{K_{xy}^2 + K_z^2} \\ &\times \delta \left(\frac{\hbar^2}{2m^*} (\mathbf{k}_i^2 + \mathbf{K}_{xy}^2 + 2k_i K_{xy} \cos \phi) - \frac{\hbar^2 \mathbf{k}_i^2}{2m^*} + \Delta \right) dK_z d\mathbf{K}_{xy} \end{aligned} \quad (9.102)$$

Making use of the fact that $\mathbf{k}^2 = k^2$, then:

$$\begin{aligned} \frac{1}{\tau_1} &= \frac{\pi e^2 \omega P'}{(2\pi)^3} \iint \frac{|G_{if}(K_z)|^2}{K_{xy}^2 + K_z^2} \\ &\times \delta \left(\frac{\hbar^2 K_{xy}^2}{2m^*} + \frac{\hbar^2 k_i K_{xy} \cos \phi}{m^*} + \Delta \right) dK_z d\mathbf{K}_{xy} \end{aligned} \quad (9.103)$$

Using $\delta(ax) = \delta(x)/a$, then:

$$\begin{aligned} \frac{1}{\tau_1} &= \frac{\pi e^2 \omega P'}{(2\pi)^3} \iint \frac{|G_{if}(K_z)|^2}{K_{xy}^2 + K_z^2} \\ &\times \frac{2m^*}{\hbar^2} \delta \left(K_{xy}^2 + 2k_i K_{xy} \cos \phi + \frac{2m^* \Delta}{\hbar^2} \right) dK_z d\mathbf{K}_{xy} \end{aligned} \quad (9.104)$$

Now the two-dimensional integral over the Cartesian in-plane phonon wave vector, as denoted by $d\mathbf{K}_{xy}$, can be changed into polar coordinates, with the radius as the modulus K_{xy} and an angle ϕ between it and the initial carrier momentum state. The new elemental area is $K_{xy} dK_{xy} d\phi$, and therefore:

$$\begin{aligned} \frac{1}{\tau_1} &= \frac{m^* e^2 \omega P'}{(2\pi)^2 \hbar^2} \int_0^{2\pi} \int_0^\infty \int_{-\infty}^{+\infty} \frac{|G_{if}(K_z)|^2}{K_{xy}^2 + K_z^2} \\ &\times \delta \left(K_{xy}^2 + 2k_i K_{xy} \cos \phi + \frac{2m^* \Delta}{\hbar^2} \right) K_{xy} dK_z dK_{xy} d\phi \end{aligned} \quad (9.105)$$

Kinsler [179] has continued the analysis by introducing a new variable $y = \cos \phi$; however, in this work we will again follow the method of Hagston, Piorek, and Harrison as documented in [180]. This will involve factorising the argument of the δ -function; for now, however, consider the integral over ϕ first.

As $\cos \phi$ is an even function, then the integral from 0 to 2π is clearly twice the integral from 0 to π , and in addition, changing the order of integration gives:

$$\frac{1}{\tau_1} = \frac{2m^* e^2 \omega P'}{(2\pi)^2 \hbar^2} \int_{-\infty}^{+\infty} \int_0^\pi \int_0^\pi \frac{|G_{if}(K_z)|^2}{K_{xy}^2 + K_z^2}$$

$$\times \delta \left(K_{xy}^2 + 2k_1 K_{xy} \cos \phi + \frac{2m^* \Delta}{\hbar^2} \right) K_{xy} \, d\phi \, dK_{xy} \, dK_z \quad (9.106)$$

Write the constant prefactor as:

$$\Upsilon'' = \frac{2m^* e^2 \omega P'}{(2\pi)^2 \hbar^2} \quad (9.107)$$

and, in addition, consider the case of a positive Δ . When $0 < \phi < \pi/2$, then $\cos \phi > 0$, which implies that the argument of the δ -function can only be factorised into the form $(K_{xy} + \alpha_1)(K_{xy} + \alpha_2)$, where α_1 and α_2 are real and positive. As K_{xy} is the *magnitude* of the phonon wave vector in the plane of the quantum wells, then it is always positive; thus, the argument of the δ -function is never zero and no contributions to the integral are made.

However, when $\pi/2 < \phi < \pi$, the factor $\cos \phi$ is negative. Consider the substitution $\phi' = \pi - \phi$, and then $\cos \phi$ becomes $-\cos \phi'$ and $d\phi = -d\phi'$. When $\phi = \pi/2$, $\phi' = \pi/2$ and when $\phi = \pi$, $\phi' = 0$; therefore:

$$\begin{aligned} \frac{1}{\tau_1} &= \Upsilon'' \int_{-\infty}^{+\infty} \int_0^\infty \int_0^{\pi/2} \frac{|G_{\text{if}}(K_z)|^2}{K_{xy}^2 + K_z^2} \\ &\times \delta \left(K_{xy}^2 - 2k_1 K_{xy} \cos \phi' + \frac{2m^* \Delta}{\hbar^2} \right) K_{xy} \, d\phi' \, dK_{xy} \, dK_z \end{aligned} \quad (9.108)$$

Now the argument of the quadratic can be factorised into the form $(K_{xy} - \alpha_1)(K_{xy} - \alpha_2)$, where it is also specified that $\alpha_1 > \alpha_2$. Thus two solutions for the in-plane phonon wave vector exist; this is illustrated for a general case in Fig. 9.7. For each absorption or emission process, two possible scattering events can occur, each one of which conserves both energy and momentum, where the latter are satisfied by mediating via two different phonon momenta, K_{xy} .

Using this factorised form in equation (9.108) gives:

$$\begin{aligned} \frac{1}{\tau_1} &= \Upsilon'' \int_{-\infty}^{+\infty} \int_0^\infty \int_0^{\pi/2} \frac{|G_{\text{if}}(K_z)|^2}{K_{xy}^2 + K_z^2} \\ &\times \delta((K_{xy} - \alpha_1)(K_{xy} - \alpha_2)) K_{xy} \, d\phi' \, dK_{xy} \, dK_z \end{aligned} \quad (9.109)$$

Now around the solution $K_{xy} = \alpha_1$, the factor $(K_{xy} - \alpha_2)$ is a constant, and vice versa, and hence:

$$\begin{aligned} \frac{1}{\tau_1} &= \Upsilon'' \int_{-\infty}^{+\infty} \int_0^\infty \int_0^{\pi/2} \frac{|G_{\text{if}}(K_z)|^2}{K_{xy}^2 + K_z^2} \\ &\times \left[\frac{\delta(K_{xy} - \alpha_1)}{K_{xy} - \alpha_2} + \frac{\delta(K_{xy} - \alpha_2)}{|K_{xy} - \alpha_1|} \right] K_{xy} \, d\phi' \, dK_{xy} \, dK_z \end{aligned} \quad (9.110)$$

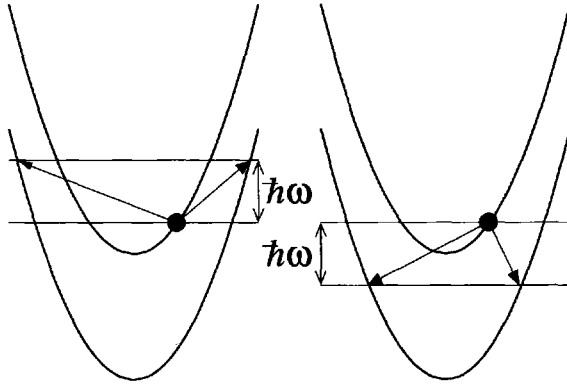


Figure 9.7 The two possible scattering events for (left) phonon absorption and (right) phonon emission

Hence, completing the integration over K_{xy} :

$$\frac{1}{\tau_1} = \Upsilon'' \int_{-\infty}^{+\infty} \int_0^{\pi/2} |G_{if}(K_z)|^2 \times \left(\frac{1}{\alpha_1^2 + K_z^2} \frac{\alpha_1}{\alpha_1 - \alpha_2} + \frac{1}{\alpha_2^2 + K_z^2} \frac{\alpha_2}{\alpha_1 - \alpha_2} \right) d\phi' dK_z \quad (9.111)$$

Recalling the quadratic for K_{xy} in the δ -function argument of equation (9.108) then:

$$K_{xy} = k_i \cos \phi' \pm \sqrt{k_i^2 \cos^2 \phi' - \frac{2m^* \Delta}{\hbar^2}} \quad (9.112)$$

where it is known that as the roots for K_{xy} are real and distinct, then the argument of the square root function is greater than zero. Furthermore, as it has been specified that $\alpha_1 > \alpha_2$, then:

$$\alpha_1 = k_i \cos \phi' + \sqrt{k_i^2 \cos^2 \phi' - \frac{2m^* \Delta}{\hbar^2}} \quad (9.113)$$

and

$$\alpha_2 = k_i \cos \phi' - \sqrt{k_i^2 \cos^2 \phi' - \frac{2m^* \Delta}{\hbar^2}} \quad (9.114)$$

Therefore:

$$\alpha_1 + \alpha_2 = 2k_i \cos \phi' \quad (9.115)$$

and

$$\alpha_1 \alpha_2 = \frac{2m^* \Delta}{\hbar^2} \quad (9.116)$$

which also come from the 'sum of the roots = $-b/a$ ' and 'product of the roots = c/a '. In addition:

$$\alpha_1 - \alpha_2 = 2\sqrt{k_i^2 \cos^2 \phi' - \frac{2m^* \Delta}{\hbar^2}} \quad (9.117)$$

Hence, if equation (9.111) can be manipulated to only contain the roots in the form of these simple constructions, then a reasonably compact expression may be obtained. Consider the term in parentheses in equation (9.111), i.e.

$$\begin{aligned} \frac{1}{\alpha_1^2 + K_z^2} \frac{\alpha_1}{\alpha_1 - \alpha_2} + \frac{1}{\alpha_2^2 + K_z^2} \frac{\alpha_2}{\alpha_1 - \alpha_2} &= \frac{(\alpha_2^2 + K_z^2)\alpha_1 + (\alpha_1^2 + K_z^2)\alpha_2}{(\alpha_1^2 + K_z^2)(\alpha_2^2 + K_z^2)(\alpha_1 - \alpha_2)} \\ &= \frac{\alpha_1 \alpha_2^2 + \alpha_1 K_z^2 + \alpha_1^2 \alpha_2 + \alpha_2 K_z^2}{(\alpha_1 - \alpha_2)(\alpha_1^2 \alpha_2^2 + \alpha_1^2 K_z^2 + \alpha_2^2 K_z^2 + K_z^4)} \end{aligned} \quad (9.118)$$

$$= \frac{(\alpha_1 + \alpha_2)(K_z^2 + \alpha_1 \alpha_2)}{(\alpha_1 - \alpha_2)\{(\alpha_1 \alpha_2)^2 + [(\alpha_1 + \alpha_2)^2 - 2\alpha_1 \alpha_2]K_z^2 + K_z^4\}} \quad (9.119)$$

Using the forms for $\alpha_1 + \alpha_2$, $\alpha_1 \alpha_2$, and $\alpha_1 - \alpha_2$ in equations (9.115), (9.116), and (9.117), then:

$$\begin{aligned} \frac{1}{\alpha_1^2 + K_z^2} \frac{\alpha_1}{\alpha_1 - \alpha_2} + \frac{1}{\alpha_2^2 + K_z^2} \frac{\alpha_2}{\alpha_1 - \alpha_2} &= \frac{2k_i \cos \phi' \left[K_z^2 + \left(\frac{2m^* \Delta}{\hbar^2} \right) \right]}{2\sqrt{k_i^2 \cos^2 \phi' - \frac{2m^* \Delta}{\hbar^2}} \left[\left(\frac{2m^* \Delta}{\hbar^2} \right)^2 + (4k_i^2 \cos^2 \phi' - 2\frac{2m^* \Delta}{\hbar^2}) K_z^2 + K_z^4 \right]} \\ &= \frac{k_i \cos \phi' \left[K_z^2 + \left(\frac{2m^* \Delta}{\hbar^2} \right) \right]}{\sqrt{k_i^2 \cos^2 \phi' - \frac{2m^* \Delta}{\hbar^2}} \left[\left(K_z^2 - \frac{2m^* \Delta}{\hbar^2} \right)^2 + 4k_i^2 K_z^2 \cos^2 \phi' \right]} \end{aligned} \quad (9.120)$$

Substituting back into equation (9.111) gives:

$$\begin{aligned} \frac{1}{\tau_i} &= \Upsilon'' \int_{-\infty}^{+\infty} \int_0^{\pi/2} |G_{if}(K_z)|^2 k_i \left[K_z^2 + \left(\frac{2m^* \Delta}{\hbar^2} \right) \right] \\ &\times \frac{\cos \phi'}{\sqrt{k_i^2 \cos^2 \phi' - \frac{2m^* \Delta}{\hbar^2}} \left[\left(K_z^2 - \frac{2m^* \Delta}{\hbar^2} \right)^2 + 4k_i^2 K_z^2 \cos^2 \phi' \right]} d\phi' dK_z \end{aligned} \quad (9.121)$$

Thus far, the derivation has followed fairly standard methods; however, it is possible to proceed further with the analytical work and evaluate the integral over the angle ϕ' by using the innovative approach of Hagston, Piorek and Harrison [180]. The remainder of this section is dedicated to this procedure, but if only the result is required, then skip to equation (9.151).

For now though, consider this integral over the angle ϕ' , where there is clearly a maximum value for ϕ' which occurs when the argument of the square root function

in equation (9.112) becomes zero, which is given by $k_i \cos \phi'_{\max} = \sqrt{2m^* \Delta / \hbar^2}$. Labelling as 'I', this is then of the form:

$$I = \int_0^{\phi'_{\max}} \frac{\cos \phi'}{\sqrt{a \cos^2 \phi' - b(c + d \cos^2 \phi')}} d\phi' \quad (9.122)$$

where:

$$a = k_i^2; \quad b = \frac{2m^* \Delta}{\hbar^2}; \quad c = \left(K_z^2 - \frac{2m^* \Delta}{\hbar^2} \right)^2; \quad d = 4k_i^2 K_z^2 \quad (9.123)$$

Then:

$$I = \int_0^{\phi'_{\max}} \frac{\sin \phi' \cos \phi'}{\sin \phi' \sqrt{a \cos^2 \phi' - b(c + d \cos^2 \phi')}} d\phi' \quad (9.124)$$

Putting $x = \cos^2 \phi'$, then $dx = -2 \sin \phi' \cos \phi' d\phi'$, and therefore:

$$I = \int_1^{\cos^2 \phi'_{\max}} \frac{-\frac{1}{2}}{\sqrt{1-x} \sqrt{ax - b(c + dx)}} dx \quad (9.125)$$

$$\therefore I = \frac{1}{2} \int_{\cos^2 \phi'_{\max}}^1 \frac{1}{\sqrt{-ax^2 + (a+b)x - b(c + dx)}} dx \quad (9.126)$$

which is the same as:

$$I = \frac{1}{2\sqrt{a}} \int_{\cos^2 \phi'_{\max}}^1 \frac{1}{\sqrt{\frac{(a-b)^2}{4a^2} - \left(x - \frac{a+b}{2a}\right)^2} (c + dx)} dx \quad (9.127)$$

Consider the substitution:

$$x - \frac{a+b}{2a} = \frac{a-b}{2a} \cos \alpha \quad (9.128)$$

and so then:

$$dx = -\left(\frac{a-b}{2a}\right) \sin \alpha d\alpha \quad (9.129)$$

when $x = \cos^2 \phi'_{\max} = b/a$; therefore:

$$\frac{b}{a} - \frac{a+b}{2a} = \frac{a-b}{2a} \cos \alpha \quad (9.130)$$

$$\therefore 2b - a - b = (a-b) \cos \alpha \quad (9.131)$$

i.e. $\cos \alpha = -1$, which implies that the lower limit of the integral becomes π . In addition, when $x = 1$, $\cos \alpha = 1$ which implies that the upper limit is 0. Substituting into equation (9.127), then obtain:

$$I = \frac{1}{2\sqrt{a}} \int_{\pi}^0 \frac{-\left(\frac{a-b}{2a}\right) \sin \alpha}{\sqrt{\left(\frac{a-b}{2a}\right)^2 - \left(\frac{a-b}{2a} \cos \alpha\right)^2} (c + d\left(\frac{a+b}{2a} + \frac{a-b}{2a} \cos \alpha\right))} d\alpha \quad (9.132)$$

Now, the factor $\left(\frac{a-b}{2a}\right)^2$ can be taken out of the square root, and cancelled with the factor in the numerator; the remainder of the argument then becomes $\sqrt{1 - \cos^2 \alpha}$, which also cancels with the numerator, and hence:

$$I = \frac{1}{2\sqrt{a}} \int_0^\pi \frac{1}{c + d\left(\frac{a+b}{2a} + \frac{a-b}{2a} \cos \alpha\right)} d\alpha \quad (9.133)$$

Writing $e = c + d(a+b)/(2a)$ and $f = d(a-b)/(2a)$, then (for later):

$$e + f = c + d \quad \text{and} \quad e - f = c + d\frac{b}{a} \quad (9.134)$$

Substituting for e and f gives:

$$I = \frac{1}{2\sqrt{a}} \int_0^\pi \frac{1}{e + f \cos \alpha} d\alpha \quad (9.135)$$

Consider the substitution $t = \tan(\alpha/2)$, then:

$$dt = \frac{1}{2} \sec^2\left(\frac{\alpha}{2}\right) d\alpha = \frac{1}{2} \left[1 + \tan^2\left(\frac{\alpha}{2}\right)\right] d\alpha = \frac{1}{2}(1 + t^2) d\alpha \quad (9.136)$$

Making use of the tan half-angle formula (see [38], p. 72):

$$\cos \alpha = \frac{1 - t^2}{1 + t^2} \quad (9.137)$$

and changing the integral limits, equation (9.135) then becomes:

$$I = \frac{1}{2\sqrt{a}} \int_0^\infty \frac{1}{e + f\left(\frac{1-t^2}{1+t^2}\right)} \frac{2}{1+t^2} dt \quad (9.138)$$

Multiplying the top and bottom of the above equation by $(1 + t^2)$ then gives the following:

$$I = \frac{1}{2\sqrt{a}} \int_0^\infty \frac{2}{e(1+t^2) + f(1-t^2)} dt \quad (9.139)$$

$$I = \frac{1}{\sqrt{a}} \int_0^\infty \frac{1}{e + f + (e-f)t^2} dt \quad (9.140)$$

This is a standard form, and given that the coefficient of t^2 is greater than zero, the result is then given by the $(\Delta = 4(e+f)(e-f) > 0)$ component of equation (2.172) in [23], i.e.

$$I = \frac{1}{\sqrt{a}} \left[\frac{2}{\sqrt{4(e+f)(e-f)}} \arctan\left(\frac{2(e-f)t}{\sqrt{4(e+f)(e-f)}}\right) \right]_0^\infty \quad (9.141)$$

which when evaluated gives:

$$I = \frac{1}{\sqrt{a}} \frac{2}{\sqrt{4(e+f)(e-f)}} \left(\frac{\pi}{2} - 0 \right) \quad (9.142)$$

Recalling the forms for a , b , c , and d given in equation (9.123) and substituting these into equations (9.134) then gives the following:

$$e + f = \left(K_z^2 - \frac{2m^* \Delta}{\hbar^2} \right)^2 + 4k_i^2 K_z^2 \quad (9.143)$$

and

$$e - f = \left(K_z^2 - \frac{2m^* \Delta}{\hbar^2} \right)^2 + 4K_z^2 \frac{2m^* \Delta}{\hbar^2} = \left(K_z^2 + \frac{2m^* \Delta}{\hbar^2} \right)^2 \quad (9.144)$$

Substituting for both $(e + f)$ and $(e - f)$ in equation (9.142), and recalling that $a = k_i^2$, then:

$$I = \frac{1}{2k_i} \frac{\pi}{\sqrt{\left(K_z^2 - \frac{2m^* \Delta}{\hbar^2} \right)^2 + 4k_i^2 K_z^2} \left(K_z^2 + \frac{2m^* \Delta}{\hbar^2} \right)} \quad (9.145)$$

$$\therefore I = \frac{1}{2k_i} \frac{\pi}{\sqrt{K_z^4 + 2K_z^2 \left(2k_i^2 - \frac{2m^* \Delta}{\hbar^2} \right) + \left(\frac{2m^* \Delta}{\hbar^2} \right)^2} \left(K_z^2 + \frac{2m^* \Delta}{\hbar^2} \right)} \quad (9.146)$$

With this analytical form for the integral over the angle ϕ' , the original equation, i.e. equation (9.121), becomes:

$$\frac{1}{\tau_i} = \frac{\Upsilon''}{2} \int_{-\infty}^{+\infty} \frac{\pi |G_{if}(K_z)|^2}{\sqrt{K_z^4 + 2K_z^2 \left(2k_i^2 - \frac{2m^* \Delta}{\hbar^2} \right) + \left(\frac{2m^* \Delta}{\hbar^2} \right)^2}} dK_z \quad (9.147)$$

This last equation represents the lifetime of a carrier in an initial subband 'i' with any in-plane wave vector k_i , strictly speaking only those k_i that satisfy energy conservation can have a lifetime. This information was really lost when the integration over the in-plane phonon wave vector K_{xy} was performed to remove the second (energy conservation) δ -function. It can be put back in with a Heaviside unit step function:

$$\Theta \left(k_i^2 - \frac{2m^* \Delta}{\hbar^2} \right)$$

Recalling that $\Delta = E_f - E_i \mp \hbar\omega$ then the Heaviside function ensures that there are only finite lifetimes τ_i when:

$$\frac{\hbar^2 k_i}{2m^*} > E_f - E_i \mp \hbar\omega \quad (9.148)$$

Remembering that the upper sign describes phonon absorption then this would imply:

$$E_i + \frac{\hbar^2 k_i}{2m^*} + \hbar\omega > E_f \quad (9.149)$$

and for emission:

$$E_i + \frac{\hbar^2 k_i}{2m^*} > E_f + \hbar\omega \quad (9.150)$$

which are clearly the desired results, hence finally:

$$\frac{1}{\tau_i} = \frac{\Upsilon''}{2} \Theta \left(k_i^2 - \frac{2m^*\Delta}{\hbar^2} \right) \int_{-\infty}^{+\infty} \frac{\pi |G_{if}(K_z)|^2}{\sqrt{K_z^4 + 2K_z^2 \left(2k_i^2 - \frac{2m^*\Delta}{\hbar^2} \right) + \left(\frac{2m^*\Delta}{\hbar^2} \right)^2}} dK_z \quad (9.151)$$

The case for a negative Δ follows a similar route and leads to the same end result. Therefore, equation (9.151) represents the final form for the lifetime τ_i of a carrier in a subband 'i' with an in-plane wave vector k_i before scattering by an LO phonon. The information regarding whether such an event is with respect to absorption of a phonon or emission, and the final state of the carrier, is incorporated within the variables Δ and P' (which is within Υ'').

This result is a particularly powerful expression because it is applicable to all two-dimensional carrier distributions, regardless of the particular form for the wave functions. Such information is wrapped up in the form factor $G_{if}(K_z)$, and thus the carrier-LO phonon scattering rate can be calculated for any semiconductor heterostructure simply by evaluating a one-dimensional integral.

9.5 APPLICATION TO CONDUCTION SUBBANDS

In the previous section, the carrier-LO phonon scattering rate, which is the reciprocal of the lifetime, was derived for a two-dimensional distribution, as found in the subbands formed in quantum well systems. In this present section, this result, as summarized in equation (9.151), will be applied to a variety of examples in order to gain an intuitive understanding of this important phenomenon.

Fig. 9.8 displays the intersubband scattering rate as a function of the total initial energy E_i^t , as defined in equation (9.97), for an electron in the second subband of an infinitely deep quantum well with respect to LO phonon emission and scattering into the ground state. (Note here the initial carrier energy domain is from the subband minimum upwards.) The scattering rate increases as the carrier approaches the subband minimum, which in this case is around 220 meV. This is a general result and occurs for subband separations greater than the LO phonon energy.

If the quantum well width is allowed to increase, then the energy separation between the initial and final subbands decreases. Fig. 9.9 illustrates the effect that such a series of calculations has on the scattering rate. It can be seen that in all cases the scattering rate increases; however, for quantum well widths greater than 300 Å, there

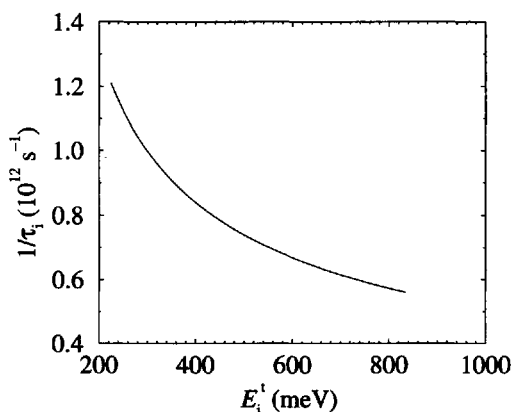


Figure 9.8 The scattering rate via LO phonon emission for an electron initially in the second subband and finally in the ground state, of a 100 Å GaAs infinitely deep quantum well at a lattice temperature of 77 K

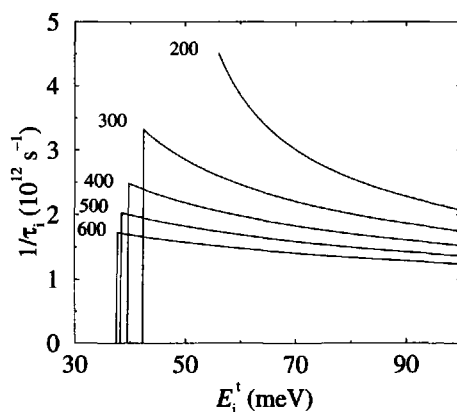


Figure 9.9 The scattering rate as given in Fig. 9.8, but for a variety of well widths (in Å), as indicated

is a small region where the scattering rate falls very rapidly to zero. This is indicative of the subband separation, $E_2 - E_1$, being less than the LO phonon energy.

The 'cut-off' in the scattering rate occurs because the electrons in the upper subband have not sufficient energy to emit an LO phonon and hence are unable to scatter. This feature is illustrated schematically in Fig. 9.10. Moving from the upper right down the curve, electrons have sufficient kinetic energy which, when combined with the potential energy from being in the upper subband, allows them to emit a phonon; however, the third electron represents the minimum kinetic energy for scattering.

Below this point, the electrons are less than an LO phonon away from the energy minimum of the *complete* system and hence can not scatter.

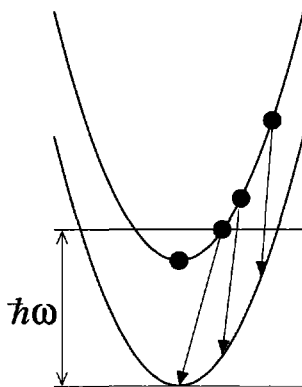


Figure 9.10 The effect of the LO phonon energy ‘cut-off’ on intersubband electron scattering

Fig. 9.11 shows the corresponding intrasubband scattering rate for an electron in the second subband. It can be seen that the behaviour of the rate with the initial energy is qualitatively similar to the intersubband case shown in Fig. 9.8. However, in addition there is a cut-off energy at 260 meV, which is an LO phonon energy above the subband minima. This is always the case for intrasubband scattering—when a carrier is within a phonon energy of the subband minima it can not emit a phonon. Note that the scattering rate is almost an order of magnitude higher for the intrasubband case than for the intersubband case. This actually represents something of a minimum, as generally the intrasubband rate is between one and two orders of magnitude higher than the intersubband rate. This is because the overlap of the wave function with itself is always complete, whereas the overlap of two distinct wave functions is often only partial. In this case, with this hypothetical infinitely deep quantum well, the overlap of the wave functions for the intersubband case is higher than the usual situation.

9.6 AVERAGING OVER CARRIER DISTRIBUTIONS

The formula shown in equation (9.151) gives the lifetime of a carrier in a particular subband with a definite in-plane wave vector k_i with respect to scattering with an LO phonon into another subband. In real situations, there isn’t just one carrier in the initial subband, plus an empty final subband; in fact, there are generally Fermi–Dirac distributions in both of the subbands (see Section 2.4). It is then more useful to know the *mean* scattering rate (or lifetime) of a carrier.

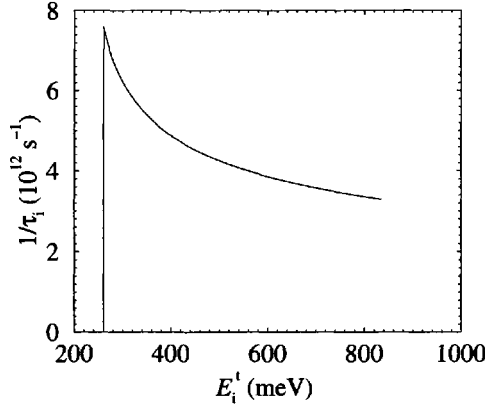


Figure 9.11 The intrasubband scattering rate via LO phonon emission for the same case as that shown in Fig. 9.8

A simple weighted mean over a distribution of carriers in the initial subband might look like the following:

$$\text{mean} \left(\frac{1}{\tau_i} \right) = \frac{\int \frac{1}{\tau_i} f_i^{\text{FD}}(E) dE}{\int f_i^{\text{FD}}(E) dE} \quad (9.152)$$

where the subscript 'i' on the distribution functions indicates the subband, i.e. to be evaluated with the 'quasi' Fermi energy of that subband. However, this still disregards the distribution in the final subband; in this case, filled states could prevent carriers from scattering into them, thus reducing the probability of an event. This effect of *final-state blocking* can be incorporated into the above to give:

$$\frac{1}{\tau_{if}} = \frac{\int \frac{1}{\tau_i} f_i^{\text{FD}}(E) f_f^{\text{FD}}(E - \hbar\omega) dE}{\int f_i^{\text{FD}}(E) dE} \quad (9.153)$$

where the double subscript 'if' is used to indicate that this scattering rate is an average over the subband populations in the initial and final states. The integrals are evaluated from the subband minimum of the initial state up to some defined maximum. In the calculations that follow, this maximum has been chosen to be the energy of the highest point in the potential profile, which means physically that any carriers above the barrier are assumed to ionise rapidly. An alternative to this may be to choose the Fermi energy plus $10kT$, or similar. Equation (9.153) may be simplified slightly, as the denominator is simply the number of carriers in the subband divided by the density of states (see equation (2.48)).

Fig. 9.12 shows this calculated mean for the same system as before, i.e. a GaAs infinitely deep quantum well, for the case of scattering via the emission of a LO phonon from the second subband to the ground subband. In this series of calculations the width of the well was varied in order to scan the difference between the energy

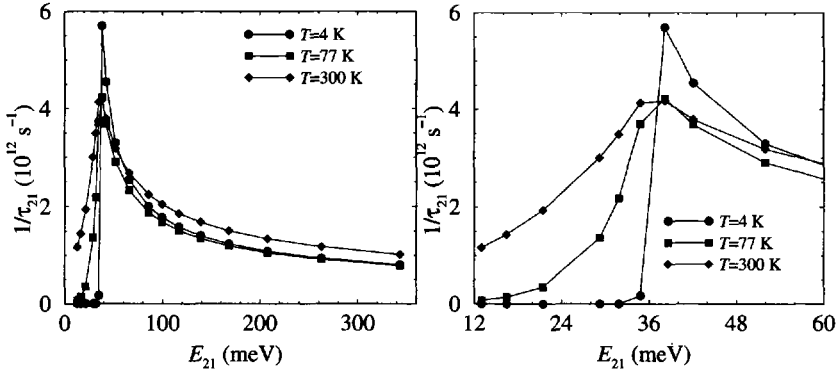


Figure 9.12 The mean scattering rate averaged over distributions in both the initial and final subbands, as a function of the subband separation, shown for three different temperatures

band minima (labelled here as $E_{21} = E_2 - E_1$) through the LO phonon energy, and the electron density in each subband was assumed to be 10^{10} cm^{-2} . The *electron temperature*, i.e. the temperature inserted into the Fermi–Dirac distribution function was taken equal to the lattice temperature—in systems under excitation where non-equilibrium distributions will be present, this assumption is almost certainly not true. Again, the infinitely deep quantum well is a good illustrative example, as the overlap of the wave functions, which is contained within the form factor G_{if} , does not change, i.e. the effect on the scattering rate is due entirely to the energy separation.

In the long-range scan, Fig. 9.12 (left), it can be seen that as the subband separation decreases, the scattering rate increases up to almost a ‘resonance’ point and then decreases rapidly. At energies above the resonance, the scattering rate has only a weak temperature dependence, but below it, the dependence is stronger. Figure 9.12 (right) illustrates this resonance effect more clearly, for a smaller range of energies. As may be expected for this fixed phonon energy, which in GaAs is 36 meV, the peak in the scattering rate occurs when the subband separation is equal to the phonon energy. The right hand figure highlights well the strong temperature dependence of the scattering rate for subband separations below the LO phonon energy. At very low temperatures, the ‘cut-off’ in the scattering is almost as complete as that shown by the single-carrier case in the previous section. However, as the temperature increases the carrier distributions broaden, so although the subband separation remains below the phonon energy, the carriers in the upper level spread up the subband, with a proportion having enough kinetic energy to be able to emit a LO phonon and scatter to the lower level. As the temperature increases, this proportion increases and hence the mean scattering rate also increases.

9.7 RATIO OF EMISSION TO ABSORPTION

In the previous section, the importance of the carrier energy for emission has been demonstrated. Converse to this, for the case of carrier scattering by absorption of a phonon, it is not the energy of the carrier that is the important issue but rather the number of phonons available—the more phonons, then the more likely an absorption process. The phonon density, given in equation (9.2), increases as the temperature rises, thus increasing the probability of an absorption.

Fig. 9.13 displays the results of calculations of the ratio of the emission to the electron–LO phonon absorption rate, for the same series of quantum wells as in the previous section. It can be seen that the emission rate is always larger than the absorption rate; thus given a carrier population in an excited subband, then when left to reach equilibrium the carriers will always emit more phonons than they absorb and hence scatter down to the ground state.

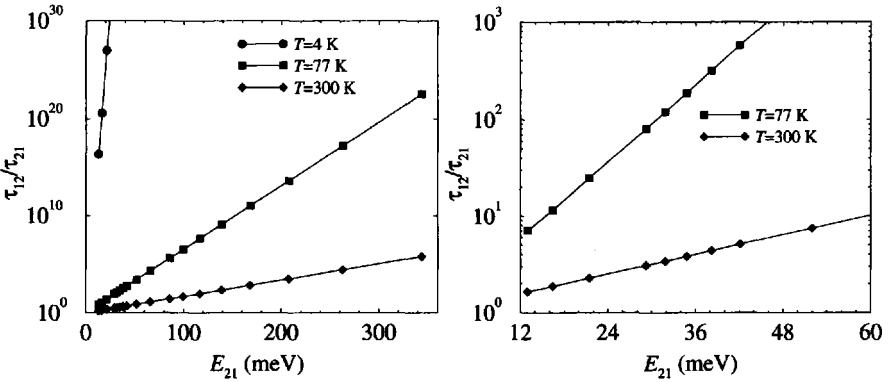


Figure 9.13 The ratio of the mean electron–LO phonon emission ($1/\tau_{21}$) to absorption ($1/\tau_{12}$) rate, as a function of the subband separation (left) and a more detailed view around the LO phonon energy (right)

The linearities of the graphs shown in Fig. 9.13 hints at a relationship of the form:

$$\frac{\tau_{12}}{\tau_{21}} \propto \exp(\text{constant} \times E_{21}) \quad (9.154)$$

and in fact, a numerical analysis of the data shows that the ‘constant’ is equal to $1/kT$, i.e.

$$\frac{\tau_{12}}{\tau_{21}} \propto \exp\left(\frac{E_{21}}{kT}\right) \quad (9.155)$$

This simple relationship between the ratio of the emission to the absorption scattering rates, the subband separation and the temperature is helpful in summarizing the data presented in Fig. 9.13. For a fixed temperature, this ratio increases as the energy separation between the two levels increases, while for a given subband separation, increasing temperature leads to a decrease in the ratio.

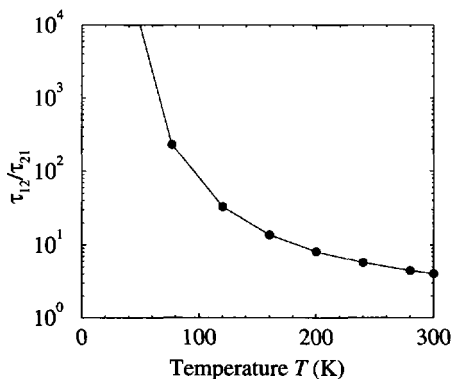


Figure 9.14 The ratio of the mean electron-LO phonon emission ($1/\tau_{21}$) to absorption ($1/\tau_{12}$) rate, as a function of temperature T for a fixed subband separation E_{21} equal to the LO phonon energy of 36 meV

These conclusions are not that obvious a priori, and go beyond the simple intuitive picture that the ratio of the scattering rates is controlled by the ratio of the phonon densities, i.e. $(N_0 + 1)/N_0$.

For a more detailed investigation into the temperature dependency of the ratio of the emission to absorption scattering rates, consider the data shown in Fig. 9.14, which corresponds to the fixed subband separation of E_{21} equal to the LO phonon energy, which in this case is 36 meV. At room temperature, the ratio is equal to the well-known result of 4; however, as the temperature decreases, the ratio of these mean scattering rates increases very rapidly and by 77 K emission is more than two orders of magnitude more likely than absorption.

9.8 SCREENING OF THE LO PHONON INTERACTION

The longitudinal optical phonon interaction is a polar interaction, thus it can be influenced by the presence of other charges. In particular, in a doped semiconductor there can be many charges which are free to move in an electromagnetic field and like Lenz's law in electromagnetic induction they move to oppose any change. This idea is known as 'screening' and its effect is to reduce the scattering rate due to LO phonons.

The screening model of Park *et al.* [181] can be implemented in a simple way in the formalism here by making the substitution:

$$K_z^2 \longrightarrow K_z^2 \left(1 + \frac{\lambda_s^2}{K_z^2} \right)^2 \quad (9.156)$$

in equation (9.151) except in the form factor $G_{if}(K_z)$. The quantity λ_s is known as the 'inverse screening length' and for systems with a majority carrier type this would

simplify to:

$$\lambda_s^2 = \frac{e^2}{\pi \hbar^2 \epsilon_s} \sum_j \left\{ \frac{\sqrt{2m^* E_j} m^* f^{\text{FD}}(E_j)}{\pi \hbar} \right\} \quad (9.157)$$

where the index 'j' includes all occupied subbands. The unusual parameters employed in Figs 9.15 and 9.16 were chosen to give direct comparisons with Figs 1 and 2 of Park *et al.* and the effects of introducing screening are very similar.

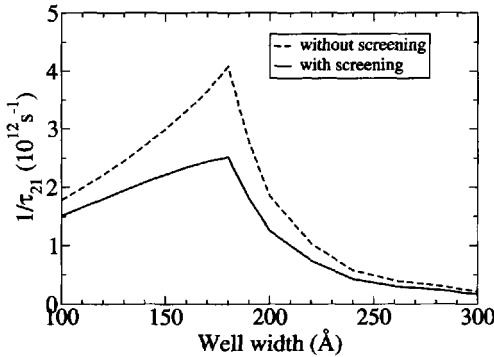


Figure 9.15 The electron-LO phonon scattering rate from the first excited to the ground subband as a function of well width for a GaAs quantum well surrounded by 60 Å $\text{Ga}_{0.7}\text{Al}_{0.3}\text{As}$ barriers. The lattice temperature was taken as 15 K, the electron temperature as 100 K and the carrier density as 10^{11} cm^{-2} .

The peaks in the curves in Fig. 9.15 occur when the energy separation between the subbands is in resonance with the LO phonon energy, in this case 36 meV. Around this energy screening at this carrier density can reduce the scattering rate by a factor of 2 or 3, though further from resonance screening has a reduced effect. Fig. 9.16 shows the effect of the temperature of the electron distributions in the subbands on the lifetime for two different carrier densities. Such elevated electron temperatures occur commonly in electronic and optoelectronic devices which are all driven by energy some of which is absorbed directly by the electron gas.

9.9 ACOUSTIC DEFORMATION POTENTIAL SCATTERING

Combining Lundstrom's [177] equations (2.56) and (2.59), summing over all bulk phonon wave functions which are of the form:

$$\frac{e^{-i\mathbf{K}_{xy} \cdot \mathbf{r}_{xy}}}{A^{\frac{1}{2}}} \frac{e^{-iK_z z}}{L^{\frac{1}{2}}} \quad (9.158)$$

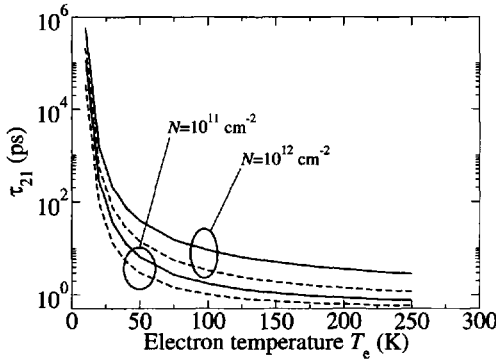


Figure 9.16 The electron-LO phonon scattering rate from the first excited to the ground subband as a function of the temperature of the electron distributions (known as the 'electron temperature') for a 220 Å GaAs quantum well surrounded by 60 Å $\text{Ga}_{0.7}\text{Al}_{0.3}\text{As}$ barriers. The lattice temperature was taken as 15 K.

and taking a carrier wave function in a heterostructure as before:

$$\psi = \psi(z) \frac{e^{-i\mathbf{k}_\bullet \cdot \mathbf{r}_{xy}}}{A^{\frac{1}{2}}} \quad (9.159)$$

gives on insertion into Fermi's Golden Rule:

$$\begin{aligned} \frac{1}{\tau_i} = & \frac{2\pi}{\hbar} \sum_{\mathbf{k}_f} \frac{D_A^2 \hbar}{2\rho\omega_s} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \\ & \times \left| \int \int \psi_f^*(z) \frac{e^{i\mathbf{k}_f \cdot \mathbf{r}_{xy}}}{A^{\frac{1}{2}}} \sum_{\mathbf{K}_{xy}} \sum_{K_z} K \frac{e^{-i\mathbf{K}_{xy} \cdot \mathbf{r}_{xy}}}{A^{\frac{1}{2}}} \frac{e^{-iK_z z}}{L^{\frac{1}{2}}} \psi_i(z) \frac{e^{-i\mathbf{k}_i \cdot \mathbf{r}_{xy}}}{A^{\frac{1}{2}}} dz d\mathbf{r}_{xy} \right|^2 \\ & \times \delta(E_f^t - E_i^t) \end{aligned} \quad (9.160)$$

which gives:

$$\begin{aligned} \frac{1}{\tau_i} = & \frac{2\pi}{\hbar} \sum_{\mathbf{k}_f} \frac{D_A^2 \hbar}{2\rho\omega_s} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \\ & \times \left| \sum_{\mathbf{K}_{xy}} \sum_{K_z} \int \psi_f^*(z) \frac{e^{-iK_z z}}{L^{\frac{1}{2}}} \psi_i(z) dz K \int \frac{e^{i\mathbf{k}_f \cdot \mathbf{r}_{xy}}}{A^{\frac{1}{2}}} \frac{e^{-i\mathbf{K}_{xy} \cdot \mathbf{r}_{xy}}}{A^{\frac{1}{2}}} \frac{e^{-i\mathbf{k}_i \cdot \mathbf{r}_{xy}}}{A^{\frac{1}{2}}} d\mathbf{r}_{xy} \right|^2 \\ & \times \delta(E_f^t - E_i^t) \end{aligned} \quad (9.161)$$

Now the integral over z is just the form factor (see equation (9.84)), hence:

$$\frac{1}{\tau_i} = \frac{2\pi}{\hbar} \sum_{\mathbf{k}_f} \frac{D_A^2 \hbar}{2\rho\omega_s} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \times \left| \sum_{\mathbf{K}_{xy}} \sum_{K_z} \frac{G_{if}(K_z)}{L^{\frac{1}{2}}} K \int \frac{e^{-i(\mathbf{k}_i - \mathbf{k}_f + \mathbf{K}_{xy}) \cdot \mathbf{r}_{xy}}}{A^{\frac{3}{2}}} d\mathbf{r}_{xy} \right|^2 \delta(E_f^t - E_i^t) \quad (9.162)$$

Converting the integral over the x - y plane (denoted by $d\mathbf{r}_{xy}$) into a δ -function gives a factor of 2π per dimension, i.e.

$$\frac{1}{\tau_i} = \frac{2\pi}{\hbar} \sum_{\mathbf{k}_f} \frac{D_A^2 \hbar}{2\rho\omega_s L A^3} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \times \left| \sum_{\mathbf{K}_{xy}} \sum_{K_z} G_{if}(K_z) K (2\pi)^2 \delta(\mathbf{k}_i - \mathbf{k}_f + \mathbf{K}_{xy}) \right|^2 \delta(E_f^t - E_i^t) \quad (9.163)$$

Now converting the summations over K_z and \mathbf{K}_{xy} into integrals introduces factors of $L/(2\pi)$ and $A/(2\pi)^2$ respectively, giving:

$$\frac{1}{\tau_i} = \frac{2\pi}{\hbar} \sum_{\mathbf{k}_f} \frac{D_A^2 \hbar}{2\rho\omega_s L A^3} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \times \left| \int \int \frac{L}{2\pi} G_{if}(K_z) K \frac{A}{(2\pi)^2} (2\pi)^2 \delta(\mathbf{k}_i - \mathbf{k}_f + \mathbf{K}_{xy}) d\mathbf{K}_{xy} dK_z \right|^2 \delta(E_f^t - E_i^t) \quad (9.164)$$

Just simplifying the constants, then:

$$\frac{1}{\tau_i} = \sum_{\mathbf{k}_f} \frac{D_A^2}{2\rho\omega_s A} \frac{L}{2\pi} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \times \left| \int \int G_{if}(K_z) K \delta(\mathbf{k}_i - \mathbf{k}_f + \mathbf{K}_{xy}) d\mathbf{K}_{xy} dK_z \right|^2 \delta(E_f^t - E_i^t) \quad (9.165)$$

Approximating the acoustic branch of the phonon dispersion curve by a linear function then the angular frequency of a sound wave is given by:

$$\omega_s = v_s K \quad (9.166)$$

hence as ω_s is dependent upon the phonon wave vector, it needs to be included within the integrals:

$$\frac{1}{\tau_i} = \sum_{\mathbf{k}_f} \frac{D_A^2}{2\rho v_s A} \frac{L}{2\pi} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right)$$

$$\times \left| \int \int G_{if}(K_z) \sqrt{K} \delta(\mathbf{k}_i - \mathbf{k}_f + \mathbf{K}_{xy}) \, d\mathbf{K}_{xy} \, dK_z \right|^2 \delta(E_f^t - E_i^t) \quad (9.167)$$

Converting the sum over final carrier momentum states \mathbf{k}_f into an integral introduces a factor of $A/(2\pi)^2$, thus:

$$\frac{1}{\tau_i} = \frac{D_A^2}{2\rho v_s A} \frac{L}{2\pi} \frac{A}{(2\pi)^2} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \times \int \left| \int \int G_{if}(K_z) \sqrt{K} \delta(\mathbf{k}_i - \mathbf{k}_f + \mathbf{K}_{xy}) \, d\mathbf{K}_{xy} \, dK_z \right|^2 d\mathbf{k}_f \delta(E_f^t - E_i^t) \quad (9.168)$$

Expanding the square:

$$\begin{aligned} \frac{1}{\tau_i} &= \frac{D_A^2}{2\rho v_s (2\pi)^2} \frac{L}{2\pi} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \\ &\times \int \left(\int \int G_{if}(K_z) \sqrt{(K_{xy}^2 + K_z^2)^{\frac{1}{2}}} \delta(\mathbf{k}_i - \mathbf{k}_f + \mathbf{K}_{xy}) \, d\mathbf{K}_{xy} \, dK_z \right) \\ &\times \left(\int \int G_{if}(K'_z) \sqrt{(K_{xy}'^2 + K_z'^2)^{\frac{1}{2}}} \delta(\mathbf{k}_i - \mathbf{k}_f + \mathbf{K}_{xy}') \, d\mathbf{K}_{xy}' \, dK_z' \right) \\ &d\mathbf{k}_f \delta(E_f^t - E_i^t) \end{aligned} \quad (9.169)$$

Performing the integration over \mathbf{K}_{xy}' first, then the second δ -function implies that there is only a contribution when $\mathbf{K}_{xy}' = -\mathbf{k}_i + \mathbf{k}_f$:

$$\begin{aligned} \frac{1}{\tau_i} &= \frac{D_A^2}{2\rho v_s (2\pi)^2} \frac{L}{2\pi} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \\ &\times \int \left(\int \int G_{if}(K_z) \sqrt{(K_{xy}^2 + K_z^2)^{\frac{1}{2}}} \delta(\mathbf{k}_i - \mathbf{k}_f + \mathbf{K}_{xy}) \, d\mathbf{K}_{xy} \, dK_z \right) \\ &\times \left(\int G_{if}(K'_z) \sqrt{(|-\mathbf{k}_i + \mathbf{k}_f|^2 + K_z'^2)^{\frac{1}{2}}} \, dK_z' \right) \\ &d\mathbf{k}_f \delta(E_f^t - E_i^t) \end{aligned} \quad (9.170)$$

Now performing the integral over the final carrier momentum \mathbf{k}_f , the first δ -function limits the integrals to contributions when $\mathbf{k}_f = \mathbf{k}_i + \mathbf{K}_{xy}$, i.e.

$$\begin{aligned} \frac{1}{\tau_i} &= \frac{D_A^2}{2\rho v_s (2\pi)^2} \frac{L}{2\pi} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \\ &\times \left(\int \int G_{if}(K_z) \sqrt{(K_{xy}^2 + K_z^2)^{\frac{1}{2}}} \, d\mathbf{K}_{xy} \, dK_z \right) \\ &\times \left(\int G_{if}(K'_z) \sqrt{(|-\mathbf{k}_i + \mathbf{k}_i + \mathbf{K}_{xy}|^2 + K_z'^2)^{\frac{1}{2}}} \, dK_z' \right) \end{aligned}$$

$$\delta(E_f^t - E_i^t) \quad (9.171)$$

and therefore:

$$\begin{aligned} \frac{1}{\tau_i} &= \frac{D_A^2}{2\rho v_s(2\pi)^2} \frac{L}{2\pi} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \\ &\times \int \int \int G_{if}(K_z) G_{if}(K'_z) \sqrt{(K_{xy}^2 + K_z^2)^{\frac{1}{2}}} \sqrt{(K_{xy}^2 + K_z'^2)^{\frac{1}{2}}} dK_z dK'_z d\mathbf{K}_{xy} \\ &\delta(E_f^t - E_i^t) \end{aligned} \quad (9.172)$$

Utilizing the 'questionable' mathematical trick as before (see Section 9.4) allows the mathematics to be forced through to yield expressions which agree with those quoted but again it is worth reiterating the statement that maybe the scattering rate should be a 'square of an integral' rather than an 'integral of a square'. The procedure equates K'_z to K_z and introduces a factor of $2\pi/L$ thus giving:

$$\begin{aligned} \frac{1}{\tau_i} &= \frac{D_A^2}{2\rho v_s(2\pi)^2} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \\ &\times \int \int (G_{if}(K_z))^2 (K_{xy}^2 + K_z^2)^{\frac{1}{2}} dK_z d\mathbf{K}_{xy} \delta(E_f^t - E_i^t) \end{aligned} \quad (9.173)$$

Now, as before, the initial and final total energies of the carrier-phonon system are given by:

$$E_i^t = E_i + \frac{\hbar^2 \mathbf{k}_i^2}{2m^*} \pm \hbar\omega_s \quad (9.174)$$

$$E_f^t = E_f + \frac{\hbar^2 \mathbf{k}_f^2}{2m^*} \quad (9.175)$$

Hence substituting into equation (9.173):

$$\begin{aligned} \frac{1}{\tau_i} &= \frac{D_A^2}{2\rho v_s(2\pi)^2} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \\ &\times \int \int (G_{if}(K_z))^2 (K_{xy}^2 + K_z^2)^{\frac{1}{2}} \delta \left(E_f + \frac{\hbar^2 \mathbf{k}_f^2}{2m^*} - E_i - \frac{\hbar^2 \mathbf{k}_i^2}{2m^*} \mp \hbar\omega_s \right) dK_z d\mathbf{K}_{xy} \end{aligned} \quad (9.176)$$

Following the prescription in Fig. 9.4 and defining the angle between the initial and final carrier momentum states as ϕ , and applying the cosine rule gives:

$$\mathbf{k}_f^2 = \mathbf{k}_i^2 + \mathbf{K}_{xy}^2 - 2k_i K_{xy} \cos(\pi - \phi) \quad (9.177)$$

$$\therefore \mathbf{k}_f^2 = \mathbf{k}_i^2 + \mathbf{K}_{xy}^2 + 2k_i K_{xy} \cos \phi \quad (9.178)$$

Hence substituting into equation (9.176), then:

$$\frac{1}{\tau_i} = \frac{D_A^2}{2\rho v_s(2\pi)^2} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \int \int (G_{if}(K_z))^2 (K_{xy}^2 + K_z^2)^{\frac{1}{2}}$$

$$\times \delta \left(E_f + \frac{\hbar^2}{2m^*} (\mathbf{k}_i^2 + \mathbf{K}_{xy}^2 + 2k_i K_{xy} \cos \phi) - E_i - \frac{\hbar^2 \mathbf{k}_i^2}{2m^*} \mp \hbar \omega_s \right) dK_z d\mathbf{K}_{xy} \quad (9.179)$$

Now the energies of phonons from the acoustic branch are generally very small, i.e. a few meV, which is small when compared to typical intersubband separations $\Delta E = E_f - E_i$, which range from several tens to several hundred meV, hence in this formalism the phonon energy $\hbar \omega_s$ will be approximated as zero. The physical implication of this is that *from the viewpoint of the calculation of scattering rates* the acoustic deformation potential scattering is assumed elastic.

Implementing this and taking the factor $\hbar^2/2m$ out of the δ -function then:

$$\begin{aligned} \frac{1}{\tau_i} &= \frac{D_A^2 m^*}{\rho v_s (2\pi)^2 \hbar^2} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \int \int (G_{if}(K_z))^2 (K_{xy}^2 + K_z^2)^{\frac{1}{2}} \\ &\times \delta \left(K_{xy}^2 + 2k_i K_{xy} \cos \phi + \frac{2m^* \Delta E}{\hbar^2} \right) dK_z d\mathbf{K}_{xy} \end{aligned} \quad (9.180)$$

Transforming the integral over the in-plane phonon wave vector \mathbf{K}_{xy} into polar coordinates gives:

$$\begin{aligned} \frac{1}{\tau_i} &= \frac{D_A^2 m^*}{\rho v_s (2\pi)^2 \hbar^2} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \int \int_0^{2\pi} \int (G_{if}(K_z))^2 (K_{xy}^2 + K_z^2)^{\frac{1}{2}} \\ &\times \delta \left(K_{xy}^2 + 2k_i K_{xy} \cos \phi + \frac{2m^* \Delta E}{\hbar^2} \right) K_{xy} dK_{xy} d\phi dK_z \end{aligned} \quad (9.181)$$

The argument within the δ -function can be factorised—as K_{xy} is a magnitude then solutions for it must be positive, hence the argument can be factorised as $(K_{xy} - \alpha_1)(K_{xy} - \alpha_2)$, where the roots α_1 and α_2 are given by:

$$\alpha_{1,2} = -k_i \cos \phi \pm \sqrt{k_i^2 \cos^2 \phi - \frac{2m^* \Delta E}{\hbar^2}} \quad (9.182)$$

where clearly $\alpha_1 > \alpha_2$. Equation (9.181) can therefore be written as:

$$\begin{aligned} \frac{1}{\tau_i} &= \frac{D_A^2 m^*}{\rho v_s (2\pi)^2 \hbar^2} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \int_0^\infty \int_0^{2\pi} \int_0^\infty (G_{if}(K_z))^2 (K_{xy}^2 + K_z^2)^{\frac{1}{2}} \\ &\times \delta((K_{xy} - \alpha_1)(K_{xy} - \alpha_2)) K_{xy} dK_{xy} d\phi dK_z \end{aligned} \quad (9.183)$$

Consider now the integration over K_{xy} , there are only two contributions to the integral, one when K_{xy} is around α_1 and the other when K_{xy} is around α_2 . Thus the δ -function can be split into two components:

$$\frac{1}{\tau_i} = \frac{D_A^2 m^*}{\rho v_s (2\pi)^2 \hbar^2} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \int_0^\infty \int_0^{2\pi} \int_0^\infty (G_{if}(K_z))^2 (K_{xy}^2 + K_z^2)^{\frac{1}{2}}$$

$$\times \left[\frac{1}{|K_{xy} - \alpha_2|} \delta(K_{xy} - \alpha_1) + \frac{1}{|K_{xy} - \alpha_1|} \delta(K_{xy} - \alpha_2) \right] K_{xy} dK_{xy} d\phi dK_z \quad (9.184)$$

and now actually performing the integration over K_{xy} gives:

$$\begin{aligned} \frac{1}{\tau_1} &= \frac{D_A^2 m^*}{\rho v_s (2\pi)^2 \hbar^2} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \int_0^\infty \int_0^{2\pi} (G_{if}(K_z))^2 \\ &\times \left(\frac{\Theta(\alpha_1) \alpha_1 \sqrt{\alpha_1^2 + K_z^2}}{|\alpha_1 - \alpha_2|} + \frac{\Theta(\alpha_2) \alpha_2 \sqrt{\alpha_2^2 + K_z^2}}{|\alpha_2 - \alpha_1|} \right) d\phi dK_z \end{aligned} \quad (9.185)$$

where the Heaviside functions ensure there are only contributions for positive α_1 and α_2 . Recalling $\alpha_1 > \alpha_2$ then finally:

$$\begin{aligned} \frac{1}{\tau_1} &= \frac{D_A^2 m^*}{\rho v_s (2\pi)^2 \hbar^2} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \int_0^\infty \int_0^{2\pi} (G_{if}(K_z))^2 \\ &\times \left(\frac{\Theta(\alpha_1) \alpha_1 \sqrt{\alpha_1^2 + K_z^2} + \Theta(\alpha_2) \alpha_2 \sqrt{\alpha_2^2 + K_z^2}}{\alpha_1 - \alpha_2} \right) d\phi dK_z \end{aligned} \quad (9.186)$$

which is the same result as Piorek [180].

9.10 APPLICATION TO CONDUCTION SUBBANDS

Fig. 9.17(a) shows the effect of the energy separation on the intersubband scattering rate due to acoustic phonon emission. It can be seen that for these thermalised distributions the scattering rate is almost linear. A comparison with the equivalent electron-electron scattering rate, as shown in Figs 9.25 and 9.26, shows that acoustic deformation potential scattering is much slower (an order of magnitude or more) than electron-electron scattering for small (less than 20 meV) intersubband energy separations at the same carrier density. However, it can be comparable or faster at larger (greater than 100 meV) energy separations. Fig. 9.17(b) compares the emission rate given in (a) with the absorption rate (shown in dashed lines). It can be seen that for the same pair of initial (2) and final (1) states the absorption rate remains a little below the emission rate due to the ratio of $N_0 + 1$ to N_0 . In contrast to this the emission and absorption rates for carriers to scatter from the ground state (1) to the first excited state (2) which are visible in the lower section of Fig. 9.17(b) reach a maximum and then decrease as the separation between the initial and final states increases. This is to be expected as the acoustic phonon can only supply a few meV of energy and as the carriers in the initial state are thermalised most of them are concentrated near the subband minimum and simply cannot gain enough energy to transfer into the higher state.

Fig. 9.18(a) shows the microscopic effect of acoustic deformation potential scattering on carriers *within* a subband. In particular it can be seen that the rate for

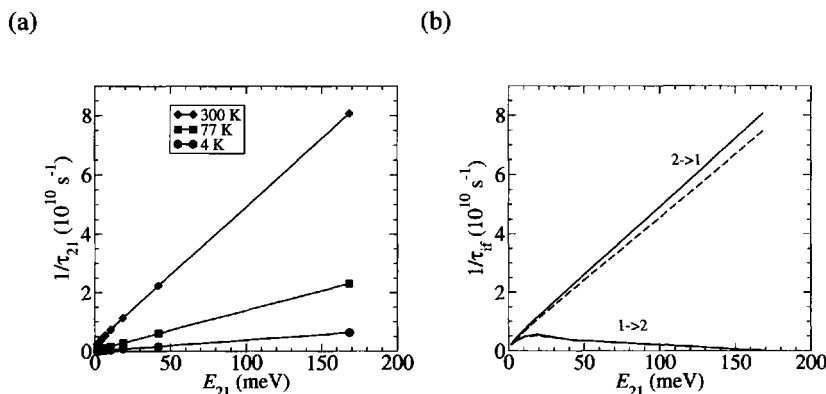


Figure 9.17 The intersubband acoustic deformation potential scattering rates as a function of the subband energy separation $E_{21} = E_2 - E_1$ for an infinitely deep quantum well with electron densities of $10 \times 10^{10} \text{ cm}^{-2}$ in each of the lowest two subbands, (a) shows the effect of temperature on the scattering rate from the second ($n = 2$) to the first ($n = 1$) subband due to the emission of acoustic phonons and (b) compares the scattering rates from the second to the first and the first to the second subbands due to emission (solid lines) and absorption (dashed lines) of acoustic phonons at the fixed temperature of 300 K.

emission of acoustic phonons is greater than that for absorption for all initial carrier energies and temperatures and this is why carriers which have a lot of kinetic energy within a subband lose this energy to the lattice and eventually form a thermalised distribution[†]. Note the presence of the subband minimum produces the ‘cut-off’ in the data at energies just larger than 50 meV.

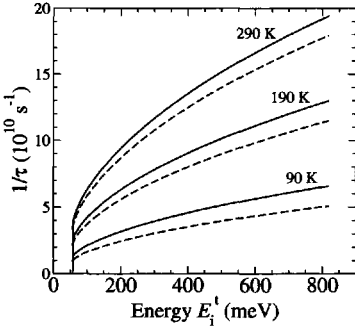
Fig. 9.18(b) illustrates the results of calculations of the effect of temperature and carrier density on the *thermally averaged* acoustic deformation potential scattering rates. Increasing temperature implies the presence of more phonons and hence the scattering rate increases, while the variations due to the carrier density changes are much smaller and are due to final state blocking. It is interesting to note that for (b) the thermally averaged phonon absorption rates (which are not shown) are almost identical to the emission rates under this assumption of thermalised (Fermi–Dirac) carrier distributions, this is a reflection of the idea of an equilibrium.

9.11 OPTICAL DEFORMATION POTENTIAL SCATTERING

Taking the optical deformation potential as a sum over all phonon wavevectors with the interaction term of Lundstrom [177] (equation 2.67) then in analogy to equa-

[†]This result helps justify the assumption which gives the thermal averaging of carriers, used so often in this chapter, some physical meaning.

(a)



(b)

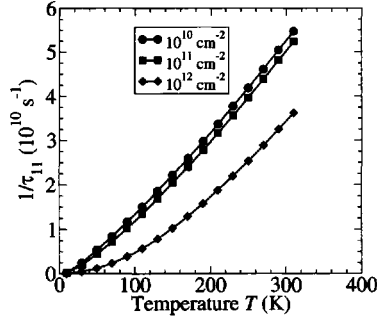


Figure 9.18 (a) The intrasubband acoustic deformation potential scattering rates due to phonon emission (solid lines) and phonon absorption (dashed lines) as a function of the total energy E_i^t of the carrier in the initial state (in this case the subband index $i = 1$) in a 100 Å GaAs infinitely deep quantum well with a carrier density of $10 \times 10^{10} \text{ cm}^{-2}$, and (b) the thermally averaged intrasubband scattering rates due to acoustic phonon emission again in the ground state of a 100 Å GaAs infinitely deep quantum well as a function of the temperature and for several different carrier densities.

tion (9.160) for acoustic deformation potential scattering, Fermi's Golden Rule gives:

$$\frac{1}{\tau_i} = \frac{2\pi}{\hbar} \sum_{\mathbf{k}_f} \frac{D_o^2 \hbar}{2\rho\omega_o} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \times \left| \int \int \psi_f^*(z) \frac{e^{i\mathbf{k}_f \cdot \mathbf{r}_{xy}}}{A^{\frac{1}{2}}} \sum_{\mathbf{K}_{xy}} \sum_{K_z} \frac{e^{-i\mathbf{K}_{xy} \cdot \mathbf{r}_{xy}}}{A^{\frac{1}{2}}} \frac{e^{-iK_z z}}{L^{\frac{1}{2}}} \psi_i(z) \frac{e^{-i\mathbf{k}_i \cdot \mathbf{r}_{xy}}}{A^{\frac{1}{2}}} dz d\mathbf{r}_{xy} \right|^2 \times \delta(E_f^t - E_i^t) \quad (9.187)$$

The derivation can be taken forward in exactly the same manner as for acoustic deformation potential scattering, except without the substitution for ω_s as given in equation (9.166), also assuming parabolic subbands then in analogy to equation (9.179) in Section 9.9:

$$\frac{1}{\tau_i} = \frac{D_o^2}{2\rho\omega_o(2\pi)^2} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \int \int (G_{if}(K_z))^2 \times \delta \left(E_f + \frac{\hbar^2}{2m^*} (\mathbf{k}_i^2 + \mathbf{K}_{xy}^2 + 2k_i K_{xy} \cos \phi) - E_i - \frac{\hbar^2 \mathbf{k}_i^2}{2m^*} \mp \hbar\omega_o \right) dK_z d\mathbf{K}_{xy} \quad (9.188)$$

Now the energies of phonons from the optical branch are not small hence the phonon energy cannot be approximated to zero and the collision considered elastic, rather the term $\mp \hbar\omega_o$ has to be retained. Labelling $E_f - E_i \mp \hbar\omega_o$ as Δ , then

$$\frac{1}{\tau_i} = \frac{D_o^2}{2\rho\omega_o(2\pi)^2} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \int \int (G_{if}(K_z))^2 \times \delta \left(\frac{\hbar^2 K_{xy}^2}{2m^*} + \frac{\hbar^2 k_i K_{xy} \cos \phi}{m^*} + \Delta \right) dK_z d\mathbf{K}_{xy} \quad (9.189)$$

and taking the factor $\hbar^2/(2m^*)$ out of the δ -function then:

$$\frac{1}{\tau_i} = \frac{D_o^2 m^*}{\rho\omega_o(2\pi)^2 \hbar^2} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \int \int (G_{if}(K_z))^2 \times \delta \left(K_{xy}^2 + 2k_i K_{xy} \cos \phi + \frac{2m^* \Delta}{\hbar^2} \right) dK_z d\mathbf{K}_{xy} \quad (9.190)$$

Transforming the integral over the in-plane phonon wave vector \mathbf{K}_{xy} into plane polar coordinates then:

$$\frac{1}{\tau_i} = \frac{D_o^2 m^*}{\rho\omega_o(2\pi)^2 \hbar^2} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \int \int_0^{2\pi} \int (G_{if}(K_z))^2 \times \delta \left(K_{xy}^2 + 2k_i K_{xy} \cos \phi + \frac{2m^* \Delta}{\hbar^2} \right) K_{xy} dK_{xy} d\phi dK_z \quad (9.191)$$

Again following previous procedures, consider the integral over K_{xy} . There is only a contribution to this integral when the argument within the δ -function is zero. Now clearly solutions for K_{xy} must be positive (since K_{xy} is the *length* of the in-plane phonon wave vector), hence the argument of the δ -function can be factorised as $(K_{xy} - \alpha_1)(K_{xy} - \alpha_2)$, where again the roots are given by:

$$\alpha_{1,2} = -k_i \cos \phi \pm \sqrt{k_i^2 \cos^2 \phi - \frac{2m^* \Delta}{\hbar^2}} \quad (9.192)$$

and again $\alpha_1 > \alpha_2$. Equation (9.191) can therefore be written:

$$\frac{1}{\tau_i} = \frac{D_o^2 m^*}{\rho\omega_o(2\pi)^2 \hbar^2} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \int \int_0^{2\pi} \int (G_{if}(K_z))^2 \times \delta((K_{xy} - \alpha_1)(K_{xy} - \alpha_2)) K_{xy} dK_{xy} d\phi dK_z \quad (9.193)$$

Again following the same procedure as for acoustic deformation potential scattering then:

$$\frac{1}{\tau_i} = \frac{D_o^2 m^*}{\rho\omega_o(2\pi)^2 \hbar^2} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \int_0^\infty \int_0^{2\pi} \int_0^\infty (G_{if}(K_z))^2$$

$$\times \left[\frac{1}{|K_{xy} - \alpha_2|} \delta(K_{xy} - \alpha_1) + \frac{1}{|K_{xy} - \alpha_1|} \delta(K_{xy} - \alpha_2) \right] K_{xy} dK_{xy} d\phi dK_z \quad (9.194)$$

and now performing the integration over K_{xy} then:

$$\begin{aligned} \frac{1}{\tau_1} = & \frac{D_o^2 m^*}{\rho \omega_o (2\pi)^2 \hbar^2} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \int_0^\infty \int_0^{2\pi} (G_{if}(K_z))^2 \\ & \times \left[\frac{\alpha_1}{|\alpha_1 - \alpha_2|} + \frac{\alpha_2}{|\alpha_2 - \alpha_1|} \right] d\phi dK_z \end{aligned} \quad (9.195)$$

and with $\alpha_1 > \alpha_2$ then:

$$\frac{1}{\tau_1} = \frac{D_o^2 m^*}{\rho \omega_o (2\pi)^2 \hbar^2} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \int_0^\infty \int_0^{2\pi} (G_{if}(K_z))^2 \frac{\alpha_1 + \alpha_2}{\alpha_1 - \alpha_2} d\phi dK_z \quad (9.196)$$

But:

$$\alpha_1 + \alpha_2 = -2k_i \cos \phi \quad \text{and} \quad \alpha_1 - \alpha_2 = 2\sqrt{k_i^2 \cos^2 \phi - \frac{2m^* \Delta}{\hbar^2}} \quad (9.197)$$

Therefore:

$$\begin{aligned} \frac{1}{\tau_1} = & -\frac{D_o^2 m^*}{\rho \omega_o (2\pi)^2 \hbar^2} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \\ & \times \int_0^\infty \int_0^{2\pi} (G_{if}(K_z))^2 \frac{k_i \cos \phi}{\sqrt{k_i^2 \cos^2 \phi - \frac{2m^* \Delta}{\hbar^2}}} d\phi dK_z \end{aligned} \quad (9.198)$$

and as these integrals are independent of each other then finally the optical deformation scattering rate for a carrier with an initial wave vector k_i into all final states of a given subband is given by:

$$\begin{aligned} \frac{1}{\tau_1} = & -\frac{D_o^2 m^*}{\rho \omega_o (2\pi)^2 \hbar^2} \left(N_0 + \frac{1}{2} \mp \frac{1}{2} \right) \\ & \times \int_0^\infty (G_{if}(K_z))^2 dK_z \int_0^{2\pi} \frac{k_i \cos \phi}{\sqrt{k_i^2 \cos^2 \phi - \frac{2m^* \Delta}{\hbar^2}}} d\phi \end{aligned} \quad (9.199)$$

9.12 CONFINED AND INTERFACE PHONON MODES

Forming quantum wells or superlattices clearly changes the electronic energy levels of a crystal from what they are in an infinite bulk crystal, which is the main subject of this book. In fact, all of the crystal properties are changed to a greater or lesser extent. Perhaps of secondary importance to the effect on the electronic energy levels, and the

subsequent changes that this induces in scattering rates, exciton energies, impurity energies, etc. are the fundamental changes introduced to the *phonon modes*. It can be appreciated that such changes are likely as the bulk LO phonon in GaAs has an energy of around 36 meV while in AlAs it is closer to 50 meV. Hence, forming a superlattice with alternating layers of GaAs and AlAs is going to have some effect on the phonon energies.

Various models have been put forward to account for this change in symmetry. At one end of the scale, some models consider each semiconductor layer as a continuum of material with macroscopic-like properties, namely the *Dielectric Continuum model*, (see for example [178]), or the *Hydrodynamic model*, (see for example [182]). Alternative approaches have considered the allowed vibrational modes calculated directly from the viewpoint of individual atomic potentials (see for example, [183]). See Adachi [14], p. 70, for an introduction.

Such improved models for phonons in heterostructures lead to modes which are confined to the individual semiconductor layers—*confined modes*—while some propagate along boundaries between the layers—the so-called *interface modes*. Recent work has shown that while the electron–phonon scattering rates from these individual modes are quite different, the *total* rate from all of the modes collectively is quite similar to that from bulk phonons (see Kinsler *et al.* [184]). However, this is still a very active area of research and future developments will need to be monitored.

9.13 CARRIER–CARRIER SCATTERING

Fermi's Golden Rule describes the lifetime of a particle in a particular state with respect to scattering by a *time-varying potential*. For phonon scattering, this harmonic potential is derived from the phonon wave function, which is itself a travelling wave. For the case of one carrier scattering against another due to the Coulomb potential, there appears to be no time dependency. The *Born approximation* is often cited in the literature when discussing carrier–carrier scattering; this is just a way of working scattering from a constant potential into Fermi's Golden Rule. This is achieved by considering that the perturbing potential is 'switched on' only when the particle reaches the same proximity. For an excellent introduction to the Born approximation, see Liboff ([185], p. 621).

Therefore, the perturbing potential appearing in Fermi's Golden Rule for the interaction of two isolated carriers is the Coulombic interaction, i.e.

$$\tilde{\mathcal{H}} = \frac{e^2}{4\pi\epsilon r} \quad (9.200)$$

where $\epsilon = \epsilon_r \epsilon_0$ is the permittivity of the material and r is the separation of the electrons. Now the initial and final states, $|i\rangle$ and $|f\rangle$, respectively, of the *system* both consist of two electron (or hole) wave functions, as carrier–carrier scattering is a two-body problem, and thus there is a much greater variety of scattering mechanisms possible than in the essentially one-body problem encountered in phonon scattering. Fig. 9.19

illustrates all of the possible mechanisms in a two-level system, where at least one of the carriers changes its subband, these are usually referred to as *intersubband* transitions. However, the distinction now is not quite so clear (see below). The central diagram in Fig. 9.19 illustrates the symmetric intersubband event, '22→11' which moves two carriers down a level. The left and right figures show *Auger-type* intersubband scattering, where one carrier relaxes down to a lower subband, giving its excess energy to another carrier which remains within its original subband

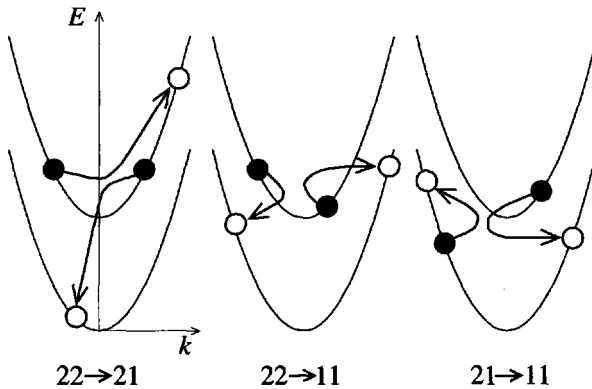


Figure 9.19 Illustration of various intersubband carrier-carrier scattering mechanisms in a two-level system

In addition to the above, there are also scattering events where the number of carries in each subband does not change. Some of these are illustrated schematically in Fig. 9.20. Clearly the first of these, '22→22' is an intrasubband event; however, the second and third events are more difficult to categorise precisely because although the number of carriers in each subband remain the same, the interaction itself is between carriers in different subbands. Pauli exclusion prevents carriers with the same spin occupying the same region of space, which therefore lowers their probability of scattering; in this work attention will be focused on collisions between particles with anti-parallel spins. Such considerations of spin-dependent scattering are often referred to by the term *exchange* [186]. Given that there are four possible carrier states involved, then in a N -level system there are 4^N different scattering events. In this two-level system these are as follows: 11-11, 11-12, 11-21, 11-22, 12-11, 12-12, 12-21, 12-22, 21-11, 21-12, 21-21, 21-22, 22-11, 22-12, 22-21, and 22-22. Note that completely different events of the type 'ij-fj' are possible in quantum wells with three or more subbands, and interactions of this type have been shown to be important in optically pumped intersubband lasers [186].

Therefore, taking a heterostructure wave function, of the form shown in equation (9.80), then the matrix element in Fermi's Golden Rule (equation (9.1)) becomes:

$$\langle f | \tilde{\mathcal{H}} | i \rangle =$$

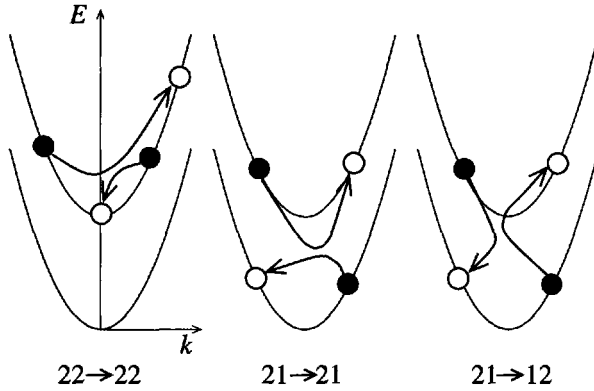


Figure 9.20 Illustration of various 'intrasubband' carrier-carrier scattering mechanisms in a two-level system

$$\left\langle \psi_f(z) \frac{e^{-i\mathbf{k}_f \cdot \mathbf{r}_{xy}}}{\sqrt{A}} \psi_g(z') \frac{e^{-i\mathbf{k}_g \cdot \mathbf{r}'_{xy}}}{\sqrt{A}} \left| \frac{e^2}{4\pi\epsilon r} \right| \psi_i(z) \frac{e^{-i\mathbf{k}_i \cdot \mathbf{r}_{xy}}}{\sqrt{A}} \psi_j(z') \frac{e^{-i\mathbf{k}_j \cdot \mathbf{r}'_{xy}}}{\sqrt{A}} \right\rangle \quad (9.201)$$

where the subband indices of the initial states are labelled 'i' and 'j' and those of the final states 'f' and 'g'. The decoupled form of the wave functions, with a component of the motion confined along the z -axis and an in-plane (x - y) travelling wave, suggests that the integrals should be evaluated across the plane and along the growth axis, and that the separation of the carriers be expressed as:

$$r = \sqrt{|\mathbf{r}_{xy} - \mathbf{r}'_{xy}|^2 + (z - z')^2} \quad (9.202)$$

Therefore obtain:

$$\begin{aligned} \langle f | \tilde{\mathcal{H}} | i \rangle &= \frac{e^2}{4\pi\epsilon A^2} \int_{-\infty}^{+\infty} \int_{-\infty}^{+\infty} \int \int \psi_f^*(z) \psi_g^*(z') \psi_i(z) \psi_j(z') \\ &\times \frac{e^{-i(\mathbf{k}_i \cdot \mathbf{r}_{xy} + \mathbf{k}_j \cdot \mathbf{r}'_{xy})} e^{i(\mathbf{k}_f \cdot \mathbf{r}_{xy} + \mathbf{k}_g \cdot \mathbf{r}'_{xy})}}{\sqrt{|\mathbf{r}_{xy} - \mathbf{r}'_{xy}|^2 + (z - z')^2}} d\mathbf{r}'_{xy} d\mathbf{r}_{xy} dz' dz \end{aligned} \quad (9.203)$$

Goodnick and Lugli [187] (later re-iterated by Smet *et al.* [188]), followed earlier methods for carrier-carrier scattering in bulk (see for example, Ziman [189], p. 170 and Takenaka *et al* [190]), and took the two-dimensional Fourier Transform of the Coulombic potential to give

$$\langle f | \tilde{\mathcal{H}} | i \rangle = \frac{2\pi e^2}{4\pi\epsilon A q_{xy}} A_{ijfg}(q_{xy}) \delta(\mathbf{k}_f + \mathbf{k}_g - \mathbf{k}_i - \mathbf{k}_j) \quad (9.204)$$

where $q_{xy} = |\mathbf{k}_i - \mathbf{k}_f|$, and A_{ijfg} is a form factor, i.e.

$$A_{ijfg} = \int_{-\infty}^{+\infty} \int_{-\infty}^{+\infty} \psi_i(z) \psi_j(z') \psi_f^*(z) \psi_g^*(z') e^{-q_{xy}|z-z'|} dz' dz \quad (9.205)$$

Using the form for the matrix element shown in equation (9.204) and then substituting directly into Fermi's Golden Rule (equation (9.1)) gives the lifetime of a carrier in subband 'i' as follows:

$$\frac{1}{\tau_i} = \frac{2\pi}{\hbar} \sum_{f,g} \left| \frac{2\pi e^2}{4\pi\epsilon A q_{xy}} A_{ijfg}(q_{xy}) \right|^2 \delta(\mathbf{k}_f + \mathbf{k}_g - \mathbf{k}_i - \mathbf{k}_j) \delta(E_f^t + E_g^t - E_i^t - E_j^t) \quad (9.206)$$

Converting the summations over both final-state wave vectors into integrals introduces a factor of $L/(2\pi)$ per dimension, thus giving a factor of $A^2/(2\pi)^4$ in total (where the general area $A = L^2$) (see Section 2.3), therefore:

$$\begin{aligned} \frac{1}{\tau_i} = \frac{2\pi}{\hbar} \frac{A^2}{(2\pi)^4} \int \int \left| \frac{2\pi e^2}{4\pi\epsilon A q_{xy}} A_{ijfg}(q_{xy}) \right|^2 \delta(\mathbf{k}_f + \mathbf{k}_g - \mathbf{k}_i - \mathbf{k}_j) \\ \times \delta(E_f^t + E_g^t - E_i^t - E_j^t) d\mathbf{k}_g d\mathbf{k}_f \end{aligned} \quad (9.207)$$

and thus:

$$\begin{aligned} \frac{1}{\tau_i} = \frac{e^4}{2\pi\hbar(4\pi\epsilon)^2} \int \int \frac{|A_{ijfg}(q_{xy})|^2}{q_{xy}^2} \delta(\mathbf{k}_f + \mathbf{k}_g - \mathbf{k}_i - \mathbf{k}_j) \\ \times \delta(E_f^t + E_g^t - E_i^t - E_j^t) d\mathbf{k}_g d\mathbf{k}_f \end{aligned} \quad (9.208)$$

Integrating over all of the states of the second carrier (given by \mathbf{k}_j) and introducing Fermi-Dirac distribution functions to account for state occupancy, then obtain:

$$\begin{aligned} \frac{1}{\tau_i} = \frac{e^4}{2\pi\hbar(4\pi\epsilon)^2} \int \int \int \frac{|A_{ijfg}(q_{xy})|^2}{q_{xy}^2} f_j^{\text{FD}}(\mathbf{k}_j) [1 - f_i^{\text{FD}}(\mathbf{k}_f)] [1 - f_g^{\text{FD}}(\mathbf{k}_g)] \\ \times \delta(\mathbf{k}_f + \mathbf{k}_g - \mathbf{k}_i - \mathbf{k}_j) \delta(E_f^t + E_g^t - E_i^t - E_j^t) d\mathbf{k}_g d\mathbf{k}_f d\mathbf{k}_j \end{aligned} \quad (9.209)$$

which is equation (49) in Smet *et al.* [188]. Following their notation, collect the distribution functions together and label them as $P_{j,f,g}(\mathbf{k}_j, \mathbf{k}_f, \mathbf{k}_g)$. The first δ -function summarizes in-plane momentum conservation and limits the integral over \mathbf{k}_g to a contribution when $\mathbf{k}_g = \mathbf{k}_i + \mathbf{k}_j - \mathbf{k}_f$. In addition, the total energy of the carriers, E_i^t , etc. are equal to the energy of the relevant subband minima, E_i say, plus the in-plane kinetic energy; thus:

$$\begin{aligned} \frac{1}{\tau_i} = \frac{e^4}{2\pi\hbar(4\pi\epsilon)^2} \int \int \frac{|A_{ijfg}(q_{xy})|^2}{q_{xy}^2} P_{j,f,g}(\mathbf{k}_j, \mathbf{k}_f, \mathbf{k}_g) \\ \times \delta \left(E_f + \frac{\hbar^2 \mathbf{k}_f^2}{2m^*} + E_g + \frac{\hbar^2 \mathbf{k}_g^2}{2m^*} - E_i + \frac{\hbar^2 \mathbf{k}_i^2}{2m^*} - E_j + \frac{\hbar^2 \mathbf{k}_j^2}{2m^*} \right) d\mathbf{k}_f d\mathbf{k}_j \end{aligned} \quad (9.210)$$

and therefore:

$$\frac{1}{\tau_i} = \frac{m^* e^4}{\pi\hbar^3(4\pi\epsilon)^2} \int \int \frac{|A_{ijfg}(q_{xy})|^2}{q_{xy}^2} P_{j,f,g}(\mathbf{k}_j, \mathbf{k}_f, \mathbf{k}_g)$$

$$\times \delta \left(\mathbf{k}_f^2 + \mathbf{k}_g^2 - \mathbf{k}_i^2 - \mathbf{k}_j^2 + \frac{2m^*}{\hbar^2} (E_f + E_g - E_i - E_j) \right) d\mathbf{k}_f d\mathbf{k}_j \quad (9.211)$$

where \mathbf{k}_g is known in terms of the other three wave vectors. It is the assumption of parabolic subbands in this last step that will be the limiting factor for the application of this method to hole-hole scattering—a point mentioned earlier in the context of the carrier-LO phonon scattering rate derivation. Now equation (9.211) represents the scattering rate of a carrier at a particular wave vector \mathbf{k}_i averaged over all of the other initial particle states \mathbf{k}_j , and hence the only unknown in this remaining δ -function is the wave vector \mathbf{k}_f . Contributions to the integral over \mathbf{k}_f occur when the argument of the δ -function is zero, and indeed given the form for this argument, it is clear that the solutions for \mathbf{k}_f map out an ellipse.

The standard procedure [187, 188] is then to introduce relative wave vectors:

$$\mathbf{k}_{ij} = \mathbf{k}_j - \mathbf{k}_i \quad (9.212)$$

$$\mathbf{k}_{fg} = \mathbf{k}_g - \mathbf{k}_f \quad (9.213)$$

and replace the integration over \mathbf{k}_f in equation (9.211) with an integration over \mathbf{k}_{fg} . Since:

$$\mathbf{k}_f = \mathbf{k}_i + \mathbf{k}_j - \mathbf{k}_g = \mathbf{k}_i + \mathbf{k}_j - \mathbf{k}_{fg} - \mathbf{k}_f \quad (9.214)$$

i.e.

$$\mathbf{k}_f = \frac{1}{2} (\mathbf{k}_i + \mathbf{k}_j - \mathbf{k}_{fg}) \quad (9.215)$$

It follows that:

$$d(\mathbf{k}_f)_x = -\frac{1}{2} d(\mathbf{k}_{fg})_x \quad \text{and} \quad d(\mathbf{k}_f)_y = -\frac{1}{2} d(\mathbf{k}_{fg})_y \quad (9.216)$$

so:

$$d\mathbf{k}_f = \frac{1}{4} d\mathbf{k}_{fg} \quad (9.217)$$

as pointed out by Moško [191], which should then be substituted into equation (9.211). In order to perform the integration over \mathbf{k}_{fg} it is converted to plane polar coordinates with $d\mathbf{k}_{fg} = k_{fg} dk_{fg} d\theta$, where θ is an angle measured from \mathbf{k}_{ij} and the trajectory in the k_{fg} - θ plane is deduced from the condition that the argument of the δ -function in equation (9.211) must be zero.

The conservation of momentum diagram therefore looks like Fig. 9.21, where:

$$\mathbf{k}_{\text{sum}} = \mathbf{k}_i + \mathbf{k}_j = \mathbf{k}_f + \mathbf{k}_g \quad (9.218)$$

In this work, the occupancy of the final states will be assumed to be small, such that the distribution functions dependent upon \mathbf{k}_f and \mathbf{k}_g can be ignored. The effect of putting $f_f^{\text{FD}}(\mathbf{k}_f)$ and $f_g^{\text{FD}}(\mathbf{k}_g)$ to zero can be seen to place an upper limit on the scattering rate. The physical interpretation of this approximation is that *final-state blocking* is ignored, i.e. the process by which a scattering event is prevented because the required final state is already occupied. This is a common simplification [192]

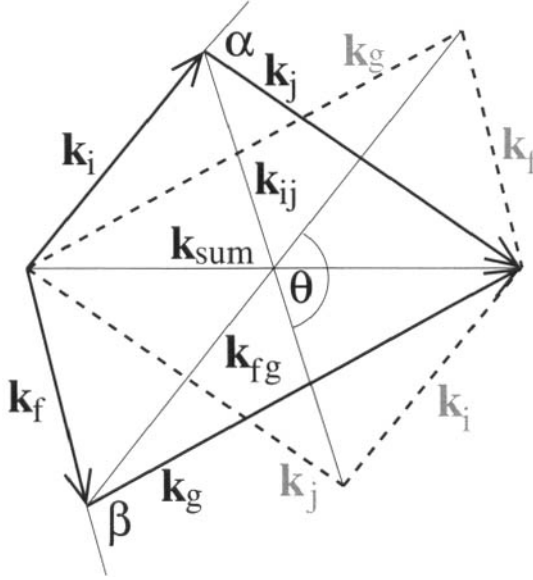


Figure 9.21 The conservation of momentum diagram for the two-body electron–electron scattering event

which allows the particular values of the final-state wave vectors to remain unknown. The approximation that this introduces is ‘good’ for the typical carrier densities encountered in devices based on quantum well heterostructures [193]. Recently, a generalisation of this approach has been put forward which allows the specific values of the final carrier wave vectors to be calculated, thus avoiding this approximation; this method is also an example of making a different choice for the angle θ [186].

Using the notation $k = |\mathbf{k}|$, and applying the cosine rule to Fig. 9.21, then:

$$k_{\text{sum}}^2 = k_i^2 + k_j^2 + 2k_i k_j \cos \alpha \quad (9.219)$$

$$k_{\text{sum}}^2 = k_f^2 + k_g^2 + 2k_f k_g \cos \beta \quad (9.220)$$

and also:

$$k_{ij}^2 = k_i^2 + k_j^2 - 2k_i k_j \cos \alpha \quad (9.221)$$

$$k_{fg}^2 = k_f^2 + k_g^2 - 2k_f k_g \cos \beta \quad (9.222)$$

Summing equation (9.219) with equations (9.221) and (9.220) with 9.222 gives:

$$k_{\text{sum}}^2 + k_{ij}^2 = 2k_i^2 + 2k_j^2 \quad (9.223)$$

$$k_{\text{sum}}^2 + k_{fg}^2 = 2k_f^2 + 2k_g^2 \quad (9.224)$$

and by eliminating k_{sum}^2 gives:

$$k_{fg}^2 = k_{ij}^2 + 2(k_f^2 + k_g^2 - k_i^2 - k_j^2) \quad (9.225)$$

Energy conservation has already been considered earlier within the δ -function, i.e.

$$E_f + \frac{\hbar^2 \mathbf{k}_f^2}{2m^*} + E_g + \frac{\hbar^2 \mathbf{k}_g^2}{2m^*} = E_i + \frac{\hbar^2 \mathbf{k}_i^2}{2m^*} + E_j + \frac{\hbar^2 \mathbf{k}_j^2}{2m^*} \quad (9.226)$$

thus giving:

$$k_f^2 + k_g^2 - k_i^2 - k_j^2 = \frac{2m^*}{\hbar^2} (E_i + E_j - E_f - E_g) \quad (9.227)$$

Substituting into equation (9.225) then gives:

$$k_{fg}^2 = k_{ij}^2 + \frac{4m^*}{\hbar^2} (E_i + E_j - E_f - E_g) = k_{ij}^2 + \Delta k_0^2, \quad \text{say} \quad (9.228)$$

where the notation of Smet *et al.* has been employed.

Consider the entity \mathbf{q}_{xy} , for which q_{xy} is the magnitude. This vector $\mathbf{q}_{xy} = \mathbf{k}_i - \mathbf{k}_f$ is illustrated in the equivalent diagram (Fig. 9.22). In addition, this figure emphasizes the difference between the relative wave vectors \mathbf{k}_{ij} and \mathbf{k}_{fg} , which turns out to be equal to $2\mathbf{q}_{xy}$, and therefore:

$$\mathbf{q}_{xy} = \frac{\mathbf{k}_{ij} - \mathbf{k}_{fg}}{2} \quad (9.229)$$

With an angle of θ between them, the cosine rule then gives:

$$(2q_{xy})^2 = k_{ij}^2 + k_{fg}^2 - 2k_{ij}k_{fg} \cos \theta \quad (9.230)$$

As the aim is to eliminate the final-state wave vectors from the definition of q_{xy} , then substitute for k_{fg} from equation (9.228), thus giving:

$$(2q_{xy})^2 = 2k_{ij}^2 + \Delta k_0^2 - 2k_{ij} \sqrt{k_{ij}^2 + \Delta k_0^2} \cos \theta \quad (9.231)$$

Summarizing then, the expression for the scattering rate of a carrier of a particular wave vector \mathbf{k}_i with another carrier, is given by:

$$\frac{1}{\tau_i} = \frac{m^* e^4}{4\pi \hbar^3 (4\pi\epsilon)^2} \int \int_0^{2\pi} \frac{|A_{ijfg}(q_{xy})|^2}{q_{xy}^2} P_{jfg}(\mathbf{k}_j, \mathbf{k}_f, \mathbf{k}_g) \, d\theta \, d\mathbf{k}_j \quad (9.232)$$

which is equation (51) in Smet *et al.* [188], but with an additional factor of 4 in the denominator [191]. In this present treatise the factor $P_{jfg}(\mathbf{k}_j, \mathbf{k}_f, \mathbf{k}_g)$ has been assumed to be dependent upon the initial-state distribution only, i.e. $P_{jfg}(\mathbf{k}_j, \mathbf{k}_f, \mathbf{k}_g) = f_j^{\text{FD}}(\mathbf{k}_j)$, and q_{xy} is given by equation (9.231). The integration over the vector \mathbf{k}_j can be performed by effectively switching to plane polar coordinates, and integrating along the length k_{ij} and around the angle α between \mathbf{k}_i and \mathbf{k}_j .

With this change in the integral, equation (9.232) then becomes:

$$\frac{1}{\tau_i} = \frac{m^* e^4}{4\pi \hbar^3 (4\pi\epsilon)^2} \int \int_0^{2\pi} \int_0^{2\pi} \frac{|A_{ijfg}(q_{xy})|^2}{q_{xy}^2} P_j(\mathbf{k}_j) \, d\theta \, d\alpha \, k_j \, dk_j \quad (9.233)$$

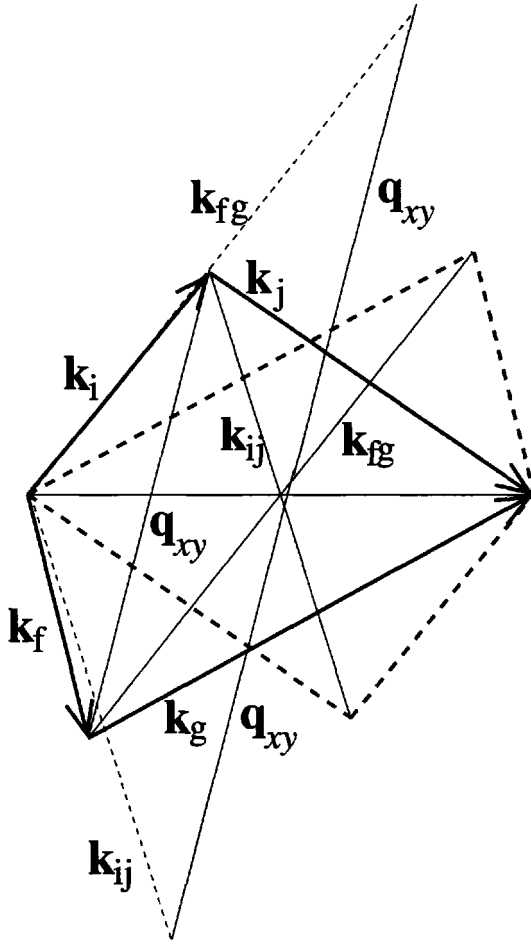


Figure 9.22 The conservation of momentum diagram for the two-body electron-electron scattering event

where q_{xy} is given by equation (9.231), Δk_0^2 is given by equation (9.228), and from Fig. 9.23:

$$k_{ij}^2 = k_i^2 + k_j^2 - 2k_i k_j \cos \alpha \quad (9.234)$$

As in the case of phonon scattering, the limits of integration of the wave vector are taken from zero (the subband minima) to that corresponding to the barrier height of the heterostructure in question.

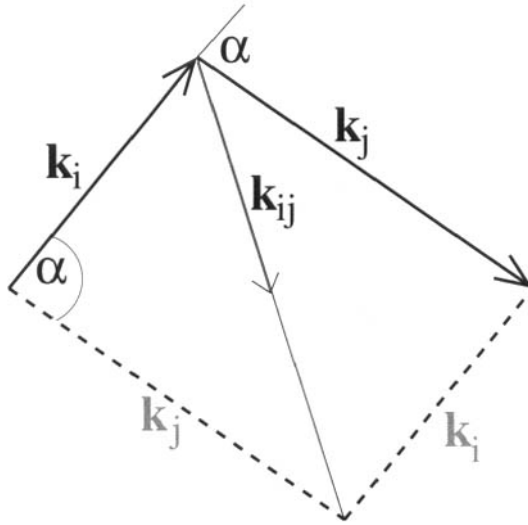


Figure 9.23 The relationship between the two initial carrier states

9.14 ADDITION OF SCREENING

So far, the theoretical treatise has only considered carrier-carrier scattering of two carriers in an empty environment. In semiconductor heterostructures, carrier-carrier scattering occurs because the system is doped, and therefore by definition there will be many carriers. In such instances, the force between any two carriers is not just the bare Coulombic repulsion, as the other mobile carriers are able to respond to any change in the electrostatic field, with the result being a reduction in the probability of scattering; the other carriers are said to *screen* the interaction.

One of the simplest models for screening [188] considers only the carriers within the same subband as the initial carrier state; it then proceeds by replacing the dielectric constant ϵ with one which is dependent upon the relative wave vector q_{xy} , i.e.

$$\epsilon_{sc} = 1 + \frac{2\pi e^2}{(4\pi\epsilon)q_{xy}} \Pi_{ii}(q_{xy}) A_{iii}(q_{xy}) \quad (9.235)$$

One consequence of this replacement is that the $q_{xy} = 0$ pole is removed from the scattering rate. Combining the results of Ando *et al.* [194] and using a Heaviside unit step function for convenience, the polarization factor at absolute zero is then given by:

$$\Pi_{ii}(q_{xy}) = \frac{m^*}{\pi\hbar^2} \left[1 - \Theta(q_{xy} - 2k_F) \sqrt{1 - \left(\frac{2k_F}{q_{xy}} \right)^2} \right] \quad (9.236)$$

where k_F is the two-dimensional Fermi wave vector [194] applicable to the initial state 'i' and is defined only at absolute zero, i.e.

$$k_F = \sqrt{\frac{2\pi N_i}{g}} \quad (9.237)$$

(see [1], p. 36, for example, for the three-dimensional equivalent). The factor g accounts for any degeneracy which can lead to multiple subband valleys; in GaAs, this is just 1. Maldague [195] showed that under certain approximations, this polarizability can be generalised to any temperature with:

$$\Pi_{ii}(q_{xy}, T) = \int_0^\infty \frac{\Pi_{ii}(q_{xy}, T=0)}{4kT \cosh^2((E_F - E)/(2kT))} dE \quad (9.238)$$

where E_F is the quasi-Fermi energy and E the minimum (called E_i here) of subband 'i' (see Section 2.4). This integral can actually be performed analytically[†]:

$$\Pi_{ii}(q_{xy}, T) = \frac{\Pi_{ii}(q_{xy}, 0)}{4kT} \int_0^\infty \frac{1}{\cosh^2((E_F - E)/(2kT))} dE \quad (9.239)$$

using the standard result $\int 1/\cosh^2 \theta d\theta = \tanh \theta$ then:

$$\Pi_{ii}(q_{xy}, T) = \frac{\Pi_{ii}(q_{xy}, 0)}{4kT} \left[\tanh \left(\frac{E_F - E}{2kT} \right) \right]_0^\infty \quad (9.240)$$

then:

$$\Pi_{ii}(q_{xy}, T) = \frac{\Pi_{ii}(q_{xy}, 0)}{2} \left\{ 1 + \tanh \left(\frac{E_F}{2kT} \right) \right\} \quad (9.241)$$

Studying this equation it can be seen that as $T \rightarrow 0$, $\Pi_{ii}(q_{xy}, T)$ recovers the low temperature limit. In the limit of high temperature it is exactly half this value.

Fig. 9.24 shows the results of a calculation of the 22–11 electron–electron scattering rate as a function of the electron energy of the initial state 'i', in an infinitely deep 400 Å wide GaAs quantum well. The figure compares the rates with and without the screening term at a temperature of 77 K and for carrier densities of $1 \times 10^{10} \text{ cm}^{-2}$ (left) followed by $100 \times 10^{10} \text{ cm}^{-2}$ (right) carriers in each level.

The first conclusion to be drawn is that the scattering rate increases as the energy of the initial electron in state 'i' decreases towards the subband minimum. In addition, the scattering rates themselves are nearly two orders of magnitude higher in the higher-carrier-density case illustrated on the right hand side of Fig. 9.24 than in the lower-density case illustrated on the left. Furthermore, it is apparent that screening reduces the scattering rate, and the higher the carrier density, then the larger the effect of screening, as might be expected.

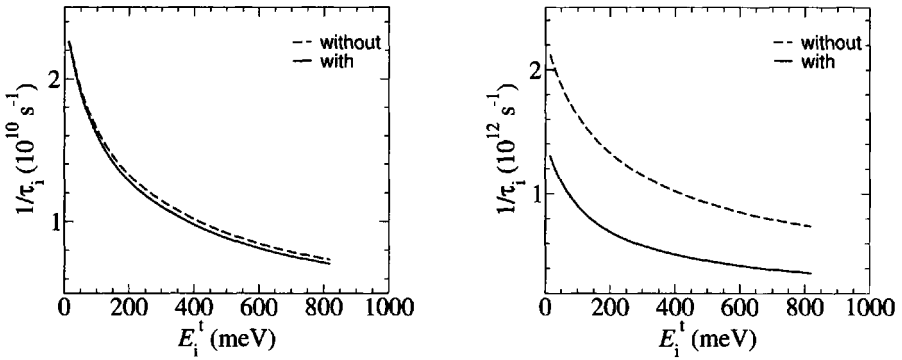


Figure 9.24 The electron–electron scattering rate as a function of the initial electron energy, with and without screening, for a carrier density of $1 \times 10^{10} \text{ cm}^{-2}$ (left) and $100 \times 10^{10} \text{ cm}^{-2}$ (right)

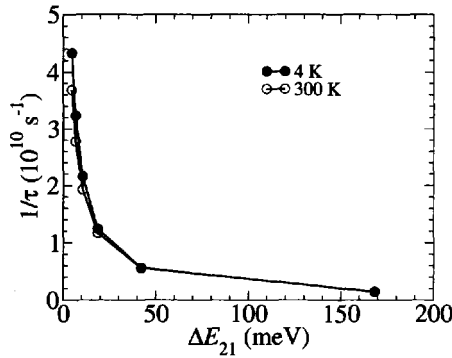


Figure 9.25 The mean 22–11 electron–electron scattering rate as a function of the subband separation, including screening and with a carrier density of $1 \times 10^{10} \text{ cm}^{-2}$ in each level

9.15 AVERAGING OVER AN INITIAL STATE POPULATION

The method documented so far gives the carrier–carrier scattering rate for a particular carrier energy averaged over another initial state distribution; as before for the carrier–

[‡]Thanks to Jim McTavish for pointing this out.

LO phonon case, it is usually more instructive to average this quantity over the Fermi–Dirac distribution of carriers in the initial state. This could be achieved by using the following mean:

$$\frac{1}{\tau} = \frac{\int \frac{1}{\tau_i} f_i^{\text{FD}}(E_i^k) dE_i^k}{\int f_i^{\text{FD}}(E_i^k) dE_i^k} \quad (9.242)$$

where the superscript k has been introduced to indicated that this is the in-plane kinetic energy associated with the wave vector \mathbf{k}_i . By using the result of equation (2.50) in Section 2.4, the denominator is then given by:

$$\int f_i^{\text{FD}}(E_i^k) dE_i^k = N_i \frac{\pi \hbar^2}{m^*} \quad (9.243)$$

As the carrier's in-plane energy is expressed in terms of a wave vector and:

$$E_i^k = \frac{\hbar^2 k_i^2}{2m^*}, \quad \text{then} \quad dE_i^k = \frac{\hbar^2 k_i}{m^*} dk_i \quad (9.244)$$

Hence the above mean can be expressed as:

$$\frac{1}{\tau} = \frac{\int \frac{1}{\tau_i} f_i^{\text{FD}}(k_i) k_i dk_i}{\pi N_i} \quad (9.245)$$

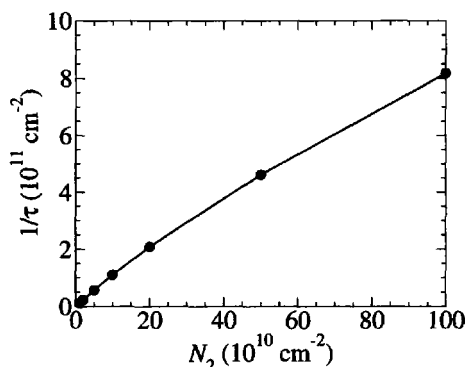


Figure 9.26 The mean 22–11 electron–electron scattering rate as a function of the carrier density in level $|2\rangle$, for a 300 Å infinitely deep GaAs quantum well at 300 K

Fig. 9.25 shows the electron–electron scattering rate averaged over the initial carrier distribution, as given by equation (9.245), for the 22–11 mechanism in a infinitely deep quantum well. The subband separation, $\Delta E_{21} = E_2 - E_1$, was varied by adjusting the quantum well width. It can be seen that the electron–electron scattering

rate increases as the subband separation decreases and that in this particular approach with the low carrier density of $1 \times 10^{10} \text{ cm}^{-2}$ the rate is only a weak function of temperature.

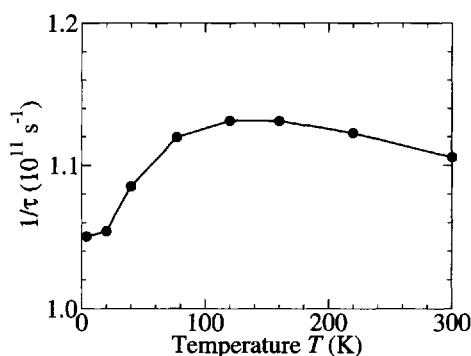


Figure 9.27 The mean 22–11 electron–electron scattering rate as a function of temperature, for a 300 Å infinitely deep GaAs quantum well, with a carrier density of $10 \times 10^{10} \text{ cm}^{-2}$

The effect of increasing carrier density is shown in Fig. 9.26. The range of carrier densities was chosen to span roughly those found in intersubband devices. It can be seen that with this model the intersubband scattering is almost proportional to the carrier density, although it must be noted that in this work any effects due to Pauli exclusion are being ignored—this took the form of assuming that the final-state populations were small, as discussed earlier. Inclusion of significant populations in a final-state could lead to final-state blocking being significant, and hence interfere with the proportionality, but at this stage of gaining an intuitive feel for carrier–carrier scattering, such investigations necessitate the introduction of yet more variables and unnecessary complication. As in the previous example, it was found that these intersubband scattering rates were quite insensitive to the temperature. This is substantiated further in Fig. 9.27, which shows the temperature dependence of the 22–11 intersubband scattering rate for $10 \times 10^{10} \text{ cm}^{-2}$ electrons in the same 300 Å infinitely deep quantum well. The rate peaks at around 10 % more than its low temperature value before decreasing towards room temperature. The temperature dependency within the model comes from the screening term as well as both of the initial-state distribution functions.

9.16 INTRASUBBAND VERSUS INTERSUBBAND

So far, only the results of calculations of intersubband carrier–carrier scattering rates have been presented, but as mentioned before intrasubband events are also possible.

The latter are characterised by no change in the numbers of carriers in each subband. Considering only the most simplistic mechanisms of the type 'ii-ii', then Fig. 9.28 plots the carrier density dependency and Fig. 9.29 the temperature dependency of the 22–22 electron–electron scattering rate, again for a typical 300 Å infinitely deep GaAs quantum well. These figures are the intrasubband equivalents of the intersubband cases of Figs 9.26 and 9.27, respectively, in the last section. Note that as all of the initial and final states are within the same subband, no dependence on an intersubband energy separation is required.

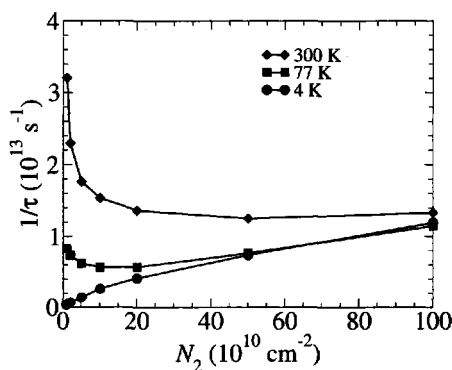


Figure 9.28 The mean 22–22 electron–electron scattering rate as a function of the carrier density, for a 300 Å infinitely deep GaAs quantum well

Therefore, comparing Fig. 9.28 with Fig. 9.26 it can be seen that the dependence of the scattering rate on the carrier density is quite different in this the intrasubband case. The ‘almost’ proportionality is only evident at the lowest of the three temperatures, and for 77 and 300 K there is a clear non-monotonic behaviour.

Fig. 9.29 shows the temperature dependence of this 22–22 rate; this is much stronger than in the intersubband case of Fig. 9.27 and indeed seems to have a linear dependence starting from the low-temperature value.

Overall, the intrasubband scattering rates are around one to two orders of magnitude higher than the closest intersubband equivalent. This is a general result which is observed in a range of semiconductor heterostructures. In fact, in this idealistic infinitely deep quantum well, the intersubband rates represent something of a maximum as the overlap of the carrier wave functions is complete, and hence the ratio of the intrasubband to intersubband rates deduced here is probably conservative.

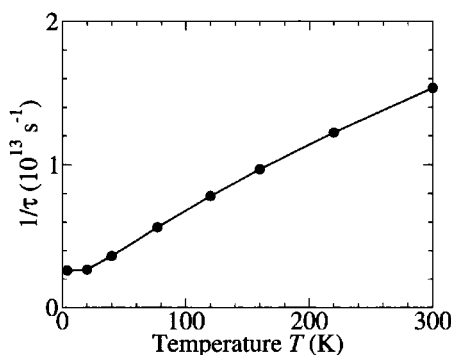


Figure 9.29 The mean 22–22 electron–electron scattering rate as a function of temperature, for a 300 Å infinitely deep GaAs quantum well, with a carrier density of $10 \times 10^{10} \text{ cm}^{-2}$

9.17 THERMALISED DISTRIBUTIONS

This last result, i.e. that the intrasubband scattering rates are much larger than their typical intersubband equivalents, is a very important result. Its implication is that given a quantum well or a quantum wire (basically any system which has carrier dispersion) with a number of subbands, then the carriers within each subband scatter much more rapidly than they do between the subbands. Thus, if the system is stimulated somehow, either by optical excitation or electrical injection of carriers, then the distributions of carriers within each subband will, in the first instance, reorganise themselves independently of each other, before intersubband scattering begins to redistribute the carriers between the levels. This justifies the initial assumption that the carrier populations in each subband can be represented by Fermi–Dirac distributions, although certainly with independent quasi-Fermi energies (see Section 2.4)—in this situation, the carriers are said to be *independently thermalised*. This is a common assumption in intersubband devices (see for example [196]). It is also possible that the temperature required to describe these distributions is not equal to the lattice temperature; such a scenario will either require a thorough analysis of the kinetics of the system or else a Monte Carlo simulation of the complete subband structure.

Should the means of excitation, i.e. the source of input energy, be removed from the quantum well system, then the carriers will relax down to a state where the populations can all be described by a single Fermi energy—this would represent thermal equilibrium. Thus, the previous case of an excited subband structure is often given the rather grand descriptive title of a *non-equilibrium carrier distribution*.

9.18 AUGER-TYPE INTERSUBBAND PROCESSES

The various carrier-carrier scattering rates were discussed earlier, and illustrated in Figs 9.19 and 9.20. Until now, the calculations themselves have only considered the 'symmetric' 22-11 process, and only within an infinitely deep quantum well. This particular heterostructure was just an illustrative example, and provided a way of reducing the number of material parameters that had to be specified. In order to calculate the scattering rates due to the 'asymmetric' 22-21 and 21-11 processes, it is necessary to base the calculations in a different quantum well system; for this purpose, a single 120 Å GaAs quantum well surrounded by $\text{Ga}_{0.8}\text{Al}_{0.2}\text{As}$ barriers was chosen. This is also an opportunity to reiterate and demonstrate that the carrier-carrier scattering rate derived is valid for any one-particle eigenstates, as calculated with the methods outlined in Chapters 2 and 3.

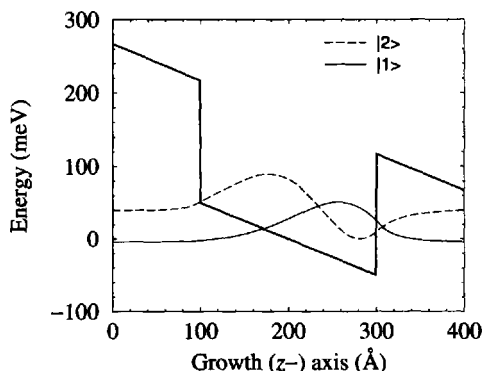


Figure 9.30 The potential profile and two lowest-energy eigenstates of a 120 Å GaAs quantum well surrounded by $\text{Ga}_{0.8}\text{Al}_{0.2}\text{As}$ barriers, at an applied electric field of 50 kVcm^{-1}

The asymmetric intersubband processes are often referred to as *Auger-type processes* because they resemble traditional interband Auger scattering, where one carrier is able to give up its potential energy to another carrier and hence relax down a level. These processes are forbidden in symmetric potentials, and the simplest way to break the symmetry of the above finite quantum well is to apply an electric field. Fig. 9.30 illustrates the potential profile and the wave functions of the ground $|1\rangle$ and first excited $|2\rangle$ states of the quantum well with the maximum applied electric field employed.

Fig. 9.31 shows the results of calculations of the mean scattering rates for all three possible intersubband mechanisms in this two-level system. The selection rules arise quite naturally in the formalism; hence, at zero field when the potential and hence the wave functions are symmetric (and anti-symmetric), only the symmetric 22-11 channel is allowed. However, as the electric field is increased and the symmetry of the potential broken, the asymmetric rates increase and the symmetric rate decreases. At the maximum value of the electric field employed, the Auger-type rates are a

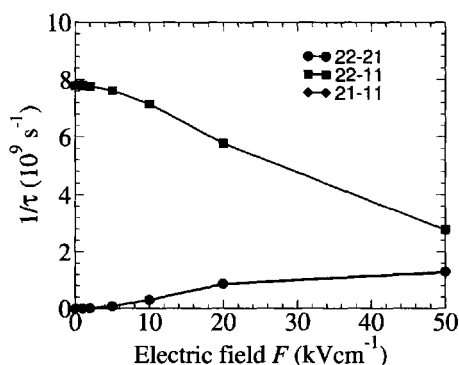


Figure 9.31 The mean intersubband electron–electron scattering rate as a function of the applied electric field for the quantum well in Fig. 9.30 with 1×10^{10} electrons per cm 2 in each level; note that the scattering rates for the 22–21 and 21–11 processes are almost identical

significant fraction of the total scattering rate, thus illustrating that their inclusion is essential in modelling such systems. Indeed, quantum well systems exist in which the Auger-type scattering rates dominate (see for example [197]). Note that the 22–11 rate moves two electrons down a level, whereas the two Auger-type rates only move one electron down at a time; this fact must be accounted for when solving subband population rate equations.

9.19 ASYMMETRIC INTRASUBBAND PROCESSES

For completeness, Fig. 9.32 shows the scattering rates of possible intrasubband mechanisms for the same finite quantum well as in the previous section. With the equal carrier densities employed in each level, the ‘asymmetric’ 21–21 and 21–12 intrasubband events are very nearly as large as the 11–11 intrasubband event. The implication of these results is that there will be a substantial interaction between the carrier distributions in the different subbands. Therefore even disregarding the transfer of carriers between levels with intersubband scattering, the distributions are coupled in some way and will not be independently thermalised, as often conjectured. The high scattering rate of the 21–21 and 21–12 events implies that there will be a redistribution of kinetic energy between the subbands without a redistribution of carriers, and thus it seems likely that although the electron temperature will not be equal to that of the lattice, the same electron temperature may be applicable to the entire multi-subband carrier distribution.

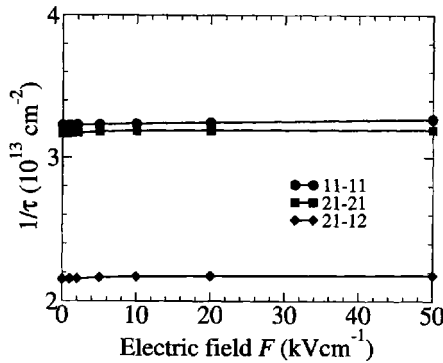


Figure 9.32 The mean intrasubband electron–electron scattering rate as a function of the applied electric field for the quantum well in Fig. 9.30 with 1×10^{10} electrons per cm^2 in each level

9.20 EMPIRICAL RELATIONSHIPS

Simple empirical relationships giving the dependency of some computationally complex quantity, such as a carrier scattering rate, on some simple material parameter, are often useful for device designers and experimentalists who need an intuitive feel for the underlying physics without having recourse to lengthy calculations. Earlier, a relationship was derived for the ratio of emission to absorption rates for carrier–LO phonon scattering (see equation (9.155)). As hinted at then, the subband separation and temperature dependencies of the 22–11 carrier–carrier scattering rate imply simple relationships (see Section 9.15). In particular, the hyperbolic temperature dependence suggests that:

$$\frac{1}{\tau} \propto \frac{1}{\Delta E_{21}} \quad (9.246)$$

and the linear carrier density dependence:

$$\frac{1}{\tau} \propto N \quad (9.247)$$

where in this simple example of a symmetric ‘ii–ff’ rate, the N refers to the carrier density in the initial state ‘i’.

The diagrams shown in Fig. 9.33 substantiate these postulated relationships, i.e. the graphs are quite linear and through the origin. Thus, given a quantum well system with a particular doping profile, and hence a specific carrier density, it is possible to answer questions such as ‘What happens to the intersubband electron–electron scattering rate if we increase the well width and hence halve the subband

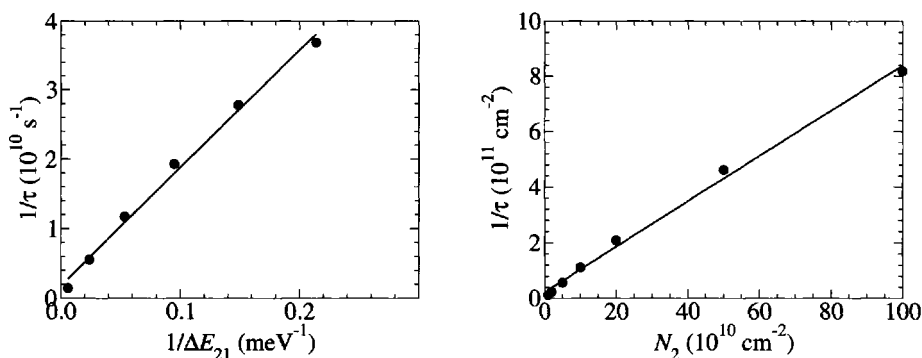


Figure 9.33 The mean 22–11 intersubband electron–electron scattering rate plotted against the reciprocal of the subband separation (left) and the carrier density (right)

separation?'. The answer is of course, that the rate will double. Similar questions can be asked about variations in doping profile and hence carrier densities for fixed layer thicknesses.

Both results *can* be combined as follows:

$$\frac{1}{\tau} \propto \frac{N}{\Delta E_{21}} \quad (9.248)$$

9.21 CARRIER–PHOTON SCATTERING

The emission (or absorption) of light by a charge carrier, whether an electron or a hole, is essentially a scattering event between an initial state ‘i’ and a final state ‘f’. The electromagnetic field is the time-dependent perturbation $\tilde{\mathcal{H}}$ which induces this event [185]. The transition rate from the initial electronic state $|i\rangle$ to the final state $|f\rangle$ is given by Fermi’s Golden Rule (see [198] for its application to light):

$$\frac{1}{\tau_i} = \frac{2\pi}{\hbar} \sum_f \left| \langle f | \tilde{\mathcal{H}} | i \rangle \right|^2 \delta(E_f^c - E_i^c \mp \hbar\omega) \quad (9.249)$$

where the superscript on the energies has been introduced to indicate that these are the total carrier energies, which contain both kinetic and potential energy components. The δ -function now explicitly contains the photon energy, with the upper sign of the \mp representing absorption and the lower emission. For example, for absorption the total initial energy is the sum of the carrier energy and the photon energy, i.e.

$$E_i^c + \hbar\omega = E_f^c, \quad \text{and hence} \quad E_f^c - E_i^c - \hbar\omega = 0 \quad (9.250)$$

and for emission:

$$E_i^c = E_f^c + \hbar\omega, \quad \text{and hence} \quad E_f^c - E_i^c + \hbar\omega = 0 \quad (9.251)$$

The photon momentum is assumed to be zero, and hence momentum conservation does not have to be considered with an additional δ -function, with the physical implication of this being that transitions on the band diagrams are always vertical whether they are between bands (interband) or within the same band (intraband or intersubband) (see Fig. 9.34).

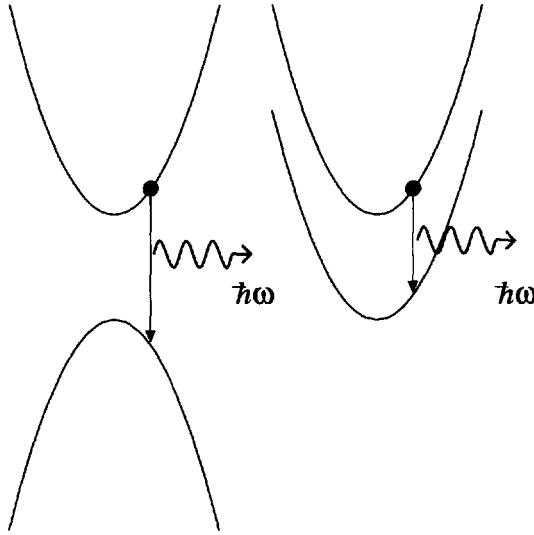


Figure 9.34 Carrier-photon scattering events result in vertical transitions; interband (left) and intraband (intersubband) (right)

Although a great deal has been achieved within the envelope function approximation, it is necessary now to reintroduce the rapidly varying component of the wave function, writing this as the product $u\psi$ (see Section 1.7). Then, as reiterated by Adachi ([14], p. 476) and Basu ([11], p. 299), Coon and Karunasiri [199] showed that the matrix element in equation 9.249 becomes:

$$\langle f | \tilde{\mathcal{H}} | i \rangle = \langle u_f | \tilde{\mathcal{H}} | u_i \rangle_{\text{cell}} \langle \psi_f | \psi_i \rangle + \langle u_f | u_i \rangle_{\text{cell}} \langle \psi_f | \tilde{\mathcal{H}} | \psi_i \rangle \quad (9.252)$$

In the case of interband transitions between the conduction and valence band, the second term gives zero since the Bloch functions u_f and u_i at the same point in the Brillouin zone, in two different bands, are orthogonal, i.e.

$$\langle u_f | u_i \rangle_{\text{cell}} = \int_{\text{cell}} u_f(\mathbf{r}) u_i(\mathbf{r}) \, d\mathbf{r} = 0 \quad (9.253)$$

and therefore:

$$\langle f | \tilde{\mathcal{H}} | i \rangle = \langle u_f | \tilde{\mathcal{H}} | u_i \rangle_{\text{cell}} \langle \psi_f | \psi_i \rangle \quad (9.254)$$

Hence the envelope function overlap integral, $\langle \psi_f | \psi_i \rangle$, determines which transitions are allowed and which are forbidden, which was the result quoted earlier in Section 6.8 during the discussion of excitons.

Radiative interband transitions are, of course, very important in both light sources (see for example [198]) and detectors (see for example works such as [200–202]). This mechanism gives access to a frequency range from the near-infrared [203] to the blue, depending on the material system employed. For a review of wide-bandgap III–V emitters and lasers, see Nakamura [13], while for II–VI see Gunshor and Nur-mikko [204]. Much discussion has already centred around interband transitions in the calculation of energy levels (see Chapters 2 and 3), exciton binding energies, and oscillator strengths (see Chapter 6), for further information see Ivchenko and Pikus ([205], p. 162), or Bastard ([18], p. 237).

For intersubband transitions the first term on the right-hand side of equation (9.252) is zero, and since the subband envelope functions ψ_f and ψ_i are both eigenfunctions of the same Hermitian operator (the conduction or valence band Hamiltonian), they are therefore orthogonal, i.e.

$$\langle \psi_f | \psi_i \rangle = \int_{\text{all space}} \psi_f(\mathbf{r}) \psi_i(\mathbf{r}) \, d\mathbf{r} = 0 \quad (9.255)$$

Therefore, the matrix element becomes:

$$\langle f | \tilde{\mathcal{H}} | i \rangle = \langle u_f | u_i \rangle_{\text{cell}} \langle \psi_f | \tilde{\mathcal{H}} | \psi_i \rangle \quad (9.256)$$

where the time-dependent perturbing potential $\tilde{\mathcal{H}}$ is given by:

$$\tilde{\mathcal{H}} = \frac{e}{m^*} \mathbf{A} \cdot \mathbf{p} \quad (9.257)$$

(see [188], [14], p. 476 and, most appropriately [206], p. 184). The vector \mathbf{A} is the vector potential of the electromagnetic field and hence consists of a magnitude and a direction, with the latter being represented by the unit polarization vector $\hat{\mathbf{e}}$. The linear momentum operator $\mathbf{p} = -i\hbar\nabla$, and therefore:

$$\langle f | \tilde{\mathcal{H}} | i \rangle \propto \langle \psi_f | \hat{\mathbf{e}} \cdot \nabla | \psi_i \rangle \quad (9.258)$$

In this case of intersubband scattering, the envelope wave functions ψ_i and ψ_f are functions of the displacement along the growth (z -) axis only. Therefore, the gradient operator will also only have a z - component, and hence:

$$\langle f | \tilde{\mathcal{H}} | i \rangle \propto \hat{e}_z \left\langle \psi_f \left| \frac{\partial \psi_i}{\partial z} \right. \right\rangle \quad (9.259)$$

This implies that transitions are only allowed when there is a component of the polarization vector $\hat{\mathbf{e}}$ along the growth (z -)axis [207, 208], which means that no

intersubband absorption occurs for normal (i.e. along the growth (z -)axis) incident light. This is illustrated in the top diagram of Fig. 9.35. For the case of normal-incidence light, there is no component of the polarization vector \hat{e} along the growth (z -) axis of the heterostructure, and hence no intersubband absorption occurs. When the incident light is at an angle, the z -component \hat{e}_z is non-zero, and some absorption is allowed. This is a major difference between intersubband and interband transitions.

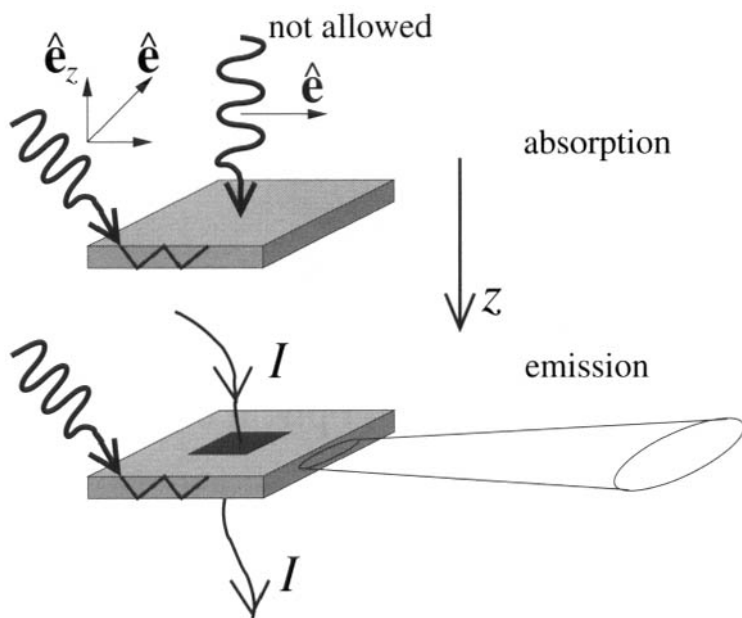


Figure 9.35 Natural intersubband device geometries

In addition any laser made from intersubband transitions, whether optically pumped or electrically injected, as illustrated schematically in the bottom diagram of Fig. 9.35, will be an edge-emitter [209, 210].

These conclusions about natural device geometries are valid for the simple single parabolic band wave functions considered in the first instance. However, there is a wealth of literature available about possible band structure effects, including non-parabolicity, anisotropy, and degeneracy which could break these selection rules (for experimental work, see, for example [211], and for a detailed theoretical analysis, see Batty and Shore [212], and references therein).

The actual lifetime for intersubband spontaneous radiative emission is obtained by summing equation (9.249) over all photon modes; this requires some assumption to the cavity, i.e. the region of dielectric that the modes occupy. Conveniently, Smet *et al.* [188] quotes the results as:

$$\frac{1}{\tau_i} = \frac{e^2 n \omega^2}{6 \pi \epsilon m^* c^3} O_{if} \quad (9.260)$$

where n is the refractive index at the emission wavelength, for a three-dimensional distribution of photon modes, and:

$$\frac{1}{\tau_i} = \frac{e^2 \omega}{4\epsilon m^* c^2 W_z} O_{if} \quad (9.261)$$

for a two-dimensional photonic density of states. The latter is a common scenario in a quantum well system which is often surrounded by barriers and/or a substrate of different dielectric material, which can act as a microcavity producing confinement of the optical modes along the growth (z -) axis, as illustrated in Fig. 9.36.

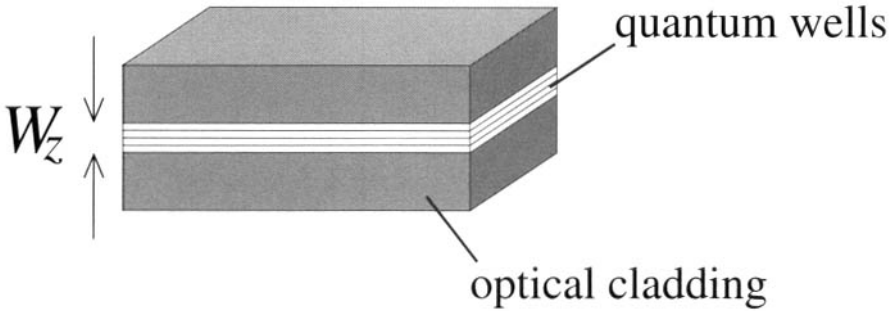


Figure 9.36 Schematic representation of a typical semiconductor quantum well system surrounded by dielectric material, thus producing confinement of optical modes

The oscillator strength O_{if} is dependent upon the dipole matrix element [209], i.e.

$$O_{if} = \frac{2m^* \omega}{\hbar} |\langle \psi_i | z | \psi_f \rangle|^2 \quad (9.262)$$

By using the approach of Burt [213] this can be shown to be equivalent to:

$$O_{if} = \frac{2\hbar}{m^* \omega} \left| \left\langle \psi_i \left| \frac{\partial \psi_f}{\partial z} \right. \right\rangle \right|^2 \quad (9.263)$$

Given that the matrix element is squared, then it is equivalent (see [12], p. 26), to that used earlier to derive the selection rules.

Fig. 9.37 compares the results of calculations of the three-dimensional and two-dimensional spontaneous radiative emission lifetimes, between the two lowest conduction band eigenstates of a GaAs infinitely deep quantum well; the emission wavelength has been varied by adjusting the quantum well width. This figure shows that for short-wavelength emission in the mid-infrared ($< 10 \mu\text{m}$), the radiative lifetime calculated by using the 3D distribution of photon modes is shorter than that calculated with the 2D distribution, with W_z taken as $3 \mu\text{m}$. However, as the wavelength increases, and the emission energy decreases, shorter lifetimes for radiative emission are obtained with the 2D equation. Fig. 9.37 basically confirms the work of Smet

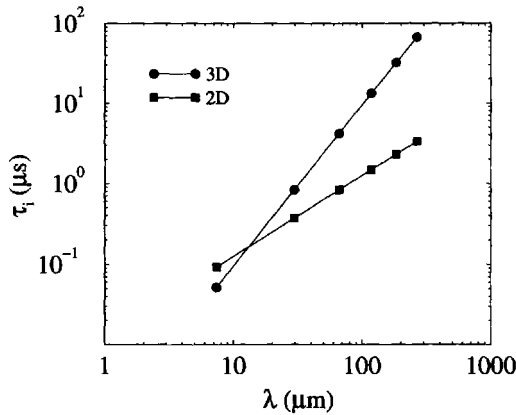


Figure 9.37 Comparison of the three-dimensional (3D) and two-dimensional (2D) spontaneous radiative emission lifetimes

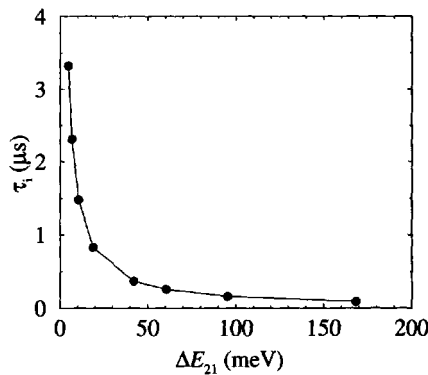


Figure 9.38 The 2D spontaneous lifetime as a function of subband separation

et al. [188]. The refractive index n was calculated by using the first-order Sellmeier equation [214] and the data of Seraphin and Bennett [215].

Fig. 9.38 plots the 2D spontaneous radiative lifetime as in the previous figure, but this time as a function of the subband separation. It can be seen that the lifetime increases as the energy separation decreases, with what appears to be a hyperbolic dependence. This is the reason why intersubband emitters and lasers are thought to be more difficult to fabricate at longer wavelengths (see, for example [216]). Fig. 9.39 confirms the hyperbolic dependence of the lifetime on the subband separation by plotting its inverse, i.e. the scattering rate against $\Delta E_{21} = E_2 - E_1$. Notice the striking linearity implying that

$$\frac{1}{\tau_i} \propto \Delta E_{21} \quad (9.264)$$

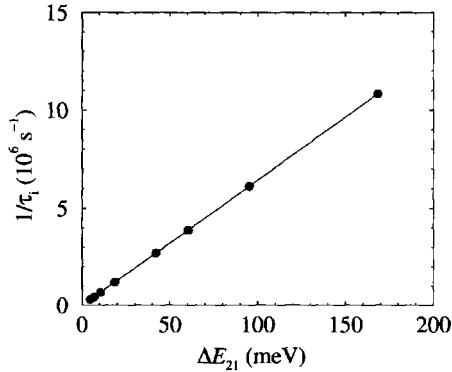


Figure 9.39 The 2D spontaneous emission rate as a function of subband separation

This is not as obvious as it seems, for in the expression for this scattering rate, the ω in the numerator cancels with the ω in the denominator. Hence, the functional dependencies are controlled by the matrix element.

The independence of the spontaneous radiative lifetime on temperature *in this model* should also be noted. In fact this is mainly because for simplicity the material dependencies on temperature have been ignored, in order to deduce as general results as are possible. In reality, the energy gaps do change slightly and hence the emission wavelength also changes (see Adachi [14] for further information). While in these models the emission frequency doesn't change, the intensity, i.e. the number of photons emitted, does change, in fact, the latter decreases. This is because of the increased competition with the non-radiative channels.

9.22 QUANTUM CASCADE LASERS

This mid-infrared (4–14 μm) device is rather esoteric and specialised, and inclusion of a discussion of its operation would seem out of place in this present work. However this is included for one particular reason, i.e. the quantum cascade laser is a *textbook* example of electron scattering. This laser is a unipolar laser in that there is only one carrier type, as opposed to the vast majority of laser diodes and light-emitting diodes which are bipolar and generate light through interband recombination of electrons with holes. Since its invention [217], development has been rapid, with improving temperature dependency [218] and increasing wavelength [193].

The simplest active region of any solid state laser would incorporate three energy levels, which in this case would be subbands, as illustrated schematically in Fig. 9.40. Each level contains a certain number of electrons, say n_1 , n_2 , and n_3 , which scatter between the subbands (as indicated by the arrows). Energy can be input to the system, either by optically exciting electrons from the ground state $|1\rangle$ to the second excited state $|3\rangle$, or by injecting them directly into the uppermost state. The quantum cascade

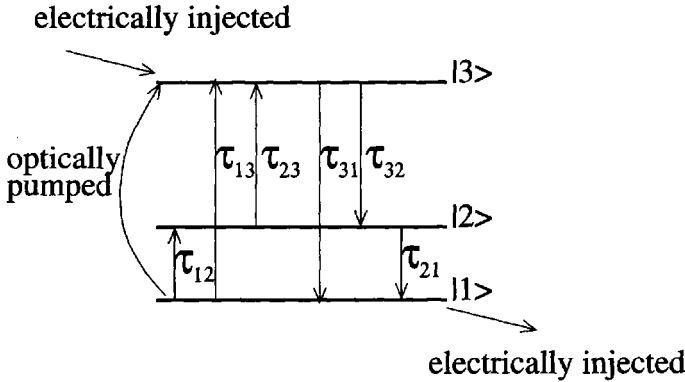


Figure 9.40 The scattering processes in a three-level laser

laser is an example of the latter, with electrons being removed from the lowest state $|1\rangle$ to be recycled, with these constituting the injected current in an additional stage. Thus, one electron can produce many photons.

Consider the rate equation for the number of electrons (population) of each level, i.e.

$$\frac{dn_3}{dt} = I_{in} + \frac{n_1}{\tau_{13}} + \frac{n_2}{\tau_{23}} - \frac{n_3}{\tau_{32}} - \frac{n_3}{\tau_{31}} \quad (9.265)$$

$$\frac{dn_2}{dt} = \frac{n_1}{\tau_{12}} + \frac{n_3}{\tau_{32}} - \frac{n_2}{\tau_{21}} - \frac{n_2}{\tau_{23}} \quad (9.266)$$

$$\frac{dn_1}{dt} = \frac{n_3}{\tau_{31}} + \frac{n_2}{\tau_{21}} - \frac{n_1}{\tau_{12}} - \frac{n_1}{\tau_{13}} - I_{out} \quad (9.267)$$

where I_{in} represents the injection rate (the number of electrons per unit time), which at equilibrium is equal to I_{out} . Next, consider the population of the second level, where at equilibrium, the net change is zero, also assume that the temperature is relatively low, and hence the absorption rates can be ignored then:

$$\frac{n_3}{\tau_{32}} = \frac{n_2}{\tau_{21}} \quad (9.268)$$

Furthermore, if:

$$\frac{1}{\tau_{21}} > \frac{1}{\tau_{32}} \quad (9.269)$$

then $n_3 > n_2$, i.e. a population inversion will exist between levels $|3\rangle$ and $|2\rangle$, thus fulfilling a necessary condition for stimulated emission. The ratio n_3/n_2 is known as the *population ratio*, which in this analysis would be given by:

$$\frac{n_3}{n_2} = \frac{\tau_{32}}{\tau_{21}} \quad (9.270)$$

Perhaps the simplest way to realise such a three-level system is within a triple quantum well structure, with an energy level in each well—the subband minima can then be altered (almost) independently merely by adjusting the quantum well widths.

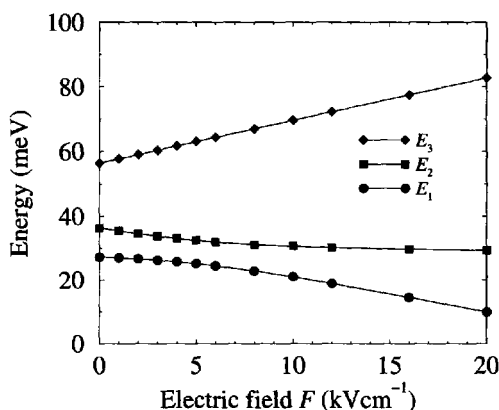


Figure 9.41 The electric field dependence of the lowest three subband minima

Equation (9.269) suggests that considering ways of enhancing the scattering rate from the second level to the first, may be a productive way of engineering a population inversion. With this in mind, consider a GaAs triple quantum well surrounded by $\text{Ga}_{0.8}\text{Al}_{0.2}\text{As}$ barriers, with well widths of 56.5, 96.1, and 84.8 Å, respectively (integral numbers of monolayers), separated by barriers of width 56.5 and 28.25 Å. The central well has been chosen to be the widest, such that at zero applied electric field it contains the ground state. As the field is increased, an anti-crossing with the state in the right-hand well (84.8 Å) will be inevitable, thus hopefully leading to an increase in the scattering rate which depopulates level $|2\rangle$.

This behaviour can be clearly seen in Fig. 9.41, where the latter plots the lowest three subband minima as a function of applied electric field. The wave functions themselves are illustrated in Fig. 9.42, just beyond the anti-crossing, at $F=10\text{ kVcm}^{-1}$. The large overlap between $|2\rangle$ and $|1\rangle$ is apparent, which will hopefully lead to a strong depopulation of the lower laser ground state, i.e. a high $1/\tau_{21}$. In comparison, the overlap between $|3\rangle$ and $|2\rangle$ is smaller, thus implying a longer carrier lifetime in the upper laser level, i.e. a small $1/\tau_{32}$.

To confirm whether there is indeed a population inversion requires a calculation of the scattering rates themselves. Using the methods outlined in the present chapter, Fig. 9.43 displays the electron–LO phonon and Fig. 9.44 the electron–electron scattering rates, as a function of the applied electric field (at 77 K).

The nature of the anti-crossing is evident from the data in Fig. 9.43, as the $|3\rangle$ to $|2\rangle$ scattering rate changes rapidly at the anti-crossing, as level $|2\rangle$ moves from being confined in the right-hand well to the central well. This increases the overlap with state $|3\rangle$ confined in the left-hand well and hence the scattering rate also increases.

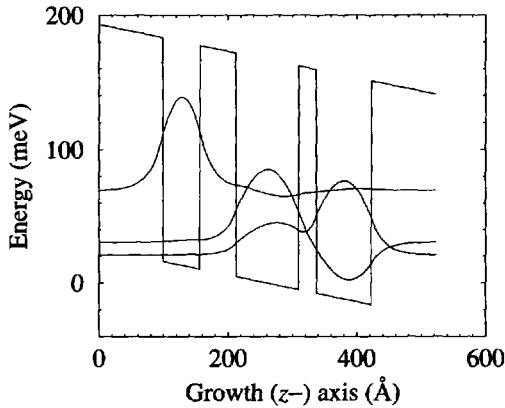


Figure 9.42 A quantum cascade laser active region at $F=10 \text{ kVcm}^{-1}$

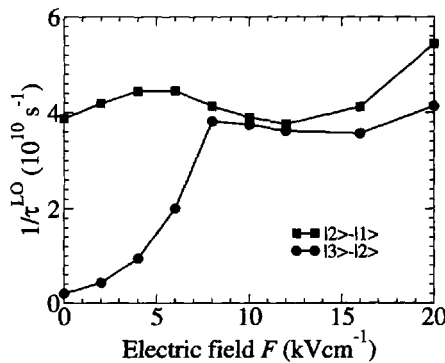


Figure 9.43 The electron-LO phonon scattering rates from level $|3\rangle$ to $|2\rangle$ and from level $|2\rangle$ to $|1\rangle$

The depopulation of the lower lasing level, i.e. the rate $|2\rangle$ to $|1\rangle$, mirrors the subband separation between these levels, and indicates that there is mixing between these states for a considerable range of electric fields.

Fig. 9.44 shows the corresponding electron-electron scattering rates; note these are total rates and include *all* contributions, i.e. the $|3\rangle$ to $|2\rangle$ rate includes 33-32, 33-22, and 32-22, while the $|2\rangle$ to $|1\rangle$ rate includes 22-21, 22-11, and 21-11. It is immediately apparent that this electron-electron depopulation rate of level $|2\rangle$, i.e. the rate at which carriers are removed from $|2\rangle$ to $|1\rangle$, is very much faster than the repopulation rate, i.e. the rate at which carriers scatter into $|2\rangle$. This is due to the

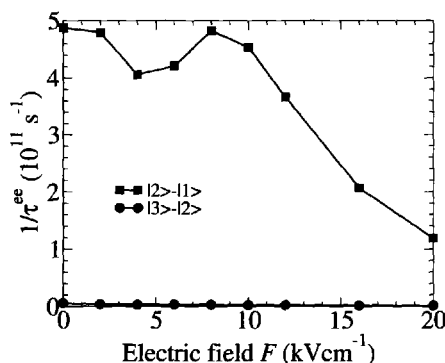


Figure 9.44 The electron–electron scattering rates from level $|3\rangle$ to $|2\rangle$ and from levels $|2\rangle$ to $|1\rangle$

close proximity in both energy and real space of levels $|1\rangle$ and $|2\rangle$. A more general observation, from comparing Figs 9.43 with 9.44, shows that the electron–electron scattering rate out of level $|2\rangle$ is very much faster than that due to scattering with LO phonons. In addition, it is also apparent that in both cases the depopulation rate is larger than the repopulation rate, and hence a population inversion is likely.

This is confirmed by Fig. 9.45, which plots the ratio of the lifetimes—a simple quantity which has been shown to approximate the population ratio between levels $|3\rangle$ and $|2\rangle$. Thus, in conclusion, using the methods outlined, this active-layer design has been shown to exhibit a population inversion between levels $|3\rangle$ and $|2\rangle$ at 77 K. The corresponding emission energy, given by the subband separation $E_3 - E_2$, varies between 33 and 53 meV, which, in turn, correspond to 37 and 23 μm , respectively. Electroluminescence has been observed at 88 μm in a quantum cascade device at low temperatures [219].

9.23 CARRIER SCATTERING IN QUANTUM WIRES AND DOTS

Carrier scattering in lower-dimensional systems, wires and dots, will be an important issue as they are introduced into future generations of opto-electronic devices. For the theoretical understanding of the carrier dynamics, scattering rate expressions will be needed.

Although the scattering rates derived here are not applicable to such systems, the techniques employed to obtain them do have some relevance. Quantum wires resemble quantum wells more closely than dots, as they still retain carrier dispersion. Therefore developing a model of carrier scattering with *bulk* LO phonons will require expressing the carrier wave function as a product of a two-dimensional envelope

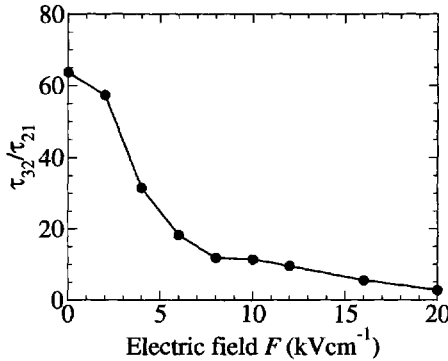


Figure 9.45 The ratio of the lifetimes τ_{32}/τ_{21} of electrons in the triple quantum well active region of the quantum cascade laser

function with a one-dimensional plane wave, as opposed to the other way around. The remainder of the derivation should follow in a similar fashion. Similar arguments for the route to carrier-carrier scattering in quantum wires may also be applicable.

However, for quantum dots the situation is quite different as the carriers never possess dispersion; hence, for scattering with phonons of fixed energies, it would appear a priori that this will only occur for resonance conditions, i.e. when the sublevel separation is equal to a phonon energy. Carrier-carrier scattering between the sublevels of quantum dots resembles the Coulomb interaction in multi-electron atoms, with the latter being an area where much work has been done.

Theoretical, and hence computational studies of carrier scattering in quantum wires and dots are still quite rare in the literature, although see for example [220–225]. For an introduction to some aspects of the particulars of optical processes (carrier-photon scattering) in wires and dots see Basu ([11], p. 343).

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CHAPTER 10

MULTIBAND ENVELOPE FUNCTION (K.P) METHOD

contributed by Z. Ikonić

10.1 SYMMETRY, BASIS STATES AND BAND STRUCTURE

The band structures of common semiconductors like GaAs, InP, Si, Ge, etc., which have the diamond/zinc blende crystal structure, are more complicated in the valence band than they are in the conduction band. This is related to the fact that looking microscopically, at the level of the crystalline unit cell, the conduction band states are mostly s-like in character (i.e. the microscopic 'Bloch' wave function has the symmetry of an atomic s orbital), while the valence band states are mostly p-like in character. There are three degenerate p-type atomic orbitals, the symmetry of which is denoted as x , y , and z , which stems from the direction along which the orbitals are aligned. It is therefore natural that all three of them will take part in valence band state wave functions.

Detailed microscopic calculations, e.g. using the empirical pseudopotential method, see Chapter 11, show that there are two valence bands degenerate at the centre of the

Brillouin zone (the Γ -point), and close by (a few tens to a few hundreds of meV below) there is a third band. The first two are called heavy-hole (HH) and light-hole (LH) bands, and they cease to be degenerate for finite values of the wave vector $\mathbf{k} = k_x, k_y, k_z$: the energy of the former descends at a slower rate as the wave vector moves away from the Γ -point, which corresponds to a larger effective mass, hence the name. The third band is called the spin-orbit split-off (SO) band. An example of the valence band dispersion is given in Fig. 10.1.

Any one of the bands is an energy eigenstate of the bulk material, and 'pure' (single-band) states with definite energy may therefore exist in bulk. However, in the case of position-dependent potentials this will no longer be true: states in quantum wells, for instance, will be 'mixtures', with *all* the bulk bands contributing to their wave functions. The contribution of a particular bulk band to a quantised state generally depends on its energy spacing from that band: the smaller the energy, the larger the relative contribution will be (this follows from quantum mechanical perturbation theory). When two or more bands are degenerate, or almost degenerate, states in their vicinity are likely to have similar contributions from these bands.

In a quantum well type of structure, for energies which are not far from the conduction band edges of the constituent materials, a quantised state wave function will have contributions mostly from the conduction bands of these materials, each of which has s-like character, and there will be a single envelope wave function (solution of the effective mass Schrödinger equation) which represents the amplitude of these, s-like Bloch functions. On the other hand, a quantised state near the valence band edges of any of the constituent materials is expected to comprise 2 or 3 of the bulk valence bands with comparable contributions, each having its own envelope function. Depending on the accuracy required in the calculation of the quantised states, and on the positions of the bands in the bulk materials, different number of bands may be included in the calculation. It may sometimes suffice to explicitly include just HH and LH bands, or, on other occasions, also the SO band in the description of the system.

Generally, the Hamiltonians which describe states in such situations are matrices, or systems of coupled Schrödinger equations, which will deliver the possible energies and wave functions expressed as a set of envelope functions (which vary slowly over a crystalline unit cell), themselves representing the amplitudes of the corresponding basis states (usually the bulk bands). For this reason the method is known as the 'multiband envelope function', or 'multiband effective mass method' and, because the interaction of bulk bands is described via the \mathbf{k}, \mathbf{p} perturbation, it is also known as the \mathbf{k}, \mathbf{p} method.

Clearly, the concept of bulk band mixing in forming quantised states of a system applies to more remote bands as well. The conduction band quantised states will thus include contributions from bulk valence band states, and vice versa, and there exist extended versions of the \mathbf{k}, \mathbf{p} method which explicitly include the HH, LH, SO, and the conduction band, or still wider variants including even more remote bands. However, in this work attention will be focussed on 4- and 6-band Hamiltonians, which explicitly include the valence band states. The number of bands the Hamiltonian is named after

is the number of bands that are explicitly included, i.e. their envelope wave functions are explicitly evaluated. However, such Hamiltonians do *implicitly* account for the existence of other, more remote bands, and their influence is incorporated via the values of the material parameters. In this context, the conventional (conduction band) effective mass Schrödinger equation is just a special case of the multiband envelope function model, where the existence of bands other than the conduction band is accounted for by using the effective, rather than the free electron mass, and only the conduction band envelope wave function is calculated explicitly. The 4- and 6-band Hamiltonians for the valence envelope wave functions were derived by Luttinger and Kohn [226] using $\mathbf{k} \cdot \mathbf{p}$ perturbation theory, while the 8-band model (that includes the conduction band) was developed by Pidgeon and Brown [227].

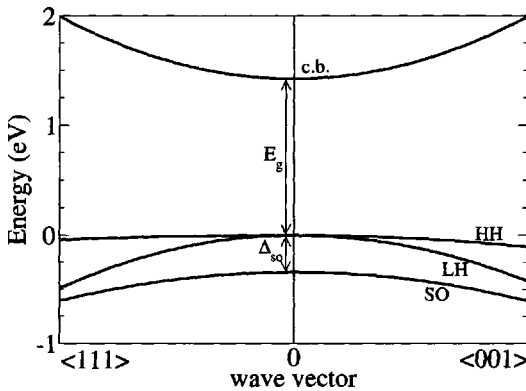


Figure 10.1 The valence and conduction bands of a group IV or III–V semiconductor near the Γ -point of the Brillouin zone.

10.2 VALENCE BAND STRUCTURE AND THE 6×6 HAMILTONIAN

It is beyond the scope of this book to derive the Hamiltonian that describes the valence band states, so its form will be stated and it will be employed in the calculation of quantised states within the valence band of nanostructures in order to illustrate how it is used. In many cases good accuracy can be obtained by using the so-called 6×6 Hamiltonian, although its shortened version, the 4×4 Hamiltonian is frequently just as good—it all depends on the energy range of interest. As mentioned above, a valid basis set can be the atomic p-like orbitals with x , y , and z -like spatial symmetry, here denoted as $|X\rangle$, $|Y\rangle$, $|Z\rangle$, each also having the spin projection along the z -axis equal to either $+1/2$ or $-1/2$, denoted as \uparrow or \downarrow respectively. However, common practice

is to change from the $|X \uparrow\rangle, |Y \uparrow\rangle, |Z \uparrow\rangle, |X \downarrow\rangle, |Y \downarrow\rangle, |Z \downarrow\rangle$ basis into another one, such that its member functions are simultaneously the eigenstates of the angular momentum operator (with eigenvalues J equal to $3\hbar/2$ or $\hbar/2$), and of its projection along the z -axis m_J (with eigenvalues equal to $\pm 3/2$ or $\pm 1/2$). This is achieved by making appropriate linear combinations of the atomic basis states. The list of this new set of $|J, m_J\rangle$ basis states is given below in equation (10.1).

The precise form of the Hamiltonian depends on the pre-factors in equation (10.1), for example the presence of the imaginary number i or -1 does not change the state properties, and it also depends on how these states are ordered in the list. There is no unique choice that is universally accepted in the literature, however one of the frequently used possibilities, see for example [228], which will be adopted here, reads:

$$\begin{aligned}
 |3/2, 3/2\rangle &= (1/\sqrt{2}) |(X + iY) \uparrow\rangle \\
 |3/2, -3/2\rangle &= (1/\sqrt{2}) |(X - iY) \downarrow\rangle \\
 |3/2, 1/2\rangle &= (1/\sqrt{6}) |(X + iY) \downarrow\rangle - (\sqrt{2}/\sqrt{3}) |Z \uparrow\rangle \\
 |3/2, -1/2\rangle &= -(1/\sqrt{6}) |(X - iY) \uparrow\rangle - (\sqrt{2}/\sqrt{3}) |Z \downarrow\rangle \\
 |1/2, 1/2\rangle &= (1/\sqrt{3}) |(X + iY) \downarrow\rangle + (1/\sqrt{3}) |Z \uparrow\rangle \\
 |1/2, -1/2\rangle &= -(1/\sqrt{3}) |(X - iY) \uparrow\rangle + (1/\sqrt{3}) |Z \downarrow\rangle
 \end{aligned} \quad (10.1)$$

Just as with the original ($|X \uparrow\rangle, |Y \uparrow\rangle, |Z \uparrow\rangle, |X \downarrow\rangle, |Y \downarrow\rangle, |Z \downarrow\rangle$) states, the new basis states are all orthogonal to each other. In this $|J, m_J\rangle$ basis the 6×6 Hamiltonian that describes the HH, LH and SO bands for the bulk reads:

$$\mathbf{H} = \begin{bmatrix} P + Q & 0 & -S & R & (1/\sqrt{2})S & \sqrt{2}R \\ 0 & P + Q & -R^\dagger & -S^\dagger & -\sqrt{2}R^\dagger & (1/\sqrt{2})S^\dagger \\ -S^\dagger & -R & P - Q & 0 & \sqrt{2}Q & \sqrt{3/2}S \\ R^\dagger & -S & 0 & P - Q & -\sqrt{3/2}S^\dagger & \sqrt{2}Q \\ (1/\sqrt{2})S^\dagger & -\sqrt{2}R & \sqrt{2}Q & -\sqrt{3/2}S & P + \Delta_{\text{SO}} & 0 \\ \sqrt{2}R^\dagger & (1/\sqrt{2})S & \sqrt{3/2}S^\dagger & \sqrt{2}Q & 0 & P + \Delta_{\text{SO}} \end{bmatrix} \quad (10.2)$$

where

$$\begin{aligned}
 P &= \left(\frac{\hbar^2}{2m_0} \right) \gamma_1 (k_x^2 + k_y^2 + k_z^2) \\
 Q &= \left(\frac{\hbar^2}{2m_0} \right) \gamma_2 (k_x^2 + k_y^2 - 2k_z^2) \\
 R &= \left(\frac{\hbar^2}{2m_0} \right) \sqrt{3} [-\gamma_2 (k_x^2 - k_y^2) + 2i\gamma_3 k_x k_y] \\
 S &= \left(\frac{\hbar^2}{2m_0} \right) 2\sqrt{3}\gamma_3 k_z k_-
 \end{aligned} \quad (10.3)$$

where $k_- = k_x - ik_y$, $\gamma_{1,2,3}$ are the Luttinger parameters, and Δ_{SO} is the spin-orbit splitting—the spacing between the HH (or LH) band and the SO band at the ($k = 0$)

Γ -point. The values of $\gamma_{1,2,3}$ and Δ_{SO} in some common semiconductors are given in Table 10.1.

In writing equation (10.2) the convention has been used that the hole energy is measured from the top of the valence band downwards (the inverted energy picture). This is because it is usually easier to look at the hole band structure in the same manner as that for electrons, and this is possible if only holes are considered. For the true energy picture (as in Fig. 10.1), all the terms in the Hamiltonian should be multiplied by -1 . Furthermore, it is important to note that the coordinate system in which this Hamiltonian is written is not oriented arbitrarily—the axes x , y , and z are aligned along the edges of the crystalline cubic unit cell.

The Schrödinger equation corresponding to this Hamiltonian may still be written as

$$\mathcal{H}\psi = E\psi \quad (10.4)$$

but $\psi = \psi(\mathbf{r})$ is a *vector* (a 6-component array). Note that the Hamiltonian is a Hermitian matrix with simple scalars as its elements (which depend on the material parameters) and wave vector components k_x , k_y , and k_z . If the eigenenergies and the corresponding eigenvectors are found, the latter will obviously be lists of (possibly complex-valued) scalar constants. What would these mean? It is implicitly assumed that the wave function has a plane wave form, i.e. all the 6 components of ψ have the common, plane wave type of spatial behaviour, i.e. $\sim \exp(i\mathbf{k}\cdot\mathbf{r})$. This makes them the components of the envelope wave function and, if interested in a more detailed form, each of them multiplies a corresponding basis state from the list in equation (10.1) and can be added together to construct the ‘true’ microscopic wave function of a state with energy E . The eigenvectors are generally ‘full’, i.e. their entries usually have non-zero values, which means that a plane-wave state has all the 6 basis states admixed. This is in contrast to the conduction band, where one usually deals with ‘pure’ spin-up or spin-down states.

To have the usual meaning of a wave function, an eigenvector ψ has to be normalised to unity, i.e. all the components of a vector may have to be multiplied by a suitable constant so that:

$$\psi^\dagger \psi = 1 \quad (10.5)$$

is satisfied. The ‘ \dagger ’ symbol means the Hermitian conjugate, i.e. the transpose (a column-vector becomes a row-vector) followed by the complex conjugate (take the complex conjugate of all the elements).

In order to find eigenenergies for a specified \mathbf{k} , solutions can be sought to $\det|H - E| = 0$, which delivers a sixth-order polynomial in the energy E . This could only be solved numerically, unless it is noted that it can be factored into two identical third-order polynomials thus allowing for analytic, though lengthy, solutions. However, the situation is quite simple at the zone centre, i.e. if we set $k_x = k_y = k_z = 0$ the

Table 10.1 Parameters relevant for the valence band structure in a few semiconductors [229, 230].

	Si	Ge	GaAs	AlAs	InAs
γ_1	4.22	13.4	6.98	3.76	20.0
γ_2	0.39	4.25	2.06	0.82	8.5
γ_3	1.44	5.69	2.93	1.42	9.2
Δ_{SO} (eV)	0.044	0.290	0.341	0.28	0.39
C_{11} (Mbar)	1.675	1.315	1.221	1.250	0.833
C_{12} (Mbar)	0.650	0.494	0.566	0.534	0.453
a_{latt} (Å)	5.431	5.657	5.653	5.661	6.058
a_v (eV)	2.46	1.24	-1.116	-2.47	-1.00
b (eV)	-2.10	-2.86	-2.0	-2.3	-1.8
VBO (eV)			-0.80	-1.33	-0.59

Hamiltonian becomes:

$$\mathcal{H}(\mathbf{k} = 0) = \begin{bmatrix} 0 & 0 & 0 & 0 & 0 & 0 \\ 0 & 0 & 0 & 0 & 0 & 0 \\ 0 & 0 & 0 & 0 & 0 & 0 \\ 0 & 0 & 0 & 0 & 0 & 0 \\ 0 & 0 & 0 & 0 & \Delta_{SO} & 0 \\ 0 & 0 & 0 & 0 & 0 & \Delta_{SO} \end{bmatrix} \quad (10.6)$$

This has a fourfold degenerate eigenvalue $E = 0$, and a twofold degenerate eigenvalue $E = \Delta_{SO}$, consistent with the concept that there are HH and LH branches degenerate at $\mathbf{k} = 0$, and an SO branch displaced by Δ_{SO} from them. As for the eigenvectors: for each of the four degenerate values $E = 0$ just one of the four amplitudes of $|3/2, 3/2\rangle$, $|3/2, -3/2\rangle$, $|3/2, 1/2\rangle$, $|3/2, -1/2\rangle$ has unity value, the others being equal to zero (though, due to degeneracy, any other linear combination of these states would do just as well). Similarly, the eigenvectors corresponding to the two degenerate eigenvalues $E = \Delta_{SO}$ may be taken to have non-zero amplitude of the $|1/2, 1/2\rangle$, and of the $|1/2, -1/2\rangle$ states. For finite values of \mathbf{k} the fourfold degeneracy splits into two pairs of double-degenerate states (HH and LH have different energies), so all the three branches remain doubly-degenerate—in analogy to the spin-degeneracy of conduction band states.

10.3 4×4 VALENCE BAND HAMILTONIAN

For non-zero \mathbf{k} the expressions for the eigenenergies would be quite lengthy, so consider in more detail a 'shortened' version, i.e. a 4×4 Hamiltonian that includes

only the HH and LH states in its basis. This is obtained by excluding the $|1/2, \pm 1/2\rangle$ states, i.e. by removing the fifth and sixth rows and columns from the Hamiltonian matrix in equation (10.2):

$$\mathcal{H} = \begin{bmatrix} P+Q & 0 & -S & R \\ 0 & P+Q & -R^\dagger & -S^\dagger \\ -S^\dagger & -R & P-Q & 0 \\ R^\dagger & -S & 0 & P-Q \end{bmatrix} \quad (10.7)$$

Finding $\det|\mathcal{H} - E|$ gives a fourth order polynomial in E , but it is easy to see that it is a square of a quadratic polynomial:

$$(P+Q-E)(P-Q-E) - |R|^2 - |S|^2 = 0 \quad (10.8)$$

Using the expressions in equation (10.3) it can be found that the two solutions read:

$$E_{\pm}(k_x, k_y, k_z) = \gamma_1 k^2 \pm 2\sqrt{\gamma_2^2 k^4 - 3(\gamma_2^2 - \gamma_3^2)(k_x^2 k_y^2 + k_x^2 k_z^2 + k_y^2 k_z^2)} \quad (10.9)$$

where $k^2 = k_x^2 + k_y^2 + k_z^2$.

It is immediately clear from equation (10.9) that both solutions, also called dispersion branches, are non-parabolic (E does not depend only on k^2) and anisotropic (E depends on $k_{x,y,z}$ in a manner different from just that of its modulus k , i.e. the energy varies differently in different directions of the wave vector space). This latter property is also called 'band warping'.

Consider now the special case when two components of the wave vector are zero, e.g. $k_x = k_y = 0$, while $k_z \neq 0$. From the definitions in equation (10.3) $R = S = 0$, hence:

$$E_+ = k_z^2 \hbar^2 (\gamma_1 + 2\gamma_2)/2m_0 \quad \text{and} \quad E_- = k_z^2 \hbar^2 (\gamma_1 - 2\gamma_2)/2m_0 \quad (10.10)$$

Clearly, the E_- branch is characterised by an effective mass in the z direction equal to $m_0/(\gamma_1 - 2\gamma_2)$, and this is larger than the effective mass of the E_+ branch, which amounts to $m_0/(\gamma_1 + 2\gamma_2)$. For this reason the E_- branch is called the 'heavy hole' (HH) branch, and its energy is denoted as E_{HH} , while the E_+ branch is referred to as the 'light hole' (LH) branch, its energy denoted as E_{LH} . Under these conditions, where in fact $E_{\text{HH}} = P+Q$, the first two equations in $(\mathcal{H} - E_{\text{HH}})\psi = 0$ have all zero entries, while the third and fourth equations do not. From this it can be concluded that elements 3 and 4 in the vector ψ are zero, while elements 1 and 2 are not. Therefore, $[1 \ 0 \ 0 \ 0]^T$ can be chosen for one HH state vector, and $[0 \ 1 \ 0 \ 0]^T$ for the other (remember that E_{HH} is double degenerate), though any pair of linear combinations of these two vectors would do just as well. However, if choosing to make such linear combinations, care should be taken to make them mutually orthogonal in order to avoid complications in any further use of the wave functions. In any case, for $k_z \neq 0$ the HH branch has only $|3/2, \pm 3/2\rangle$ basis states as its constituents. Similar reasoning for the E_{LH} branch shows that for $k_z \neq 0$ the LH branch includes only $|3/2, \pm 1/2\rangle$ states.

For finite k_x and/or k_y , however, the system of linear equations is generally full, which indicates that a hole state has a finite contribution from all the basis states. This may seem puzzling, in view of the fact that there is essentially no difference between the x and z directions in the bulk crystal. However, this difference arises because of the choice of basis states: each of them has a definite projection of momentum and spin along the z -axis, and therefore (as is known from quantum mechanics) does not have the same property along the x -axis.

10.4 COMPLEX BAND STRUCTURE

While equation (10.8) and its solutions in equation (10.9) give the possible values of hole energy for a particular wave vector \mathbf{k} , it is interesting to consider the reverse problem: what values of k_z may holes have if their energy E and the other two components of the wave vector (k_x and k_y) are specified? For sake of simplicity k_y is set to zero, while k_x and E are generally non-zero and are real. Equation (10.8) in its expanded form then reads:

$$(\gamma_1^2 - \gamma_2^2)k_z^4 + [(2\gamma_1^2 + 8\gamma_2^2 - 12\gamma_3^2)k_x^2 - 2E\gamma_1]k_z^2 + [E - (\gamma_1 + 2\gamma_2)k_x^2][E - (\gamma_1 - 2\gamma_2)k_x^2] \equiv Ak_z^4 + Bk_z^2 + C = 0 \quad (10.11)$$

which is, of course, quadratic in k_z^2 , with the two solutions:

$$(k_z^2)_{1,2} = \frac{-B \pm \sqrt{B^2 - 4AC}}{2A} \quad (10.12)$$

each of which delivers two k_z 's as a positive and a negative root of k_z^2 . In real semiconductors the values of the Luttinger parameters γ_1 and γ_2 are such that A is always positive, while B and C may be of either sign, depending on E and k_x , as well as the material parameters.

Consider the case of $4AC < 0$, i.e. $C < 0$. From the form used to write C it is clear that this will happen when k_x and E are such that $(\gamma_1 - \gamma_2)k_x^2 < E < (\gamma_1 + 2\gamma_2)k_x^2$ is satisfied. The $\sqrt{B^2 - 4AC}$ is then larger than B , and regardless of the sign of B one of the roots $(k_z^2)_{1,2}$ will be positive and the other one negative. Hence there will be a pair of real and a pair of purely imaginary k_z values.

Consider now the case of $4AC > B^2$: equation (10.12) then contains the square root of a negative number, which will imply that both roots will be fully complex numbers (with both the real and imaginary parts non-zero). Finally, for $0 < 4AC < B^2$ both values of $(k_z^2)_{1,2}$ are either positive or negative, depending on the sign of B , i.e. all roots are either real or imaginary.

Real-valued solutions for k_z imply conventional plane-wave envelope wave functions, which is an allowed state in an infinite bulk. Complex k_z implies an 'evanescent' wave, which decays in one direction and increases in the opposite direction, and which may simultaneously oscillate (if the real part of $k_z \neq 0$). Because of the infinite length in the z -direction, and the inability to normalise the wave function, such states are

not allowed in an infinite bulk crystal, but are perfectly allowed in finite regions, just as is the case with electrons, and can therefore appear in structures of finite extent.

The above considerations were presented for the 4×4 model, however the general conclusions about the possible types of evanescent waves remain for more elaborate models as well. The list of all possible states at a particular energy E , that behave exponentially is called the 'complex band structure'. The wave functions vary along a particular direction, say z , as $\exp(ik_z z)$, where k_z is a real or complex wave vector.

The case of electrons in the conduction band is quite different as they have a scalar effective mass and really a very simple complex band structure: their wave vector can be only real or imaginary: $k_z = \pm \sqrt{2m^* E / \hbar^2 - k_x^2 - k_y^2}$, but never fully complex.

For holes this situation only occurs at $k_x = k_y = 0$, when $(k_z^2)_1 = E / (\gamma_1 - 2\gamma_2)$ and $(k_z^2)_2 = E / (\gamma_1 + 2\gamma_2)$, so all k_z 's are real for $E > 0$ and imaginary for $E < 0$.

10.5 BLOCK-DIAGONALISATION OF THE HAMILTONIAN

There is an elegant method which allows one to simplify the 6×6 Hamiltonian, by recasting it into a block-diagonal form: it then has two 3×3 matrices as its diagonal elements and the remaining two off-diagonal elements are zero-matrices. This is achieved by a suitable change of basis, i.e. by creating a new basis from linear combinations of the existing basis states. Essentially this means that there are two completely independent sets of states, which are eigenstates of either one or the other 3×3 block. Since this block-diagonalisation reduces the size of the system to be considered at any time, it is useful when attempting to obtain a result in analytical form. Block-diagonalisation can also be useful in cases which are more complicated than straightforward bulk material, for example, strained bulk or two-dimensional quantum-well systems.

The derivation of the transformation of the Hamiltonian into a block-diagonal form will not be reproduced here, the result will merely be stated, i.e. upon introducing the new basis:

$$\begin{aligned}
 |F_1\rangle &= \alpha |3/2, -3/2\rangle - \alpha^* |3/2, 3/2\rangle \\
 |F_2\rangle &= \beta |3/2, 1/2\rangle + \beta^* |3/2, -1/2\rangle \\
 |F_3\rangle &= \beta |1/2, 1/2\rangle + \beta^* |1/2, -1/2\rangle \\
 |F_4\rangle &= \alpha |3/2, -3/2\rangle + \alpha^* |3/2, 3/2\rangle \\
 |F_5\rangle &= \beta |3/2, 1/2\rangle - \beta^* |3/2, -1/2\rangle \\
 |F_6\rangle &= \beta |1/2, 1/2\rangle - \beta^* |1/2, -1/2\rangle
 \end{aligned} \tag{10.13}$$

where:

$$\begin{aligned}
 \alpha &= \frac{1}{\sqrt{2}} \exp[i(\phi/2 + \eta + \pi/4)], & \beta &= \frac{1}{\sqrt{2}} \exp[i(\phi/2 - \eta - 3\pi/4)] \\
 \phi &= \arctan(k_y/k_x), & \eta &= \frac{1}{2} \arctan[(\gamma_3/\gamma_2) \tan(2\phi)]
 \end{aligned} \tag{10.14}$$

the Hamiltonian takes a block-diagonal form:

$$\mathbf{H} = \begin{bmatrix} \mathbf{H}_+ & 0 \\ 0 & \mathbf{H}_- \end{bmatrix} \quad (10.15)$$

The two 3×3 blocks read:

$$\mathbf{H}_{\pm} = \begin{bmatrix} P + Q & R \mp iS & \sqrt{2}R \pm iS/\sqrt{2} \\ R \pm iS^\dagger & P - Q & \sqrt{2}Q \mp i\sqrt{3/2}S \\ \sqrt{2}R \mp iS^\dagger/\sqrt{2} & \sqrt{2}Q \pm i\sqrt{3/2}S^\dagger & P + \Delta_{\text{SO}} \end{bmatrix} \quad (10.16)$$

where P and Q are the same as above, while R and S now read:

$$R = -\sqrt{3} \left(\frac{\hbar^2}{2m_0} \right) \gamma_\phi k_{||}^2, \quad S = 2\sqrt{3} \left(\frac{\hbar^2}{2m_0} \right) \gamma_3 k_{||} k_z$$

$$\gamma_\phi = \sqrt{\bar{\gamma}^2 + \mu^2 - 2\bar{\gamma}\mu \cos \phi}, \quad \bar{\gamma} = \frac{1}{2}(\gamma_2 + \gamma_3), \quad \mu = \frac{1}{2}(\gamma_3 - \gamma_2) \quad (10.17)$$

where $k_{||}^2 = k_x^2 + k_y^2$ is the in-plane wave vector.

Without going into detail, it should be noted that in order to make this transformation possible, the element R from equation (10.2) had to be approximated as given in equation (10.17). A simpler approximation, with $\gamma_\phi = (\gamma_2 + \gamma_3)/2$ has previously been in use, but the one given above is much better. The important point to note is that the transformation in equation (10.13) does not depend on k_z , and when this becomes an operator $k_z \rightarrow -i\partial/\partial z$ (for application to quantum wells) there are no difficulties in changing from one basis to another.

To find the band energies and wave functions the block-diagonal form of the Hamiltonian may then be used. Solving one 3×3 block at a time, the wave function is now a vector of length 3, i.e.

$$\mathbf{H}_{\pm} \psi_{\pm} = E \psi_{\pm}$$

$$\psi_+(z) = \begin{bmatrix} F_1 \\ F_2 \\ F_3 \end{bmatrix}, \quad \psi_-(z) = \begin{bmatrix} F_4 \\ F_5 \\ F_6 \end{bmatrix} \quad (10.18)$$

Block-diagonalisation may also be applied to the 4×4 Hamiltonian of equation (10.7), using the basis states F_1, F_2, F_4 , and F_5 defined in equation (10.13), which transforms the full 4×4 matrix into one which has two 2×2 blocks. Alternatively, one can simply cut out the third row and third column from the two blocks defined in equation (10.16) to obtain the same result. Taking for example the upper 2×2 block:

$$\begin{bmatrix} P + Q - E & R - iS \\ R + iS & P - Q - E \end{bmatrix} \cdot \begin{bmatrix} F_1 \\ F_2 \end{bmatrix} = 0 \quad (10.19)$$

and evaluating $\det|H_+ - E| = 0$ gives:

$$E^2 - 2PE + P^2 - Q^2 - |R - iS|^2 = 0 \quad (10.20)$$

which has two non-degenerate solutions:

$$E_{\pm} = P \pm \sqrt{Q^2 + |R - iS|^2} \quad (10.21)$$

in which it can be immediately recognised that $E_- = E_{\text{HH}}$ and $E_+ = E_{\text{LH}}$ from the previous section. The same values would be obtained from the lower 2×2 block.

Putting these solutions (one at a time!) back into either of the two equations of the homogeneous linear system in equation (10.19), as convenient, and solving it, gives the eigenvectors for the two eigenenergies as:

$$\begin{bmatrix} F_1 \\ F_2 \end{bmatrix}_{E_-} = \begin{bmatrix} P - Q - E_- \\ -(R + iS) \end{bmatrix} \quad \begin{bmatrix} F_1 \\ F_2 \end{bmatrix}_{E_+} = \begin{bmatrix} (R - iS) \\ E_+ - P - Q \end{bmatrix} \quad (10.22)$$

These eigenvectors are not yet normalised to unity, which can be done simply by dividing each by the square root of the sum of its component moduli squared. In the special case when $k_x = k_y = 0$ and $k_z \neq 0$ the HH state vector is simply $[1 \ 0]^T$ (i.e. only a particular linear combination of $|3/2, \pm 3/2\rangle$ states is present), while the LH state vector is $[0 \ 1]^T$ (only $|3/2, \pm 1/2\rangle$ states are present).

10.6 THE VALENCE BAND IN STRAINED CUBIC SEMICONDUCTORS

In this section the influence of strain on the valence band structure will be considered. Strain may be created in several different ways. One way is to subject a semiconductor to hydrostatic pressure, in which case the cubic unit cell would become compressed equally in all three directions. Another is to apply a uniaxial pressure along some axis (e.g. z), while leaving the surfaces perpendicular to the other two axes free, which would result in compressive strain along z and tensile strain along x and y . Yet another way of straining a semiconductor is to grow an epitaxial lattice-mismatched layer of a material on top of a substrate (i.e. grow a layer of a material with a different lattice constant from that of the substrate). The thin (epitaxial) layer is forced to acquire the in-plane lattice constant of the substrate, and the perpendicular lattice constant will then also change.

Strain is described by 6 components: ϵ_{xx} , ϵ_{yy} , ϵ_{zz} , ϵ_{xy} , ϵ_{xz} , and ϵ_{yz} . It modifies various terms in the Hamiltonian (see equation (10.2)), which then acquire strain-dependent contributions, see for example [231]:

$$\begin{aligned} P &\rightarrow P + P_{\epsilon} \quad , \quad P_{\epsilon} = -a_v(\epsilon_{xx} + \epsilon_{yy} + \epsilon_{zz}) \\ Q &\rightarrow Q + Q_{\epsilon} \quad , \quad Q_{\epsilon} = -\frac{b}{2}(\epsilon_{xx} + \epsilon_{yy} - 2\epsilon_{zz}) \\ R &\rightarrow R + R_{\epsilon} \quad , \quad R_{\epsilon} = \frac{\sqrt{3}}{2}b(\epsilon_{xx} - \epsilon_{yy}) - id\epsilon_{xy} \\ S &\rightarrow S + S_{\epsilon} \quad , \quad S_{\epsilon} = -d(\epsilon_{zx} - i\epsilon_{yz}) \end{aligned} \quad (10.23)$$

where a_v and b , and d are the Pikus-Bir deformation potentials, describing the influence of hydrostatic, uniaxial, and shear strain, respectively.

In the (practically most important) case of biaxial strain generated by lattice-mismatched growth of a semiconductor layer on a [001] oriented substrate, the components of strain become:

$$\begin{aligned}\epsilon_{xx} &= \epsilon_{yy} \neq \epsilon_{zz}, \\ \epsilon_{xy} &= \epsilon_{yz} = \epsilon_{zx} = 0\end{aligned}\quad (10.24)$$

where the non-zero strain components can be evaluated from:

$$\begin{aligned}\epsilon_{xx} &= \epsilon_{yy} = \frac{a_0 - a_{\text{latt}}}{a_{\text{latt}}}, \\ \epsilon_{zz} &= -\frac{2C_{12}}{C_{11}}\epsilon_{xx}\end{aligned}\quad (10.25)$$

and where C_{11} and C_{12} are the stiffness constants, a_0 is the lattice constant of the substrate (which is taken to be unstrained, because it is much thicker than the layer grown on top of it), and a_{latt} is the lattice constant of the unstrained epitaxial material.

Looking again at the special case of $\mathbf{k} = 0$, the Hamiltonian takes the form:

$$\mathbf{H} = \begin{bmatrix} P_\epsilon + Q_\epsilon & 0 & 0 & 0 & 0 & 0 \\ 0 & P_\epsilon + Q_\epsilon & 0 & 0 & 0 & 0 \\ 0 & 0 & P_\epsilon - Q_\epsilon & 0 & \sqrt{2}Q_\epsilon & 0 \\ 0 & 0 & 0 & P_\epsilon - Q_\epsilon & 0 & \sqrt{2}Q_\epsilon \\ 0 & 0 & \sqrt{2}Q_\epsilon & 0 & P_\epsilon + \Delta_{\text{SO}} & 0 \\ 0 & 0 & 0 & \sqrt{2}Q_\epsilon & 0 & P_\epsilon + \Delta_{\text{SO}} \end{bmatrix}\quad (10.26)$$

This is a bit more complicated than the unstrained case. For example, the presence of non-zero off-diagonal matrix elements on positions which link LH and SO states implies that these two bands are coupled even at $\mathbf{k} = 0$, however the HH band remains independent. Away from the zone centre, however, all the three bands are coupled. If the size of the system is reduced to a 4×4 Hamiltonian, by removing the $|1/2, \pm 1/2\rangle$ states (the fifth and sixth rows and columns), a simple diagonal matrix is obtained and its eigenenergies can readily be written, each of which is two-fold degenerate:

$$\begin{aligned}E_{\text{HH}}(0) &= P_\epsilon + Q_\epsilon \\ E_{\text{LH}}(0) &= P_\epsilon - Q_\epsilon\end{aligned}\quad (10.27)$$

Consider the case when a_{latt} is larger than a_0 , so that the thin epitaxial layer experiences a compressive in-plane strain (i.e. $\epsilon_{xx} < 0$). Therefore according to equation (10.25) the result of this accommodation of the in-plane lattice constant to the value of the substrate is that ϵ_{zz} must be greater than zero. The result is a finite value of P_ϵ which, as it only occurs along the diagonal terms in equations (10.2) or (10.15), it has the trivial effect in bulk of a rigid energy shift of all the bands by that amount, with no other physical consequences.

The term Q_ϵ , however, has non-trivial effects. The value of b in semiconductors is negative, therefore $Q_\epsilon < 0$, and the HH band edge will therefore decrease, while the

LH band edge will increase (with the inverted energy picture), hence the HH and LH bands cease to be degenerate, with the light-holes at the zone centre having an energy greater than that of the heavy-holes. In the opposite case with a_{latt} smaller than a_0 , the thin layer acquires a tensile in-plane strain, and the above effects are reversed.

Returning to the 6×6 Hamiltonian, the eigenvalue equation is clearly a sixth-order polynomial in the energy E . It can be written as a square of a third-order polynomial, which can be solved analytically, but this is not an easy thing to do. Again, looking at the special case of $\mathbf{k} = 0$ summarised in equation (10.26), notice that the diagonal elements $P_\epsilon + Q_\epsilon$ stand alone in both their rows and columns, which means that this is a doubly degenerate eigenvalue and can be factored out. The remaining problem fourth-order polynomial is then easier to solve and in fact it can be seen that it is the square of a quadratic polynomial. The eigenenergies, which are the edges of the three valence bands in a strained semiconductor, follow as:

$$\begin{aligned} E_{\text{HH}}(0) &= P_\epsilon + Q_\epsilon \\ E_{\text{LH}}(0) &= P_\epsilon - \frac{1}{2} \left(Q_\epsilon - \Delta_{\text{SO}} + \sqrt{\Delta_{\text{SO}}^2 + 2\Delta_{\text{SO}}Q_\epsilon + 9Q_\epsilon^2} \right) \\ E_{\text{SO}}(0) &= P_\epsilon - \frac{1}{2} \left(Q_\epsilon - \Delta_{\text{SO}} - \sqrt{\Delta_{\text{SO}}^2 + 2\Delta_{\text{SO}}Q_\epsilon + 9Q_\epsilon^2} \right) \end{aligned} \quad (10.28)$$

It is important to note, for the purpose of a later discussion on the band offset at an interface between different materials, that the average ('centre of mass') of these three valence band edges shifts only because of the hydrostatic component P_ϵ , from its unstrained value of $\Delta_{\text{SO}}/3$. In the case of a large spin-orbit splitting ($\Delta_{\text{SO}} \gg Q_\epsilon$), by taking the first two terms in Taylor series expansions of the roots in equation (10.28) the results in equation (10.27) are recovered, while the SO band in this limit appears relatively insensitive to strain.

Values of the parameters necessary for strain calculations in a few common semiconductors are given in Table 10.1.

10.7 HOLE SUBBANDS IN HETEROSTRUCTURES

In a heterostructure, where the material composition is modulated, and hence all the material parameters, as well as the valence band edge energies become position dependent, the Hamiltonian in equation (10.2) has to be modified appropriately. For some time it has been accepted that equation (10.2) is modified in analogy with the effective mass Schrödinger equation: in a 2D system, for example, k_x and k_y remain as they are, because only a plane-wave type of solution is allowed in the non-quantising* directions, along which the potential is constant, however the wave vector k_z along the quantising (growth) direction is substituted for its quantum mechanical operator $-i\partial/\partial z$, and then, in order to preserve the hermiticity of the Hamiltonian, the terms

*In the plane of the quantum well.

of the type γk_z^2 become $k_z \gamma k_z$, while terms of the type γk_z become $(\gamma k_z + k_z \gamma)/2$ (note that the γ 's are position dependent in a heterostructure).

Furthermore, the diagonal elements of the Hamiltonian are amended with the potential V , which could have contributions from the valence band offset in the particular material, the potential from an external electrostatic field, or the self-consistent space-charge electrostatic potential. Following this it was shown by Foreman [228] that further modifications are necessary which improve the accuracy of the method, and under these developments the Hamiltonian now reads:

$$\mathbf{H} =$$

$$\begin{bmatrix} P+Q+V & 0 & -S_- & R & (1/\sqrt{2})S_- & \sqrt{2}R \\ 0 & P+Q+V & -R^\dagger & -S_+ & -\sqrt{2}R^\dagger & (1/\sqrt{2})S_+ \\ -S_-^\dagger & -R & P-Q+V & C & \sqrt{2}Q & \sqrt{3/2}\Sigma_- \\ R^\dagger & -S_+^\dagger & C^\dagger & P-Q+V & -\sqrt{3/2}\Sigma_+ & \sqrt{2}Q \\ (1/\sqrt{2})S_-^\dagger & -\sqrt{2}R & \sqrt{2}Q & -\sqrt{3/2}\Sigma_+^\dagger & P+\Delta_{so}+V & -C \\ \sqrt{2}R^\dagger & (1/\sqrt{2})S_+^\dagger & \sqrt{3/2}\Sigma_-^\dagger & \sqrt{2}Q & -C^\dagger & P+\Delta_{so}+V \end{bmatrix} \quad (10.29)$$

where:

$$\begin{aligned} P &= \left(\frac{\hbar^2}{2m_0} \right) \gamma_1 (k_x^2 + k_y^2 + k_z^2) \\ Q &= \left(\frac{\hbar^2}{2m_0} \right) \gamma_2 (k_x^2 + k_y^2 - 2k_z^2) \\ R &= \sqrt{3} \left(\frac{\hbar^2}{2m_0} \right) (-\bar{\gamma} k_-^2 + \mu k_+^2) \\ S_\pm &= 2\sqrt{3} \left(\frac{\hbar^2}{2m_0} \right) k_\pm [(\sigma - \delta)k_z + k_z \pi] \\ \Sigma_\pm &= 2\sqrt{3} \left(\frac{\hbar^2}{2m_0} \right) k_\pm \left\{ \left[\frac{1}{3}(\sigma - \delta) + \frac{2}{3}\pi \right] k_z + k_z \left[\frac{2}{3}(\sigma - \delta) + \frac{1}{3}\pi \right] \right\} \\ C &= 2 \left(\frac{\hbar^2}{2m_0} \right) k_- [k_z(\sigma - \delta - \pi) - (\sigma - \delta - \pi)k_z] \\ k_\pm &= k_x \pm ik_y, \quad \bar{\gamma} = \frac{1}{2}(\gamma_2 + \gamma_3), \quad \mu = \frac{1}{2}(\gamma_3 - \gamma_2) \\ \sigma &= \bar{\gamma} - \frac{1}{2}\delta, \quad \pi = \mu + \frac{3}{2}\delta, \quad \delta = \frac{1}{9}(1 + \gamma_1 + \gamma_2 - 3\gamma_3) \end{aligned} \quad (10.30)$$

It is possible to show that equation (10.29) reduces to equation (10.2) if the Luttinger parameters (γ_1 , γ_2 and γ_3) are set as constants.

As for the case of the bulk Hamiltonian, equation (10.29) can also be block-diagonalised into two 3×3 blocks, which read:

$$\mathbf{H}_\pm = \begin{bmatrix} P+Q+V & R \mp iS & \sqrt{2}R \pm iS/\sqrt{2} \\ R \pm iS^\dagger & P-Q \mp iC+V & \sqrt{2}Q \mp i\sqrt{3/2}\Sigma \\ \sqrt{2}R \mp iS^\dagger/\sqrt{2} & \sqrt{2}Q \pm i\sqrt{3/2}\Sigma^\dagger & P+\Delta_{so} \pm iC+V \end{bmatrix} \quad (10.31)$$

where P and Q are the same as above, but R , S , Σ and C now read:

$$\begin{aligned}
 R &= -\sqrt{3} \left(\frac{\hbar^2}{2m_0} \right) \gamma_\phi k_{||}^2, & S &= 2\sqrt{3} \left(\frac{\hbar^2}{2m_0} \right) k_{||} [(\sigma - \delta)k_z + k_z\pi] \\
 \Sigma &= 2\sqrt{3} \left(\frac{\hbar^2}{2m_0} \right) k_{||} \left\{ \left[\frac{1}{3}(\sigma - \delta) + \frac{2}{3}\pi \right] k_z + k_z \left[\frac{2}{3}(\sigma - \delta) + \frac{1}{3}\pi \right] \right\} \\
 C &= 2 \left(\frac{\hbar^2}{2m_0} \right) k_{||} [k_z(\sigma - \delta - \pi) - (\sigma - \delta - \pi)k_z] \\
 \gamma_\phi &= \sqrt{\bar{\gamma}^2 + \mu^2 - 2\bar{\gamma}\mu \cos \phi}, & k_{||}^2 &= k_x^2 + k_y^2
 \end{aligned} \tag{10.32}$$

It would now appear unclear what values of γ_2 and γ_3 should be used in γ_ϕ , however it is a good approximation to use their average values across the structure.

For a layer-type structure (i.e. with a constant composition inside any of the layers) use equation (10.29), or equation (10.31), to find the boundary conditions for the wave function at interfaces. This is achieved by formal integration across the interface, in the same way as with the effective mass Schrödinger equation (resulting in the conclusion that ψ , as well as $(1/m^*)\partial\psi/\partial z$, is conserved across the interface). Such integration of equations (10.29) or (10.31) shows that the amplitudes F_i in the wave function vector are individually conserved, and also that there are particular linear combinations of both the derivatives and the amplitudes of all F_i components that are conserved across the interface (see later).

Methods of calculating the eigenstates of heterostructures may be divided in two groups. One of them, which is practical only for one-dimensional heterostructure potentials (like quantum wells), uses a 'layer approach' and first finds the relevant properties of each single layer in the structure, before proceeding to find its 'global' properties—in particular, the eigenstates.

The other approach considers directly the heterostructure as a whole. It is computationally more demanding than the layer approach, but is equally applicable to quantum wells, wires and dots. In either case, the final ingredient we need for the calculation is the position-dependent potential $V(z)$ (or $V(\mathbf{r})$ in multidimensional structures) that is to be used in the Hamiltonian equations (10.29) or (10.31).

10.8 VALENCE BAND OFFSET

In case of an unstrained system the valence band offset has a meaning analogous to that in the conduction band: V here shows the valence band edge (of both the HH and LH branches) in any layer. This is precisely how it enters the Hamiltonian. The valence band offsets at heterointerfaces are generally available from the literature, but there is a considerable amount of scatter in this data. In Table 10.1 values are given of the valence band offset (VBO) with respect to vacuum, for a few common semiconductors. For alloys it is usual to use linear interpolation to approximate.

Having found the VBO values for two materials of interest, we subtract the two in order to get the relative offset.

The VBO data in Table 10.1 are given in the real energy scale, hence the material with higher VBO will be the quantum well—in other words, the relative band offset should be multiplied by -1 before being used in either of the Hamiltonians in equation (10.29) or (10.31), which are written in the inverted energy picture. It should be noted, however, that the valence band offsets determined in this way are only approximate (the exception being GaAs/AlAs), and for more accurate values the literature should be consulted.

The case of strained structures is more complicated, and requires some care in using the available data on valence band offsets. The potential V which should be inserted into the Hamiltonian is *not* any of the band edges, defined by equation (10.28), because these expressions already include the effects of strain and cannot be filtered out of the Hamiltonian. Instead, V in the Hamiltonian is the potential before the addition of strain.

In many cases data can be found on the so-called average valence band energy E_{av} , which is the weighted mean of the three valence bands. In an unstrained material, where HH and LH band edges are degenerate, the weighted mean (in the inverted energy picture!) is $\Delta_{SO}/3$ above the valence band edge. Alternatively, the interface of two semiconductors may be characterised by the average valence band offset, ΔE_{av} , which is the difference of E_{av} 's in the two materials. Since Δ_{SO} is material dependent, the ΔE_{av} is not the same as the valence band offset, even in an unstrained material. It has been established that ΔE_{av} is roughly constant with strain, and can therefore be used as a single parameter to describe the interface.

If a strained semiconductor layer is grown on an unstrained substrate (made of a different material), and the values of ΔE_{av} and Δ_{SO} in the epitaxial layer material are known, then the difference $\Delta E_{av} - \Delta_{SO}/3$ clearly gives the energy of the HH/LH valence band edge in this material *without* strain, measured from the E_{av} value in the substrate. In the same manner the valence band edge can be obtained in all layers of a multilayer structure, measured from the same reference point, regardless of whether a particular layer is in direct contact with the substrate or not. This is the potential that should be used in the Hamiltonian. It can be subtracted from any desired reference energy, if it is preferred to have output energies measured from that point. Since the valence band edge in a layer before strain has no physical significance in the strained system, one reasonable choice for a reference point might be the lowest valence band edge in the quantum well—either HH or LH, whichever came out to be lower (if there are different quantum wells, the deepest one could be chosen). Another choice might be the valence band edge in the substrate, which is unstrained, and still has its HH and LH band edges degenerate.

In some cases, for a particular materials interface and particular strain conditions, data may be quoted like 'AB on CD has the valence band offset of ΔE_v (eV)'. This means that strained material AB grown on unstrained substrate CD has such an offset between its valence band edge (either HH or LH, whichever is the lower in this case) and the valence band edge in the substrate. To use this data for a calculation within

the same material system, but under different strain conditions, it will be necessary to find whether the HH or LH band in the strained AB material makes its valence band edge (which depends on the sign of the strain), then find P_e and Q_e and then use the appropriate expression from equation (10.28) to get the AB valence band edge without strain.

10.9 THE LAYER (TRANSFER MATRIX) METHOD

In order to describe the procedure of finding the quantised state energies and wave functions, the block-diagonal form of the Hamiltonian will be used, i.e. attention will be focussed on just one of the 3×3 blocks. However, the method is straightforwardly applicable to any other size of the Hamiltonian, e.g. the 2×2 , or one of the non-block-diagonalised 6×6 or 4×4 forms, or in fact any other form, see [232].

Consider a structure modulated along one dimension (i.e. a quantum well) with arbitrarily varying material composition and potential (see Figs. 1.9–1.13 for a few examples). For the purpose of finding its bound states, or the tunnelling probability, the heterostructure of interest is subdivided into a number (N_z , number of coordinate points) of thin layers, and within each layer the potential (which includes any self-consistent potential, if such a calculation was performed) is taken to be constant, as are the values of the Luttinger parameters (note that under these conditions $S = \Sigma$ and $C = 0$ in equations (10.31–10.32)). If the structure is step-graded, comprising some number of layers of different material composition and width (perhaps like the structure in Fig. 3.15), and provided any continuously varying potentials (e.g. self-consistent potential) are absent, the computational layers in this calculation coincide with actual material layers, and need not be very thin.

Similarly to the case of quantised states of electrons, the aim is to describe the wave function of a quantised state in terms of its form within each layer. These are then joined at the interfaces using appropriate boundary conditions. Within such an approach the need is not to generate all the energy eigenstates of the bulk Hamiltonian for a specified wave vector, but rather all the solutions that correspond to a definite energy, i.e. the complex band structure has to be found.

In order to find the complex band structure in the valence band, consider a structure grown in the [001] direction, which it is convention to define as the z -axis. Since the structure composition and the potential are modulated (varied) along z , but are constant along the x - and y -axes, the wave function in the x - y plane must behave like a plane wave, hence k_x and k_y must be real, while k_z is arbitrary. The Hamiltonian is written as (to make the writing shorter, the factor $\hbar^2/2m_0$ is taken to be absorbed into the γ parameters):

$$[\mathbf{H}] = [\mathbf{H}_2 k_z^2 + \mathbf{H}_1 k_z + \mathbf{H}_0] \quad (10.33)$$

where \mathbf{H}_0 , \mathbf{H}_1 , and \mathbf{H}_2 are the 3×3 matrices that are associated with the corresponding powers of the wave vector k_z , that is:

$$\begin{aligned}\mathbf{H}_2 &= \begin{bmatrix} (\gamma_1 - 2\gamma_2) & 0 & 0 \\ 0 & (\gamma_1 + 2\gamma_2) & -2\sqrt{2}\gamma_2 \\ 0 & -2\sqrt{2}\gamma_2 & \gamma_1 \end{bmatrix} \\ \mathbf{H}_1 &= \begin{bmatrix} 0 & \mp i2\sqrt{3}\gamma_3 k_{||} & \pm i\sqrt{6}\gamma_3 k_{||} \\ \pm i2\sqrt{3}\gamma_3 k_{||} & 0 & \mp i3\sqrt{2}\gamma_3 k_{||} \\ \mp i\sqrt{6}\gamma_3 k_{||} & \pm i3\sqrt{2}\gamma_3 k_{||} & 0 \end{bmatrix} \\ \mathbf{H}_0 &= \begin{bmatrix} (\gamma_1 + \gamma_2)k_{||}^2 + P_e + Q_e + V & -\sqrt{3}\gamma_\phi k_{||}^2 & -\sqrt{6}\gamma_\phi k_{||}^2 \\ -\sqrt{3}\gamma_\phi k_{||}^2 & (\gamma_1 - \gamma_2)k_{||}^2 + P_e - Q_e + V & \sqrt{2}(\gamma_2 k_{||}^2 + Q_e) \\ -\sqrt{6}\gamma_\phi k_{||}^2 & \sqrt{2}(\gamma_2 k_{||}^2 + Q_e) & \gamma_1 k_{||}^2 + P_e + \Delta_{so} + V \end{bmatrix}\end{aligned}\quad (10.34)$$

At any specified value of the energy E , the values of the complex wave vector k_z may be viewed as eigenvalues of the 3×3 non-linear eigenvalue problem $[\mathbf{H}(k_z) - E][F] = 0$, where $[F]$ is the eigenfunction vector of length 3. This eigenproblem is non-linear because eigenvalues (k_z , not E !) appear in powers of both 1 and 2, in contrast to the standard linear eigenproblem where the energy appears only linearly on the diagonal of the matrix. Non-linear eigenproblems of this type are solved by a trick which converts them to a doubled-in-size, 6×6 linear eigenvalue problem, for which well-developed techniques exist (the method is readily generalised to handle any polynomial-type non-linear eigenproblem). The linear problem to be solved thus reads:

$$\begin{bmatrix} \mathbf{0} & \mathbf{1} \\ -\mathbf{H}_2^{-1}(\mathbf{H}_0 - E) & -\mathbf{H}_2^{-1}\mathbf{H}_1 \end{bmatrix} \begin{bmatrix} u \\ k_z u \end{bmatrix} = k_z \begin{bmatrix} u \\ k_z u \end{bmatrix} \quad (10.35)$$

where $\mathbf{0}$ and $\mathbf{1}$ are the 3×3 null and unity matrices. Note that the first row in equation (10.35) is just an identity. The solution of this non-Hermitian but linear matrix eigenproblem, by standard diagonalisation routines [233], delivers the six, generally complex-valued wave vectors k_z , and the corresponding six eigenstates. These states, denoted as $[u]$, are expressed in the basis F_1 – F_3 for the upper block. The $[u]$ -states are thus particular linear combinations of F_1 – F_3 that, at energy E , behave as plane waves, i.e. $\exp(ik_z z)$, within a layer. It should be noted that the first three components of an eigenvector of equation (10.35) are the amplitudes of the basis states, and the other three, when multiplied by i , will be their derivatives. In further considerations the notation $[u]$ will denote the vector of length 6, with the amplitudes and derivatives.

For convenience, these states may then be divided into two groups, as follows: a state which has purely real k_z may be classified according to the sign of k_z^\dagger , if it is positive the state is 'forward', otherwise it is 'backward'; if k_z is complex, including purely imaginary, it is classified as forward if the wave function decays to the right,

[†]Strictly speaking, it should be tested for the current it carries, where the current density operator is $(2\mathbf{H}_2 k_z + \mathbf{H}_1)/\hbar$, but for the purpose here, the sign test suffices.

otherwise it is backward. There are three forward and three backward states in any one layer and any wave function in the structure may in this layer be written as a linear combination of these six states. The vector containing the three coefficients of forward states will be denoted as $[a_j]$, and that for backward states as $[b_j]$, where the subscript j denotes that these correspond to layer j in the structure. In a vector of length 6, which gives the amplitudes of these states in a wave function at a particular energy, the choice is to write $[a_j]$ first and $[b_j]$ below it.

Having the wave function at the beginning (left hand side) of the j -th layer, written in the basis of eigen- k_z states u , it may be propagated to the beginning of the next layer, by first multiplying the state vector by the corresponding exponentials, i.e. by left-multiplying the state vector with the diagonal matrix $D_j = \text{diag}(\dots, \exp(ik_{zj}d_j), \dots)$, where d_j is the width of the j -th layer, and then using the boundary conditions (the interface matching matrix) to propagate the wave function just across the interface. However, the interface matching matrix is written in the F -basis, equation (10.1), as noted above, and not in the eigen- k_z states basis. It relates the wave function amplitudes and derivatives, written in the F -basis, and is obtained from equation (10.31) in our case (or, if not using the block-diagonal form, the corresponding form would be obtained from equation (10.29)). It demands that:

$$\begin{bmatrix} \mathbf{1} & \mathbf{0} \\ \mathbf{B}_1 & \mathbf{B}_2 \end{bmatrix} \cdot \begin{bmatrix} F \\ F' \end{bmatrix} = \text{const} \quad (10.36)$$

across the interface, where:

$$\mathbf{B}_1 = \begin{bmatrix} 0 & \pm 2\sqrt{3}\pi k_{||} & \mp \sqrt{6}\pi k_{||} \\ \mp 2\sqrt{3}(\sigma - \delta)k_{||} & \pm 2(\sigma - \delta - \pi)k_{||} & \pm \sqrt{2}(2\sigma - 2\delta + \pi)k_{||} \\ \pm \sqrt{6}(\sigma - \delta)k_{||} & \mp \sqrt{2}(\sigma - \delta + 2\pi)k_{||} & \mp 2(\sigma - \delta - \pi)k_{||} \end{bmatrix}$$

$$\mathbf{B}_2 = \begin{bmatrix} \gamma_1 - 2\gamma_2 & 0 & 0 \\ 0 & \gamma_1 + 2\gamma_2 & -2\sqrt{2}\gamma_2 \\ 0 & -2\sqrt{2}\gamma_2 & \gamma_1 \end{bmatrix} \quad (10.37)$$

and:

$$\begin{bmatrix} F \\ F' \end{bmatrix} = \begin{bmatrix} F_1 \\ F_2 \\ F_3 \\ F'_1 \\ F'_2 \\ F'_3 \end{bmatrix} \quad (10.38)$$

where the '+' sign applies to the upper block, and the '-' would apply to the lower block (with F_4 - F_6 functions) when required. The matrix in equation (10.36) will be denoted as \mathbf{I}_j (the interface matrix, corresponding to the layer j). From $\mathbf{I}_{j+1} \begin{bmatrix} F \\ F' \end{bmatrix}_{j+1}^T = \mathbf{I}_j \begin{bmatrix} F \\ F' \end{bmatrix}_j^T$ it follows that the matrix $\mathbf{I}_{j+1}^{-1} \mathbf{I}_j$ transfers a state written in the F -basis from the left into the right-hand side of the interface of layers j and $j+1$.

A wave function written in the u basis, can be re-expressed in the F -basis (amplitudes and derivatives) by multiplying it with the matrix U , made by stacking side-by-side all the eigen- k_z states u_1 – u_6 (i.e. eigenvectors of equation (10.35) in which the lower components have been turned into derivatives, as noted above):

$$U = [u_1|u_2|\dots|u_6], \quad [F] = U[u], \quad [u] = U^{-1}[F] \quad (10.39)$$

Therefore, the transfer matrix $T^{j,j+1}$, relating the wave functions at the beginning of layer j and the beginning of layer $j+1$ (with the wave function written in the u -basis) is given by:

$$T^{j,j+1} = U_{j+1}^{-1} I_{j+1}^{-1} I_j U_j D_j, \quad \begin{bmatrix} a_{j+1} \\ b_{j+1} \end{bmatrix} = T^{j,j+1} \begin{bmatrix} a_j \\ b_j \end{bmatrix} \quad (10.40)$$

To give a more verbose description of equation (10.40): having a state at the beginning of layer j written in the u -basis, first propagate it to the end of that layer, but still inside it (D_j), then express it in the F -basis (U_j), then transfer it across the interface into the beginning of layer $j+1$ ($I_{j+1}^{-1} I_j$), and finally express it back in the u -basis this time of layer $j+1$ (U_{j+1}^{-1}).

The total transfer matrix of a structure, consisting of N layers, may then clearly be calculated as:

$$T^{0,N} = T^{N-1,N} \dots T^{1,2} T^{0,1} \quad (10.41)$$

How can the bound states of a system be found if the transfer matrix is known? First note that it can be written in terms of its four 3×3 blocks as:

$$T = \begin{bmatrix} T_{11} & T_{12} \\ T_{21} & T_{22} \end{bmatrix} \quad (10.42)$$

so that:

$$\begin{bmatrix} a_N \\ b_N \end{bmatrix} = \begin{bmatrix} T_{11} & T_{12} \\ T_{21} & T_{22} \end{bmatrix} \begin{bmatrix} a_0 \\ b_0 \end{bmatrix} \quad (10.43)$$

This is similar to the familiar case of electrons in the conduction band, except that the transfer matrix there is of size 2×2 and its elements are numbers, while here these numbers are replaced by matrix blocks.

Now, assume that, for a particular in-plane wave vector and energy, the complex band structure calculation for the outermost two layers of a quantum well delivers a pair of purely real-valued k_z . In this case (which is analogous to the case of the energy being above the barrier for electrons) there are no bound states. However, if all the k_z 's in some energy range turn out to be complex-valued or simply imaginary, there may be bound states because the wave function composed of such states can be made to decay away from the quantum well region. To find these energies and wave functions it must be noted that on the left-hand side a bound state (whose wave function can be normalised to unity) must have $a_0 = 0$, while b_0 is finite. Therefore, from equation (10.43), on the right-hand side the wave function has the component $a_N = T_{12}b_0$ which decays further to the right, and the component $b_N = T_{22}b_0$

which grows to the right. However, this latter term cannot exist in a bound state wave function, hence it must be the case that:

$$\mathbf{T}_{22}\mathbf{b}_0 = 0 \quad (10.44)$$

For electronic bound states, where \mathbf{T}_{22} and \mathbf{b}_0 are both scalars (numbers), the method would be to search for energies which make $T_{22} = 0$ in which case b_0 is then arbitrary, and eventually determined from the normalisation of the wave function. For holes equation (10.44) is in fact a homogeneous system of linear equations in the components of \mathbf{b}_0 . To have a nontrivial solution the determinant of the matrix \mathbf{T}_{22} must be equal to zero, and this will happen only for some particular values of the energy. Therefore, when searching for bound states the energy is varied and the value of $|\det[\mathbf{T}_{22}(E)]| = 0$ is monitored (note that this determinant is generally a complex number). Upon finding such value(s) of the energy E , the system of equations for \mathbf{b}_0 is solved which can be done to within an arbitrary multiplier (its value is eventually determined from normalisation). Finally, starting with this \mathbf{b}_0 , and $\mathbf{a}_0 = 0$ and using the transfer matrix once again, the wave function for each particular bound state can be generated.

In the energy range where at least one of the wave vectors k_z is real there are no bound states, but the transmission or reflection coefficient of the structure can be calculated. In this case it would be necessary to specify the incident state (all its components, not just one amplitude) for which the transmission is sought.

Despite a more complicated procedure, which has been described above, the layer method is usually much faster than any direct numerical procedure for finding the bound states. This is because it handles only small matrices, while direct diagonalisation uses very large matrices which, in the case of holes, are not tridiagonal. Furthermore, this was a nice illustration of the application of the transfer matrix method to more complex systems, which is why it was considered in detail.

10.10 QUANTUM WELL SUBBANDS

In this section examples of hole subbands will be given, calculated for GaAs quantum wells embedded in the $\text{Al}_{0.5}\text{Ga}_{0.5}\text{As}$ bulk alloy. The valence band edge of GaAs is lower than that of the alloy (in the inverted energy picture), so the GaAs layer forms the quantum well and $\text{Al}_{0.5}\text{Ga}_{0.5}\text{As}$ the barrier. GaAs and AlAs have such similar values of the lattice constant that the structure is taken to be lattice-matched, i.e. unstrained. The material parameters were taken from Table 10.1.

Subband dispersion curves (E versus k dependence) in two structures with different quantum well widths are given in Fig. 10.2. The wave vector in both examples was varied along two different directions in the k_x - k_y plane: the $\langle 01 \rangle$ direction ($k_x \neq 0$ and $k_y = 0$, or vice versa), and the $\langle 11 \rangle$ direction ($k_x = k_y$). It is immediately apparent that the dispersion is quite anisotropic.

The subbands are denoted as HH or LH according to their composition at $k_x = k_y = 0$. For finite in-plane wave vectors, particularly in the outer parts of Fig. 10.2,

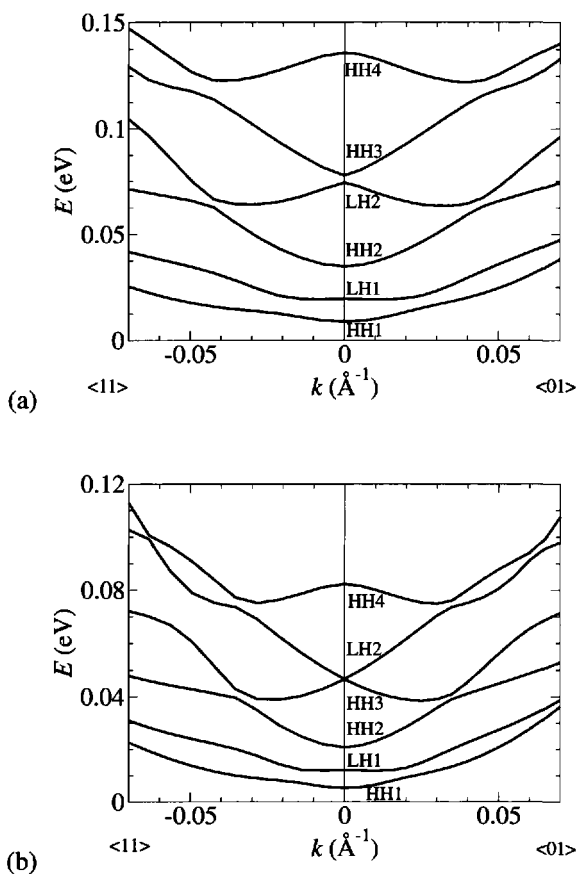


Figure 10.2 The dispersion of the first 6 hole subbands in (a) an 85 \AA wide and (b) a 113 \AA wide GaAs quantum well surrounded by $\text{Al}_{0.5}\text{Ga}_{0.5}\text{As}$ barriers. The energy is measured from the valence band edge in the well.

the mixing of $|3/2, \pm 3/2\rangle$ and $|3/2, \pm 1/2\rangle$ states becomes so strong that it would be difficult to tell, by looking at the wave function composition alone, whether a subband is of HH or LH type. The label HH, for instance, then just means that this state originated as HH at the Γ -point.

The wave vector dependent mixing of the bulk bands is the reason that the subband dispersion along any particular direction is very non-parabolic, even with occasional reversal ($\partial E/\partial k$ becoming negative, which can be interpreted as a negative mass). The ordering of the first few subbands is the same for both well widths: HH1 is the lowest, then comes LH1, then HH2 (this remains so for any well width), but

differences appear for higher subbands. It is also interesting to point out that in the (relatively low) energy range shown in Fig. 10.2 the content of bulk SO band ($|1/2, \pm 1/2\rangle$ basis states) in the subband wave functions is very small. From $<1\%$ at $k_x = k_y = 0$ in the LH subband, it increases but stays typically within the range of $<10\text{--}15\%$ in either the HH or the LH subbands almost throughout the range shown in Fig. 10.2.

10.11 THE INFLUENCE OF STRAIN

Calculation of the hole subband structure in strained quantum wells proceeds by the same method as in the unstrained case, the only (quantitative) difference being the presence of strain terms in the Hamiltonian. In this section the effects of strain on states in quantum wells will be discussed briefly.

Consider the case of a compressively strained quantum well between two tensile strained barriers. An example of this would be a $\text{Si}_{1-x}\text{Ge}_x$ layer embedded in Si, grown on a substrate with a composition between the two (this is a common configuration in order to achieve strain balance). The valence band edge in the SiGe alloy is lower than in Si (in the inverted energy picture), so the former is the well and the latter the barrier. The larger lattice constant of Ge and hence the compressive strain in the $\text{Si}_{1-x}\text{Ge}_x$ layer, imply that the HH band edge in it will decrease from the unstrained value, while the LH band edge would increase (note again the use of the inverted energy picture). The opposite applies in the tensile strained Si barriers. Therefore, the quantum well becomes deeper for heavy- and shallower for light-holes. This is a direct consequence of the uniaxial component of strain, which splits the HH and LH band edges apart.

The hydrostatic component also influences the barrier heights, because the hydrostatic deformation potential has different values in the two materials, but this does not affect the bound state energies very much. If the barriers are not very shallow, the bound state energies *measured from the band edge* mostly depend on the well width, and one can perceive that the strain-induced shift of the band edge (HH or LH) will essentially 'drag' with it all of 'its own' bound states (however, due to mixing of the bulk states, it has to be admitted that the meaning of this is somewhat vague for non-zero values of $k_{||}$). In the unstrained case the lowest HH state is below the lowest LH state, therefore compressive strain will displace the two sets of states, leaving HH as the lowest.

In the opposite case of a quantum well under tensile strain, the valence band edges behave in exactly the opposite manner, and this can lead to the unusual situation of the lowest LH state sinking below the lowest HH state. In the SiGe system it is not possible to get the LH subband as the lowest as this would require a tensilely strained well, and in this case the well (the layer with the larger Ge content) is always compressively strained. However, there are other materials (based on III-V alloys) where the well can be put under tensile strain by the appropriate choice of parameters, and the LH1 subband can be made to be the lowest in the system.

10.12 STRAINED QUANTUM WELL SUBBANDS

As an example of a strained quantum well take a $\text{Si}_{0.6}\text{Ge}_{0.4}$ layer embedded between Si layers, grown on a $\text{Si}_{0.7}\text{Ge}_{0.3}$ substrate. The calculated subband dispersion in two structures with different quantum well widths (different $\text{Si}_{0.6}\text{Ge}_{0.4}$ layer widths), using the material parameters from Table 10.1, and the value $\Delta E_{\text{av}} = 0.56$ eV, is shown in Fig. 10.3. Along with the anisotropic dispersion, just as in the case of the unstrained structure in Fig. 10.2, notice that the ordering of subbands now depends on the well width: it is possible to get two (or even more) HH subbands below the lowest LH subband. This occurs because of the strain-induced displacement of the HH and LH band edges. Another interesting feature in Fig. 10.3(b) is the dispersion of the HH2 subband: starting from the zone centre its energy first decreases before acquiring the expected increase with the wave vector, i.e. it shows an inverted-mass feature, which is brought about by mixing with the LH1 subband, lying just above it.

In contrast to the GaAs/AlGaAs system, in this example the SO band contributes significantly to the composition of all subbands. At the Γ -point its contribution to the LH subbands is of the order of $\sim 10\%$ (this is a consequence of strain alone), and then increases significantly for larger in-plane wave vectors for all the subbands.

10.13 DIRECT NUMERICAL METHODS

An alternative method of finding the quantised states is to solve equation (10.29) or (10.31) when written for the whole structure. One possibility is to use the finite difference approximation which was introduced for approximating derivatives in the effective mass Schrödinger equation of Chapter 3. In the valence band the wave function has more than a single component, hence finite differences never lead to a tridiagonal matrix, even for one-dimensional problems (quantum wells), but rather to a band matrix having non-zero elements along the main and some number of remote diagonals. In the usual implementation the finite difference method implicitly assumes hard wall (box) boundary conditions, as in Fig. 3.6.

Another possibility is to use a plane wave method and expand the wave function components in a Fourier series, hence implicitly assuming the structure to be a superlattice, even though it may be non-periodic—for example a single quantum well embedded in thick barriers. To prevent interaction between the adjacent wells in implicitly existing neighbouring periods care has to be taken to ensure that the barriers are thick enough. The plane-wave method is explained in the chapter on empirical pseudopotentials (Chapter 11), and will not be elaborated on here in any more detail.

The advantage of these methods is that they may be used equally well for structures of higher dimensionality (i.e. quantum wires and quantum dots) with no essential complications in the formulation. However, these methods require considerable amounts of computer memory in order to store the necessary matrices, especially in multidimensional systems, and such matrices also require large computational time for diagonalisation. An alternative is to employ a suitable set of basis functions that

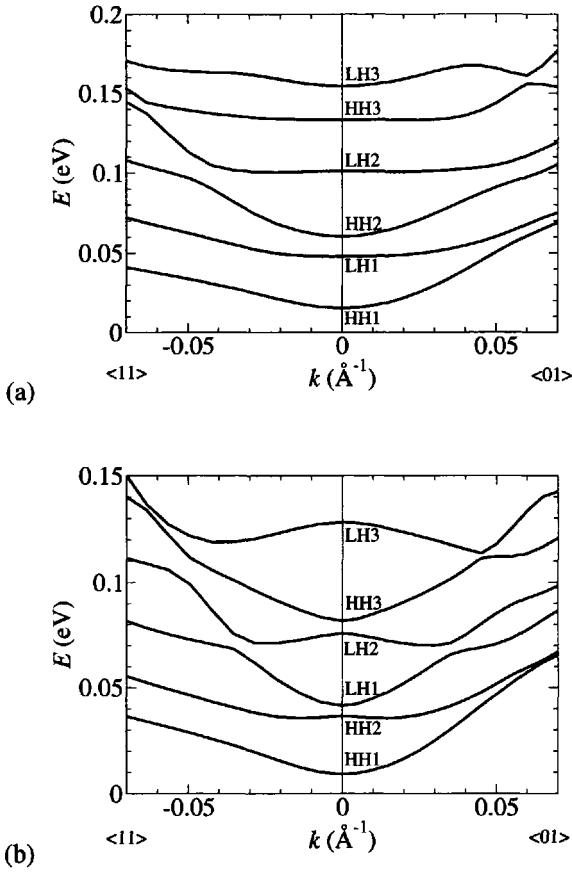


Figure 10.3 The dispersion of the first 6 hole subbands in (a) an 83 Å wide and (b) a 111 Å wide $\text{Si}_{0.6}\text{Ge}_{0.4}$ quantum well embedded in Si barriers, grown on a $\text{Si}_{0.7}\text{Ge}_{0.3}$ substrate. The energy is measured from the HH band edge in the strained well material.

are adapted to the problem at hand, in the sense that using a relatively small number of these functions suffices for good convergence.

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CHAPTER 11

EMPIRICAL PSEUDOPOTENTIAL THEORY

11.1 PRINCIPLES AND APPROXIMATIONS

The envelope function and effective mass approximations, which as stated earlier can be thought of as an approximation to the band structure of a crystal, rather than to the quantum mechanics, are very successful theories, which have allowed many of the most fundamental properties of semiconductors and their heterostructures to be explained. However, it is clear that it is at least worthwhile considering more complex, perhaps more fundamental models to see if they can offer more insight. In particular, given the approximations that the 'single band' effective mass and envelope function models made to the crystal potential, an obvious improvement would be to consider the potential microscopically, i.e. including the potentials of the atoms individually.

The complexity of such a procedure—solving an exact solid, is illustrated by the form of the complete Hamiltonian:

$$\mathcal{H} = \mathcal{H}_{\text{electrons}} + \mathcal{H}_{\text{nuclei}} + \mathcal{H}_{\text{electrons-nuclei}} \quad (11.1)$$

where

$$\mathcal{H}_{\text{electrons}} = \sum_{\mu} \left(-\frac{\hbar^2}{2m_0} \nabla_{\mu}^2 + \sum_{\lambda < \mu} \frac{e^2}{|\mathbf{r}_{\lambda} - \mathbf{r}_{\mu}|} \right) \quad (11.2)$$

and \mathbf{r}_{μ} are the positions of the electrons and m_0 is the mass. In addition:

$$\mathcal{H}_{\text{nuclei}} = \sum_{\nu} \left(-\frac{\hbar^2}{2M_{\nu}} \nabla_{\nu}^2 + \sum_{\lambda < \nu} \frac{Z_{\lambda} Z_{\nu} e^2}{|\mathbf{R}_{\lambda} - \mathbf{R}_{\nu}|} \right) \quad (11.3)$$

where \mathbf{R}_{ν} are the positions, Z_{ν} the atomic numbers and M_{ν} the masses of the nuclei. Finally:

$$\mathcal{H}_{\text{electrons-nuclei}} = - \sum_{\mu, \nu} \frac{Z_{\nu} e^2}{|\mathbf{R}_{\nu} - \mathbf{r}_{\mu}|} \quad (11.4)$$

In a typical macroscopic sample of semiconducting crystal, there are a lot of nuclei, and correspondingly a lot of electrons! In fact for Si, there are about 5×10^{22} atoms per cm^3 . The problem, as stated above, is therefore insolvable. There are however some approximations [234, 235] that can be made which have proved to be acceptably accurate and thus lead to a more manageable problem.

- i. By assuming that the electrons below the outer shell are tightly bound to the nucleus simplifies the system slightly—the atoms can then be treated as separate entities and hence the number of particles is reduced. In addition, these inner-shell electrons screen the outer valence electrons from the central nuclear charge, which has the effect of smoothing out the potential of what would otherwise be a rapidly varying term. Thinking ahead slightly, the smoother the potential then the less terms may be required if the electron wave functions were to be constructed from some Fourier series.
- ii. The *adiabatic approximation* assumes that a change in the coordinates of a nucleus passes no energy to the electrons, i.e. the electrons respond adiabatically, which then allows the decoupling of the motions of the nuclei and the electrons.
- iii. The *independent electron approximation* removes the complications of the electron–electron interactions and replaces them with a time averaged potential.

These approximations allow the system, i.e. the electron wave functions that represent the complete self-interacting electron cloud within the ‘plasma’ of the crystal, to be collapsed down to a one-electron problem with the following Hamiltonian:

$$\mathcal{H} = -\frac{\hbar^2}{2m_0} \nabla^2 + V_c \quad (11.5)$$

where V_c is some, as yet undetermined, crystal potential that represents not only the interaction between the electrons and the nuclei of the atoms that constitute the lattice, but also the interaction between the electrons themselves.

11.2 ELEMENTAL BAND STRUCTURE CALCULATION

Recognising that it is expected a priori that the solutions obtained from a model which will rely on the periodicity of the potential, are going to be dependent on the electron momentum \mathbf{k} , since it is known that electric current flows in semiconductors and from experiences with the Kronig-Penney model. In addition, as more than one solution may be obtained, then the electron wave function may be most generally labelled as $\psi_{n,\mathbf{k}}$. Given these features, then employment of the time-independent (as the potential is static) Schrödinger equation gives:

$$\mathcal{H}\psi_{n,\mathbf{k}} = E_{n,\mathbf{k}}\psi_{n,\mathbf{k}} \quad (11.6)$$

Being influenced by all the previous success in expanding periodic functions in terms of linear combination of a set of basis functions, where each have the periodicity of the system, and by utilising the idea of Bloch ([1], p. 133), for example, it is tempting to expand the wave function $\psi_{n,\mathbf{k}}$ in terms of the complete orthonormal set of plane waves:

$$u_{\mathbf{G},\mathbf{k}} = \frac{1}{\sqrt{\Omega}} e^{i(\mathbf{G}+\mathbf{k})\cdot\mathbf{r}} \quad (11.7)$$

and then:

$$\psi_{n,\mathbf{k}}(\mathbf{r}) = \sum_{\mathbf{G}} a_{n,\mathbf{k}}(\mathbf{G}) u_{\mathbf{G},\mathbf{k}} \quad (11.8)$$

$$\therefore \psi_{n,\mathbf{k}}(\mathbf{r}) = \frac{1}{\sqrt{\Omega}} \sum_{\mathbf{G}} a_{n,\mathbf{k}}(\mathbf{G}) e^{i(\mathbf{G}+\mathbf{k})\cdot\mathbf{r}} \quad (11.9)$$

If \mathbf{R} is a Bravais lattice vector, then:

$$\psi_{n,\mathbf{k}}(\mathbf{r} + \mathbf{R}) = \frac{1}{\sqrt{\Omega}} \sum_{\mathbf{G}} a_{n,\mathbf{k}}(\mathbf{G}) e^{i(\mathbf{G}+\mathbf{k})\cdot(\mathbf{r}+\mathbf{R})} \quad (11.10)$$

which gives:

$$\psi_{n,\mathbf{k}}(\mathbf{r} + \mathbf{R}) = \frac{1}{\sqrt{\Omega}} \sum_{\mathbf{G}} a_{n,\mathbf{k}}(\mathbf{G}) e^{i(\mathbf{G}+\mathbf{k})\cdot\mathbf{R}} e^{i(\mathbf{G}+\mathbf{k})\cdot\mathbf{r}} \quad (11.11)$$

$$\therefore \psi_{n,\mathbf{k}}(\mathbf{r} + \mathbf{R}) = e^{i\mathbf{k}\cdot\mathbf{R}} \frac{1}{\sqrt{\Omega}} \sum_{\mathbf{G}} a_{n,\mathbf{k}}(\mathbf{G}) e^{i\mathbf{G}\cdot\mathbf{R}} e^{i(\mathbf{G}+\mathbf{k})\cdot\mathbf{r}} \quad (11.12)$$

which yields:

$$\psi_{n,\mathbf{k}}(\mathbf{r} + \mathbf{R}) = e^{i\mathbf{k}\cdot\mathbf{R}} \psi_{n,\mathbf{k}}(\mathbf{r}) \quad (11.13)$$

i.e. Bloch's theorem, provided that a set of wave vectors \mathbf{G} can be found for the original expansion, which satisfy:

$$\mathbf{G}\cdot\mathbf{R} = 2\pi n, \quad \text{where } n \in \mathbb{Z} \quad (11.14)$$

Of course, such a set of vectors do exist—they are merely the reciprocal lattice vectors describing the periodicity of the lattice (see Chapter 1).

Therefore, the proposed expansion of the wave function in terms of a linear combination of plane waves satisfies Bloch's theorem and thus it is worthwhile proceeding further in order to deduce the consequences. Hence, substituting the expansion for $\psi_{n,\mathbf{k}}$ given in equation (11.9) into the Schrödinger equation (equation (11.6)), then:

$$\mathcal{H} \sum_{\mathbf{G}} a_{n,\mathbf{k}}(\mathbf{G}) u_{\mathbf{G},\mathbf{k}} = E_{n,\mathbf{k}} \sum_{\mathbf{G}} a_{n,\mathbf{k}}(\mathbf{G}) u_{\mathbf{G},\mathbf{k}} \quad (11.15)$$

Utilising the *linearity* of the Hamiltonian, i.e. it acts on each $u_{\mathbf{G},\mathbf{k}}$ in turn, then the above equation can be written:

$$\sum_{\mathbf{G}} a_{n,\mathbf{k}}(\mathbf{G}) \mathcal{H} u_{\mathbf{G},\mathbf{k}} = E_{n,\mathbf{k}} \sum_{\mathbf{G}} a_{n,\mathbf{k}}(\mathbf{G}) u_{\mathbf{G},\mathbf{k}} \quad (11.16)$$

Multiplying through by a particular plane wave $u_{\mathbf{G}',\mathbf{k}}^*$ and integrating over all space, then:

$$\sum_{\mathbf{G}} a_{n,\mathbf{k}}(\mathbf{G}) \int u_{\mathbf{G}',\mathbf{k}}^* \mathcal{H} u_{\mathbf{G},\mathbf{k}} \, d\tau = \sum_{\mathbf{G}} a_{n,\mathbf{k}}(\mathbf{G}) E_{n,\mathbf{k}} \int u_{\mathbf{G}',\mathbf{k}}^* u_{\mathbf{G},\mathbf{k}} \, d\tau \quad (11.17)$$

Writing:

$$\mathcal{H}_{\mathbf{G}',\mathbf{G}} = \int u_{\mathbf{G}',\mathbf{k}}^* \mathcal{H} u_{\mathbf{G},\mathbf{k}} \, d\tau \quad (11.18)$$

and using the orthonormality property of the basis set, i.e.

$$\int u_{\mathbf{G}',\mathbf{k}}^* u_{\mathbf{G},\mathbf{k}} \, d\tau = \delta_{\mathbf{G}',\mathbf{G}} \quad (11.19)$$

equation (11.17) then becomes:

$$\sum_{\mathbf{G}} a_{n,\mathbf{k}}(\mathbf{G}) \mathcal{H}_{\mathbf{G}',\mathbf{G}} = \sum_{\mathbf{G}} a_{n,\mathbf{k}}(\mathbf{G}) E_{n,\mathbf{k}} \delta_{\mathbf{G}',\mathbf{G}} \quad (11.20)$$

An equation such as this exists for each value of the electron wave vector \mathbf{k} . The solution is therefore reduced to finding the eigenvalues and eigenvectors of the square matrix $\mathcal{H}_{\mathbf{G}',\mathbf{G}}$. In principle, the matrix is of infinite order, but in practice the expansion is limited to a finite set of N plane waves \mathbf{G} , and hence the problem can be solved by direct diagonalisation using one of the many computer libraries that are available. This yields N eigenvalues $E_{n,\mathbf{k}}$ and the corresponding eigenvectors $a_{n,\mathbf{k}}(\mathbf{G})$.

It remains now to construct the individual matrix elements $\mathcal{H}_{\mathbf{G}',\mathbf{G}}$, given by equation (11.18). The Hamiltonian for an electron of rest mass m_0 , moving in the crystal potential V_c , is given by:

$$\mathcal{H} = -\frac{\hbar^2}{2m_0} \nabla^2 + V_c \quad (11.21)$$

Using the form of the plane waves given in equation (11.7), then:

$$\begin{aligned}\mathcal{H}_{\mathbf{G}',\mathbf{G}} &= \frac{1}{\Omega} \int e^{-i(\mathbf{G}'+\mathbf{k})\cdot\mathbf{r}} \left(-\frac{\hbar^2}{2m_0} \nabla^2 \right) e^{i(\mathbf{G}+\mathbf{k})\cdot\mathbf{r}} d\tau \\ &\quad + \frac{1}{\Omega} \int e^{-i(\mathbf{G}'+\mathbf{k})\cdot\mathbf{r}} V_c e^{i(\mathbf{G}+\mathbf{k})\cdot\mathbf{r}} d\tau\end{aligned}\quad (11.22)$$

Therefore:

$$\begin{aligned}\mathcal{H}_{\mathbf{G}',\mathbf{G}} &= \frac{\hbar^2}{2m_0\Omega} \int |\mathbf{G}+\mathbf{k}|^2 e^{i(\mathbf{G}-\mathbf{G}')\cdot\mathbf{r}} d\tau \\ &\quad + \frac{1}{\Omega} \int e^{-i(\mathbf{G}'+\mathbf{k})\cdot\mathbf{r}} V_c e^{i(\mathbf{G}+\mathbf{k})\cdot\mathbf{r}} d\tau\end{aligned}\quad (11.23)$$

Now the integral over all space in the first term only has a value if $\mathbf{G}=\mathbf{G}'$, in which case it is equal to the normalisation volume Ω introduced earlier. This can be summarized by using a Kronecker delta, i.e.

$$\mathcal{H}_{\mathbf{G}',\mathbf{G}} = \frac{\hbar^2}{2m_0} |\mathbf{G}+\mathbf{k}|^2 \delta_{\mathbf{G},\mathbf{G}'} + V \quad (11.24)$$

where V is 'the potential' given by:

$$V = \frac{1}{\Omega} \int e^{-i(\mathbf{G}'+\mathbf{k})\cdot\mathbf{r}} V_c e^{i(\mathbf{G}+\mathbf{k})\cdot\mathbf{r}} d\tau \quad (11.25)$$

Now, the aim of this exercise is to derive a *microscopic* or *atomistic* model of the crystal, so it is clear that the potential V should be written as a sum of some other, as yet undefined, potential that is situated at every atom site \mathbf{r}_a , i.e.

$$V_c = \sum_{\mathbf{r}_a} V_a(\mathbf{r} - \mathbf{r}_a) \quad (11.26)$$

Using this form for V_c in equation (11.25) gives:

$$V = \frac{1}{\Omega} \int e^{-i(\mathbf{G}'+\mathbf{k})\cdot\mathbf{r}} \sum_{\mathbf{r}_a} V_a(\mathbf{r} - \mathbf{r}_a) e^{i(\mathbf{G}+\mathbf{k})\cdot\mathbf{r}} d\tau \quad (11.27)$$

$$\therefore V = \frac{1}{\Omega} \sum_{\mathbf{r}_a} \int V_a(\mathbf{r} - \mathbf{r}_a) e^{i(\mathbf{G}-\mathbf{G}')\cdot\mathbf{r}} d\tau \quad (11.28)$$

As the origin for the summation over the atoms is undefined, then it is possible, and indeed mathematically convenient, to perform the transformation $\mathbf{r} \rightarrow \mathbf{r} + \mathbf{r}_a$, then obtain:

$$V = \frac{1}{\Omega} \sum_{\mathbf{r}_a} \int V_a(\mathbf{r}) e^{i(\mathbf{G}-\mathbf{G}')\cdot(\mathbf{r}+\mathbf{r}_a)} d\tau \quad (11.29)$$

$$\therefore V = \frac{1}{\Omega} \sum_{\mathbf{r}_a} e^{i(\mathbf{G}-\mathbf{G}') \cdot \mathbf{r}_a} \int V_a(\mathbf{r}) e^{i(\mathbf{G}-\mathbf{G}') \cdot \mathbf{r}} d\mathbf{r} \quad (11.30)$$

The term $\sum_{\mathbf{r}_a} e^{i(\mathbf{G}-\mathbf{G}') \cdot \mathbf{r}_a}$ is known as the geometrical structure factor S (see for example, [1], p. 104). Utilising the property discussed earlier that the atomic sites within the crystal are constructed from the Bravais lattice plus the basis, then the sum over the atom sites can be replaced with two separate summations, i.e. one over the Bravais lattice points and one over the points in each basis:

$$S = \sum_{\mathbf{R}} \sum_{\mathbf{t}} e^{i(\mathbf{G}-\mathbf{G}') \cdot (\mathbf{R}+\mathbf{t})} \quad (11.31)$$

where \mathbf{R} is a Bravais lattice vector and \mathbf{t} is a basis vector. Hence:

$$S = \sum_{\mathbf{R}} e^{i(\mathbf{G}-\mathbf{G}') \cdot \mathbf{R}} \sum_{\mathbf{t}} e^{i(\mathbf{G}-\mathbf{G}') \cdot \mathbf{t}} \quad (11.32)$$

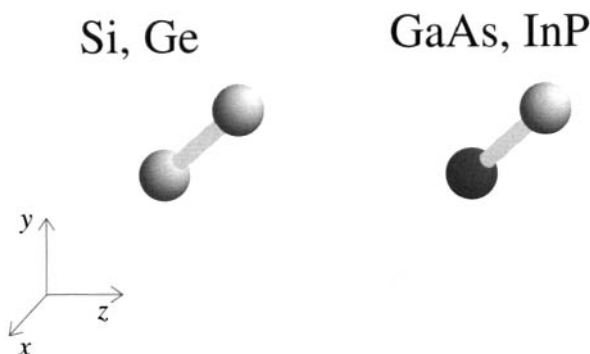


Figure 11.1 The two-atom basis of the diamond structure (left), e.g. Si, Ge, and the zinc blende structure (right), e.g. GaAs, InP

Now the difference between two reciprocal lattice vectors, i.e. $\mathbf{G} - \mathbf{G}'$, must be another reciprocal lattice vector, \mathbf{G}'' . In addition, the scalar product of a Bravais lattice vector with a reciprocal lattice vector is just an integral number of 2π , i.e.

$$\mathbf{R} \cdot \mathbf{G}'' = 2\pi n, \quad \text{where } n \in \mathcal{Z} \quad (11.33)$$

Therefore, $e^{i(\mathbf{G}-\mathbf{G}') \cdot \mathbf{R}} = 1$, and hence for N Bravais lattice points in the crystal:

$$S = N \sum_{\mathbf{t}} e^{i(\mathbf{G}-\mathbf{G}') \cdot \mathbf{t}} \quad (11.34)$$

As discussed earlier in Chapter 1, *most* of the important semiconductors, both elemental and compound, have a face-centred cubic Bravais lattice with a two-atom basis, as shown in Fig. 11.1. The atomic positions within the basis are described by the *basis vector*, $\mathbf{t} = \pm \mathbf{T}$ (say), where:

$$\mathbf{T} = \frac{A_0}{8}(\hat{\mathbf{i}} + \hat{\mathbf{j}} + \hat{\mathbf{k}}) \quad (11.35)$$

Hence, the summation for the structure factor in equation (11.34) contains just two terms, thus giving:

$$S = N \left[e^{i(\mathbf{G}-\mathbf{G}') \cdot \mathbf{T}} + e^{-i(\mathbf{G}-\mathbf{G}') \cdot \mathbf{T}} \right] \quad (11.36)$$

and therefore:

$$S = N \left[\cos (\mathbf{G} - \mathbf{G}') \cdot \mathbf{T} + i \sin (\mathbf{G} - \mathbf{G}') \cdot \mathbf{T} \right. \\ \left. + \cos (\mathbf{G} - \mathbf{G}') \cdot \mathbf{T} - i \sin (\mathbf{G} - \mathbf{G}') \cdot \mathbf{T} \right] \quad (11.37)$$

$$\therefore S = 2N \cos (\mathbf{G} - \mathbf{G}') \cdot \mathbf{T} \quad (11.38)$$

By using this form for the structure factor S , the potential term V , as presented above in equation (11.30), then becomes:

$$V = \frac{2N}{\Omega} \cos (\mathbf{G} - \mathbf{G}') \cdot \mathbf{T} \int V_a(\mathbf{r}) e^{i(\mathbf{G}-\mathbf{G}') \cdot \mathbf{r}} d\tau \quad (11.39)$$

Now the total volume, Ω , divided by the number of Bravais lattice points in the crystal N , is equal to the volume occupied by a single Bravais lattice point, Ω_c (say). Note for the face-centred cubic crystal, the primitive cube of side A_0 (the lattice constant) contains four Bravais lattice points (i.e. eight atoms), and hence:

$$\Omega_c = \frac{A_0^3}{4} \quad (11.40)$$

Furthermore, relabelling $\mathbf{G}' - \mathbf{G}$ as the vector \mathbf{q} , then:

$$V = \frac{2}{\Omega_c} \cos \mathbf{q} \cdot \mathbf{T} \int V_a(\mathbf{r}) e^{-i\mathbf{q} \cdot \mathbf{r}} d\tau \quad (11.41)$$

The factor:

$$\frac{1}{\Omega_c} \int V_a(\mathbf{r}) e^{-i\mathbf{q} \cdot \mathbf{r}} d\tau = V_f(\mathbf{q}) \quad (11.42)$$

is the Fourier transform of the, as yet, undefined atomic potential V_a , and can be labelled conveniently as $V_f(\mathbf{q})$ where $q = |\mathbf{q}|$. This term is known as the *pseudopotential form factor* [236].

It should be noted that as q (the magnitude of \mathbf{q}) is a scalar, then the form factors are spherically symmetric. Physically they represent the central nuclear potential, plus the potential of the inner electron shells, and so for Si, for example, this would include

Table 11.1 The first 65 reciprocal lattice vectors of a face-centred cubic crystal (in units of $2\pi/A_0$)

Direction		Permutations					Magnitude
$\langle 000 \rangle$	$[000]$						0
$\langle 111 \rangle$	$[111]$	$[\bar{1}\bar{1}1]$	$[1\bar{1}\bar{1}]$	$[11\bar{1}]$	$[\bar{1}\bar{1}1]$	$[1\bar{1}1]$	$\sqrt{3}$
	$[\bar{1}\bar{1}\bar{1}]$	$[\bar{1}\bar{1}1]$					
$\langle 200 \rangle$	$[200]$	$[020]$	$[002]$	$[\bar{2}00]$	$[0\bar{2}0]$	$[00\bar{2}]$	2
$\langle 220 \rangle$	$[220]$	$[202]$	$[022]$	$[\bar{2}20]$	$[20\bar{2}]$	$[02\bar{2}]$	$\sqrt{8}$
	$[\bar{2}\bar{2}0]$	$[\bar{2}02]$	$[0\bar{2}2]$	$[\bar{2}20]$	$[\bar{2}02]$	$[0\bar{2}2]$	
$\langle 311 \rangle$	$[311]$	$[131]$	$[113]$	$[\bar{3}11]$	$[\bar{1}31]$	$[\bar{1}13]$	$\sqrt{11}$
	$[\bar{3}\bar{1}\bar{1}]$	$[\bar{1}\bar{3}\bar{1}]$	$[\bar{1}\bar{1}\bar{3}]$	$[\bar{3}11]$	$[\bar{1}31]$	$[\bar{1}13]$	
	$[\bar{3}11]$	$[\bar{1}31]$	$[\bar{1}\bar{1}3]$	$[\bar{3}11]$	$[\bar{1}31]$	$[\bar{1}\bar{1}3]$	
	$[\bar{3}\bar{1}\bar{1}]$	$[\bar{1}31]$	$[\bar{1}\bar{1}\bar{3}]$	$[\bar{3}11]$	$[\bar{1}31]$	$[\bar{1}\bar{1}3]$	
$\langle 222 \rangle$	$[222]$	$[\bar{2}\bar{2}\bar{2}]$	$[\bar{2}22]$	$[\bar{2}\bar{2}2]$	$[\bar{2}22]$	$[\bar{2}\bar{2}2]$	$\sqrt{12}$
	$[\bar{2}\bar{2}\bar{2}]$	$[\bar{2}\bar{2}\bar{2}]$					
$\langle 400 \rangle$	$[400]$	$[040]$	$[004]$	$[\bar{4}00]$	$[0\bar{4}0]$	$[00\bar{4}]$	4

the $1s^2 2s^2 2p^6$ electrons. The remaining four valence electrons, which in an isolated Si atom are found in the 3s and 3p orbitals, are the subject of the investigation; it is their energy levels and charge distributions which determine the electronic properties of the crystal and thus provide the motivation for this theoretical derivation.

The *empirical* nature of the pseudopotential method is incorporated by adjusting the values of $V_f(q)$ in order to achieve the closest agreement of the calculated energy levels with those measured by experimental methods. Note, therefore, that $V_f(q)$ summarizes many of the microscopic electrostatic properties of the crystal. For example, it accounts for the nuclear charge, the inner-shell electrons, the screening provided by these electrons, and under the auspices of the independent electron approximation described earlier, it also accounts for the electron-electron interaction experienced between the valence electrons.

As mentioned above, \mathbf{q} (the difference between two reciprocal vectors), is also a reciprocal lattice vector, and in a bulk crystal, such as a face-centred cubic, it takes discrete values as deduced in Chapter 1 (see Table 11.1). The pseudopotential form factor $V_f(q)$ is, therefore, also a discrete function, only having non-zero values for particular q . Cohen and Bergstresser [236] found that the experimentally determined band structure features of Si could be reproduced by using the values of $V_f(q)$ given in Table 11.2.

It should be noted that as $V_f(q)$ is truncated for $q > \sqrt{11}$ and as $V_f(0)$ only has the effect of shifting the energies up or down then $V_f(q)$ has only three non-zero values, which occur for $q = \sqrt{3}, \sqrt{8}$, and $\sqrt{11}$. Note also that for $q = 2$, the structure factor

Table 11.2 The form factors (in eV) of the common group-IV semiconductor elements, converted from the original values of Cohen and Bergstresser [236]

Material	A_0 (Å)	$V_f(\sqrt{3})$	$V_f(\sqrt{8})$	$V_f(\sqrt{11})$
Si	5.43	-1.43	+0.27	+0.54
Ge	5.66	-1.57	+0.07	+0.41

for the face-centred cubic crystals of interest here, which is given by equation (11.38) becomes:

$$S = 2N \cos \left(\mathbf{q} \cdot \left(\frac{A_0}{8} \hat{\mathbf{i}} + \frac{A_0}{8} \hat{\mathbf{j}} + \frac{A_0}{8} \hat{\mathbf{k}} \right) \right) \quad (11.43)$$

With $\mathbf{q} = (2\pi/A_0)2\hat{\mathbf{i}}$, for example, this then gives:

$$S = 2N \cos \left(\frac{2\pi}{A_0} \frac{A_0}{8} 2\hat{\mathbf{i}} \cdot \hat{\mathbf{i}} \right) = 2N \cos \frac{\pi}{2} \quad (11.44)$$

which is zero, and hence the value of $V_f(q)$ is irrelevant.

Reverting back to the mathematics, then:

$$V = 2V_f(q) \cos (\mathbf{G} - \mathbf{G}') \cdot \mathbf{T} \quad (11.45)$$

which when substituted back into equation (11.24) finally gives the complete form for the matrix elements as

$$\mathcal{H}_{\mathbf{G}', \mathbf{G}} = \frac{\hbar^2}{2m_0} |\mathbf{G} + \mathbf{k}|^2 \delta_{\mathbf{G}, \mathbf{G}'} + 2V_f(q) \cos (\mathbf{G} - \mathbf{G}') \cdot \mathbf{T} \quad (11.46)$$

Fig. 11.2 displays the results of calculations of the bulk band structure of Si by using the form factors of Table 11.2, together with the 65-element plane wave basis set in Table 11.1. The graph shows the three highest-energy valence-band levels, which (in this simple approach, ignoring spin-orbit coupling) are all degenerate at $k = 0$, and two of which remain degenerate across all k . In addition, the two lowest-energy conduction-band states can be clearly seen. In this first calculation, the energy levels are plotted along one of the $\langle 100 \rangle$ directions. The continuous energy *band* nature of the solutions is visible, as expected, which is in contrast to the solutions of the Schrödinger equation in heterostructures, the focus of the majority of this work so far. Furthermore, Fig. 11.2 demonstrates the periodic nature of the band structure, with equal energy levels separated by an electron wave vector equal to $[200]$, i.e. the smallest reciprocal lattice vector along the direction of interest (see Table 11.1). Such periodicity is, of course, merely reflecting the Brillouin zone symmetry structure of the crystal [1] and the edge of the *first* Brillouin is at exactly half of the reciprocal lattice vector in question, i.e. $k = 2\pi/A_0$. Note that there is a slight discrepancy in

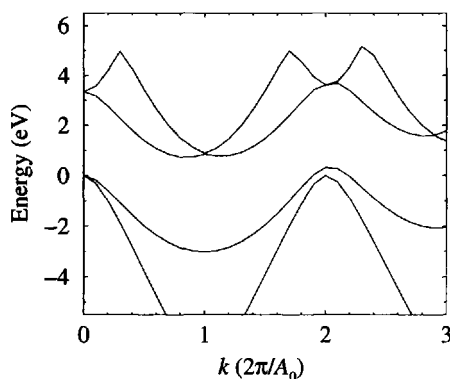


Figure 11.2 The band structure of bulk Si along the $\langle 100 \rangle$ direction, obtained by using the pseudopotential form factors of Cohen and Bergstresser [236], adjusted to set the top of the valence band to zero

the energy levels, e.g. exact degeneracy is not reproduced at the top of the valence band at $k = 2 \times 2\pi/A_0$; this is simply a computational deficiency, and as the vast majority of calculations are confined to the first Brillouin zone, is not an issue here.

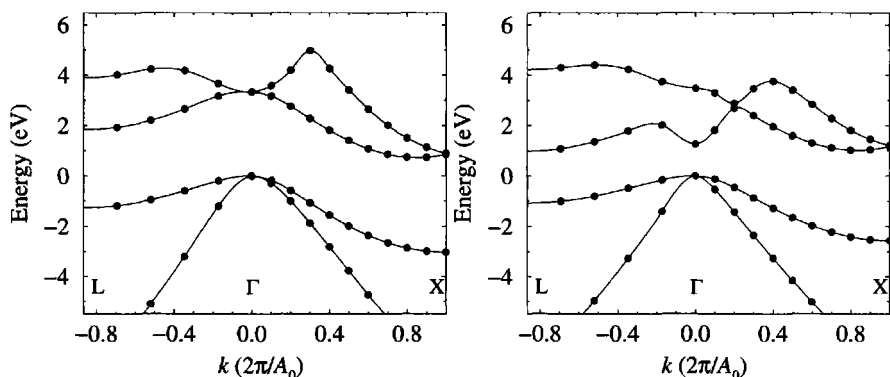


Figure 11.3 The band structures of bulk Si (left) and Ge (right) obtained by using the pseudopotential form factors of Cohen and Bergstresser [236], adjusted to set the top of the valence band to zero

Of course, there is a periodicity in the energy bands along other crystal directions, e.g. the edge of the Brillouin zone along the $\langle 111 \rangle$ direction is at $k = \sqrt{3}/2 \times 2\pi/A_0$. The data from these two principle axes of symmetry are often gathered together to produce a standard form for displaying the band structure, Fig. 11.3 illustrates this for the two most common group-IV elemental semiconductors, i.e. Si and Ge. Again, the form factors given in Table 11.2 were employed, together with the 65-element basis set of Table 11.1. The indirect nature of both materials can be seen from the band-gap

minimum for Si occurring towards the X point within the Brillouin zone, while for Ge it occurs towards L. The choice of the basis set, i.e. plane waves, has been discussed, and the number of 65 can be justified by the data presented in Table 11.3. This table displays the results of calculations of the energy of the lowest conduction band in Si at two electron wave vectors \mathbf{k} . The first is for the Γ point ($\mathbf{k} = 0$), and the second is near the X minima ($\mathbf{k} = 0, 0, 0.8$). In addition, the table shows the discretization of the face-centred reciprocal lattice vectors, plus the largest vector included in the set.

Table 11.3 The energies of the lowest conduction band of Si at Γ and $[0,0,0.8]$ (near the X minima) as a function of the number of plane waves in the basis set

Number of plane waves	Largest \mathbf{G} ($2\pi/A_0$)	E_Γ (eV)	E_X (eV)
15	$\langle 200 \rangle$	3.212	0.926
27	$\langle 220 \rangle$	3.453	0.982
51	$\langle 311 \rangle$	3.552	1.085
59	$\langle 222 \rangle$	3.389	0.827
65	$\langle 400 \rangle$	3.344	0.741
89	$\langle 331 \rangle$	3.382	0.781
113	$\langle 420 \rangle$	3.412	0.825
137	$\langle 422 \rangle$	3.409	0.821

The convergence of the band structure is perhaps most visible through a plot of the above data, as in Fig. 11.4. The data corresponding to a 65-element basis set is just at the bottom of the shoulder, i.e. the beginning of the plateau. It is this accuracy which is generally chosen for simple bulk calculations, such as in Fig. 11.3.

In summary, the *empirical* pseudopotential method is a technique for inputting the experimentally determined features into a theoretical model of the electronic structure of the crystal. At this level, the features that are reproduced are the fundamental band-structure properties, such as the band gap at the centre of the Brillouin Zone, Γ , and at indirect gaps such as the X-valley in Si.

11.3 SPIN-ORBIT COUPLING

Speaking simplistically the four valence electrons of a Si (or Ge) atom, three of which occupy p states and one an s state, produce four equivalent covalent bonds (sp^3 hybridisation) directed towards the corners of a tetrahedron, upon crystallization. Four neighbouring atoms each supply a single electron, and hence any one atom completes its outer shell of eight electrons by forming four doubly occupied bonds. The same is also true for a compound semiconductor, but in this case, however, one

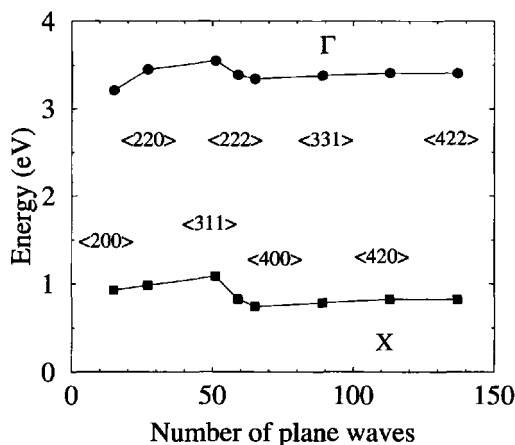


Figure 11.4 The energies of the lowest conduction band of Si at Γ and $[0,0,0.8]$ (near the X minima) as a function of the number of plane waves in the basis set

species, the cation (e.g. Ga) supplies three electrons and the anion (e.g. As) supplies five.

Within an infinite crystal, as has already been shown with the pseudopotential calculations in the previous section, these bonds form *bands*. Fig. 11.5 illustrates the three uppermost of these *valence bands* for Si and Ge. The lowest-energy valence band is far off scale to the bottom of the figure and represents a well localised state around the atomic cores, and has an s-type nature.

The other three bands, in the calculations performed so far, are degenerate for stationary electrons (holes). These bands resemble the p-type atomic orbitals and have an orbital angular momentum, $L=1$ (see for example [18], p. 41). This couples with the spin angular momentum, $S = \frac{1}{2}$ to give $J = L + S = \frac{3}{2}$ and $J = L - S = \frac{1}{2}$, thus lifting the degeneracy. This is analogous to LS coupling in a low-atomic-number atom (see any atomic physics book for an introduction, for example, [3] p. 428 and 441).

In III–V compounds, it has been found that the quadruplet $J = \frac{3}{2}$ is higher in energy than the $J = \frac{1}{2}$ doublet. Note that the degeneracy, whether two fold or four fold, arises from the number of possible values that the projection J_z of J can take. As the total energy remains the same, the change in the energy of the quadruplet $J = \frac{3}{2}$ must be half that of the $J = \frac{1}{2}$ doublet.

The quadruplet is degenerate in bulk systems at the zone centre, but for an increasing electron wave vector it splits into two separate spin degenerate bands. The lower-energy band corresponds to the $m_J = \frac{3}{2}$ projection and because of the effective mass interpretation of the E – k curves, it has a higher effective mass than the higher energy $m_J = \frac{1}{2}$ band. Hence, the two $J = \frac{3}{2}$ bands are labelled as the heavy-hole (HH) ($m_J = \frac{3}{2}$) and the light-hole (LH) ($m_J = \frac{1}{2}$) bands, as shown in Fig. 11.5.

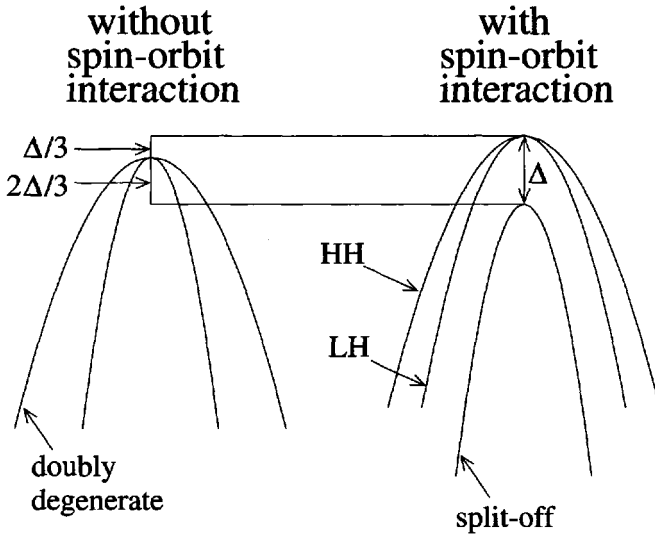


Figure 11.5 The effect of the spin-orbital interaction on the valence band structure

The spin-orbit interaction can be accounted for within the empirical pseudopotential approach [237, 238] and amounts to an addition term in the potential V as originally defined in equation (11.25), i.e. the crystal potential V_c is replaced with $V_c + V_{so}$.

Inclusion of the spin-orbit interaction is essential if detailed calculations of valence band structure are to be performed, and this will be returned to after the elemental calculation has been generalised to compound semiconductors.

11.4 COMPOUND SEMICONDUCTORS

The generalisation of the calculation to compound semiconductors manifests itself merely as a change in the atomic basis.

Recalling equation (11.30):

$$V = \frac{1}{\Omega} \sum_{\mathbf{r}_a} e^{i(\mathbf{G}-\mathbf{G}') \cdot \mathbf{r}_a} \int V_a(\mathbf{r}) e^{i(\mathbf{G}-\mathbf{G}') \cdot \mathbf{r}} d\mathbf{r} \quad (11.47)$$

then again, the structure factor can be written as:

$$S = \sum_{\mathbf{r}_a} e^{i(\mathbf{G}-\mathbf{G}') \cdot \mathbf{r}_a} = \sum_{\mathbf{R}} \sum_{\mathbf{t}} e^{i(\mathbf{G}-\mathbf{G}') \cdot (\mathbf{R}+\mathbf{t})} \quad (11.48)$$

giving as before:

$$S = N \sum_{\mathbf{t}} e^{i(\mathbf{G}-\mathbf{G}') \cdot \mathbf{t}} \quad (11.49)$$

However, for a compound semiconductor the sum over \mathbf{t} is over two dissimilar atoms (the cation is at $-\mathbf{T}$ and the anion is at $+\mathbf{T}$, where $\mathbf{T} = \frac{A_0}{8}(\hat{\mathbf{i}} + \hat{\mathbf{j}} + \hat{\mathbf{k}})$) and therefore it cannot be made independent of the form of the atomic potential $V_a(\mathbf{r})$. Hence, the structure factor S must be substituted back into equation (11.47) at this point, thus giving:

$$V = \frac{N}{\Omega} \sum_{\mathbf{t}} e^{i(\mathbf{G}-\mathbf{G}')\cdot\mathbf{t}} \int V_a(\mathbf{r}) e^{i(\mathbf{G}-\mathbf{G}')\cdot\mathbf{r}} d\tau \quad (11.50)$$

Performing the sum over the two basis positions, $\mathbf{T}^{\text{cat}} = -\mathbf{T}$, and $\mathbf{T}^{\text{an}} = +\mathbf{T}$, then:

$$V = \frac{N}{\Omega} e^{-i(\mathbf{G}-\mathbf{G}')\cdot\mathbf{T}} \int V_a^{\text{cat}}(\mathbf{r}) e^{i(\mathbf{G}-\mathbf{G}')\cdot\mathbf{r}} d\tau + \frac{N}{\Omega} e^{i(\mathbf{G}-\mathbf{G}')\cdot\mathbf{T}} \int V_a^{\text{an}}(\mathbf{r}) e^{i(\mathbf{G}-\mathbf{G}')\cdot\mathbf{r}} d\tau \quad (11.51)$$

Recalling that $N/\Omega = 1/\Omega_c$, and again writing $\mathbf{q} = \mathbf{G}' - \mathbf{G}$, then obtain:

$$V = e^{-i(\mathbf{G}-\mathbf{G}')\cdot\mathbf{T}} \frac{1}{\Omega_c} \int V_a^{\text{cat}}(\mathbf{r}) e^{-i\mathbf{q}\cdot\mathbf{r}} d\tau + e^{i(\mathbf{G}-\mathbf{G}')\cdot\mathbf{T}} \frac{1}{\Omega_c} \int V_a^{\text{an}}(\mathbf{r}) e^{-i\mathbf{q}\cdot\mathbf{r}} d\tau \quad (11.52)$$

Now the definition of the pseudopotential form factor (from equation (11.42)) is as follows:

$$V_f(q) = \frac{1}{\Omega_c} \int V_a(\mathbf{r}) e^{-i\mathbf{q}\cdot\mathbf{r}} d\tau \quad (11.53)$$

$$\therefore V = V_f^{\text{cat}}(q) e^{-i(\mathbf{G}-\mathbf{G}')\cdot\mathbf{T}} + V_f^{\text{an}}(q) e^{i(\mathbf{G}-\mathbf{G}')\cdot\mathbf{T}} \quad (11.54)$$

where the form factors associated with the cation and anion atomic potentials, $V_a^{\text{cat}}(\mathbf{r})$ and $V_a^{\text{an}}(\mathbf{r})$, have been labelled as $V_f^{\text{cat}}(q)$ and $V_f^{\text{an}}(q)$, respectively. This equation can be manipulated further, to give:

$$V = [V_f^{\text{an}}(q) + V_f^{\text{cat}}(q)] \cos(\mathbf{G} - \mathbf{G}')\cdot\mathbf{T} + i [V_f^{\text{an}}(q) - V_f^{\text{cat}}(q)] \sin(\mathbf{G} - \mathbf{G}')\cdot\mathbf{T} \quad (11.55)$$

A quick check on the validity of this extension is to put the cation potential $V_f^{\text{cat}}(q)$ and the anion potential $V_f^{\text{an}}(q)$ equal to the same potential $V_f(q)$, thus reproducing the potential term of the elemental calculation, as in equation (11.45).

Often the 'sum' $[V_f^{\text{an}}(q) + V_f^{\text{cat}}(q)]$ and 'difference' $[V_f^{\text{an}}(q) - V_f^{\text{cat}}(q)]$ potentials are relabelled as the symmetric and anti-symmetric form factors, respectively, i.e. $V_f^S(q)$ and $V_f^A(q)$, this gives:

$$V = V_f^S(q) \cos(\mathbf{G} - \mathbf{G}')\cdot\mathbf{T} + i V_f^A(q) \sin(\mathbf{G} - \mathbf{G}')\cdot\mathbf{T} \quad (11.56)$$

Substituting for the potential V into equation (11.24) finally gives all of the elements of the Hamiltonian matrix as:

$$\mathcal{H}_{\mathbf{G}',\mathbf{G}} = \frac{\hbar^2}{2m_0} |\mathbf{G} + \mathbf{k}|^2 \delta_{\mathbf{G},\mathbf{G}'} + V_f^S(q) \cos(\mathbf{G} - \mathbf{G}')\cdot\mathbf{T}$$

$$+iV_f^A(q) \sin(\mathbf{G} - \mathbf{G}') \cdot \mathbf{T} \quad (11.57)$$

Table 11.4 gives examples of the symmetric ($V_f^S(q)$) and anti-symmetric ($V_f^A(q)$) form factors as deduced *empirically* by Cohen and Bergstresser [236]. Note that the value of the symmetric form factor of $q = 2$, i.e. $V_f^S(2)$, is again unnecessary because of a zero structure factor, as in the elemental (group IV) calculation. In the anti-symmetric case, however, the structure factor corresponding to $q = \sqrt{8}$ is zero, thus allowing this form factor to remain undefined.

Table 11.4 The form factors of a selection of III-V compound semiconductor in eV, converted from the original values of Cohen and Bergstresser [236]

Material	A_0 (Å)	$V_f^S(q)$			$V_f^A(q)$		
		$\sqrt{3}$	$\frac{\sqrt{3}}{\sqrt{8}}$	$\sqrt{11}$	$\sqrt{3}$	2	$\sqrt{11}$
GaAs	5.64	-3.13	+0.14	+0.82	+0.95	+0.68	+0.14
InAs	6.04	-2.99	0.00	+0.68	+1.09	+0.68	+0.41
GaP	5.44	-2.99	+0.41	+0.95	+1.63	+0.95	+0.27
InP	5.86	-3.13	+0.14	+0.82	+0.95	+0.68	+0.14

Fig. 11.6 displays the results of calculations of the band structures of bulk GaAs and InAs. In contrast to Si and Ge the minimum band gaps are direct, and thus an electron and hole can recombine without recourse to a momentum change resulting in efficient light emission. Hence, the use of both of these materials as part of the quaternary compound, $\text{In}_{1-x}\text{Ga}_x\text{As}_y\text{P}_{1-y}$ [239] in light-emitting diodes [240].

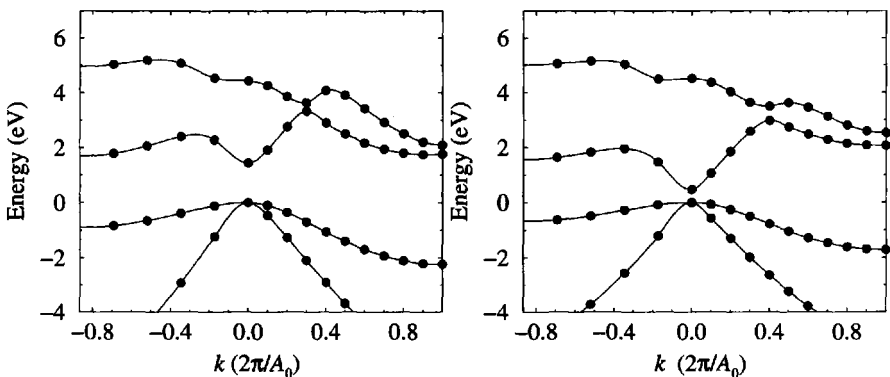


Figure 11.6 The band structures of bulk GaAs (left) and InAs (right) obtained by using the pseudopotential form factors of Cohen and Bergstresser [236], shifted to set the top of the valence band to zero

In recent years, a new material, namely GaN (and its alloys with In), has come to prominence with the development of room-temperature blue-light-emitting diodes [241] and continuous-wave lasers [242, 243]. GaN is particularly interesting from the viewpoint of electronic band-structure calculations in that it does not take the face-centred-cubic zinc blende or diamond structures employed so far in this work, rather the natural crystal structure of GaN being the hexagonal-close-packed wurtzite structure (see, for example [2], p. 47).

Such a change in crystal symmetry does not represent a problem for the analysis derived above, as it is only necessary to change the basis vectors, the reciprocal lattice vectors and, of course, employ the appropriate pseudopotential form factors (for such a treatise see, for example [244]). Many examples will appear in the following which illustrate pseudopotential calculations of systems of symmetry other than zinc blende, e.g. those involving isolated impurities, superlattices, and quantum wires.

11.5 CHARGE DENSITIES

In addition to yielding the eigenvalues (energy levels), the diagonalisation of equations (11.46) and (11.57) also gives the corresponding eigenvectors, i.e. the expansion coefficients $a_{n,\mathbf{k}}$ for each one of the basis vectors \mathbf{G} . It is straightforward to generate the wave function from these eigenvectors by using equation (11.9), i.e.

$$\psi_{n,\mathbf{k}}(\mathbf{r}) = \sum_{\mathbf{G}} a_{n,\mathbf{k}}(\mathbf{G}) e^{i(\mathbf{G}+\mathbf{k})\cdot\mathbf{r}} \quad (11.58)$$

Utilising the probability interpretation of the wave function, i.e. the probability of finding a particle at a point is proportional to $\psi^*\psi$, then the *charge density* is given by:

$$\rho = \psi_{n,\mathbf{k}}^*(\mathbf{r}) \psi_{n,\mathbf{k}}(\mathbf{r}) \quad (11.59)$$

and can be easily evaluated.

Fig. 11.7 illustrates the results of summing the charge density over all of the four bulk valence band states, across a series of x - y planes through a single face-centred cube of Si. Given the coordinate system employed thus far, the Bravais lattice points are, of course, at the corners and on the faces of the cube of side A_0 , with a pair of atoms at $(-\frac{1}{8}, -\frac{1}{8}, -\frac{1}{8})$ and $(+\frac{1}{8}, +\frac{1}{8}, +\frac{1}{8})$ respectively. Therefore, the first ($z = 0$), third ($z = 0.250A_0$), and fifth ($z = 0.5A_0$) planes are cross-sections through the bond centres. The symmetry of the sp^3 -hybridised bonds can be seen most clearly across the planes intersecting the centre of atoms, namely the second ($z = 0.125A_0$) and fourth ($z = 0.375A_0$) planes. In addition, this figure illustrates the equality of the atoms in this, the diamond structure, where the symmetry between the planes containing the atoms is merely translational.

In contrast to this Fig. 11.8 displays the corresponding charge-density plot, again summed over all four valence-band states, but this time for the zinc blende structure of GaAs. Again there is a symmetry between the planes intersecting the bond centres,

i.e. the first ($z = 0$), third ($z = 0.250A_0$), and fifth ($z = 0.5A_0$). However, the two planes intersecting the centres of the atoms are quite dissimilar. In particular, the second ($z = 0.125A_0$) plane contains the anions, in this case As, and is quite different from the fourth ($z = 0.375A_0$) plane which contains the cations, i.e. Ga. The fourth plane displays an increase in the charge density (represented by the lighter colour) around the Ga atoms, while the second plane displays a decrease in the charge density around the As atoms, in comparison to the Si structure of Fig. 11.7. This is a reflection of the increased *ionicity* in the Ga–As bond, in comparison with the completely covalent bond in bulk Si. Instead of having an equal share of the two valence electrons which sit between the atomic cores and form the bond, as occurs in the diamond structure, the Ga atom takes a larger share, thus becoming a partially negatively charged centre (i.e. a cation); the As atom therefore has a smaller share, thus becoming a partially positively charged centre (i.e. an anion).

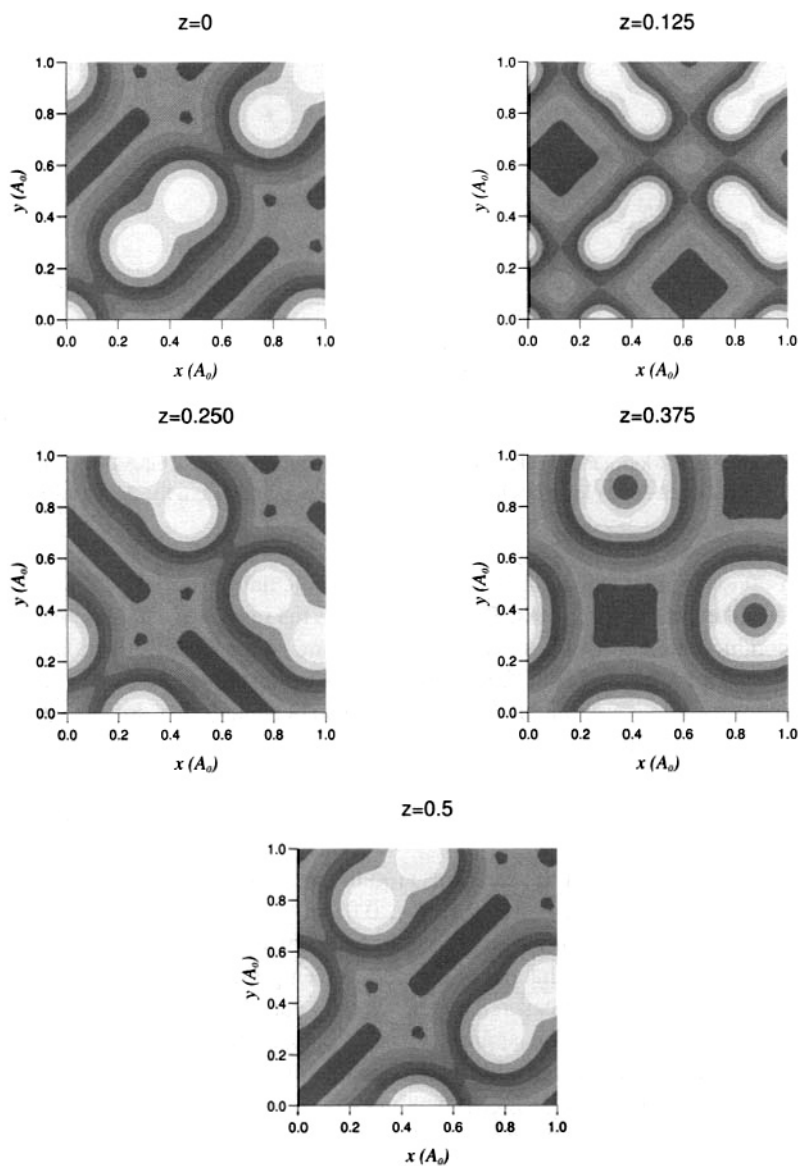


Figure 11.7 Total valence-band charge density for Si across the x - y plane of a single face-centred-cubic unit cell, for a variety of z values (given in A_0); note that the lighter the colour, then the higher the charge density. The Bravais lattice points are at the corners and centre of the $z = 0$ plane and on the edges of the $z = 0.5A_0$ plane

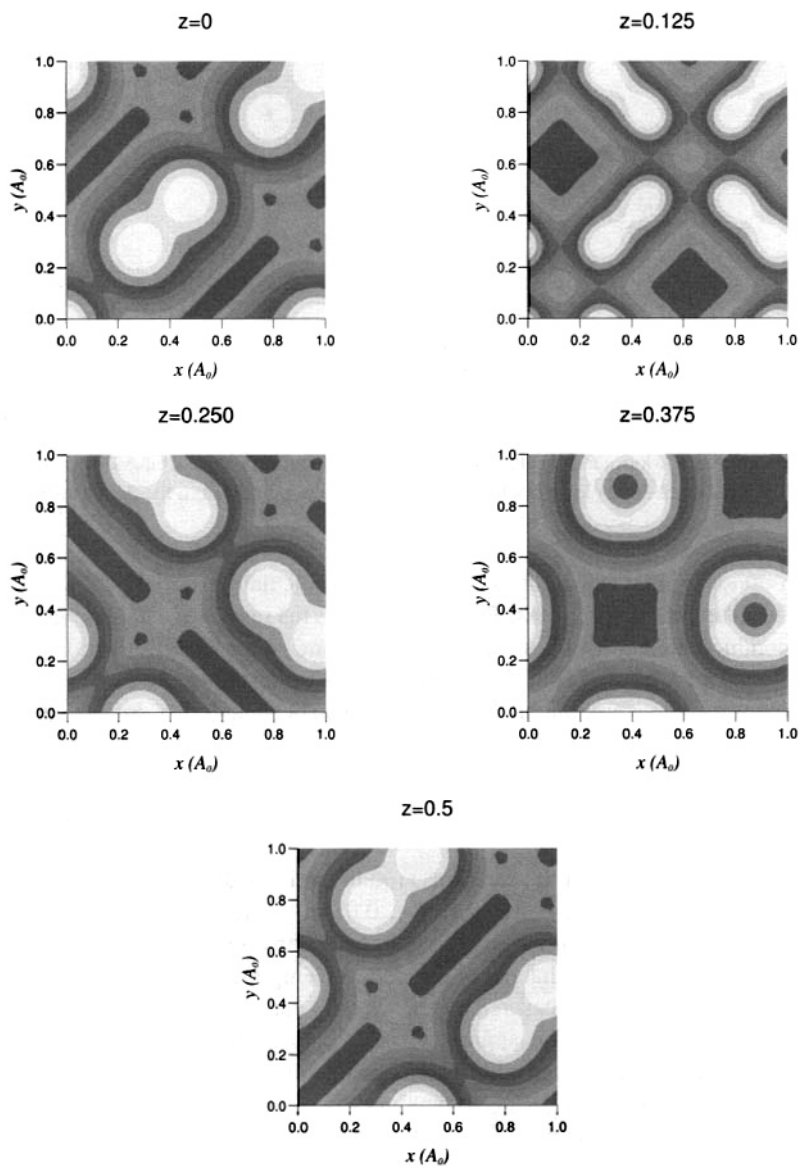


Figure 11.8 Total valence-band charge density for GaAs across the x - y plane of a single face-centred-cubic unit cell, for a variety of z values (given in A_0); note that the lighter the colour, then the higher the charge density. The Bravais lattice points are at the corners and centre of the $z = 0$ plane and on the edges of the $z = 0.5A_0$ plane

11.6 CALCULATING THE EFFECTIVE MASS

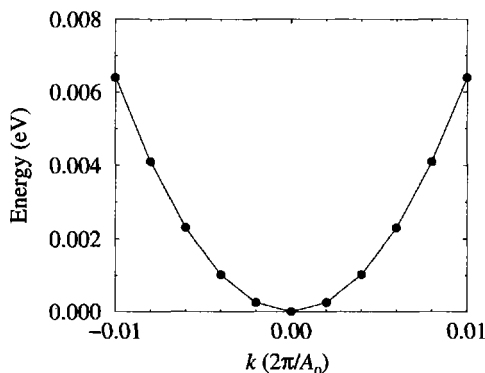


Figure 11.9 Empirical pseudopotential calculations of the band structure of bulk GaAs around the centre (Γ) of the Brillouin zone (solid circles), together with a parabolic fit to the data (line)

Empirical pseudopotential calculations can be used to calculate material parameters other than just the main energy gaps. Fig. 11.9 displays the results of calculations of the lowest conduction band of GaAs at points along one of the $\langle 100 \rangle$ directions around the centre of the Brillouin zone (the minimum has been adjusted to zero). In addition to the data points, a parabolic fit $E = Ak^2$ has also been included, suggested by the definition of the effective mass, i.e.

$$E = \frac{\hbar^2 k^2}{2m^*} \quad (11.60)$$

From this, the effective mass m^* of the electrons around the Γ conduction band minimum can be calculated, in this case giving $m^* = 0.074m_0$. Given the crudity of the form factors employed, this compares well with the accepted value of $0.067m_0$ [14]. This implies that form factors that have been deduced empirically by comparing calculated energy gaps with those measured experimentally can also give good information as to the effective masses. This approach has been used successfully in other materials, and in one case has produced additional insight into the behaviour of excitons in quantum wells [102]. More sophisticated potentials will be introduced later, which in addition to allowing for more accurate determination of effective masses, will also allow for the calculation of deformation potentials in strained crystals.

11.7 ALLOYS

Mixing two binary (or indeed elemental) semiconductors together to form an alloy is a very common and often used technique for producing a whole new range of

materials whose fundamental properties, e.g. the bandgap and the Γ valley effective mass, can be tuned by adjusting the proportions of the constituents [14]. The simplest semiconductor alloy would be that formed from the two common elemental semiconductors Si and Ge [245], but perhaps the most common are the alloys of GaAs, such as $\text{Ga}_{1-x}\text{Al}_x\text{As}$, a ternary alloy formed from GaAs and AlAs.

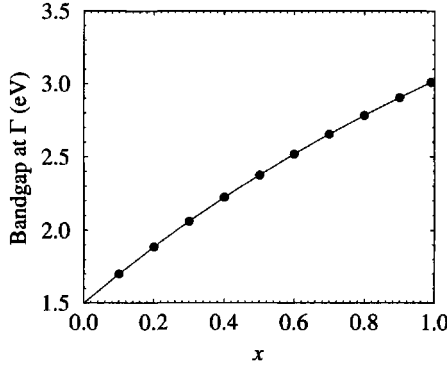


Figure 11.10 The zone-centre bandgap of a $\text{Ga}_{1-x}\text{Al}_x\text{As}$ alloy calculated by linearly interpreting the pseudopotential form factors

Generally such alloys are assumed to have properties which vary linearly between the two constituents, a procedure which is known as the *virtual crystal approximation*, thus giving material parameters which are simply taken in proportion from their components. For example, the bandgap of $\text{Ga}_{1-x}\text{Al}_x\text{As}$ is given by:

$$E_g^{\text{Ga}_{1-x}\text{Al}_x\text{As}} = (1-x)E_g^{\text{GaAs}} + xE_g^{\text{AlAs}} \quad (11.61)$$

However, in some instances such a linear interpolation is only approximate and the actual dependence is more complicated, e.g. it is quite surprising that there is still some disagreement as to the particular form of the bandgap of $\text{Ga}_{1-x}\text{Al}_x\text{As}$, as highlighted by the range of measured values quoted by Adachi ([14], p. 146). A non-linear dependence is often written as:

$$E_g^{\text{Ga}_{1-x}\text{Al}_x\text{As}} = (1-x)E_g^{\text{GaAs}} + xE_g^{\text{AlAs}} + C_{\text{GaAl}}x(x-1) \quad (11.62)$$

where $C_{\text{GaAl}}x(x-1)$ is known as the *bowing parameter*.

Such detailed reviews of the properties of alloys lay beyond the focus of this present work, although it is suffice to say here that the particulars are well documented (see for example [14, 245, 246]) and that in this work it is sufficient to know that equation (11.61) can be generalised to represent any material parameter, e.g. the lattice constant:

$$A_0^{\text{Ga}_{1-x}\text{Al}_x\text{As}} = (1-x)A_0^{\text{GaAs}} + xA_0^{\text{AlAs}} \quad (11.63)$$

or the pseudopotential form factors:

$$V_f(q)^{\text{Ga}_{1-x}\text{Al}_x\text{As}} = (1-x)V_f(q)^{\text{GaAs}} + xV_f(q)^{\text{AlAs}} \quad (11.64)$$

Fig. 11.10 displays the results of a simple demonstration calculation showing the bandgap at the centre of the Brillouin zone as a function of the proportion of Al in the $\text{Ga}_{1-x}\text{Al}_x\text{As}$ alloy; note that a slight degree of bowing is evident here.

The reader may be interested to know that the microscopic properties of alloys are themselves a thriving research topic, one aspect of which is whether the constituent atoms, for example Si and Ge or Ga and Al, are arranged randomly or otherwise (see for example [247–249]).

11.8 ATOMIC FORM FACTORS

It is interesting to decompose the symmetric and anti-symmetric form factors to yield the individual *atomic* form factors, which are given simply by:

$$V_f^{\text{an}}(q) = \frac{1}{2} [V_f^S(q) + V_f^A(q)] \quad (11.65)$$

$$V_f^{\text{cat}}(q) = \frac{1}{2} [V_f^S(q) - V_f^A(q)] \quad (11.66)$$

The term ‘atomic’ is introduced here to specify unambiguously that these are pseudopotential form factors that can be identified with an individual atom, unlike the symmetric, $V_f^S(q)$, and anti-symmetric, $V_f^A(q)$, form factors which are identified with a cation–anion basis.

Table 11.5 The atomic form factors (in eV) of a selection of III–V compound semiconductors, decomposed from their original symmetric and anti-symmetric values as in Table 11.4

Cation	Anion	$V_f^{\text{cat}}(q)$		$V_f^{\text{an}}(q)$	
		$\sqrt{3}$	$\sqrt{11}$	$\sqrt{3}$	$\sqrt{11}$
Ga	As	−2.04	+0.34	−1.09	+0.48
In	As	−2.04	+0.14	−0.95	+0.54
Ga	P	−2.31	+0.34	−0.68	+0.61
In	P	−2.04	+0.34	−1.09	+0.48

Table 11.5 gives the atomic form factors for the cations (Ga, In) and anions (As, P) calculated in this manner. Note at this point that it is not possible to deduce the atomic form factors $V_f^{\text{an}}(q)$ and $V_f^{\text{cat}}(q)$ at $q = 2$ and $\sqrt{8}$ as this would rely on *both* the symmetric ($V_f^S(q)$) and anti-symmetric ($V_f^A(q)$) form factors being known at these points. There are quite some differences between the potentials of the same species in different materials, e.g. the As associated with GaAs, compared to the As associated with InAs. This appears to be a general result, with the physical interpretation being that the *core* potential i.e. nucleus plus inner electron shells, of the As atom is indeed

different in both materials, with this difference arising from the crystal environment. Such a result is disappointing, as a single well-defined atomic potential suitable for all compounds would be a very powerful parameter indeed, allowing the properties of new materials to be predicted without recourse to empirical deductions of the form factors.

11.9 GENERALISATION TO A LARGE BASIS

Thus far, the background to the empirical pseudopotential method has been explored and applied to the calculation of the electronic band structure of bulk elemental and compound semiconductors. It is convenient now to generalise this method to allow for the calculation of the electronic properties of systems of different symmetries, in particular quantum wells, wires and dots. The application of this theory will be covered in the following two chapters.

Consider now the first, and hence most general expression for the crystal potential (equation 11.30), i.e.

$$V = \frac{1}{\Omega} \sum_{\mathbf{r}_a} e^{i(\mathbf{G}-\mathbf{G}') \cdot \mathbf{r}_a} \int V_a(\mathbf{r}) e^{i(\mathbf{G}-\mathbf{G}') \cdot \mathbf{r}} d\tau \quad (11.67)$$

Again writing $\mathbf{q} = \mathbf{G}' - \mathbf{G}$, then:

$$V = \frac{\Omega_c}{\Omega} \sum_{\mathbf{r}_a} e^{-i\mathbf{q} \cdot \mathbf{r}_a} \frac{1}{\Omega_c} \int V_a(\mathbf{r}) e^{-i\mathbf{q} \cdot \mathbf{r}} d\tau \quad (11.68)$$

where Ω_c is again the volume of the primitive cell, which in the case of a face-centred-cubic crystal is $A_0^3/4$. Now, clearly the normalised integral of the atomic potential $V_a(\mathbf{r})$ is still the pseudopotential form factor, although now it is acknowledged that the generalisation may allow for many atom types at, as yet, unspecified positions. Hence, it is important to write:

$$\frac{1}{\Omega_c} \int V_a(\mathbf{r}) e^{-i\mathbf{q} \cdot \mathbf{r}} d\tau \quad \text{as} \quad V_f^{\mathbf{r}_a}(\mathbf{q}) \quad (11.69)$$

implying the empirical pseudopotential form factor of the atom at position \mathbf{r}_a . Therefore:

$$V = \frac{\Omega_c}{\Omega} \sum_{\mathbf{r}_a} e^{-i\mathbf{q} \cdot \mathbf{r}_a} V_f^{\mathbf{r}_a}(\mathbf{q}) \quad (11.70)$$

As with bulk semiconductors, the atomic positions can always be written as a sum of a Bravais lattice vector and a basis vector, i.e.

$$\mathbf{r}_a = \mathbf{R} + \mathbf{t} \quad (11.71)$$

which as before replaces the summation over the atomic positions \mathbf{r}_a with two summations, i.e. one over \mathbf{R} and one over \mathbf{t} , thus giving the following:

$$V = \frac{\Omega_c}{\Omega} \sum_{\mathbf{R}} e^{-i\mathbf{q} \cdot \mathbf{R}} \sum_{\mathbf{t}} e^{-i\mathbf{q} \cdot \mathbf{t}} V_f^{\mathbf{r}_a}(\mathbf{q}) \quad (11.72)$$

Clearly, whatever the symmetry the expansion set of plane waves will always have the periodicity of the system, so just as before the scalar product of a reciprocal lattice vector, $\mathbf{q} = \mathbf{G}' - \mathbf{G}$, with a Bravais lattice vector, \mathbf{R} , will be equal to an integral multiple of 2π , and hence $e^{-i\mathbf{q} \cdot \mathbf{R}} = 1$. If there are N of these new generalised bases in the total volume of the crystal, then:

$$V = \frac{\Omega_c N}{\Omega} \sum_{\mathbf{t}} e^{-i\mathbf{q} \cdot \mathbf{t}} V_f^{\mathbf{r}_a}(q) \quad (11.73)$$

Now the total volume of the crystal, Ω , divided by the number of the new general Bravais lattice points, N , is equal to the volume occupied by one of the bases, Ω_b (say), and therefore:

$$V = \frac{\Omega_c}{\Omega_b} \sum_{\mathbf{t}} e^{-i\mathbf{q} \cdot \mathbf{t}} V_f^{\mathbf{r}_a}(q) \quad (11.74)$$

The diamond, zinc blende and wurtzite crystals discussed so far all have a two-atom basis, and hence Ω_c (the volume of the primitive cell) has been the volume occupied by two atoms. In this more general basis with N_a atoms (say), the volume occupied by its primitive cell would be:

$$\Omega_b = \frac{N_a}{2} \Omega_c \quad (11.75)$$

Therefore, the final expression for the crystal potential follows as:

$$V = \frac{2}{N_a} \sum_{\mathbf{t}} e^{-i\mathbf{q} \cdot \mathbf{t}} V_f^{\mathbf{r}_a}(q) \quad (11.76)$$

and the full expression for the Hamiltonian matrix elements is:

$$\mathcal{H}_{\mathbf{G}', \mathbf{G}} = \frac{\hbar^2}{2m_0} |\mathbf{G} + \mathbf{k}|^2 \delta_{\mathbf{G}, \mathbf{G}'} + \frac{2}{N_a} \sum_{\mathbf{t}} e^{-i\mathbf{q} \cdot \mathbf{t}} V_f^{\mathbf{r}_a}(q) \quad (11.77)$$

At this point, the final equation appears to be merely a re-expression of the Hamiltonian matrix elements of the elemental and compound bulk semiconductors, and it can indeed be shown to reproduce those expressions. For example, consider a ($N_a=$) 2-atom basis, with a cation at $-\mathbf{T} = -\frac{A_0}{8}(\hat{\mathbf{i}} + \hat{\mathbf{j}} + \hat{\mathbf{k}})$ and an anion at $+\mathbf{T}$; then the potential term in equation (11.77) gives:

$$V = \frac{2}{2} [e^{i\mathbf{q} \cdot \mathbf{T}} V_f^{\text{cat}}(q) + e^{-i\mathbf{q} \cdot \mathbf{T}} V_f^{\text{an}}(q)] \quad (11.78)$$

which recalling that $\mathbf{q} = \mathbf{G}' - \mathbf{G}$, is equivalent to equation (11.54).

However, the above is much more than just a generalised form for bulk semiconductor calculations; by thoughtful choice of the atomic basis and the primitive cell, equation (11.77) can be used to calculate the electronic structure of heterostructures of all dimensions, i.e. quantum wells, wires and dots. Such calculations are often referred to as *large-cell* calculations, however, the computational method of summing

over atoms in a more extensive basis suggests the term '*large-basis*' calculations to be more appropriate. The promise of this generalisation will be explored fully in subsequent chapters, however, for now, it is worthwhile pursuing these ideas for bulk systems.

Table 11.6 The form factors (in eV) of a selection of III-V compound semiconductor, converted from the original values of Cohen and Bergstresser [236], where the symmetric form factor at $q = 2$ and the anti-symmetric at $q = \sqrt{8}$ have been deduced by linear interpolation from the two adjacent values

	$\sqrt{3}$	$V_f^S(q)$			$\sqrt{3}$	$V_f^A(q)$		
		2	$\sqrt{8}$	$\sqrt{11}$		2	$\sqrt{8}$	$\sqrt{11}$
GaAs	-3.13	-2.33	+0.14	+0.82	+0.95	+0.68	+0.34	+0.14
InAs	-2.99	-2.26	0.00	+0.68	+1.09	+0.68	+0.51	+0.41
GaP	-2.99	-2.16	+0.41	+0.95	+1.63	+0.95	+0.52	+0.27
InP	-3.13	-2.33	+0.14	+0.82	+0.95	+0.68	+0.34	+0.14

In particular, the brief digress into *atomic form factors* was in anticipation of this formalism—this large-basis method relies on pseudopotential form factors which are associated with individual atoms, and hence in order to perform calculations of bulk compound semiconductors, the symmetric and anti-symmetric form factors listed in Table 11.4 need to be decomposed (as in Section 11.8), into their atomic components. In addition, however, the 'structure factor' associated with these atomic potentials is now of the form $e^{-i\mathbf{q}\cdot\mathbf{t}}$, which is never zero, and hence the *atomic* form factors in this method need to be specified at all values of \mathbf{q} that can arise from the basis set of reciprocal lattice vectors \mathbf{G} .

For a compound semiconductor, this implies that it is necessary to know the atomic form factors at $q = 2$ and $\sqrt{8}$. For the example III-V materials given earlier in Table 11.4, this can be achieved as a zeroth-order approximation simply by linearly interpolating between the existing symmetric and anti-symmetric potentials (as in Table 11.6), and then decomposing the resulting form factors to give the data presented in Table 11.7.

Fig. 11.11 shows the band structures of GaP and InP, calculated by using this large-basis method with a 65-element plane-wave basis set and the atomic form factors of Table 11.7. The validity of this approach, which turns out to be a small generalisation for the bulk case, is substantiated by the close agreement obtained with the original calculations of Cohen and Bergstresser [236].

Table 11.7 The atomic form factors (in eV) of a selection of III-V compound semiconductors, decomposed from their original symmetric and anti-symmetric values and now including the interpolated potentials for $q = 2$ and $\sqrt{8}$ (as indicated by italics)

	$\sqrt{3}$	$V_f^{\text{cat}}(q)$ 2	$\sqrt{8}$	$\sqrt{11}$	$\sqrt{3}$	$V_f^{\text{an}}(q)$ 2	$\sqrt{8}$	$\sqrt{11}$
GaAs	-2.04	-1.51	-0.10	+0.34	-1.09	-0.83	+0.24	+0.48
InAs	-2.04	-1.47	-0.26	+0.14	-0.95	-0.79	+0.26	+0.55
GaP	-2.31	-1.56	-0.06	+0.34	-0.68	-0.61	+0.47	+0.61
InP	-2.04	-1.51	-0.10	+0.34	-1.09	-0.83	+0.24	+0.48

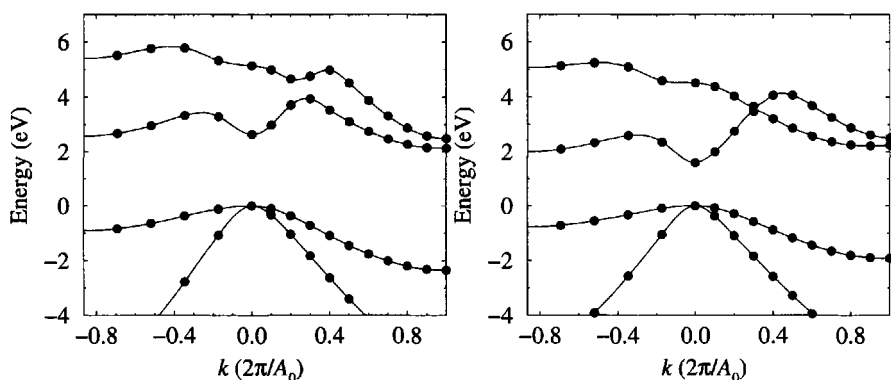


Figure 11.11 The band structures of bulk GaP (left) and InP (right) obtained by using the interpolated atomic form factors given in Table 11.7, shifted to set the top of the valence band to zero

11.10 SPIN-ORBIT COUPLING WITHIN THE LARGE BASIS APPROACH

In constructing spin-dependent solutions the basis set must contain spin-dependent (s or s') terms, hence the matrix elements given in equation (11.18) i.e.

$$\mathcal{H}_{G',G} = \int u_{G',\mathbf{k}}^* \mathcal{H} u_{G,\mathbf{k}} d\tau \quad (11.79)$$

become:

$$\mathcal{H}_{(G',s'),(G,s)} = \int u_{G',\mathbf{k},s'}^* \mathcal{H} u_{G,\mathbf{k},s} d\tau \quad (11.80)$$

The additional potential V_{so} representing the spin-orbit interaction in the total Hamiltonian \mathcal{H} , leads to an additional term within each matrix element [237, 250, 251]. Under the auspices of the large basis approach this term is again a sum over the

contributions from all the N_a atoms within the basis, and in analogy to the crystal potential of equation (11.76), is given by:

$$V_{so} = \frac{1}{N_a} \sum_t e^{-i\mathbf{q} \cdot \mathbf{t}} [-i\lambda^{\mathbf{r}_a} (\mathbf{G}' + \mathbf{k}) \times (\mathbf{G} + \mathbf{k}) \cdot \sigma_{s',s}] \quad (11.81)$$

where $\lambda^{\mathbf{r}_a}$ is a parameter quantifying the magnitude of the interaction and $\mathbf{q} = \mathbf{G}' - \mathbf{G}$. The superscript \mathbf{r}_a indicates that λ is dependent upon the atomic species at all the basis sites within the unit cell. The entity $\sigma_{s',s} = \langle s' | \sigma | s \rangle$ where:

$$\sigma = \sigma_1 \hat{\mathbf{i}} + \sigma_2 \hat{\mathbf{j}} + \sigma_3 \hat{\mathbf{k}} \quad (11.82)$$

and σ_1, σ_2 and σ_3 are the Pauli spin matrices:

$$\sigma_1 = \begin{pmatrix} 0 & 1 \\ 1 & 0 \end{pmatrix} \quad \sigma_2 = \begin{pmatrix} 0 & -i \\ i & 0 \end{pmatrix} \quad \sigma_3 = \begin{pmatrix} 1 & 0 \\ 0 & -1 \end{pmatrix} \quad (11.83)$$

Thus:

$$\sigma = \begin{pmatrix} \hat{\mathbf{k}} & \hat{\mathbf{i}} - i\hat{\mathbf{j}} \\ \hat{\mathbf{i}} + i\hat{\mathbf{j}} & -\hat{\mathbf{k}} \end{pmatrix} \quad (11.84)$$

As partial analytical verification for this representation presented here, consider the case of a zinc-blende basis consisting of 2 atoms, one cation and one anion. In this example, N_a would be 2 and $\mathbf{t} = \pm \mathbf{T}$, where, again, $\mathbf{T} = \frac{A_0}{8}(\hat{\mathbf{i}} + \hat{\mathbf{j}} + \hat{\mathbf{k}})$, i.e.

$$V_{so} = \frac{1}{2} [(-i\lambda^{\text{cat}})e^{i\mathbf{q} \cdot \mathbf{T}} + (-i\lambda^{\text{an}})e^{-i\mathbf{q} \cdot \mathbf{T}}] (\mathbf{G}' + \mathbf{k}) \times (\mathbf{G} + \mathbf{k}) \cdot \sigma_{s',s} \quad (11.85)$$

which gives:

$$V_{so} = \frac{1}{2} [-i\lambda^{\text{cat}} (\cos \mathbf{q} \cdot \mathbf{T} + i \sin \mathbf{q} \cdot \mathbf{T}) - i\lambda^{\text{an}} (\cos \mathbf{q} \cdot \mathbf{T} - i \sin \mathbf{q} \cdot \mathbf{T})] \times (\mathbf{G}' + \mathbf{k}) \times (\mathbf{G} + \mathbf{k}) \cdot \sigma_{s',s} \quad (11.86)$$

Gathering terms in cos and sin, then:

$$V_{so} = \frac{1}{2} [-i(\lambda^{\text{cat}} + \lambda^{\text{an}}) \cos \mathbf{q} \cdot \mathbf{T} - i(\lambda^{\text{cat}} - \lambda^{\text{an}}) i \sin \mathbf{q} \cdot \mathbf{T}] \times (\mathbf{G}' + \mathbf{k}) \times (\mathbf{G} + \mathbf{k}) \cdot \sigma_{s',s} \quad (11.87)$$

which gives:

$$V_{so} = \left[-i \left(\frac{\lambda^{\text{cat}} + \lambda^{\text{an}}}{2} \right) \cos \mathbf{q} \cdot \mathbf{T} + \left(\frac{\lambda^{\text{cat}} - \lambda^{\text{an}}}{2} \right) \sin \mathbf{q} \cdot \mathbf{T} \right] \times (\mathbf{G}' + \mathbf{k}) \times (\mathbf{G} + \mathbf{k}) \cdot \sigma_{s',s} \quad (11.88)$$

The half sum and half difference of the λ parameters are often labelled as symmetric λ^S and anti-symmetric λ^A contributions to the spin-orbit interaction, thus giving:

$$V_{so} = (-i\lambda^S \cos \mathbf{q} \cdot \mathbf{T} + \lambda^A \sin \mathbf{q} \cdot \mathbf{T}) (\mathbf{G}' + \mathbf{k}) \times (\mathbf{G} + \mathbf{k}) \cdot \sigma_{s',s} \quad (11.89)$$

which is the same result as Chelikowsky and Cohen [251].

11.11 COMPUTATIONAL IMPLEMENTATION

With the aim of a computational implementation of this extension to the pseudopotential method*, first consider the new extended basis set. The addition of spin information means that the basis set of plane waves has to be doubled (one set for spin 'up' and one set for spin 'down'). The procedure is to split the new matrix vertically and horizontally into four blocks each one of which is the same size as the original matrix:

$$\left(\begin{array}{c|c} \text{Block 1} & \text{Block 2} \\ \hline s = +\frac{1}{2}, s' = +\frac{1}{2} & s = +\frac{1}{2}, s' = -\frac{1}{2} \\ \hline \text{Block 3} & \text{Block 4} \\ \hline s = -\frac{1}{2}, s' = +\frac{1}{2} & s = -\frac{1}{2}, s' = -\frac{1}{2} \end{array} \right) \quad (11.90)$$

Each block has exactly the same crystal potential components as the original matrix but with the additional spin-dependent component given in equation (11.81). A Kronecker delta $\delta_{s',s}$ acts upon the original components of each matrix element just to ensure that the only spin-dependent part is the new term V_{so} . Thus the complete matrix elements including the spin-orbit interaction within the large basis approach are given by:

$$\mathcal{H}_{(\mathbf{G}',s'),(\mathbf{G},s)} = \mathcal{H}_{\mathbf{G}',\mathbf{G}} \delta_{s',s} + V_{so} \quad (11.91)$$

where for the large basis approach $\mathcal{H}_{\mathbf{G}',\mathbf{G}}$ is given by equation (11.77). In full this would be:

$$\begin{aligned} \mathcal{H}_{(\mathbf{G}',s'),(\mathbf{G},s)} = & \frac{\hbar^2}{2m_0} |\mathbf{G} + \mathbf{k}|^2 \delta_{\mathbf{G}',\mathbf{G}} \delta_{s',s} + \frac{2}{N_a} \sum_t e^{-i\mathbf{q}_t \cdot \mathbf{r}} V_f^{\mathbf{r}_a}(q) \delta_{s',s} + \\ & \frac{1}{N_a} \sum_t e^{-i\mathbf{q}_t \cdot \mathbf{r}} [-i\lambda^{\mathbf{r}_a} (\mathbf{G}' + \mathbf{k}) \times (\mathbf{G} + \mathbf{k}) \cdot \boldsymbol{\sigma}_{s',s}] \end{aligned} \quad (11.92)$$

Thus computationally the quickest way to generate this new extended matrix is to calculate the original crystal potential terms for just the lower triangle of the first 'block' and then copy it into the lower triangle of the fourth block. The spin-orbit term can then be added, with the only other \mathbf{k} -dependent part, the kinetic energy term, added along the leading diagonal last.

For computational reasons it may be worthwhile considering the form of the spin-orbit contribution to the matrix elements, as this can be simplified for more efficient evaluation. With this in mind consider the forms of $\sigma_{s',s}$. The spin states are represented by the spinors $(1, 0)$ and $(0, 1)$, hence:

$$\sigma_{\frac{1}{2}, \frac{1}{2}} = (1, 0) \begin{pmatrix} \hat{\mathbf{k}} & \hat{\mathbf{i}} - i\hat{\mathbf{j}} \\ i + \hat{\mathbf{j}} & -\hat{\mathbf{k}} \end{pmatrix} \begin{pmatrix} 1 \\ 0 \end{pmatrix} = (1, 0) \begin{pmatrix} \hat{\mathbf{k}} \\ i + \hat{\mathbf{j}} \end{pmatrix} = \hat{\mathbf{k}} \quad (11.93)$$

$$\sigma_{\frac{1}{2}, -\frac{1}{2}} = (1, 0) \begin{pmatrix} \hat{\mathbf{k}} & \hat{\mathbf{i}} - i\hat{\mathbf{j}} \\ i + \hat{\mathbf{j}} & -\hat{\mathbf{k}} \end{pmatrix} \begin{pmatrix} 0 \\ 1 \end{pmatrix} = (1, 0) \begin{pmatrix} \hat{\mathbf{i}} - i\hat{\mathbf{j}} \\ -\hat{\mathbf{k}} \end{pmatrix} = \hat{\mathbf{i}} - i\hat{\mathbf{j}} \quad (11.94)$$

*The author would like to thank Fei Long for his assistance

$$\sigma_{-\frac{1}{2}, \frac{1}{2}} = (0, 1) \begin{pmatrix} \hat{\mathbf{k}} & \hat{\mathbf{i}} - i\hat{\mathbf{j}} \\ \hat{\mathbf{i}} + i\hat{\mathbf{j}} & -\hat{\mathbf{k}} \end{pmatrix} \begin{pmatrix} 1 \\ 0 \end{pmatrix} = (0, 1) \begin{pmatrix} \hat{\mathbf{k}} \\ \hat{\mathbf{i}} + i\hat{\mathbf{j}} \end{pmatrix} = \hat{\mathbf{i}} + i\hat{\mathbf{j}} \quad (11.95)$$

$$\sigma_{-\frac{1}{2}, -\frac{1}{2}} = (0, 1) \begin{pmatrix} \hat{\mathbf{k}} & \hat{\mathbf{i}} - i\hat{\mathbf{j}} \\ \hat{\mathbf{i}} + i\hat{\mathbf{j}} & -\hat{\mathbf{k}} \end{pmatrix} \begin{pmatrix} 0 \\ 1 \end{pmatrix} = (0, 1) \begin{pmatrix} \hat{\mathbf{i}} - i\hat{\mathbf{j}} \\ -\hat{\mathbf{k}} \end{pmatrix} = -\hat{\mathbf{k}} \quad (11.96)$$

The vector product $(\mathbf{G}' + \mathbf{k}) \times (\mathbf{G} + \mathbf{k})$, it is clearly equal to some other vector $A_x \hat{\mathbf{i}} + A_y \hat{\mathbf{j}} + A_z \hat{\mathbf{k}}$, say. Then in Block 1 of the Hamiltonian matrix:

$$(\mathbf{G}' + \mathbf{k}) \times (\mathbf{G} + \mathbf{k}) \cdot \sigma_{s', s} = A_z \quad (11.97)$$

In Block 2:

$$(\mathbf{G}' + \mathbf{k}) \times (\mathbf{G} + \mathbf{k}) \cdot \sigma_{s', s} = A_x - iA_y \quad (11.98)$$

In Block 3:

$$(\mathbf{G}' + \mathbf{k}) \times (\mathbf{G} + \mathbf{k}) \cdot \sigma_{s', s} = A_x + iA_y \quad (11.99)$$

In Block 4:

$$(\mathbf{G}' + \mathbf{k}) \times (\mathbf{G} + \mathbf{k}) \cdot \sigma_{s', s} = -A_z \quad (11.100)$$

11.12 DEDUCING THE PARAMETERS AND APPLICATION

The primary aim of the book is to demystify the theory of semiconductor heterostructures, and make such calculations more *accessible*. This current topic of spin-orbit coupling within the empirical pseudopotential approach is one example where this is paramount, as the literature can be quite confusing. The spin-orbit interaction V_{so} given in equation (11.81) is an energy, and λ is usually quoted in Rydbergs (a unit of energy), thus the vectors \mathbf{G} and \mathbf{k} must be dimensionless. The question is what dimensions? In this work the result of the vector and scalar product, i.e. $(\mathbf{G}' + \mathbf{k}) \times (\mathbf{G} + \mathbf{k}) \cdot \sigma_{s', s}$, is made dimensionless by dividing by $(2\pi/A_0)^2$. As the structure factor $e^{-i\mathbf{q} \cdot \mathbf{r}}$ is dimensionless then the contribution V_{so} to the matrix element takes the units of the parameter λ .

Not all authors use this convention and this is reflected in the literature by the usual choice to quote the spin-orbit splitting Δ_0 between the doubly-degenerate light- and heavy-hole bands and the split-off band. In some instances the symmetric λ^S and anti-symmetric λ^A parameters are quoted for binary compounds and it is a simple task to obtain the spin-orbit parameters for the individual atoms as required here:

$$\lambda^{\text{cat}} = \lambda^S - \lambda^A \quad \lambda^{\text{an}} = \lambda^S + \lambda^A \quad (11.101)$$

Fig. 11.12 shows the results of a calculation of the band structure of CdTe using the original form factors of Cohen and Bergstresser [236] with the spin-orbit parameters

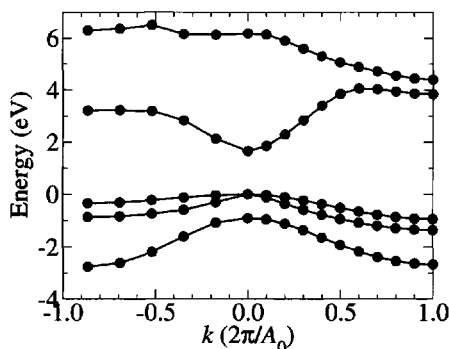


Figure 11.12 The band structure of bulk CdTe calculated with 169 plane waves and adjusted to set the top of the valence band to zero

λ^S and λ^A quoted by Bloom and Bergstresser [252]. It was found that the λ values had to be multiplied by a factor of 0.343 in order to obtain a splitting Δ_0 of 905 meV in agreement with the values of 910 and 900 meV quoted by Bloom and Bergstresser. This is a relatively quick process because to *first-order* the valence-band splitting is proportional to the λ parameters, so if the result for Δ_0 is half the experimental value, scaling the lambda parameters by a factor of 2 will 'get you close'. The splitting of the light- and heavy-hole at the X ($k = (0, 0, 2\pi/A_0)$) is 415 meV and at the L point ($k = (\pi/A_0, \pi/A_0, \pi/A_0)$) 525 meV, which are both in reasonable agreement with the values quoted by Chelikowsky and Cohen [251] of 380 and 530 meV respectively. These results serve to substantiate the validity of the spin-orbit extension. Note though that there is some discrepancy in the calculations here and disagreement between the results quoted [251, 252], this serves as an illustration of the uncertainty surrounding the knowledge of remote areas of the band structure (away from high symmetry points and band edges) but also it must be noted that the spin-orbit parameters should be chosen self-consistently with the atomic form factors—a tedious process that is not going to be attempted here.

Taking μ' as some adjustable parameter (which can again absorb any differences in units employed) Chelikowsky and Cohen [251] give the spin-orbit splittings for a binary compound AB as:

$$\lambda^A = \mu' \quad \lambda^B = \alpha\mu' \quad (11.102)$$

where α is the ratio of the spin-orbit splittings of the free atoms, where the latter can be looked up in a book [254]. Unfortunately the book only gives data for atoms found in II–VI compounds, so it is difficult to know what α should be for any III–V material. Thus any calculation for a III–V material, such as GaAs, would be dependent upon

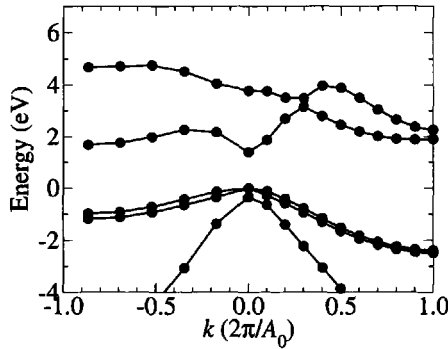


Figure 11.13 The band structure of bulk GaAs using the spin-free form factors of Mader and Zunger [253] together with the spin-orbit parameters described in the text

two parameters which means a lot of work may be required to limit the values they take and produce good agreement with experiment.

To 'cut a corner' consider just *interpolating* the spin-orbit splittings of the free Ga and As atoms from their neighbouring entries in the tabulated data of Herman and Skillman (p. 2-6), using their notation:

$$E(S-O)(GA) = \frac{E(S-O)(Zn) + E(S-O)(Ge)}{2} \quad (11.103)$$

$$E(S-O)(AS) = \frac{E(S-O)(Ge) + E(S-O)(Se)}{2} \quad (11.104)$$

i.e.

$$E(S-O)(GA) = \frac{-0.00596 - 0.00515}{2} = -0.00556 \quad (11.105)$$

$$E(S-O)(AS) = \frac{-0.00515 - 0.01026}{2} = -0.00771 \quad (11.106)$$

Giving:

$$\alpha = \frac{E(S-O)(AS)}{E(S-O)(GA)} = 1.38 \quad (11.107)$$

Using this, then the value of μ' was adjusted (and found to be 0.000402) to give a valence-band splitting of 350 meV. This in itself is not enough to substantiate the calculation, however at the same time the light-hole-heavy-hole splitting at X is 97 meV which is incredibly close to the 100 meV quoted by Chelikowsky and Cohen, see Fig. 11.13.

This shows the robustness of the method—while this GaAs calculation has been forced through and is not necessarily scientifically rigorous, this calculation does

demonstrate that the technique can be used to obtain the valence band structure of III–V materials to some degree of acceptability and should further accuracy be required then more time could justifiably be invested in determining a better set of parameters.

It is a much simpler task to deduce the spin–orbit parameters of the elemental group IV semiconductors because with only one atom type, there is only one parameter. Using the methods outlined above λ was found to be 0.000106 and 0.00058 Rydberg in order to give valence band splittings of 44 and 296 meV for Si and Ge respectively. This gave rise to the band structures plotted in Fig. 11.14.

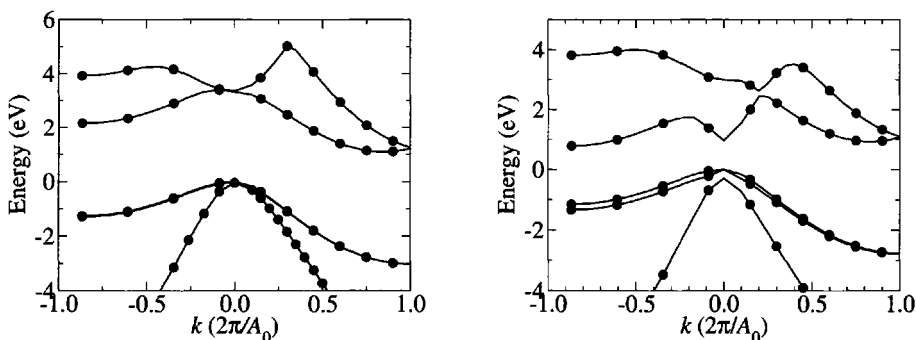


Figure 11.14 The band structures of bulk Si (left) and Ge (right) including spin–orbit coupling along the usual $L \rightarrow \Gamma \rightarrow X$ line

The calculations are based on the spin-free atomic form factors for Si and Ge heterostructures given by Friedel *et al.* [255]. They are in excellent agreement with the data presented by Chelikowsky and Cohen and in particular for Ge, the light-hole–heavy-hole splitting at the L-point is 186 meV in comparison with the quoted value of 200 meV.

11.13 ISOELECTRONIC IMPURITIES IN BULK

A generalisation of the standard two-atom basis within the empirical pseudopotential method was demonstrated in Section 11.9, although hitherto the calculations presented have merely seemed more complex than previous and the investment seems to have produced no visible benefit. In fact, the generalisation was motivated by the desire to study more complex distributions of atoms within a crystal, where primarily this implies quantum well heterostructures, wires and dots (the latter having their own chapters). However, in addition, some bulk systems of interest still remain and the techniques are now at hand for their solutions.

The first case in question is that of a single impurity atom within a host semiconductor. This topic is itself extensive, and has already been given a substantial chapter in this text within the envelope function and effective mass approximations. The interest here lies now with the ability to explore the microscopic electronic properties from a more fundamental point of view. Indeed, such a detailed study has already been undertaken by Jaros [235]—here, a very brief demonstration will be given as to how the large basis calculation *may* be applied to such instances.

The first point to be addressed is: ‘What is the nature of the atomic basis?’. Clearly, a host crystal needs to be specified and a single atom, representing the impurity, needs to be placed substitutionally on one of the lattice sites. It is perhaps convenient in this scenario to consider a cube of Si, and in this case the impurity atom chosen will be Ge, which would, of course, be *isoelectronic*.



Figure 11.15 The non-primitive eight atom basis of a group-IV face-centred-cubic crystal (diamond structure)

Ashcroft and Mermin ([1], p. 75), give the non-primitive face-centred cubic four-point basis vectors as:

$$\mathbf{T}_0 = \mathbf{0}; \quad \mathbf{T}_1 = \frac{A_0}{2}(\hat{\mathbf{i}} + \hat{\mathbf{j}}); \quad \mathbf{T}_2 = \frac{A_0}{2}(\hat{\mathbf{j}} + \hat{\mathbf{k}}); \quad \mathbf{T}_3 = \frac{A_0}{2}(\hat{\mathbf{k}} + \hat{\mathbf{i}}) \quad (11.108)$$

These are chosen to represent the cubic symmetry of the face-centred-cubic lattice more clearly. When applied to the diamond or zinc blende structures, such a set of vectors would then imply an eight-atom basis, with atoms at $\pm \mathbf{T} = \pm \frac{A_0}{8}(\hat{\mathbf{i}} + \hat{\mathbf{j}} + \hat{\mathbf{k}})$ for each of the above, i.e.

$$\mathbf{r}_a = \mathbf{T}_i \pm \frac{A_0}{8}(\hat{\mathbf{i}} + \hat{\mathbf{j}} + \hat{\mathbf{k}}), \quad \text{where } i = 0, 1, 2, 3 \quad (11.109)$$

as displayed in Fig. 11.15. The Bravais lattice vectors then simply follow as:

$$\mathbf{a}_1 = A_0\hat{\mathbf{i}}; \quad \mathbf{a}_2 = A_0\hat{\mathbf{j}}; \quad \mathbf{a}_3 = A_0\hat{\mathbf{k}} \quad (11.110)$$

Such an alternative way of constructing the crystal is useful for defining large cells within a host semiconductor. In this case, where a description of a single isolated impurity is the aim, it is necessary to generalise this to produce a larger cube. For example, if n_x , n_y , and n_z of these cubes were stacked together along the x -, y -, and z - axes, respectively then the atomic positions within this larger basis would be given by:

$$\mathbf{r}_a = m_x A_0 \hat{\mathbf{i}} + m_y A_0 \hat{\mathbf{j}} + m_z A_0 \hat{\mathbf{k}} + \mathbf{T}_i \pm \frac{A_0}{8} (\hat{\mathbf{i}} + \hat{\mathbf{j}} + \hat{\mathbf{k}}) \quad (11.111)$$

where again the index $i = 0, 1, 2$ and 3 and the additional indices $m_{x,y,z} = 0, 1, 2, \dots, n_{x,y,z}$. Fig. 11.16 illustrates a large basis produced by stacking eight face-centred-cubic cells together, i.e. $n_x = n_y = n_z = 2$, thus giving 64 atoms in total.

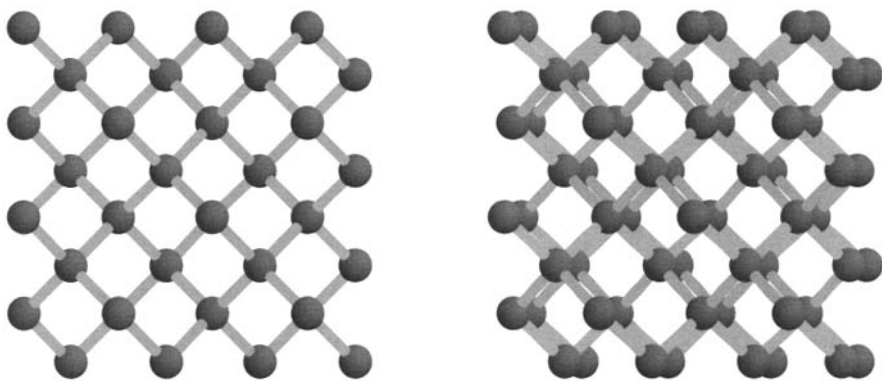


Figure 11.16 A 64-atom basis for the diamond structure, shown looking along one of the $\langle 100 \rangle$ directions (left) and slightly off-axis (right)

For this large ($n_x \times n_y \times n_z$) cuboid, the primitive reciprocal lattice vectors would simply generalise to:

$$\mathbf{b}_1 = \frac{1}{n_x} \frac{2\pi}{A_0} \hat{\mathbf{i}}; \quad \mathbf{b}_2 = \frac{1}{n_y} \frac{2\pi}{A_0} \hat{\mathbf{j}}; \quad \mathbf{b}_3 = \frac{1}{n_z} \frac{2\pi}{A_0} \hat{\mathbf{k}} \quad (11.112)$$

Hence, the basis- \mathbf{G} set for expansion is given by all linear combinations of the above, i.e.

$$\mathbf{G} = \beta_1 \mathbf{b}_1 + \beta_2 \mathbf{b}_2 + \beta_3 \mathbf{b}_3, \quad \text{where } \beta_i \in \mathcal{Z} \quad (11.113)$$

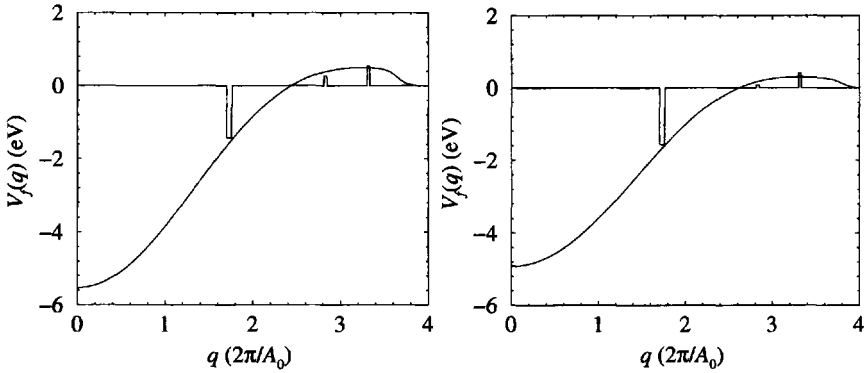


Figure 11.17 The discrete (δ -function like) pseudopotential form factors of the face-centred-cubic lattice [236] compared with the continuous form necessary for systems of reduced dimensionality, for Si (left) and Ge (right) [255]

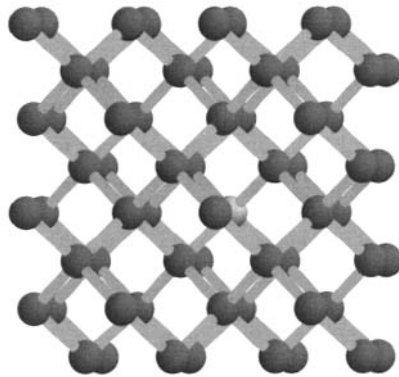


Figure 11.18 A 64-atom basis for the diamond structure, with a single substitutional Ge impurity (lighter colour, near centre)

Note that this set of reciprocal lattice vectors implies that the magnitude of the difference vector, $\mathbf{q} = \mathbf{G}' - \mathbf{G}$, can take many more values than the discrete set of the face-centred-cubic structure (given in Table 11.1). For this reason (and many others), more recently deduced empirical pseudopotential form factors are often represented by a continuous curve rather than a discrete set of points. Fig. 11.17 illustrates this for the Si and Ge form factors of Friedel *et al.* [255], and compares them to the

original discrete form factors of Cohen and Bergstresser [236], employed earlier in this chapter.

It can be seen that the agreement between the two representations of the pseudopotential is close and both have been truncated at $q = 4$, although at the higher end of the domain there is a slight difference. The continuous potentials of Friedel *et al.* [255] have been deduced by comparison with a variety of experimental results and have been shown to reproduce a large amount of experimental data, including bulk-band structure, deformation potentials, and heterostructure band offsets. Truncating the reciprocal lattice vectors of equation (11.112) at the same point as the potential ($q = 4$), gives a basis set of 2109 plane waves. Hence, the Hamiltonian matrix $\mathcal{H}_{\mathbf{G}',\mathbf{G}}$ has $(2109)^2$ elements, each of which is a complex number. Using double-precision arithmetic (8 bytes per number), this then implies the Hamiltonian matrix will occupy $(2109^2 \times 2 \times 8)$ bytes ≈ 71 Mbytes. Even when allowing for the operating system and the code to diagonalise it, the memory requirement for this size of calculation is now well within the reach of a standard desktop computer. However, it can be seen that the memory demands for this type of calculation can grow very quickly and this is one of the detrimental points of the large-basis method. While it is very simple, convenient and flexible, simply employing direct diagonalisation of the matrix as the computational solution is quite inefficient. However, a calculation of *this size* is manageable and all is now in place to proceed with it.

In this analysis one of the 64 Si atoms in the basis is substituted by Ge, as shown in Fig. 11.18; the choice of which atom is not important as the periodic nature of the pseudopotential calculations insures that they are all equivalent. Just as the two-atom basis of the simple elemental Si calculation produces four valence-band states, this more complex 64-atom basis gives $64 \times 2 (= 128)$ valence-band states. Fig. 11.19 shows the change in the *total* valence-band charge density on substitution of the single Ge isoelectronic impurity at an electron wave vector $\mathbf{k}=0$; note the lighter the colour, then the larger the increase.

In this particular example, the Si atom with coordinates $(0.875, 0.875, 0.875)A_0$ was substituted, and hence the third plane, with $z = 0.875 A_0$, intersects the centre of the Ge atom. Focusing on this plot, the charge density on the Ge atomic site is higher than the surrounding crystal and is fairly well localised, which (just about) justifies the initial choice of a 64-atom basis. The remaining four figures display the change in the charge density above and below the plane of this isolated Ge atom, where the symmetry is apparent equidistant from the centre of the impurity and it can be seen that a quarter of a lattice constant away (1.4 \AA), as shown on the first and fifth planes, the charge density has returned to its background (Si) value.

11.14 THE ELECTRONIC STRUCTURE AROUND POINT DEFECTS

Such an approach can also be used to model point defects. For example, instead of substituting a Si atom for an isoelectronic Ge atom, the atom can be removed completely to represent a vacancy. Now this is a simplified picture, in that for this

example it is assumed that all of the other atoms remain on their diamond lattice sites. This, of course, will not be the case, as there will be some localised relaxation of the crystal around the vacancy, i.e. some movement of the Si atoms.

Again employing the 64-atom cubic basis of above, Fig. 11.20 displays the change in the electronic structure around a single vacancy when compared to the unperturbed Si host crystal. The plot centres the vacancy site and represents a cross-section of four (2×2) face-centred-cubic unit cells. The absence of charge is clearly illustrated by the darker shading, and interestingly suggests a 'halo' of increased charge density around the vacancy. Such a movement of charge will aid shielding of the 'hole' from the remainder of the crystal.

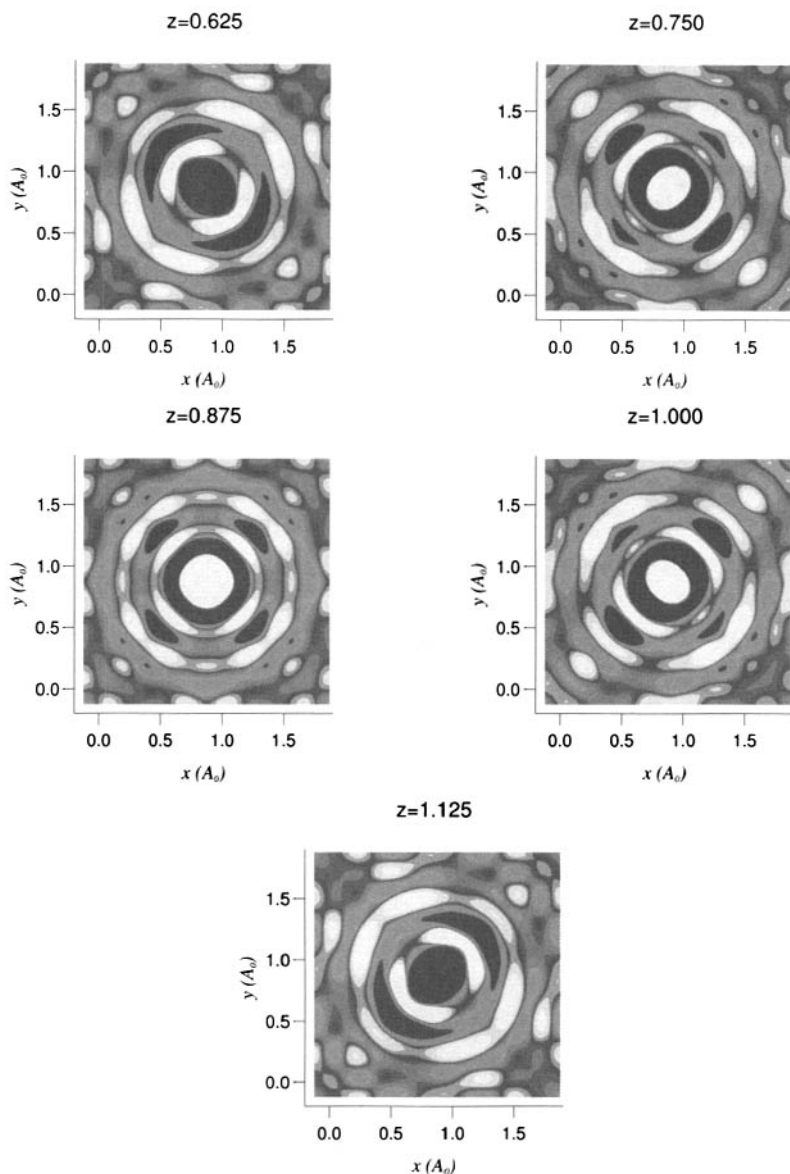


Figure 11.19 The change in the total valence-band charge density (ignoring spin-orbit coupling) around a single Ge atom within a Si host across the x - y plane of four face-centred-cubic unit cells, for a variety of z values (given in A_0); note the lighter the colour, then the higher the increase in the charge density

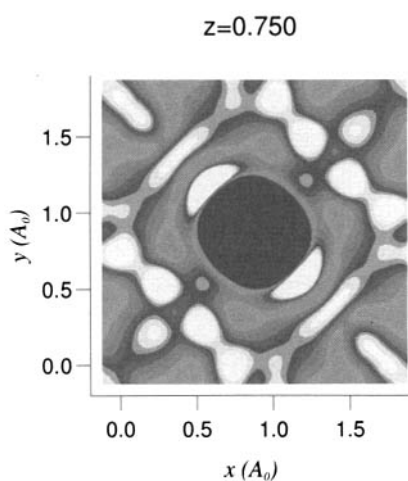


Figure 11.20 The change in the total valence-band charge density (ignoring spin-orbit coupling) around a vacancy within a Si host across the x - y plane of four face-centred-cubic unit cells; note that the lighter the colour, then the higher the increase in the charge density

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CHAPTER 12

MICROSCOPIC ELECTRONIC PROPERTIES OF HETEROSTRUCTURES

12.1 THE SUPERLATTICE UNIT CELL

The large-basis approach to a pseudopotential calculation *can* be applied to semiconductor heterostructures, where in particular, the short-period nature of the repeating unit cell in a superlattice is an ideal example. In order to proceed with such a calculation, it is necessary only to deduce the Bravais lattice vectors describing the crystal symmetry of the superlattice and from these derive the reciprocal lattice vectors.

Fig. 12.1 displays the now familiar zinc blende crystal lattice along one of the $\langle 100 \rangle$ directions. The crystal *can* be thought of as being composed of a series of 'one-dimensional' spirals which project into the plane of the paper. Such a spiral begins on an atom (in this case a cation) labelled '1' and then proceeds anticlockwise through atoms '2', '3' and '4', with the fifth atom in the spiral being exactly one lattice constant behind the first. This is illustrated more clearly in Fig. 12.2. Choosing the z -axis along the length of the spiral, then the atomic positions, in units of A_0 , follow

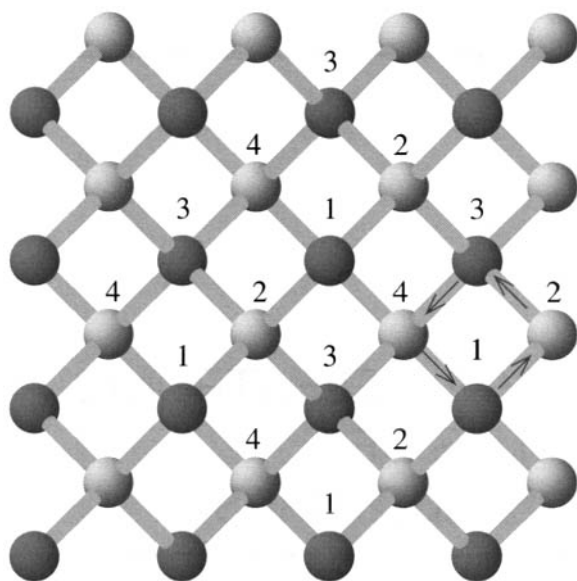


Figure 12.1 Looking along one of the $\langle 100 \rangle$ axes of a zinc blende crystal lattice

as:

$$\mathbf{r}_1 = \left(-\frac{1}{8}, -\frac{1}{8}, -\frac{1}{8} \right) \quad (12.1)$$

$$\mathbf{r}_2 = \left(\frac{1}{8}, \frac{1}{8}, \frac{1}{8} \right) \quad (12.2)$$

$$\mathbf{r}_3 = \left(-\frac{1}{8}, \frac{3}{8}, \frac{3}{8} \right) \quad (12.3)$$

$$\mathbf{r}_4 = \left(\frac{3}{8}, \frac{1}{8}, \frac{5}{8} \right) \quad (12.4)$$

The spiral then continues with a period (in this bulk case) of $A_0 \hat{\mathbf{k}}$, and thus:

$$\mathbf{r}_5 = \mathbf{r}_1 + (0, 0, 1) \quad (12.5)$$

$$\mathbf{r}_6 = \mathbf{r}_2 + (0, 0, 1) \quad (12.6)$$

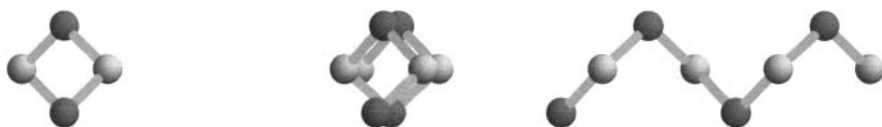


Figure 12.2 A single spiral of zinc blende: (left) Looking along one of the $\langle 100 \rangle$ directions; (centre) slightly off axis; (right) side view

This is easily generalised to a superlattice by simply defining the atomic positions with the appropriate choice of species. Fig. 12.3 illustrates this for a diamond (just to keep the illustration simple) heterostructure. Again, the atomic positions within the unit cell are repetitions of the four primitive vectors \mathbf{r}_1 to \mathbf{r}_4 given in equations (12.1) to (12.4) above. Thus for a superlattice of period $n_z A_0$, the position vectors of the atoms within this larger basis are given by:

$$\mathbf{t} = \mathbf{r}_i + i_{n_z} A_0 \hat{\mathbf{k}}, \quad i = 1, 2, 3, 4 \quad \text{and} \quad i_{n_z} = 0, 1, 2, \dots, n_z - 1 \quad (12.7)$$

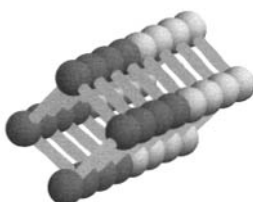


Figure 12.3 Off-axis view along the single spiral unit cell of a diamond heterostructure

The Bravais lattice vector describing the periodicity of the superlattice along the axis of symmetry (usually defined as the z -axis) is therefore clearly just $n_z A_0 \hat{\mathbf{k}}$. Referring back to Fig. 12.1, then the Bravais lattice vectors describing the in-plane (x - y) periodicity are those linking the atoms labelled '1', i.e. in units of A_0 , $(\frac{1}{2}, \frac{1}{2}, 0)$ and $(\frac{1}{2}, -\frac{1}{2}, 0)$. In summary, the Bravais lattice vectors for a superlattice are:

$$\mathbf{a}_1 = A_0 \left(\frac{1}{2} \hat{\mathbf{i}} + \frac{1}{2} \hat{\mathbf{j}} \right) \quad (12.8)$$

$$\mathbf{a}_2 = A_0 \left(\frac{1}{2} \hat{\mathbf{i}} - \frac{1}{2} \hat{\mathbf{j}} \right) \quad (12.9)$$

$$\mathbf{a}_3 = n_z A_0 \hat{\mathbf{k}} \quad (12.10)$$

The primitive reciprocal lattice vectors then follow from the usual vector product relationships (see Chapter 1), as:

$$\mathbf{b}_1 = \frac{2\pi}{A_0} (\hat{\mathbf{i}} + \hat{\mathbf{j}}) \quad (12.11)$$

$$\mathbf{b}_2 = \frac{2\pi}{A_0} (\hat{\mathbf{i}} - \hat{\mathbf{j}}) \quad (12.12)$$

$$\mathbf{b}_3 = \frac{2\pi}{A_0} \frac{1}{n_z} \hat{\mathbf{j}} \quad (12.13)$$

The complete set of reciprocal lattice vectors required for the pseudopotential calculation is created by taking linear combinations of the above primitive vectors. Although this could be worked out analytically, it is quicker to take the short cut of generating this set numerically with a simple computer program. In this manner, Fig. 12.4 plots the number of plane waves in the expansion set (when limiting the maximum magnitude to $4 \times (2\pi/A_0)$) as a function of the superlattice period.

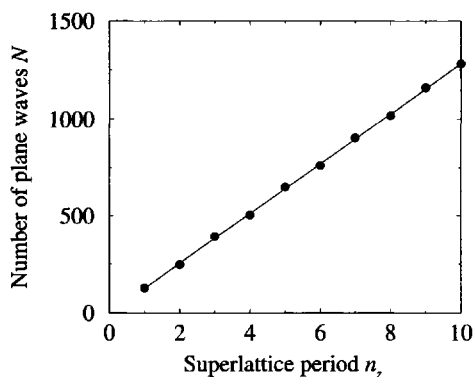


Figure 12.4 The number of plane waves required in the expansion set for the large-basis approach to superlattices

A linear fit to the data shown in Fig. 12.4 gives the number of plane waves as:

$$N \approx 128n_z \quad (12.14)$$

which, as can be seen, can become quite large for relatively short periods. In particular, a superlattice of period $n_z=10A_0$ corresponds to 56.5 \AA in the GaAs/Ga_{1-x}Al_xAs system. This simplistic approach generates 1281 plane waves thus giving a Hamiltonian matrix $\mathcal{H}_{G',G}$ of order 1281. With each element being a complex (2×) double (8 bytes), then this matrix requires $2 \times 8 \times 1281^2 \sim 26 \text{ Mbytes}$ of storage, which, is now well within the reach of a modern desktop computer.

12.2 APPLICATION OF LARGE BASIS METHOD TO SUPERLATTICES

While not perhaps being the most computationally 'slick' method, the large-basis approach to the empirical pseudopotential calculation of an extended crystallographic unit cell is straightforward and offers an excellent introduction to such calculations for researchers new to the field. In the last section, the application of this technique has been introduced for a particular classification of semiconductor heterostructures, namely superlattices with a relatively short period. In these first example calculations, this period will be considered to be constructed from a single layer of GaAs coupled to a single layer of AlAs, as illustrated schematically in Fig. 12.5.

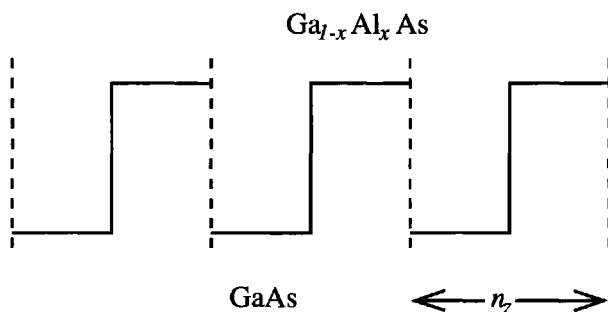


Figure 12.5 Schematic illustration of the repeating unit cell, of length $n_z A_0$, within a superlattice

As mentioned often before, a knowledge of the pseudopotentials for superlattices is required at many more points than the small discrete set akin to bulk materials. With this aim, the pseudopotentials of Mäder and Zunger [253], which have been shown to reproduce many experimentally measured parameters as well as the results of *ab initio* calculations, will be employed. The required continuous nature of these potentials is illustrated in Fig. 12.6 for both atomic species in GaAs and AlAs.

Again, notice how the atomic potential of As in GaAs is different from that in AlAs. This difference, $\Delta V_f(q) = V_f^{\text{As in GaAs}}(q) - V_f^{\text{As in AlAs}}(q)$, is plotted in Fig. 12.7, just

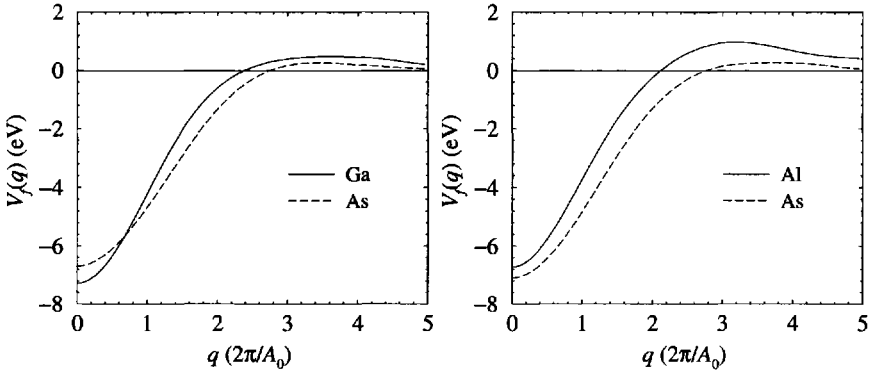


Figure 12.6 The continuous atomic pseudopotentials of GaAs (left) and AlAs (right) of Mäder and Zunger [253]

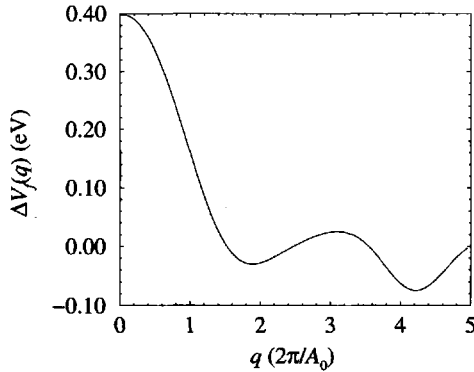


Figure 12.7 The difference in the atomic potentials of As in GaAs and AlAs

to re-emphasise the point made in Section 11.8, namely that unfortunately, universal atomic potentials do not actually exist.

Fig. 12.8 shows the band structures of bulk GaAs and AlAs, *ignoring spin-orbit coupling*, as produced from the continuous potentials. Both band structures have been translated upwards in energy such that the top of the valence band of GaAs is at zero. At the same time, the top of the valence band for AlAs is at -0.51 eV, which gives the offset in the valence band as 33 % of the total, in line with experimentally determined values (see for example [14], p. 179), and recent calculations (see for example [256]). This illustrates an important point, namely that the correct choice of potentials can not only just be used to reproduce experimental data for bulk material, but in addition they can retain information on the alignment (or band offset) of two dissimilar materials.

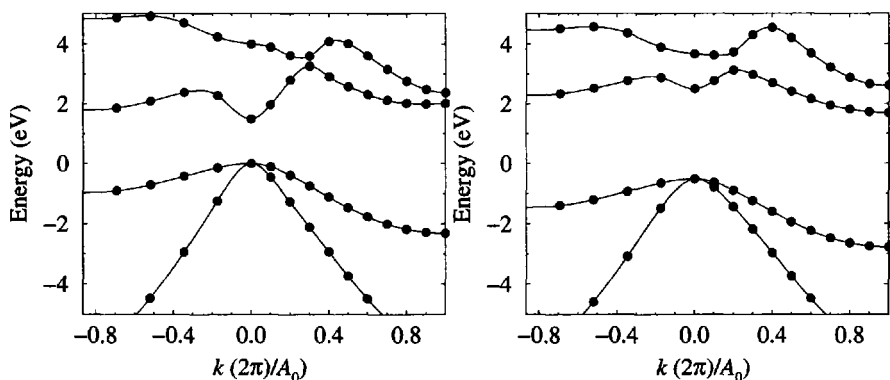


Figure 12.8 The bulk band structures of GaAs (left) and AlAs (right), illustrating the valence band offset built into the potentials of Mäder and Zunger [253]

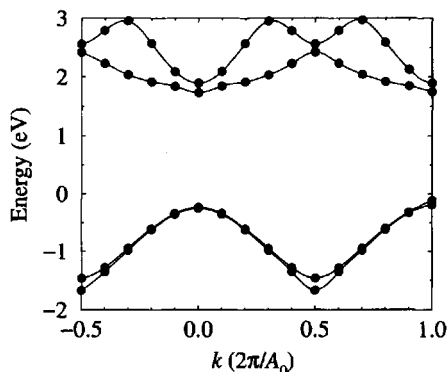


Figure 12.9 The band structure of a 1 ml GaAs/1 ml AlAs superlattice around the fundamental gap

Therefore, all is in place for the calculation to proceed: the superlattice unit cell has been defined, the reciprocal lattice vectors calculated, and the pseudopotentials have been shown to reproduce the bulk band structure *and* the band alignment between the materials of interest. Fig. 12.9 displays the results of calculations of the energy level structure around the fundamental gap of a 1 ml (monolayer) GaAs/1 ml AlAs superlattice (sometimes written as $(\text{GaAs})_1(\text{AlAs})_1$), i.e. $n_z \approx 1$, for a range of electron wave vectors k along the z -axis (parallel to the growth direction). The cyclic nature of the dispersion curves is evident with a period of $2\pi/A_0$, which represents the primitive reciprocal lattice vector along that direction, as given above in equation (12.13). Thus, the edges of the superlattice Brillouin zone are at $k = \pm\pi/(n_z A_0)$, which is again the commonly found result.

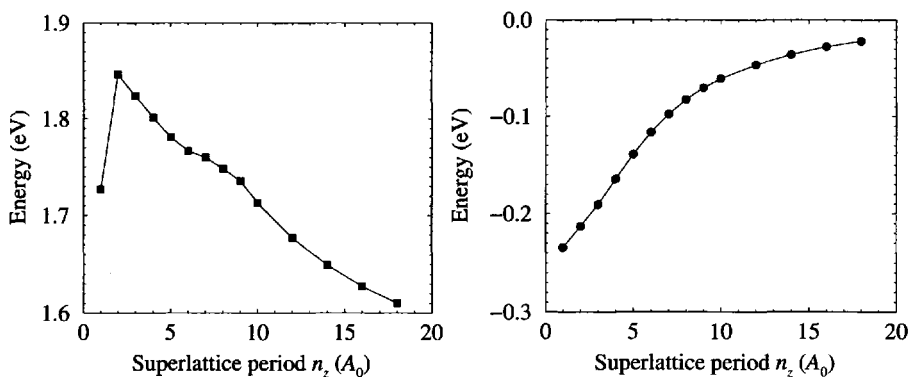


Figure 12.10 The energies of the lowest conduction-band state (left) and highest valence-band state (right) as functions of the superlattice period

Following on from this, Fig. 12.10 displays the results of calculations of the energies at the centre of the superlattice Brillouin zone ($k = 0$) for the uppermost valence band and the lower-most conduction band, as a function of the superlattice period (expressed in units of A_0). In both cases, the quantum-confinement energy decreases as the superlattice period increases, an effect which was commonly observed in the discussions in Chapters 2 and 3. Note that the energies are given in relation to the top of the valence band of bulk GaAs, which is set to zero, with the bottom of the conduction band coming out as 1.499 eV in these calculations. One interesting feature is the sudden reduction in the quantum-confinement energy of the conduction-band state when the period decreases to just A_0 . In fact, it would be expected a priori that the conduction band minimum would tend towards the band edge of the bulk alloy $\text{Ga}_{0.5}\text{Al}_{0.5}\text{As}$ in the limit of short-period GaAs/AlAs superlattices. By taking a linear combination of the bulk band structures as suggested by the virtual crystal approximation, this would imply an energy of around 2 eV. This behaviour will be discussed in more detail in the next section.

Finally for this series of calculations, Fig. 12.11 plots the charge density along the centre ($x = y = 0$) of the $n_z=10$ superlattice spiral, for the uppermost valence-band state. The confinement in the GaAs layers is clear, and in addition to this, the lowest conduction-band state is also confined in these layers, thus illustrating the Type-I nature of the band alignment. It can be seen from this figure that the wave function does consist of two components, where one is rapidly varying and the second is an envelope. This reflects well the idea behind the envelope-function approximation which was discussed (and then used extensively) earlier. The rapidly varying component has a period which is the same as the atomic spacing, and indeed along the axis used in the plot, i.e. the $x = y = 0$ axis, the peaks in the charge density correspond to bond centres. The envelope function is obtained by joining together the peaks in the charge-density plot by using the eye, with its form being familiar from Chapter 2.

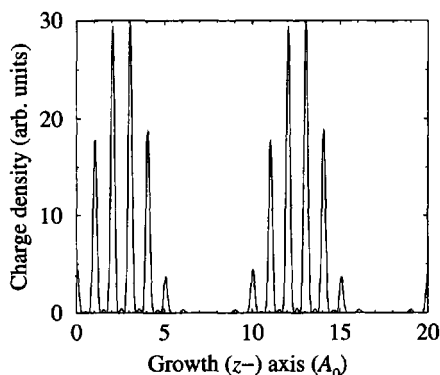


Figure 12.11 The charge density of the uppermost valence-band state along the $x = y = 0$ (z -) axis for two complete superlattice periods

12.3 COMPARISON WITH ENVELOPE-FUNCTION APPROXIMATION

Fig. 12.12 compares the results of a Kronig-Penney superlattice calculation under the envelope function/effective mass approximations (see Section 2.13), with the empirical pseudopotential calculations of the lowest-energy conduction band state as a function of the superlattice period, as previously displayed in Fig. 12.10. It is this comparison which draws attention to the complexity of the data. For guidance, it is necessary to refer to the specialist treatise on empirical pseudopotential calculations of GaAs/AlAs superlattices from which the pseudopotentials employed in these calculations are taken (see [253], p. 17396).

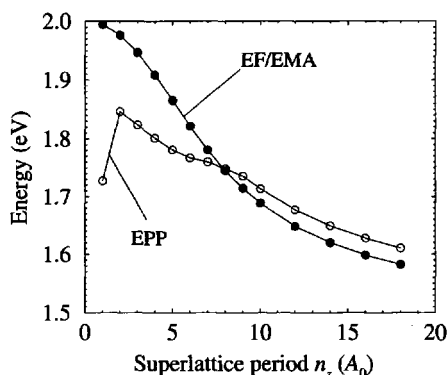


Figure 12.12 Comparison of the lowest-energy conduction band state given by the empirical pseudopotential (EPP) calculation in the previous section with that given by the Kronig-Penney model of a superlattice under the envelope function/effective mass approximations (EF/EMA)

Mäder and Zunger indicate that these simple bulk-like potentials are not sufficient to describe the microscopic structure of such a short-period superlattice and hence the result for $n_z=1$ could be prone to error. In particular, no account has been made for the change in the coordination of the interface anions (see Section 12.5). For $2 \leq n_z \leq 8$, the conduction-band minimum for the superlattice originates from the X valleys of the bulk and not from the Γ valley—this is possible because of the indirect nature of AIAs. As the X valley in bulk AIAs lies below that of the Γ valley, then narrow quantum wells can produce such high confinement energies that the eigenstate is influenced by these outlying valleys (in essence!). For $n_z > 8$, the simpler Kronig–Penney model describes the behaviour of the energy state with superlattice period quite well *qualitatively*.

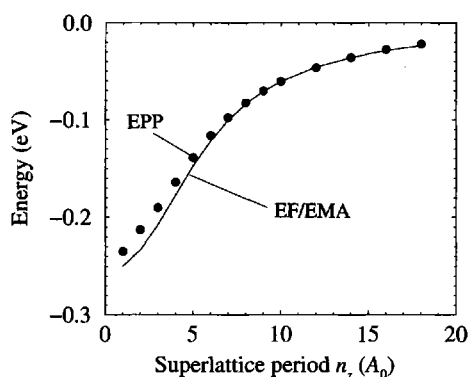


Figure 12.13 Comparison of the highest-energy valence-band state given by the empirical pseudopotential (EPP) calculation in the previous section with that given by the Kronig–Penney model of a superlattice under the envelope function/effective mass approximations (EF/EMA)

Fig. 12.13 displays the results of an identical series of calculations for the uppermost (heavy-hole) valence-band state. The effective mass in the AIAs barriers was taken (and fixed) as the bulk value of $0.51 m_0$ (Adachi [14], p. 254). In this case, however, the hole effective mass in the GaAs well regions was used as a parameter and varied in order to produce the best fit to the empirical pseudopotential (EPP) results. It can be seen that the Kronig–Penney (EF/EMA) results for an effective mass of $0.45m_0$ fit the pseudopotential data very well at the larger well and barrier widths, but the match is poorer at narrower widths. This is the result found by Long *et al.* [21], and indeed should be expected, as the envelope-function approximation (see Chapter 1) hinges on the point that the wave function can be considered as a product of two components, with one being a rapidly varying Bloch function (which is factorised out) and the second a slowly varying envelope. It stands to reason that when the period of the envelope approaches that of the Bloch function, as happens here in short-period superlattices, the approximation becomes poorer.

In essence, the result here can be summarized by saying that, the effective mass of a particle in a quantum well is a function of the well width. The fact that a constant effective mass predicts too high a confinement energy allows the further deduction that the effective mass must be increased as the well width decreases in order to produce agreement with the empirical pseudopotential calculations.

In conclusion, the microscopic nature of the pseudopotential calculation gives more detail and allows for more complexity in the eigenstates of heterostructures than methods based on the envelope function/effective mass approximation. In particular, use of the constant-effective-mass approximation has been shown to breakdown for short-period superlattices.

12.4 IN-PLANE DISPERSION

In addition to computing the dispersion along the line of symmetry, i.e. the growth (z -) axis, the large-basis method, when applied to superlattices as in Section 12.2, can be used to calculate the in-plane (x - y) dispersion. Such knowledge is fundamental for describing the transport properties of electronic devices which exploit in-plane transport for their operation, such as High Electron Mobility Transistors (HEMTs), as well as optical devices which are influenced by the carriers populating the in-plane momentum states, such as intersubband lasers (see Section 9.22).

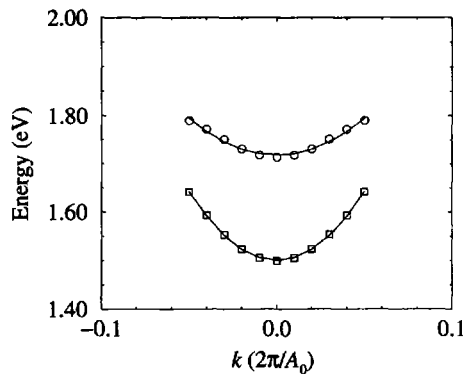


Figure 12.14 The in-plane [100] dispersion curve around the centre of the Brillouin zone for a $(\text{GaAs})_{10}(\text{AlAs})_{10}$ superlattice (top) in comparison with that for bulk (bottom)

Fig. 12.14 displays the results of calculations of the in-plane dispersion (top curve), in this case along the [100] direction, for the $(\text{GaAs})_{10}(\text{AlAs})_{10}$ superlattice of the previous section. On the same axes the bulk dispersion curve (bottom) is also shown for comparison. The subband minima are both measured from the top of the valence band and hence the minimum of the bottom curve represents the band gap of the bulk material, while the difference in the minima represents the quantum confinement energy of this, i.e. the lowest confined state in the superlattice.

Remembering that for low electron momenta the $E-k$ dispersion curves are parabolic, then as before the effective mass is given by:

$$m^* = \hbar^2 \left(\frac{\partial^2 E}{\partial k^2} \right)^{-1} \quad (12.15)$$

The fitting of parabolas (the solid curves) to the data points in Fig. 12.14 thus allows the effective masses to be calculated (as before in Chapter 11). Using the potentials of Mäder and Zunger [253], this procedure gives the effective mass along any of the $\langle 100 \rangle$ directions in the bulk crystal as $0.082m_0$. In contrast to this, the in-plane electron effective mass for the superlattice is $0.15m_0$, which is obviously quite different.

This would seem to be a general result, at least for relatively short period superlattices; note the period for this example is $10A_0 = 56.5 \text{ \AA}$. A priori, it might be expected that as there is no confinement in the $x-y$ plane (i.e. parallel to the layers), that the dispersion curves would resemble that of the bulk. In fact, as there is confinement along the z -direction, which leads to a shift in the band minimum the band structure around the minima is clearly different from that of the bulk. The immediate consequence of which is that the effective mass increases.

12.5 INTERFACE COORDINATION

In bulk zinc blende material, each anion, e.g. As in GaAs, is bonded to four Ga cations. It has been shown that the atomic pseudopotential of this As is different from the As'' (say) in AlAs, which is merely a reflection of the different chemical nature of the bonding between Ga atoms and Al atoms arising from their different electronegativities. Although disappointing in that universal atomic potentials cannot be deduced, the calculations thus far have shown how to deal with such a problem.

Consider now a heterojunction between two compound semiconductors which share a common species, e.g. GaAs/AlAs, as illustrated in Fig. 12.15. The As anion at the interface, as indicated, is bonded to two Ga atoms and two Al atoms, and hence they have neither the character of a fully tetrahedrally coordinated As or As''. In fact, they have a character which is intermediate between the two, which can be described by the mean:

$$As' = \frac{As + As''}{2} \quad (12.16)$$

and hence the interface properties *can* be described better by the mean in the atomic pseudopotentials, i.e.

$$V_f^{As'}(q) = \frac{V_f^{As}(q) + V_f^{As''}(q)}{2} \quad (12.17)$$

12.6 STRAIN-LAYERED SUPERLATTICES

As already mentioned, the more recently deduced pseudopotentials generally take account of a much greater diversity of experimental information than just the main

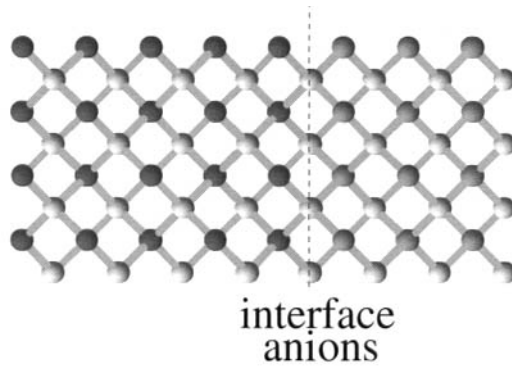


Figure 12.15 The atoms near the heterojunction between two compound semiconductors

energy gaps. In particular, one of these additional points is the reproducibility of the deformation potentials from the pseudopotential calculations (see for example [253,256] for III–V compounds and [255] for SiGe). One consequence of this development is that the pseudopotentials can be used to accurately describe the microscopic electronic structure in *strain-layered* systems.

Strain arises in heterosystems which are formed from materials with difference lattice constants (see Kelly [7], p. 317, or Adachi [14], p. 271, for an introduction). For example, the lattice constant of GaAs is often quoted as 5.653 Å and that of AlAs as 5.660 Å, i.e. a difference of just 0.1 %. Hence, heterosystems formed from these materials (and all of the intermediate $\text{Ga}_{1-x}\text{Al}_x\text{As}$ alloys) have a quite constant lattice spacing, and for this reason strain in this material system is usually ignored.

The lattice constant of Si is 5.43 Å and that of Ge 5.66 Å, which represents a lattice mismatch of $(5.66 - 5.43)/5.43 = 4\%$. While this may appear to be a low value, this is not the case for a semiconductor structure (for an introduction to strain in SiGe, see Meyerson [257]). If a layer of Ge is grown on top of a Si substrate, then the first few layers of Ge will assume the in-plane lattice constant of Si, i.e. the Ge atoms will be squeezed beyond their normal equilibrium separation by the crystal potential of Si. To compensate for this, the lattice constant along the growth (z -) axis increases, thus distorting the usual cubic cell into a cuboid. Given the pseudopotentials now available, it would suffice to calculate the atomic positions, in order to calculate the effect of strain on the electronic structure. One very simplistic model, perhaps suitable for thin strained layers, would be to assume that the material attempts to keep its unit cell at constant volume. Hence, if A_z is the lattice constant along the growth axis then:

$$A_x A_y A_z = A_0^3 \quad (12.18)$$

Therefore, considering Ge on Si then the in-plane lattice constants of the Ge layer will equate to Si, i.e. 5.43 \AA , and hence:

$$A_z = \frac{5.66^3}{5.43^2} \text{ \AA} = 6.15 \text{ \AA} \quad (12.19)$$

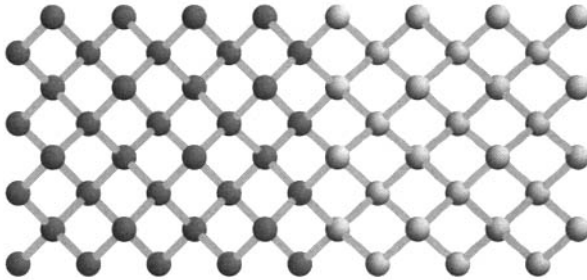


Figure 12.16 The atomic positions at a Si–Ge heterojunction

The application of this model is illustrated for just such a Si–Ge heterojunction in Fig. 12.16—there is an increase in the atomic spacing along the growth (z -) axis as the heterojunction is crossed from the left-hand region of Si atoms to the Ge atoms on the right-hand side. More sophisticated models for the atomic positions in strain-layered semiconductors are available. In particular, Morrison *et al.* [258] compared the results of pseudopotential calculations of SiGe superlattices, keeping the lattice constant equal to that of a SiGe alloy buffer layer, with those produced with the atomic spacing calculated by minimizing the elastic energy [259].

If the strained Ge layer is allowed to grow thicker than the *critical thickness* then dislocations form which relieve the strain, thus leading to a reduction in the total structural energy of the crystal. Empirical pseudopotential calculations of the electronic structure around dislocations, and indeed any in-depth study of the particulars of strain-layered superlattices lie beyond the scope of this present work. The interested reader is therefore referred to the literature (see for example [260, 261] for SiGe, and [262] for $\text{In}_x\text{Ga}_{1-x}\text{As}/\text{GaAs}$).

The latter of these two material systems is exploited in a relatively common electronic device, namely the pseudomorphic-High Electron Mobility Transistor, or p-HEMT [263].

12.7 THE SUPERLATTICE AS A PERTURBATION

The direct approach to the solution of superlattices by the large-basis method works well, as shown so far; however, this is not always the case. The large number of plane waves required in the expansion generates a very large Hamiltonian matrix, $\mathcal{H}_{G',G}$, which in turn requires a computer with a large amount of memory for solution. Such machines are not always available and thus it may be necessary to resort to alternative methods. In the case of simple quantum wells and superlattices, one such method is to treat the superlattice as a perturbation on a bulk semiconductor. For the case of quantum dots and wires, an even more powerful computational technique may be required, which will be discussed later in Chapter 13.

Concentrating upon one-dimensional periodic potentials as generated by a pseudopotential description of a quantum well, then as already mentioned, the superlattice can be thought of as a perturbation on a bulk 'host' semiconductor [264]. For example, for a GaAs/Ga_{1-x}Al_xAs superlattice the original host semiconductor may be chosen as GaAs, with the superlattice being formed by exchanging some of the Ga cations for Al. As the word *perturbation* implies, the superlattice wave function is then constructed from a linear combination of the orthogonal bulk eigenfunctions.

Recalling the bulk Schrödinger equation, i.e.

$$\mathcal{H}\psi_{n,\mathbf{k}} = E_{n,\mathbf{k}}\psi_{n,\mathbf{k}} \quad (12.20)$$

then introduction of a perturbation, V_{sl} , which is acknowledged to represent the additional potential introduced on formation of the superlattice (sl), yields a new Schrödinger equation with a new set of eigenfunctions, i.e.

$$(\mathcal{H} + V_{sl})\Psi_{N,\xi} = E_{N,\xi}\Psi_{N,\xi} \quad (12.21)$$

where the indices N and ξ represent the principal quantum number and the wave vector, respectively, of the superlattice eigenfunctions $\Psi_{N,\xi}$. As this is a perturbative approach, it is expected that these eigenfunctions will be expanded in terms of the bulk eigenfunctions, ψ_{n,\mathbf{k}_ξ} , i.e.

$$\Psi_{N,\xi} = \sum_{n,\mathbf{k}_\xi} A_{n,\mathbf{k}_\xi}^N \psi_{n,\mathbf{k}_\xi} \quad (12.22)$$

The additional qualification ξ on the wave vectors of the bulk eigenstates indicates that the superlattice eigenstate $\Psi_{N,\xi}$ is summed, not just over a particular set of eigenfunctions at a fixed \mathbf{k} , but also over the set of bulk wave vectors which are *zone-folded* into the superlattice Brillouin Zone ([5], p. 95). This is illustrated in Fig. 12.17.

The smallest reciprocal lattice vector directed along one of the $\langle 100 \rangle$ directions, as highlighted by Table 11.1, is the $[200]$ vector. Hence, the von Laue condition, i.e.

$$\mathbf{k} \cdot \hat{\mathbf{G}} = \frac{1}{2}|\mathbf{G}| \quad (12.23)$$

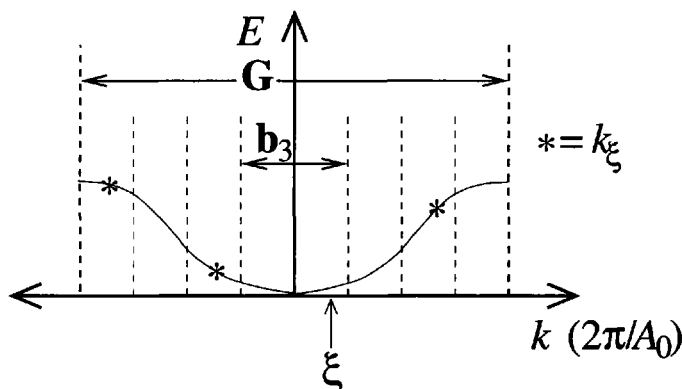


Figure 12.17 Schematic representation of the effect of zone-folding of the bulk Brillouin zone to produce the reduced superlattice mini-zone

(see [1], p. 98) is fulfilled for an electron wave vector \mathbf{k} equal to [100], and therefore the edge of the Brillouin zone along the z -axis is at [001], as indicated in the figure. A superlattice, of period $A_{sl} = n_z A_0$ along the z -axis, has its symmetry described by a new set of reciprocal lattice vectors, and indeed the principle primitive is given by:

$$\mathbf{b}_3 = \frac{2\pi}{A_{sl}} \hat{\mathbf{k}} = \frac{1}{n_z} \frac{2\pi}{A_0} \hat{\mathbf{k}} \quad (12.24)$$

which is the same result as that obtained for the simple cubic lattice (see equation (11.112)). Hence, the edge of the superlattice Brillouin zone, which is often referred to as the *mini-zone* is at:

$$\frac{1}{2} |\mathbf{b}_3| = \frac{2\pi}{2A_{sl}} = \frac{1}{2n_z} \frac{2\pi}{A_0} \quad (12.25)$$

Fig. 12.17 illustrates the mini-zone for a superlattice with period $A_{sl} = 2A_0$, i.e. $n_z = 2$. Thus, the edge of the mini-zone is a quarter of the wave vector that it is for bulk, along this direction. The set of wave vectors \mathbf{k}_ξ are those points within the bulk Brillouin zone which are separated by a superlattice reciprocal lattice vector \mathbf{b}_3 from the electron wave vector within the mini-zone. The former are illustrated by the asterisks (*), while the latter is indicated by ξ in the figure.

Therefore, a pseudopotential calculation of a superlattice of period $A_{sl} = n_z A_0$ requires $N_{\mathbf{k}_\xi} = 2n_z$ wave vector points from the bulk Brillouin zone. The points themselves are given by:

$$\mathbf{k}_\xi = \xi + (-n_z + i_{\mathbf{k}_\xi}) \mathbf{b}_3, \quad \text{where } i_{\mathbf{k}_\xi} = 0, 1, 2, \dots, 2n_z - 1 \quad (12.26)$$

and \mathbf{b}_3 is given by equation (12.24).

Following standard perturbative theory expansion procedures, substitute Ψ_{N, \mathbf{k}_ξ} from equation (12.22) into the perturbed Schrödinger equation (equation 12.21), to

give the following:

$$\sum_{n, \mathbf{k}_\xi} A_{n, \mathbf{k}_\xi}^N (\mathcal{H} + V_{sl}) \psi_{n, \mathbf{k}_\xi} = E_{N, \xi} \sum_{n, \mathbf{k}_\xi} A_{n, \mathbf{k}_\xi}^N \psi_{n, \mathbf{k}_\xi} \quad (12.27)$$

Using equation (12.20), then:

$$\sum_{n, \mathbf{k}_\xi} A_{n, \mathbf{k}_\xi}^N (E_{n, \mathbf{k}_\xi} + V_{sl}) \psi_{n, \mathbf{k}_\xi} = E_{N, \xi} \sum_{n, \mathbf{k}_\xi} A_{n, \mathbf{k}_\xi}^N \psi_{n, \mathbf{k}_\xi} \quad (12.28)$$

Multiplying by $\psi_{n', \mathbf{k}'_\xi}^*$, and integrating over all space:

$$\sum_{n, \mathbf{k}_\xi} A_{n, \mathbf{k}_\xi}^N \int \psi_{n', \mathbf{k}'_\xi}^* (E_{n, \mathbf{k}_\xi} + V_{sl}) \psi_{n, \mathbf{k}_\xi} d\tau = E_{N, \xi} \sum_{n, \mathbf{k}_\xi} A_{n, \mathbf{k}_\xi}^N \int \psi_{n', \mathbf{k}'_\xi}^* \psi_{n, \mathbf{k}_\xi} d\tau \quad (12.29)$$

Using the orthonormality of the bulk eigenvectors, i.e. the integral over all space of the product $\psi_{n', \mathbf{k}'_\xi}^* \psi_{n, \mathbf{k}_\xi}$ is non-zero only when $n = n'$ and $\mathbf{k}_\xi = \mathbf{k}'_\xi$, which in mathematical language would be written as:

$$\int \psi_{n', \mathbf{k}'_\xi}^* \psi_{n, \mathbf{k}_\xi} d\tau = \delta_{n'n} \delta_{\mathbf{k}'_\xi, \mathbf{k}_\xi} \quad (12.30)$$

then:

$$\sum_{n, \mathbf{k}_\xi} A_{n, \mathbf{k}_\xi}^N \left(E_{n, \mathbf{k}_\xi} \delta_{n'n} \delta_{\mathbf{k}'_\xi, \mathbf{k}_\xi} + \int \psi_{n', \mathbf{k}'_\xi}^* V_{sl} \psi_{n, \mathbf{k}_\xi} d\tau \right) = E_{N, \xi} \sum_{n, \mathbf{k}_\xi} A_{n, \mathbf{k}_\xi}^N \delta_{n'n} \delta_{\mathbf{k}'_\xi, \mathbf{k}_\xi} \quad (12.31)$$

Write this as:

$$\sum_{n, \mathbf{k}_\xi} A_{n, \mathbf{k}_\xi}^N \mathcal{H}'_{(n', \mathbf{k}'_\xi), (n, \mathbf{k}_\xi)} = E_{N, \xi} \sum_{n, \mathbf{k}_\xi} A_{n, \mathbf{k}_\xi}^N \delta_{n'n} \delta_{\mathbf{k}'_\xi, \mathbf{k}_\xi} \quad (12.32)$$

where the perturbed Hamiltonian matrix elements are given by:

$$\mathcal{H}'_{(n', \mathbf{k}'_\xi), (n, \mathbf{k}_\xi)} = E_{n, \mathbf{k}_\xi} \delta_{n'n} \delta_{\mathbf{k}'_\xi, \mathbf{k}_\xi} + \int \psi_{n', \mathbf{k}'_\xi}^* V_{sl} \psi_{n, \mathbf{k}_\xi} d\tau \quad (12.33)$$

or

$$\mathcal{H}'_{(n', \mathbf{k}'_\xi), (n, \mathbf{k}_\xi)} = E_{n, \mathbf{k}_\xi} \delta_{n'n} \delta_{\mathbf{k}'_\xi, \mathbf{k}_\xi} + V' \quad (12.34)$$

where the potential term has been relabelled as follows:

$$V' = \int \psi_{n', \mathbf{k}'_\xi}^* V_{sl} \psi_{n, \mathbf{k}_\xi} d\tau \quad (12.35)$$

The problem is therefore reduced to finding the eigenvalues, $E_{N, \xi}$, and eigenvectors, A_{n, \mathbf{k}_ξ}^N , of the square matrix, $\mathcal{H}'_{(n', \mathbf{k}'_\xi), (n, \mathbf{k}_\xi)}$. This is a direct analogy of the original

equation (equation (11.20)), for bulk, and can again be solved by direct diagonalisation. Furthermore, if N_n bulk energy levels at any one of the $N_{\mathbf{k}_\xi}$ bulk wave vectors are included, then $\mathcal{H}'_{(n', \mathbf{k}'_\xi), (n, \mathbf{k}_\xi)}$ is of the order $N_n N_{\mathbf{k}_\xi}$.

It remains then to calculate the matrix elements of \mathcal{H}' (for short), and given that the bulk has already been solved, this manifests itself merely as deducing the integral constituting the second term of equation (12.34). Consider substituting the bulk eigenvectors for their plane wave summations, as detailed in the previous chapter, i.e.

$$\psi_{n, \mathbf{k}} = \frac{1}{\sqrt{\Omega}} \sum_{\mathbf{G}} a_{n, \mathbf{k}}(\mathbf{G}) e^{i(\mathbf{G} + \mathbf{k}) \cdot \mathbf{r}} \quad (12.36)$$

then:

$$V' = \frac{1}{\Omega} \sum_{\mathbf{G}', \mathbf{G}} a_{n', \mathbf{k}'_\xi}^*(\mathbf{G}') a_{n, \mathbf{k}_\xi}(\mathbf{G}) \int e^{-i(\mathbf{G}' + \mathbf{k}'_\xi) \cdot \mathbf{r}} V_{sl} e^{i(\mathbf{G} + \mathbf{k}_\xi) \cdot \mathbf{r}} d\tau \quad (12.37)$$

$$\therefore V' = \frac{1}{\Omega} \sum_{\mathbf{G}', \mathbf{G}} a_{n', \mathbf{k}'_\xi}^*(\mathbf{G}') a_{n, \mathbf{k}_\xi}(\mathbf{G}) \int e^{-i(\mathbf{G}' - \mathbf{G} + \mathbf{k}'_\xi - \mathbf{k}_\xi) \cdot \mathbf{r}} V_{sl} d\tau \quad (12.38)$$

The perturbation V_{sl} can be written as a sum of the atomic potentials over all of the sites where atoms have been exchanged. If $V_{a'}$ is the atomic potential of the newly introduced species at a position \mathbf{r}'_a , and V_a is the atomic potential of the original species, then the potential due to the creation of the superlattice is just the difference between the two summed over all of the sites, i.e.

$$V_{sl} = \sum_{\mathbf{r}'_a} [V_{a'}(\mathbf{r} - \mathbf{r}'_a) - V_a(\mathbf{r} - \mathbf{r}'_a)] \quad (12.39)$$

Again, specifying *all* of the atomic positions \mathbf{r}'_a as a combination of a Bravais lattice vector \mathbf{R}_{sl} and a basis \mathbf{t}_{sl} (note that the set of basis vectors \mathbf{t}_{sl} are the position vectors of the exchanged atom positions within a superlattice unit cell), i.e.

$$\mathbf{r}'_a = \mathbf{R}_{sl} + \mathbf{t}_{sl} \quad (12.40)$$

then:

$$V_{sl} = \sum_{\mathbf{R}_{sl}} \sum_{\mathbf{t}_{sl}} [V_{a'}(\mathbf{r} - \mathbf{R}_{sl} - \mathbf{t}_{sl}) - V_a(\mathbf{r} - \mathbf{R}_{sl} - \mathbf{t}_{sl})] \quad (12.41)$$

Consider the vector $\mathbf{G}' - \mathbf{G} + \mathbf{k}'_\xi - \mathbf{k}_\xi = \mathbf{g}$ (say), in equation (12.38). From Fig. 12.17, $\mathbf{k}'_\xi - \mathbf{k}_\xi$ is clearly a reciprocal lattice vector of the superlattice, and in addition, as both \mathbf{G}' and \mathbf{G} are bulk reciprocal lattice vectors, then they are also reciprocal lattice vectors of the reduced symmetry system of the superlattice, albeit some large multiple of the primitive superlattice reciprocal lattice vectors. Hence, \mathbf{g} is also a reciprocal lattice vector of the superlattice.

Substituting for \mathbf{g} and V_{sl} from equation (12.41) into equation (12.38) then gives:

$$V' = \frac{1}{\Omega} \sum_{\mathbf{G}', \mathbf{G}} a_{n', \mathbf{k}'_\xi}^*(\mathbf{G}') a_{n, \mathbf{k}_\xi}(\mathbf{G})$$

$$\times \int e^{-i\mathbf{g}\cdot\mathbf{r}} \sum_{\mathbf{R}_{sl}} \sum_{\mathbf{t}_{sl}} [V_{a'}(\mathbf{r} - \mathbf{R}_{sl} - \mathbf{t}_{sl}) - V_a(\mathbf{r} - \mathbf{R}_{sl} - \mathbf{t}_{sl})] d\tau \quad (12.42)$$

Making the transformation $\mathbf{r} - \mathbf{R}_{sl} - \mathbf{t}_{sl} \rightarrow \mathbf{r}$ then:

$$V' = \sum_{\mathbf{R}_{sl}} e^{-i\mathbf{g}\cdot\mathbf{R}_{sl}} \sum_{\mathbf{t}_{sl}} e^{-i\mathbf{g}\cdot\mathbf{t}_{sl}} \times \frac{1}{\Omega} \sum_{\mathbf{G}', \mathbf{G}} a_{n', \mathbf{k}'_\xi}^*(\mathbf{G}') a_{n, \mathbf{k}_\xi}(\mathbf{G}) \int e^{-i\mathbf{g}\cdot\mathbf{r}} [V_{a'}(\mathbf{r}) - V_a(\mathbf{r})] d\tau \quad (12.43)$$

Now the product of any Bravais lattice vector of the superlattice \mathbf{R}_{sl} and a reciprocal lattice vector \mathbf{g} is an integral number of 2π , and hence for N_{sl} superlattice unit cells within the total volume Ω of the crystal:

$$\sum_{\mathbf{R}_{sl}} e^{-i2\pi n} = N_{sl}, \quad \text{where } n \in \mathbb{Z} \quad (12.44)$$

and:

$$\frac{N_{sl}}{\Omega} = \frac{1}{\Omega_{sl}} \quad (12.45)$$

and therefore:

$$V' = \frac{1}{\Omega_{sl}} \sum_{\mathbf{G}', \mathbf{G}} a_{n', \mathbf{k}'_\xi}^*(\mathbf{G}') a_{n, \mathbf{k}_\xi}(\mathbf{G}) \sum_{\mathbf{t}_{sl}} e^{-i\mathbf{g}\cdot\mathbf{t}_{sl}} \int e^{-i\mathbf{g}\cdot\mathbf{r}} [V_{a'}(\mathbf{r}) - V_a(\mathbf{r})] d\tau \quad (12.46)$$

Note that, as in the bulk calculation, the integral over all space of the atomic potential multiplied by the exponential factor is assigned as the empirical pseudopotential form factor, i.e.

$$\frac{1}{\Omega_c} \int e^{-i\mathbf{g}\cdot\mathbf{r}} V_a(\mathbf{r}) d\tau = V_f^{r'_a}(g) \quad (12.47)$$

and:

$$\frac{1}{\Omega_c} \int e^{-i\mathbf{g}\cdot\mathbf{r}} V_{a'}(\mathbf{r}) d\tau = V_f^{r'_a}(g) \quad (12.48)$$

where $g = |\mathbf{g}|$.

As already mentioned, \mathbf{g} is a superlattice reciprocal lattice vector, and as the period of the superlattice is much greater than the period of the bulk crystal, then \mathbf{g} is much smaller than the bulk analogy \mathbf{q} . Thus, the pseudopotential form factor $V_f(g)$ needs to be known at intermediate points between the discrete bulk reciprocal lattice vectors—just as in the case of the large-basis calculation.

Therefore, the potential term becomes:

$$V' = \frac{\Omega_c}{\Omega_{sl}} \sum_{\mathbf{G}', \mathbf{G}} a_{n', \mathbf{k}'_\xi}^*(\mathbf{G}') a_{n, \mathbf{k}_\xi}(\mathbf{G}) \sum_{\mathbf{t}_{sl}} e^{-i\mathbf{g}\cdot\mathbf{t}_{sl}} [V_f^{r'_a}(g) - V_f^{r_a}(g)] \quad (12.49)$$

Hence, the final form of the Hamiltonian matrix elements \mathcal{H}' is obtained by substituting the potential matrix elements V' given in equation (12.49) into equation (12.34):

$$\begin{aligned} \mathcal{H}'_{(n', \mathbf{k}'_\xi), (n, \mathbf{k}_\xi)} &= E_{n, \mathbf{k}_\xi} \delta_{n'n} \delta_{\mathbf{k}'_\xi, \mathbf{k}_\xi} \\ &+ \frac{\Omega_c}{\Omega_{sl}} \sum_{\mathbf{G}', \mathbf{G}} a_{n', \mathbf{k}'_\xi}^*(\mathbf{G}') a_{n, \mathbf{k}_\xi}(\mathbf{G}) \sum_{\mathbf{t}_{sl}} e^{-i\mathbf{g}_s \cdot \mathbf{t}_{sl}} \left[V_{f'}^{\mathbf{r}'_a}(g) - V_f^{\mathbf{r}_a}(g) \right] \end{aligned} \quad (12.50)$$

Recalling the discussion in Section 11.8 that there is no single As potential for all semiconductors, it is important to realise that when creating a $\text{Ga}_{1-x}\text{Al}_x\text{As}/\text{GaAs}$ superlattice for solution by this perturbative approach, e.g. by substituting atoms onto some of the bulk sites, it is necessary to substitute both Al cations and the As anions associated with AlAs, as these are different from those associated with GaAs.

Equation (12.50) represents the elements of the square matrix $\mathcal{H}'_{(n', \mathbf{k}'_\xi), (n, \mathbf{k}_\xi)}$, the eigenvalues and eigenvectors of which represent, respectively, the energy levels and wave functions of the superlattice, where they are found by direct diagonalisation.

12.8 APPLICATION TO GAAS/ALAS SUPERLATTICES

In this section, the series of calculations given in Section 12.2 will be revisited and the lowest conduction-band energy calculated for the same range of GaAs/AlAs superlattices, again as a function of period, but this time by using the perturbative approach in order to implement an empirical pseudopotential calculation rather than the more direct large-basis method.

While the superlattice period controls the number of bulk electron wave vectors, given by the set \mathbf{k}_ξ , there is freedom as to the number of *bands* included in the perturbative expansion set. Such a choice is usually made by considering the system of interest and its relation to the bulk band structure. In particular, calculations of the lowest conduction-band state in a superlattice *could* be thought of as being adequately represented by considering just the lowest bulk-conduction-band state, whereas the triply degenerate (when excluding spin-orbit coupling) valence band suggests that all three may be needed for the calculation of the uppermost superlattice valence states.

Fig. 12.18 compares the results of calculations obtained by using both the perturbative method and the large-basis (LB) method, for the lowest-conduction-band state in a GaAs/AlAs superlattice as a function of the period ($n_z(A_0)$). As expected for a perturbative technique, the energy levels obtained are higher than those given by the direct large-basis solution.

It can be seen that some of the details of the data, previously commented upon in Section 12.2, are not reproduced when just one bulk conduction band ($N_n=1$) is used to provide the expansion set. In particular, the low-energy state of the $(\text{GaAs})_1(\text{AlAs})_1$ short-period superlattice is not reproduced, although there is some evidence of a change in the nature of the states, as evidenced by the functional form of the $E-n_z$ data, at around $n_z = 8$ or 9 , as discussed earlier.

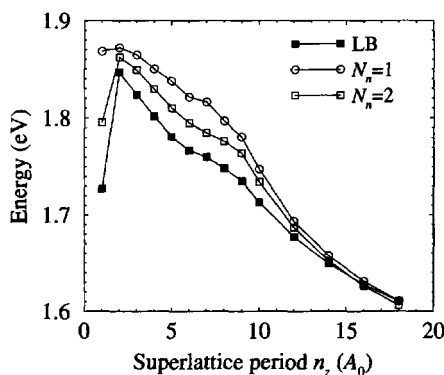


Figure 12.18 Comparison of the perturbative approach to the pseudopotential calculation of a superlattice, with $N_n=1$ and 2 conduction band(s), with that of the large-basis (LB) method of earlier

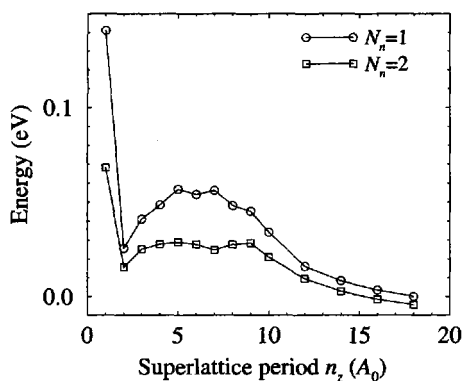


Figure 12.19 The energy differences between the perturbative and large basis approaches of Fig. 12.18

When the expansion set is increased to include a second bulk conduction band, the results are much closer to those of the large-basis calculation, as illustrated by the $N_n = 2$ data in Fig. 12.18. Thus, the obvious conclusion can be drawn that the more bulk bands included in the expansion set, then the better the approximation produced by the perturbative approach. These data are summarized in Fig. 12.19, which displays the energy difference between the calculations. Analysis of this figure allows a second conclusion to be drawn, i.e. that the larger the superlattice period, then the better the approximation given by the perturbative approach.

Table 12.1 takes this line of enquiry further, displaying the results of calculations of the lowest conduction-band state of the $(\text{GaAs})_{10}(\text{AlAs})_{10}$ superlattice, (i.e. $n_z = 10$), as a function of the number of bulk-conduction bands (N_n) included in the basis set. The improvement, evidenced by the lowering in the energy level, as implied by

Table 12.1 The lowest conduction-band energy level of a $(\text{GaAs})_{10}(\text{AlAs})_{10}$ superlattice as a function of the number of bulk conduction bands N_n included in the basis set

N_n	E_1 (eV)
1	1.747636
2	1.734434
3	1.734429
4	1.734421
5	1.734421

the variational principle, is clear when moving from just one bulk conduction band $N_n=1$ to $N_n=2$. The data show that further improvements are marginal, i.e. less than 1 meV, and are really not worth the additional computational time. The 'best' value offered by this approach, i.e. 1.734 eV, compares well with the 1.713 eV given by the large basis calculation. The perturbative approach attempts to replace the rather crude set of plane waves with a more sophisticated basis set. However, a discrepancy still exists because no matter how many bulk bands are included in the superlattice expansion set, each one is still only constructed from the same set of N (in this case, 65) plane waves. This difference between the two approaches diminishes as the superlattice period increases.

The calculations in this section lead to the conclusion that the large-basis approach gives lower, and therefore more accurate (from the variational principle) energy levels. However, good approximations can be made by using the perturbative approach, particularly for the longer superlattice periods. The further consideration of computational expense is discussed below in Section 12.11.

12.9 INCLUSION OF REMOTE BANDS

It has been concluded in the previous section that, for these particular GaAs/AlAs superlattices, if the basis set for the perturbative approach for the superlattice calculation is *limited* to bulk-conduction bands, then two bands are sufficient.

However, the question remains, *Can an extended basis including more remote bands, give improved energies?*. This implies, for the present series of calculations, the necessary inclusion of one or more of the bulk-valence bands in the basis set.

Table 12.2 compares the results of just such calculations, including the uppermost bulk-valence-band state, for the series of $(\text{GaAs})_{n_z}(\text{AlAs})_{n_z}$ superlattices of earlier. It can be seen that the difference in the lowest conduction-band energy level, calculated including the remote valence band (2CB+1VB), is always less than 1 meV than the original (2CB) data, across the entire range of superlattice periods being consid-

Table 12.2 The effect of the inclusion of the uppermost bulk-valence band in the basis set of a $(\text{GaAs})_{n_z}(\text{AlAs})_{n_z}$ superlattice

n_z	$E_1^{2\text{CB}}$ (eV)	$E_1^{2\text{CB}+1\text{VB}}$ (eV)
1	1.795507	1.796218
2	1.861993	1.862025
3	1.849019	1.849090
4	1.829507	1.829511
5	1.809960	1.809997
6	1.794834	1.794835
7	1.784877	1.784882
8	1.776331	1.776331
9	1.764226	1.764241
10	1.734433	1.734456
12	1.686980	1.687005
14	1.652077	1.652094

ered. Therefore, the inclusion of remote bands for these particular superlattices is unnecessary.

12.10 THE VALENCE BAND

As mentioned earlier, the triple degeneracy of the valence band at the centre of the bulk Brillouin zone, produced in this series of calculations without spin-orbit coupling, suggests that all three bands may be needed in any valence-band calculation. Fig. 12.20 displays the highest-valence-band state of the GaAs/AlAs superlattices, calculated by using the three uppermost bulk-valence-bands, compared with the results of earlier produced by using the large-basis (LB) method.

Again, the same conclusions can be drawn as for the conduction-band calculations, namely that the perturbative approach does not give energies as low as the large basis method, while the discrepancy between the two decreases as the superlattice period increases. This is highlighted in Fig. 12.21.

12.11 COMPUTATIONAL EFFORT

The main computational difficulty with the large-basis approach is the demands on computer memory necessary to store the large Hamiltonian matrix. In this respect, the perturbative method can be quite computationally efficient. As mentioned above, the order of the Hamiltonian matrix requiring diagonalisation in this latter method is

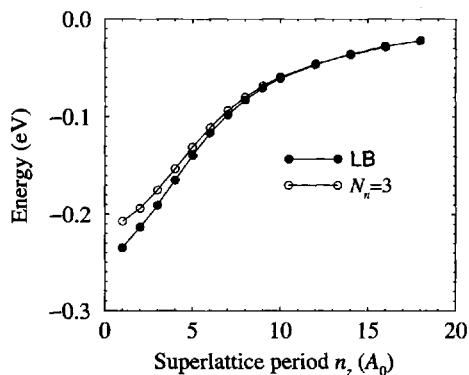


Figure 12.20 Comparison of the perturbative approach to the pseudopotential calculation of a superlattice, with $N_n = 3$ valence bands, with that of the large-basis (LB) method of earlier

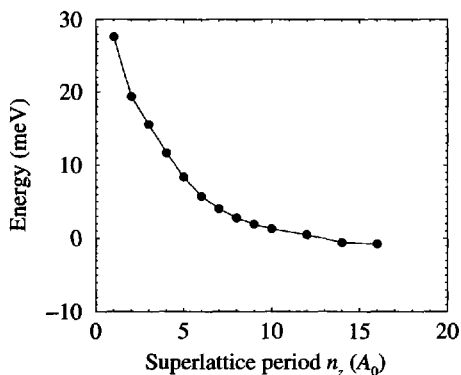


Figure 12.21 The difference between the energies of Fig. 12.20

given by:

$$O(\mathcal{H}'_{(n', \mathbf{k}'_\xi), (n, \mathbf{k}_\xi)}) = N_n N_{\mathbf{k}_\xi} = 2N_n n_z \quad (12.51)$$

where n_z is the period of the superlattice in lattice constants. In comparison, the number of plane waves required in the expansion set for the large-basis calculation was deduced empirically in Section 12.2 as approximately $128n_z$; hence, for the large basis approach:

$$O(\mathcal{H}_{G', G}) \approx 128n_z \quad (12.52)$$

Thus, it can be seen that the order of both matrices scales linearly with the superlattice period n_z . However, it is clear that if the number of bands N_n required in the perturbative calculation is less than 64, then the Hamiltonian matrix $\mathcal{H}'_{(n', \mathbf{k}'_\xi), (n, \mathbf{k}_\xi)}$ will be smaller than $\mathcal{H}_{G', G}$.

The computational effort associated with the actual process of diagonalisation of these matrices scales as the square of the order, and hence this part of the perturbative approach will require less computer time, as well as less memory, than the large basis approach.

12.12 SUPERLATTICE DISPERSION AND THE INTERMINIBAND LASER

As yet, the dispersion curves for electron movement along the growth (z -) axis, i.e., perpendicular transport, have not been calculated by using the perturbative approach to the superlattice.

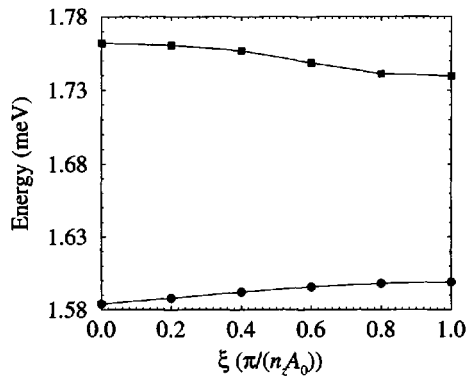


Figure 12.22 The two lowest energy dispersion curves (minibands) along the growth (z -) axis of a $(\text{GaAs})_{10}(\text{Ga}_{0.8}\text{Al}_{0.2}\text{As})_{10}$ superlattice

Using the virtual crystal approximation to obtain the form factors for the alloy, Fig. 12.22 plots the lowest two conduction bands of the superlattice, for a range of electron wave vectors ξ across the Brillouin zone of a $(\text{GaAs})_{10}(\text{Ga}_{0.8}\text{Al}_{0.2}\text{As})_{10}$ superlattice. Just as the superlattice Brillouin zone is often referred to as the *minizone*, these superlattice dispersion curves are called *minibands* (see Section 2.13). As discovered in Section 2.13, when using the Kronig–Penney model, the lowest-energy miniband has its minimum at the zone centre, while the second miniband has its minimum at the zone edges.

Thus, if carriers are injected into the upper of these two minibands then they will rapidly lower their energy, via the fast non-radiative *intraminiband* scattering processes of phonon emission and carrier scattering. These have times similar to those characterising intrasubband processes, perhaps of the order of 100 fs (see Chapter 9). Given the functional form of the E - \mathbf{k} curves within the superlattice Brillouin zone, this energy lowering implies an increasing momentum and the carriers thus move towards the zone edge.

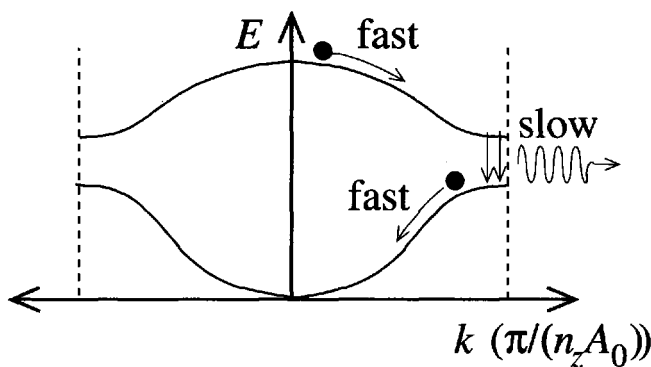


Figure 12.23 The carrier dynamics within an interminiband laser

At this point, the carriers have reached the bottom of the miniband and can no longer lower their energy through intraminiband relaxation processes, and they must wait the relatively long time for an *interminiband* scattering event to occur. Following this, any carrier scattering to the states near the zone edge of the lowest miniband, will scatter rapidly to lower momentum states. Such a scenario of a long lifetime in the upper level, compared to a short lifetime in the lower level, for the selection of electron momenta near the superlattice zone edge, is reminiscent of the requirements of the intersubband laser of Chapter 9. Indeed, the combination of these scattering processes within the two lowest minibands of a superlattice forms the basis of a new form of semiconductor laser known as the *interminiband laser* [41,42]. One exciting feature of this device is that it lases without an overall population inversion: it is only necessary to obtain a localised inversion in the states near the zone edge for stimulated emission to occur.

12.13 ADDITION OF ELECTRIC FIELD

An electric field can be considered merely as a perturbation on the semiconductor system, just as the superlattice itself was. However, given the inherent periodic nature of the pseudopotential method, the electric field itself has to be periodic and repeated within each unit cell, as shown in Fig. 12.24.

Taking the first and last atomic positions within the unit cell of interest as the extent of the electric field, and defining the zero-field point as the centre of the cell,

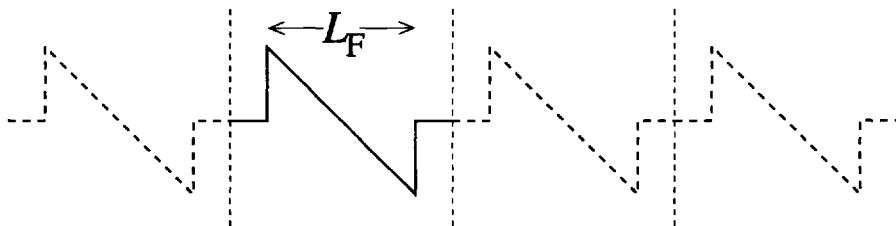


Figure 12.24 The necessary periodic nature of any electric field accounted for with pseudopotential theory

i.e. $(z_{N_a} + z_1)/2$, then the potential energy due to the electric field would be:

$$V_F = 0, \quad z < z_1 \quad (12.53)$$

$$V_F = -qF \left(z - \frac{z_{N_a} + z_1}{2} \right), \quad z_1 < z < z_{N_a} \quad (12.54)$$

$$V_F = 0, \quad z_{N_a} < z \quad (12.55)$$

where z_1 and z_{N_a} are the z -coordinates of the first (1) and last (N_a) atoms, respectively. With this definition, the field does look like that in Fig. 12.24, with small zero-field regions of width $A_0/2$ in between the regions of linear sloping potential.

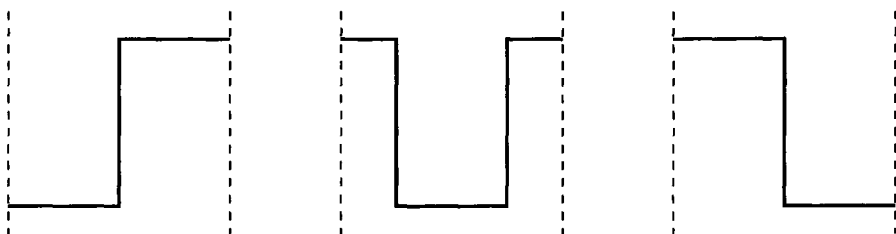


Figure 12.25 The possible unit cells for a pseudopotential study of a superlattice

The pseudopotential calculations so far have all centred around true superlattices, i.e. systems of quantum wells with significant overlap between the wave functions of adjacent wells. Single isolated quantum wells (SQWs) can be considered by using pseudopotential theory just by making the barriers within each period thick in size, thus producing a large distance between the wells. Hitherto it was not relevant where the atoms were exchanged within the unit cell to produce a superlattice, for example, as shown in Fig. 12.25 the well could be formed at the beginning of the unit cell, the middle or the end. However, when incorporating an electric field as well, it is important that the field extends either side of the 'region of interest'—this is achieved simply by ensuring the quantum wells are centred in the unit cell.

The effect of an electric field can be calculated for a single quantum well, or a system of several quantum wells, by the appropriate choice of unit cell (see for example Fig. 12.26).

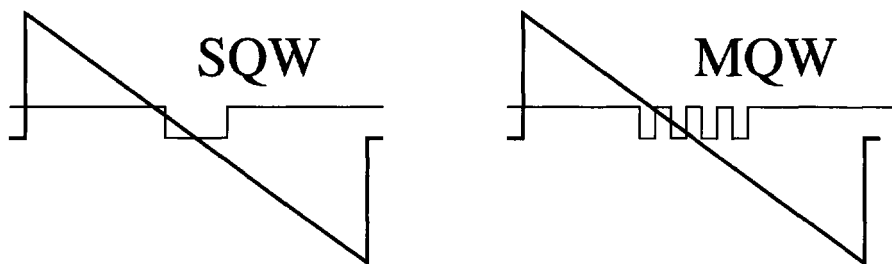


Figure 12.26 The unit cells required to study the effect of an electric field on a single quantum well (SQW) or a multiple quantum well (MQW)

Therefore, an electric field can be considered as an additional perturbation with the same periodicity as the superlattice unit cell, whether that cell contains one or more quantum wells. Thus, the original Schrödinger equation for the superlattice (equation 12.21) i.e.

$$(\mathcal{H} + V_{sl})\Psi_{N,\xi} = E_{N,\xi}\Psi_{N,\xi} \quad (12.56)$$

would have an addition term representing the perturbation due to the electric field, i.e.

$$(\mathcal{H} + V_{sl} + V_F)\Psi_{N,\xi} = E_{N,\xi}\Psi_{N,\xi} \quad (12.57)$$

the consequence of which is that the potential term in equation (12.38) becomes:

$$V'' = \frac{1}{\Omega} \sum_{\mathbf{G}', \mathbf{G}} a_{n', \mathbf{k}'_\xi}^*(\mathbf{G}') a_{n, \mathbf{k}_\xi}(\mathbf{G}) \int e^{-i(\mathbf{G}' - \mathbf{G} + \mathbf{k}'_\xi - \mathbf{k}_\xi) \cdot \mathbf{r}} (V_{sl} + V_F) d\tau \quad (12.58)$$

The manipulation of the first term (V_{sl}) clearly proceeds as before, thus giving the original perturbing potential V' due to the superlattice potential, as defined in equation (12.49), so therefore:

$$V'' = V' + \frac{1}{\Omega} \sum_{\mathbf{G}', \mathbf{G}} a_{n', \mathbf{k}'_\xi}^*(\mathbf{G}') a_{n, \mathbf{k}_\xi}(\mathbf{G}) \int e^{-i(\mathbf{G}' - \mathbf{G} + \mathbf{k}'_\xi - \mathbf{k}_\xi) \cdot \mathbf{r}} V_F d\tau \quad (12.59)$$

Consider just the integral component, and again writing $\mathbf{g} = \mathbf{G}' - \mathbf{G} + \mathbf{k}'_\xi - \mathbf{k}_\xi$, then:

$$\frac{1}{\Omega} \int e^{-i(\mathbf{G}' - \mathbf{G} + \mathbf{k}'_\xi - \mathbf{k}_\xi) \cdot \mathbf{r}} V_F d\tau = \frac{1}{\Omega} \int e^{-i\mathbf{g} \cdot \mathbf{r}} V_F d\tau \quad (12.60)$$

Given the form of V_F in equation (12.55) and writing the origin of the electric field potential as $z_0 = (z_{Na} + z_1)/2$, then obtain:

$$\frac{1}{\Omega} \int e^{-i\mathbf{g} \cdot \mathbf{r}} V_F d\tau = -\frac{qF}{\Omega} \int_{-\infty}^{+\infty} \int_{-\infty}^{+\infty} \int_{-\infty}^{+\infty} (z - z_0) e^{-i\mathbf{g} \cdot \mathbf{r}} dz dx dy \quad (12.61)$$

The in-plane (x - y) integrals only have value when the x - and y -components of \mathbf{g} are zero, and are then equal to the length of the crystal in that dimension, i.e.

$$\frac{1}{\Omega} \int e^{-i\mathbf{g}_\bullet \cdot \mathbf{r}} V_F d\tau = -\frac{qFL_xL_y}{\Omega} \int_{-\infty}^{+\infty} (z - z_0) e^{-ig_z z} dz \delta_{0,g_x} \delta_{0,g_y} \quad (12.62)$$

where the vector coefficients g_x and g_y are defined by $\mathbf{g} = g_x \hat{\mathbf{i}} + g_y \hat{\mathbf{j}} + g_z \hat{\mathbf{k}}$, and thus the Dirac δ -functions ensure that the integral is non-zero for \mathbf{g} vectors along the axis of the field only. If there are N_{sl_z} unit cells along the z -direction, then:

$$\frac{1}{\Omega} \int e^{-i\mathbf{g}_\bullet \cdot \mathbf{r}} V_F d\tau = -\frac{qFL_xL_yN_{sl_z}}{\Omega} \int_{z_1}^{z_{N_a}} (z - z_0) e^{-ig_z z} dz \delta_{0,g_x} \delta_{0,g_y} \quad (12.63)$$

However, the total volume of the crystal $\Omega = L_xL_yN_{sl_z}n_zA_0$, where, of course, n_zA_0 is the superlattice period. Hence:

$$\frac{1}{\Omega} \int e^{-i\mathbf{g}_\bullet \cdot \mathbf{r}} V_F d\tau = -\frac{qF}{n_zA_0} \int_{z_1}^{z_{N_a}} (z - z_0) e^{-ig_z z} dz \delta_{0,g_x} \delta_{0,g_y} \quad (12.64)$$

Integrating by parts, then obtain:

$$\begin{aligned} \therefore \frac{1}{\Omega} \int e^{-i\mathbf{g}_\bullet \cdot \mathbf{r}} V_F d\tau = & -\frac{qF}{n_zA_0} \left\{ \left[(z - z_0) \frac{e^{-ig_z z}}{-ig_z} \right]_{z_1}^{z_{N_a}} \right. \\ & \left. - \frac{1}{-ig_z} \int_{z_1}^{z_{N_a}} e^{-ig_z z} dz \right\} \delta_{0,g_x} \delta_{0,g_y} \end{aligned} \quad (12.65)$$

$$\therefore \frac{1}{\Omega} \int e^{-i\mathbf{g}_\bullet \cdot \mathbf{r}} V_F d\tau = -\frac{qF}{n_zA_0} \left[(z - z_0) \frac{ie^{-ig_z z}}{g_z} + \frac{e^{-ig_z z}}{g_z^2} \right]_{z_1}^{z_{N_a}} \delta_{0,g_x} \delta_{0,g_y} \quad (12.66)$$

and:

$$\frac{1}{\Omega} \int e^{-i\mathbf{g}_\bullet \cdot \mathbf{r}} V_F d\tau = -\frac{qF}{n_zA_0} \left[\left(\frac{i(z - z_0)}{g_z} + \frac{1}{g_z^2} \right) e^{-ig_z z} \right]_{z_1}^{z_{N_a}} \delta_{0,g_x} \delta_{0,g_y} \quad (12.67)$$

which upon evaluation, gives:

$$\begin{aligned} \therefore \frac{1}{\Omega} \int e^{-i\mathbf{g}_\bullet \cdot \mathbf{r}} V_F d\tau = & -\frac{qF}{n_zA_0} \left\{ \left(\frac{i(z_{N_a} - z_0)}{g_z} + \frac{1}{g_z^2} \right) e^{-ig_z z_{N_a}} \right. \\ & \left. - \left(\frac{i(z_1 - z_0)}{g_z} + \frac{1}{g_z^2} \right) e^{-ig_z z_1} \right\} \delta_{0,g_x} \delta_{0,g_y} \end{aligned} \quad (12.68)$$

The total potential term for both the superlattice perturbation and the electric field perturbation is obtained by substituting equation (12.68) into equation (12.59), thus giving:

$$V'' = V' + \sum_{\mathbf{G}', \mathbf{G}} a_{n', \mathbf{k}'}^*(\mathbf{G}') a_{n, \mathbf{k}}(\mathbf{G}) \left(-\frac{qF}{n_zA_0} \right)$$

$$\times \left\{ \left(\frac{i(z_{N_a} - z_0)}{g_z} + \frac{1}{g_z^2} \right) e^{-ig_z z_{N_a}} - \left(\frac{i(z_1 - z_0)}{g_z} + \frac{1}{g_z^2} \right) e^{-ig_z z_1} \right\} \delta_{0,g_x} \delta_{0,g_y} \quad (12.69)$$

Using the definition for the superlattice perturbing potential V' in equation (12.49), then the final form for the Hamiltonian matrix elements *including* an electric field is:

$$\begin{aligned} \mathcal{H}'_{(n',k'_\xi),(n,k_\xi)} &= E_{n,k_\xi} \delta_{n'n} \delta_{k'_\xi,k_\xi} + \sum_{G',G} a_{n',k'_\xi}^*(G') a_{n,k_\xi}(G) \\ &\times \left\{ \frac{\Omega_c}{\Omega_{sl}} \sum_{t_{sl}} e^{-ig_\bullet t_{sl}} \left[V_{f'}^{r'_a}(g) - V_f^{r'_a}(g) \right] - \frac{qF}{n_z A_0} \delta_{0,g_x} \delta_{0,g_y} \right. \\ &\times \left. \left\{ \left(\frac{i(z_{N_a} - z_0)}{g_z} + \frac{1}{g_z^2} \right) e^{-ig_z z_{N_a}} - \left(\frac{i(z_1 - z_0)}{g_z} + \frac{1}{g_z^2} \right) e^{-ig_z z_1} \right\} \right\} \quad (12.70) \end{aligned}$$

which is an extension of the earlier form in equation (12.50). Inspection of equation (12.70) does raise a small problem, namely divergence of the electric field perturbation when $g_z = 0$. For this particular instance, it is necessary to revisit equation (12.64) and put $g_z = 0$, i.e.

$$\frac{1}{\Omega} \int e^{-ig_\bullet \mathbf{r}} V_F \, d\tau = -\frac{qF}{n_z A_0} \int_{z_1}^{z_{N_a}} (z - z_0) \, dz \, \delta_{0,g_x} \delta_{0,g_y} \quad (12.71)$$

and then:

$$\frac{1}{\Omega} \int e^{-ig_\bullet \mathbf{r}} V_F \, d\tau = -\frac{qF}{n_z A_0} \left[\frac{z^2}{2} - z z_0 \right]_{z_1}^{z_{N_a}} \delta_{0,g_x} \delta_{0,g_y} \quad (12.72)$$

Recalling that $z_0 = (z_{N_a} + z_1)/2$, then this becomes:

$$\frac{1}{\Omega} \int e^{-ig_\bullet \mathbf{r}} V_F \, d\tau = 0 \quad (12.73)$$

Therefore, it has been shown that an electric field can be included in the pseudopotential formalism, with the result being an additional potential term in the Hamiltonian matrix.

Consider the application of an electric field to a single 10 ml (28.25 Å) thick GaAs quantum well surrounded by 20 ml (56.5 Å) thick $\text{Ga}_{0.8}\text{Al}_{0.2}\text{As}$ barriers. Thus, in the periodic formalism characteristic of pseudopotentials, this would imply a separation of 2×20 ml (113 Å), which should be enough for them to act as independent quantum wells.

Fig. 12.27 displays the results of calculations of the change in the energy (ΔE) of the lowest conduction-band energy level as a function of the electric field, as deduced by the empirical pseudopotential (EPP) method. For comparison the figure shows also data obtained by the envelope function approximation (EFA) for the same system. The parabolic nature of this, i.e. the quantum-confined-Stark effect, is clearly

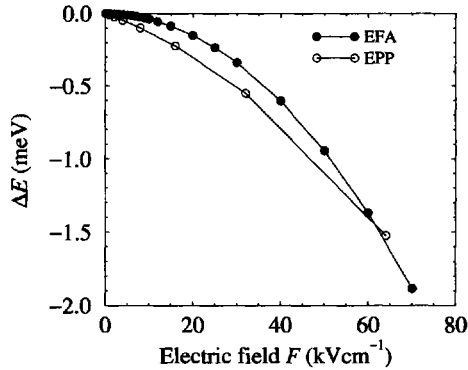


Figure 12.27 The change in the energy of the lowest eigenstate of a 28.25 Å GaAs single quantum well surrounded by Ga_{0.8}Al_{0.2}As barriers, as a function of the applied electric field

evidenced in the empirical pseudopotential calculation, and in fact, the energy changes are remarkably similar over the range of electric fields employed. Thus, Fig. 12.27 seems to imply, at first sight, that the computationally 'long-winded' approach of the empirical pseudopotential method is no better than much quicker methods based on the envelope function approximation. Indeed, if Stark effect energy level changes are all that is of interest, then this would be the case. However, the pseudopotential method gives much more, e.g. the same calculation can also be used to yield the in-plane dispersion curve, and the pseudopotential method can account much more fundamentally for valence band mixing and strain, *at the same time* as modelling the electric field.

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CHAPTER 13

APPLICATION TO QUANTUM WIRES AND DOTS

13.1 RECENT PROGRESS

The difficulties associated with applying the techniques outlined in this work to lower-dimensional quantum wires and dots, have been hinted at already. In the work on point defects in bulk material (Section 11.13), in which a single impurity atom was placed within a larger cube of bulk material, more than 2000 plane waves were needed in the expansion set. In effect, such a system represents the ultimate quantum dot, with the confining potential existing on account of the single atom, after all, quantum dots are often referred to as *artificial atoms* (see, for example [265]). Single-atom quantum dots are also *single-electron* quantum dots [266, 267]. To deal with larger quantum dots, remembering that real dots often have a base length in the range 100–400 Å, requires a very large number of plane waves, and indeed the problem becomes too large to handle with present day computers. The perturbative approach, as derived for the superlattice in Chapter 12, can be extended to deal with quantum dots [268], and this appears to be a promising improvement for modest-sized unit cells.

Recently, Zunger and co-workers have demonstrated a new computational technique which deals with this large-matrix problem. Instead of seeking solutions of the

usual Schrödinger equation, i.e.

$$\mathcal{H}\psi_{n,\mathbf{k}} = E_{n,\mathbf{k}}\psi_{n,\mathbf{k}} \quad (13.1)$$

solutions are sought to the alternative expression:

$$(\mathcal{H} - E_{\text{ref}})^2 \psi_{n,\mathbf{k}} = (E_{n,\mathbf{k}} - E_{\text{ref}})^2 \psi_{n,\mathbf{k}} \quad (13.2)$$

where the reference energy, E_{ref} , can be chosen to lie within the fundamental gap, and hence the valence- and conduction- band-edge states, which are often those of primary interest, are transformed from being arbitrarily high-energy states to being the lowest states. The technique is to minimize the expectation value:

$$\langle \psi | (\mathcal{H} - E_{\text{ref}})^2 | \psi \rangle \quad (13.3)$$

where the standard empirical pseudopotential operator of equation (11.5) can be employed. The approach has become known as the *folded spectrum method*; for more details of the theory, see Wang and Zunger [269], while for an example of its application, see Wang *et al.* [270].

At the present time, such a technique lies beyond the scope of this introductory text, and although limited, the large-basis method coupled with direct diagonalisation of the Hamiltonian matrix, will be explored to its limits with application to the reduced-dimensional systems of quantum wires and dots.

13.2 THE QUANTUM-WIRE UNIT CELL

The unit cell that has to be employed for a quantum wire is just a special case of the extended cube utilized earlier in Section 11.13 for impurities in bulk material. The periodic nature of the pseudopotential method implies that, in fact the crystal potential will be an infinite sequence of parallel quantum wires, as shown in Fig. 13.1. However making use of the symmetry properties of the wire, the latter need only be 1 lattice constant in length, i.e. assuming the wire to lie along the z -axis, then the number of lattice constants in this direction is given by $n_z = 1$ (see Fig. 13.2).

A barrier or 'cladding' material surrounds the quantum wire; in this method, it is important that this is of sufficient thickness to isolate the wires from their hypothetical neighbours—just as in the case of modelling single quantum wells using pseudopotential theory. This thickness depends on the confinement of the carriers within the wire. For example, for wide wires, the confinement is high and hence the overlap of the wave function with that in an adjacent wire is low. Therefore, only a thin barrier layer may be needed.

For the purpose of this illustrative example, consider a square cross-sectional Ge quantum wire surrounded by Si barriers. The atomic positions within the unit cell are illustrated in Fig. 13.3.

The unit cell is just five lattice constants square, i.e. $n_x=5$, $n_y=5$, and as mentioned above, just one lattice constant deep, i.e. $n_z=1$. About the only useful wire that

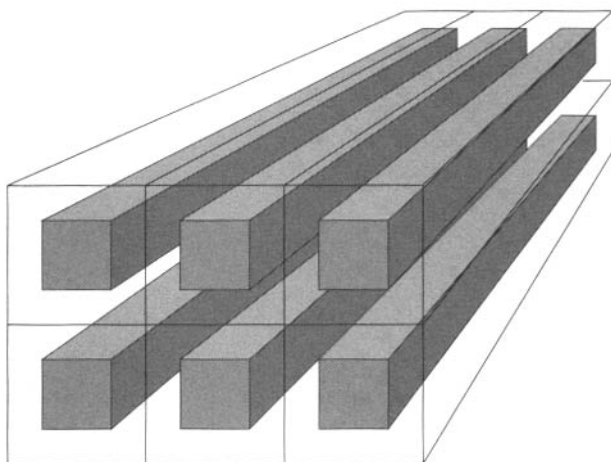


Figure 13.1 The periodic nature of the quantum wire unit cell

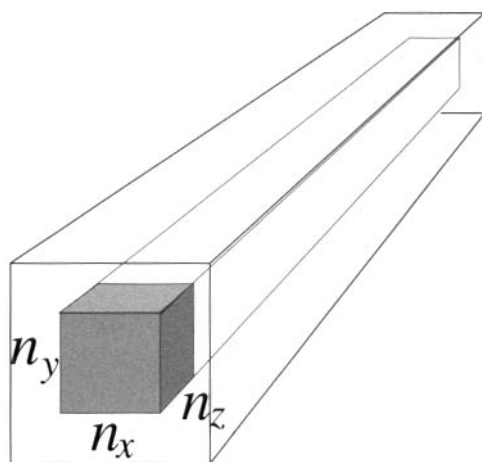


Figure 13.2 The quantum-wire unit cell; note the depth of 1 lattice constant, with sufficient barrier to encompass the wire, and if modelling a single wire, to localise the charge

can be accommodated in this box would be three lattice constants square with a single lattice constant barrier, thus giving a total of two lattice constants between the wires; the reason for these limitations on the wire geometry being that for even for this small cross-section wire, the number of plane waves required in the expansion would normally be 6625 , which would give a Hamiltonian matrix $\mathcal{H}_{G,G'}$ occupying $6625^2 \times 8 = 351$ Mbytes of computer memory. This is only just in reach of high-end desktop computers at the present time, so for the purpose of these illustrative calculations, the expansion set will be reduced by truncating at a maximum reciprocal

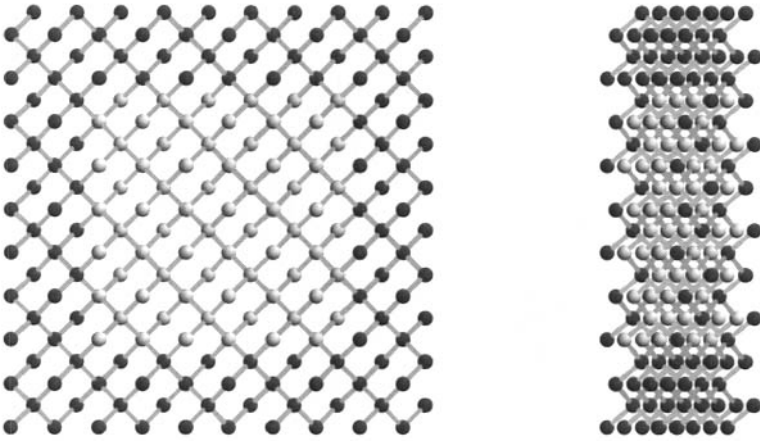


Figure 13.3 The quantum-wire unit cell; note the depth of 1 lattice constant

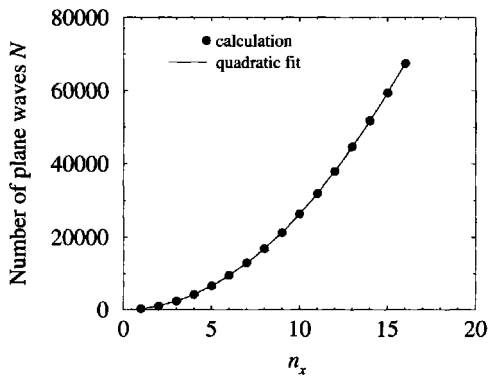


Figure 13.4 The number of plane waves in the expansion set versus the length of the wire side n_x

lattice vector of $3 \times (2\pi/A_0)$, rather than the usual 4. This reduces the expansion set to 2751, which still represents 60 Mbytes.

Fig. 13.4 illustrates how the untruncated expansion set increases with the number of lattice constants along the side of the wire unit cell. The accuracy of the fit indicates that the expansion set increases as the square of n_x . Given that the number of atoms within the unit cell is also proportional to the area, which in this case of a *square* unit cell is proportional to n_x^2 , then the expansion set may also be expected to be proportional to the number of atoms in the unit cell. This is confirmed by the linear fit shown in Fig. 13.5.

As machine specifications increase, larger wire unit cells will be tractable, thus making this straightforward direct diagonalisation method more useful. For example,

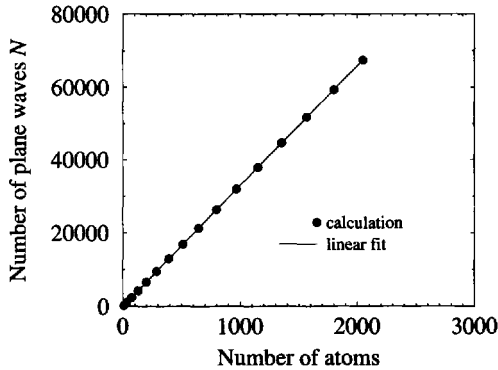


Figure 13.5 The number of plane waves in the expansion set versus the number of atoms in the unit cell

at the time of writing, desktop machines with 1 Gbyte of RAM are becoming more common, which would allow the full expansion set to be used, or a quantum-wire unit cell of twice the side (four times the area) to be tackled.

13.3 CONFINED STATES

The results of a direct diagonalisation of the Hamiltonian matrix for the quantum-wire unit cell in Fig. 13.3 are shown in Fig. 13.6. The latter illustrates the charge density of the lowest conduction-band state for an area the size of the unit cell and across the $z = 0$ plane. The origin of the plot has been shifted slightly in order to centralise the wire within the unit cell.

13.4 V-GROOVED QUANTUM WIRES

Due to the limitations on unit-cell size, some ‘artistic license’ has to be invoked so as to produce a rough approximation to a cross-section of a ‘V-groove’ quantum wire (illustrated in Fig. 13.7).

The same method of calculation as used in the previous section can be employed to generate the corresponding charge density, again for the lowest conduction-band state, as now given in Fig. 13.8. The change in the charge distribution in comparison with a square cross-section wire can be seen by referring back to Fig. 13.6, i.e. a relatively small change in the number of atoms in the wire unit cell, which changes the symmetry of the wire, leads to a quite different charge distribution.

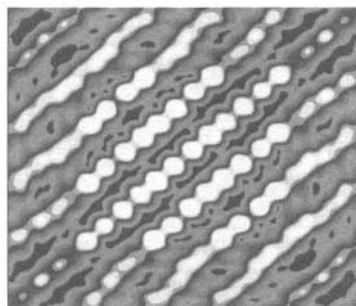


Figure 13.6 The charge density of the lowest conduction-band state over the cross-section of a Ge quantum wire embedded in a Si host

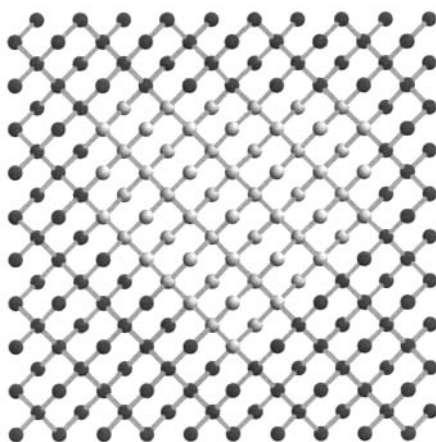


Figure 13.7 A 'V-grooved' quantum-wire unit cell

13.5 ALONG-AXIS DISPERSION

The shortest reciprocal lattice vector along any of the mutually perpendicular Cartesian axes in bulk is of the form $(0,0,2)$ when expressed in units of $2\pi/A_0$ (see, for example, Table 11.1). Therefore the edge of the Brillouin zone in this direction is $(0,0,1)$, i.e. the point usually referred to as 'X'. However, for quantum wires, the shortest 'along-axis' reciprocal lattice vector is actually $(0,0,1)$, and the edge of the Brillouin zone is therefore $(0,0,\frac{1}{2})$. This is a little counter-intuitive, since along the axis of the wire, the material might be considered as being just infinitely extended

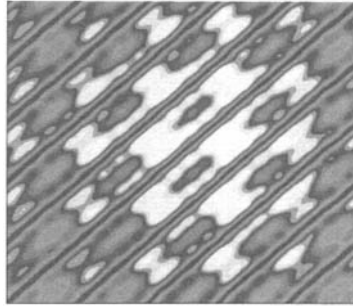


Figure 13.8 The charge density of the lowest conduction-band state over the cross-section of a 'V-grooved' Ge quantum wire embedded in a Si host

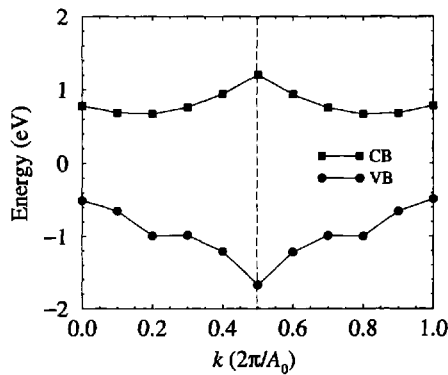


Figure 13.9 The along-axis dispersion of a Ge quantum wire: CB, conduction band; VB, valence band

bulk, and therefore be expected to exhibit bulk-material properties. In fact, as the calculations of the near-band-edge dispersion curves displayed in Fig. 13.9 show, this is not the case, and indeed zone-folding does occur.

13.6 TINY QUANTUM DOTS

The problem of the very large basis set is even more acute for a quantum dot, which requires a three-dimensional cubic (or cuboid) unit cell to encase it. Again, with the present computational methods, only small calculations can be attempted, which do, however, illustrate how such calculations are set up.

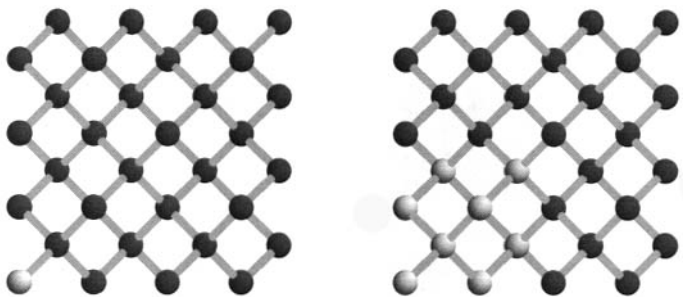


Figure 13.10 Two of the units cells for the 'tiny' quantum dot calculations: containing 1 Ge atom (left); and 8 Ge atoms (right)

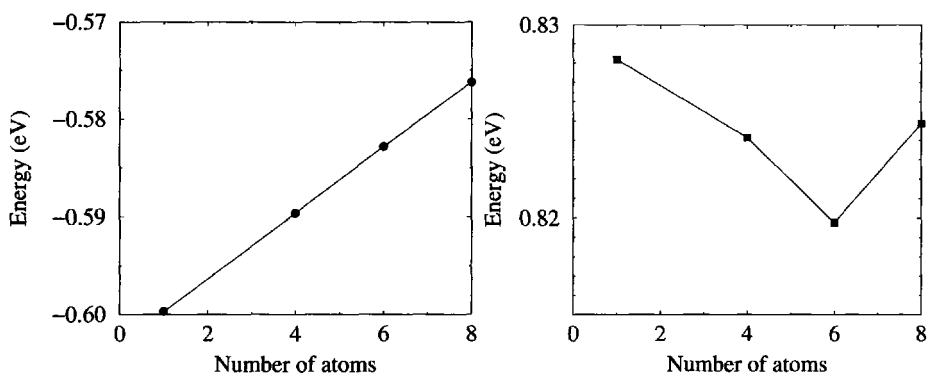


Figure 13.11 The energy of the uppermost valence state (left) and lower most conduction state (right) as a function of the number of Ge atoms embedded in a Si host

Consider a unit cell of $2 \times 2 \times 2$ lattice constants; this contains 64 atoms. The (almost ridiculously small) simplest quantum dot that could be placed within this unit cell would be just one atom. Taking silicon as the host material and germanium as the dot, then this particular case is just the isoelectronic impurity dealt with earlier (see Chapter 11). However, in this case, the single Ge atom will be placed at the extreme corner of the unit cell (see Fig. 13.10), which will show the equivalence with the previous example, and will also be the approach used for subsequent calculations. This method of choosing the unit cell is somewhat easier than that used in the quantum-wire calculations described earlier in this chapter.

After choosing the unit cell, which in turn determines the reciprocal lattice vectors in the expansion set, then the empirical pseudopotential method can yield information such as the energy of the uppermost valence-band state or the lowest conduction-band state. These are illustrated for this simple example in Fig. 13.11. In addition, it is possible to plot the charge density for any state or number of states (see Fig. 13.12).

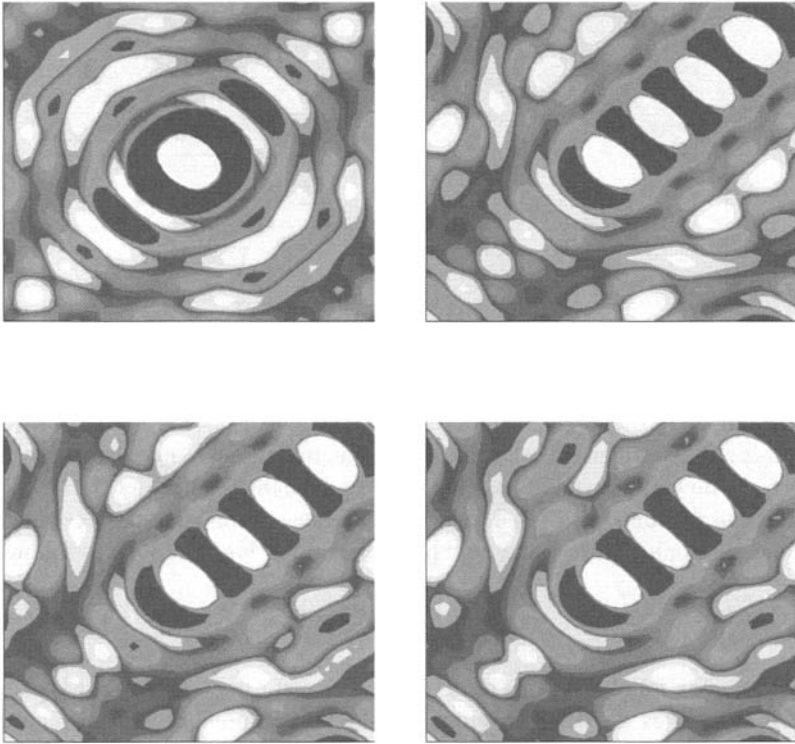


Figure 13.12 The difference in the total valence-band charge density between the ‘tiny’ quantum dots and bulk Si for different numbers of Ge atoms: (top left) 1; (top right) 4; (bottom left) 6; and (bottom right) 8

Note that the plots are centred on the point occupied by the single Ge atom in Fig. 13.10 (left).

13.7 PYRAMIDAL QUANTUM DOTS

As mentioned in Chapter 8, the deposition of a thin layer of one material on top of a substrate where there is a large difference in lattice constants, *can* lead to the formation of pyramidal-shaped quantum dots. These quantum dots can be quite large, perhaps a few hundred Angstroms across, and are totally beyond the range of direct diagonalisation of the empirical pseudopotential Hamiltonian matrix. However, solution has been achieved by using the folded-spectrum method [271].

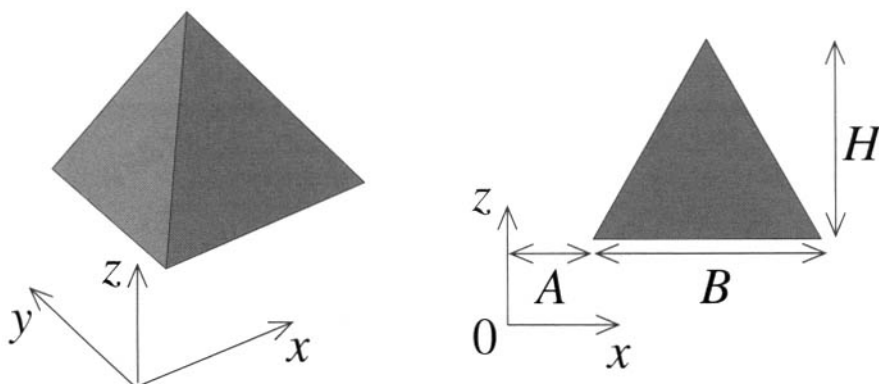


Figure 13.13 Schematic representations of a pyramidal quantum dot

The setting up of such a calculation requires a knowledge of the atomic positions, so with this aim, consider the schematic representation of a pyramidal dot shown in Fig. 13.13. Allowing the dot of base length B and height H to be encased within a cubic unit cell of side $A + B + A$, then the equations of the planes represented by the diagonally sloping sides in the right hand diagram of Fig. 13.13 are as follows:

$$z = \frac{2H}{B}x - \frac{2H}{B} + A \quad \text{and} \quad z = -\frac{2H}{B}x + \frac{2HA}{B} + 2H + A \quad (13.4)$$

Similar equations follow, but this time in terms of y for the remaining two planes. Fig. 13.14 illustrates the resulting atomic positions for an elemental quantum dot, such as Ge, as might occur when deposited on a Si substrate (the Si atoms have been removed for clarity). Note that at certain ratios of the height and base, horizontal ridges form around the circumference of the pyramid.

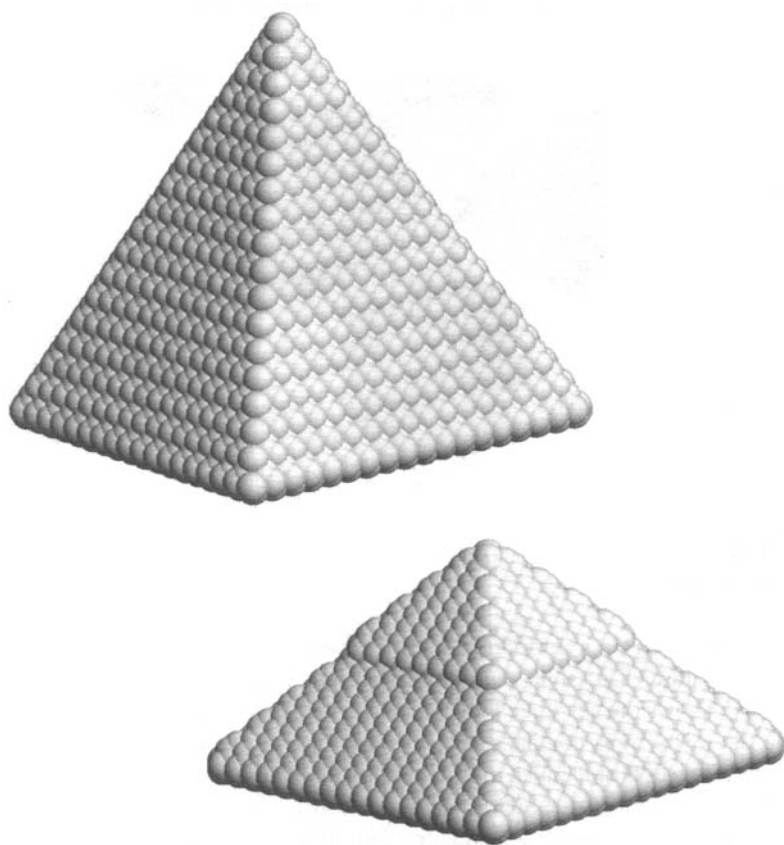


Figure 13.14 The atomic positions for self-assembled pyramidal quantum dots formed from face-centred-cubic materials

13.8 TRANSPORT THROUGH DOT ARRAYS

Thus far, only the zone-centre, i.e. $k = 0$ states have been looked at, but clearly it is possible to calculate transport or other properties which relate to larger momentum states. For example, an earlier figure, i.e. Fig. 13.9, showed the along-axis dispersion curve of a quantum wire. Transport through the dots is possible, and could be studied by using the empirical pseudopotential method, although given the necessary periodicity of the technique, solvable problems would be limited to *arrays* of dots.

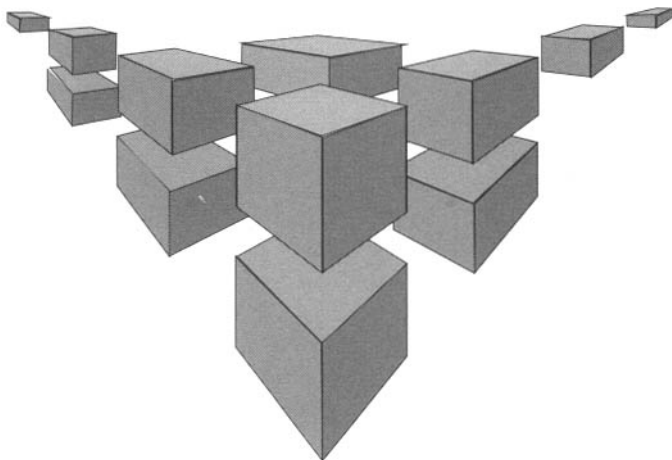


Figure 13.15 A schematic representation of how an array of cubic quantum dots embedded in a host crystal might look

Fig. 13.15 illustrates a periodic array of cubic quantum dots, while Fig. 13.16 indicates the nature of possible solutions for such a system.

Fig. 13.17 shows the calculated dispersion curves for the uppermost valence-band state and the lower most conduction-band state for a periodic array of cubic eight-atom Ge quantum dots embedded in a Si host crystal. The dispersion curves along this [001] axis resemble those of a SiGe superlattice, with a zone-folded conduction-band minimum which is nearer the zone-centre than in the bulk material; the difference here though is that the [010] and [100] directions, i.e. the in-plane directions of the superlattices, also have this same dispersion curve. Thus, the outlying X-valleys in the conduction band are brought nearer the zone centre *in all directions*. This contrasts with those superlattices which retain valleys near the zone boundary for electron motion in the conduction band. It may be expected, therefore, that introducing periodic potentials in all directions, as occurs with arrays of quantum dots, might lead to a more direct bandgap in SiGe, and hence a better quantum efficiency for light emission.

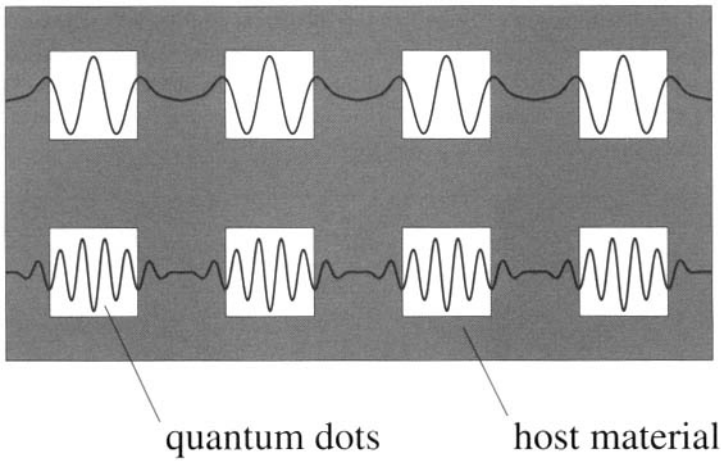


Figure 13.16 Possible transport modes through arrays of quantum dots

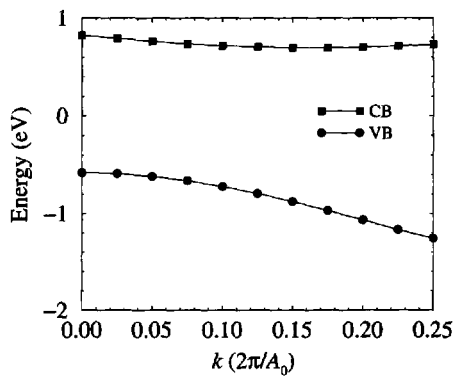


Figure 13.17 Lowest conduction-band (CB) and highest valence band (VB) dispersion curves through a periodic array of Ge quantum dots embedded in a Si host crystal

13.9 ANTI-WIRES AND ANTI-DOTS

The 'reversing' of the materials from which the quantum dots were grown, e.g. depositing a thin wetting layer of Si on to a Ge substrate, could lead to the formation of Si quantum dots. However, as Si has a wider bandgap than Ge, these would repel carriers rather than attracting them, thus leading to use of the term *anti-dots*. *Anti-wires* represent a similar concept in one-dimensional systems. Fig. 13.18 shows possible transport trajectories for the former case.

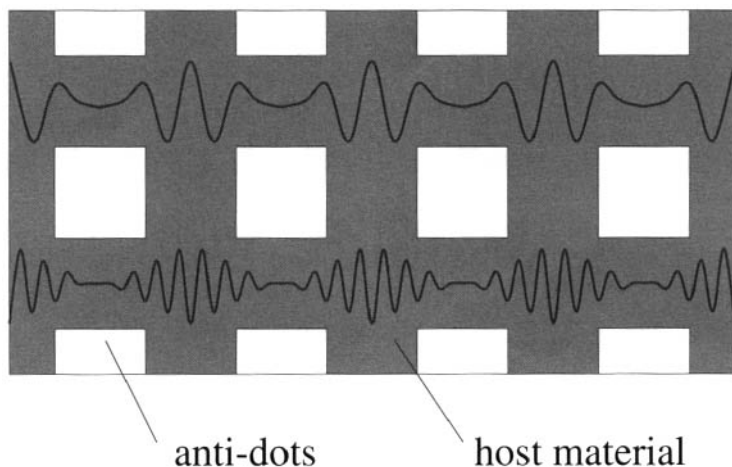


Figure 13.18 Possible transport modes through arrays of anti-dots

The development of dot arrays is really still in its infancy, although work has been reported of transport through both dot [272] and anti-dot [273,274] structures. These systems of dots may have applications in future generations of logic [275,276] and *Coulomb blockade* based [277] devices.

CONCLUDING REMARKS

A detailed study of this book will have revealed that besides containing quite a considerable amount of information, there is also a fair bit missing, and indeed there remains one or two question marks over some of the content that has been presented. The reason for this is that the whole area is still a very active field. It is not 'all done and dusted', and there still remain some major pieces of theory that need to be developed. It is hoped that this present work will provide a stimulus to other workers in the field to fill the gaps—gaps that may well have only become apparent upon reading this summary.

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APPENDIX A

MATERIALS PARAMETERS

The results of the calculations in the main text depend on the assumed material parameters, a brief summary of which are presented here.

GaAs/Ga_{1-x}Al_xAs

- Bandgap, $E_g = (1.426 + 1.247x)$ eV
- Band alignment: 33% of total discontinuity in valence band, i.e. $\Delta V_{VB} = 0.33$; $\Delta V_{CB} = 0.67$
- Electron effective mass, $m^* = (0.067 + 0.083x) m_0$
- Heavy-hole effective mass, $m^* = (0.62 + 0.14x) m_0$
- Lattice constant $A_0 = 5.65$ Å
- Low frequency (static) dielectric constant $\epsilon_s = 13.18\epsilon_0$
- High frequency dielectric constant $\epsilon_\infty = 10.89\epsilon_0$

- Material density $\rho = 5317.5 \text{ kgm}^{-3}$
- Longitudinal Optical (LO) phonon energy $E_{LO} = 36 \text{ meV}$
- Deformation potential $D_A = 7.0 \text{ eV}$
- Velocity of sound $v_s = 5117.0 \text{ m}^{-1}$

CdTe/Cd_{1-x}Mn_xTe

- Bandgap $E_g = (1.606 + 1.587x) \text{ eV}$
- Band alignment: 30% of total discontinuity in valence band, i.e. $\Delta V_{VB} = 0.30$; $\Delta V_{CB} = 0.70$
- Electron effective mass, $m^* = (0.11 + 0.067x) m_0$
- Heavy-hole effective mass, $m^* = (0.60 + 0.21x + 0.15x^2) m_0$

In_{1-x-y}Al_xGa_yAs/AlAs

- Total band discontinuity, $\Delta V = [2.093x + 0.629y + 0.577x^2 + 0.436y^2 + 1.013xy - 2.0x^2(1 - x - y)] \text{ eV}$
- Band alignment: 47% of total discontinuity in valence band, i.e. $\Delta V_{VB} = 0.47$; $\Delta V_{CB} = 0.53$
- Electron effective mass, $m^* = (0.0427 + 0.0685x) m_0$

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